

# Investigation of magnetostatic interactions in NiFe/Au/Co/Au multilayers

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In cooperation between:

Institute of Molecular Physics, Polish Academy of Sciences, Poznań, Poland

Department of Physics , University of Kassel, Kassel, Germany

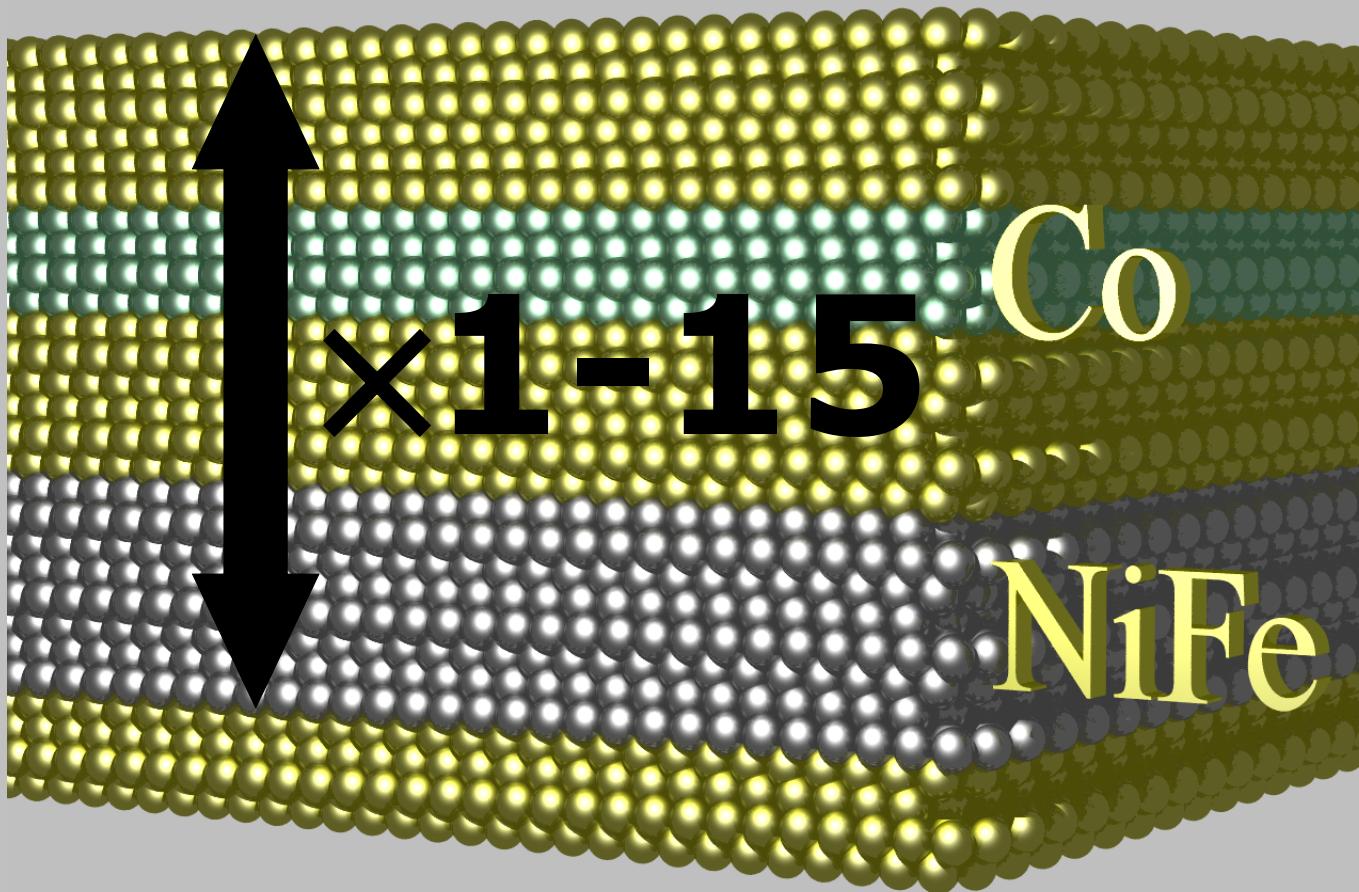
Institute of Electronic Materials Technology, Warszawa, Poland

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# Investigation of magnetostatic interactions in NiFe/Au/Co/Au multilayers

- Introduction
- NiFe/Au/Co/Au multilayers
- Magnetic and transport properties
- Mössbauer spectroscopy investigations
- SXRMS measurements
- Conclusions

# Introducing $[NiFe/Au/Co/Au]_N$ – the structure



magnetron sputtering

Substrate:  
naturally oxidized  
Si(100)

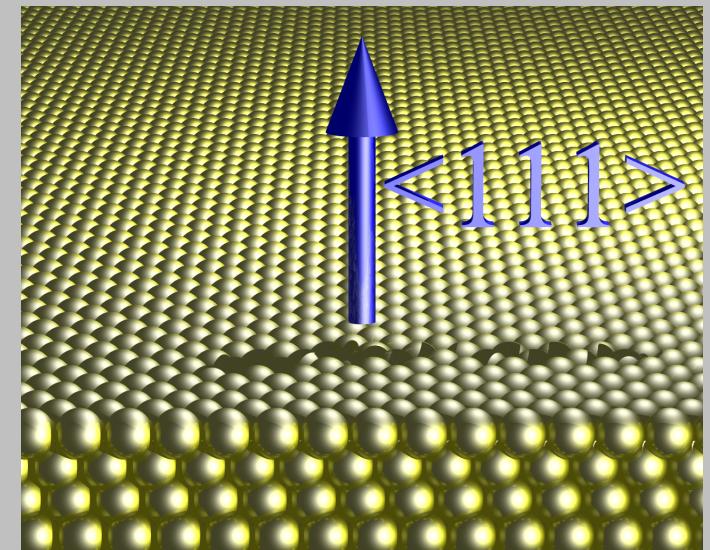
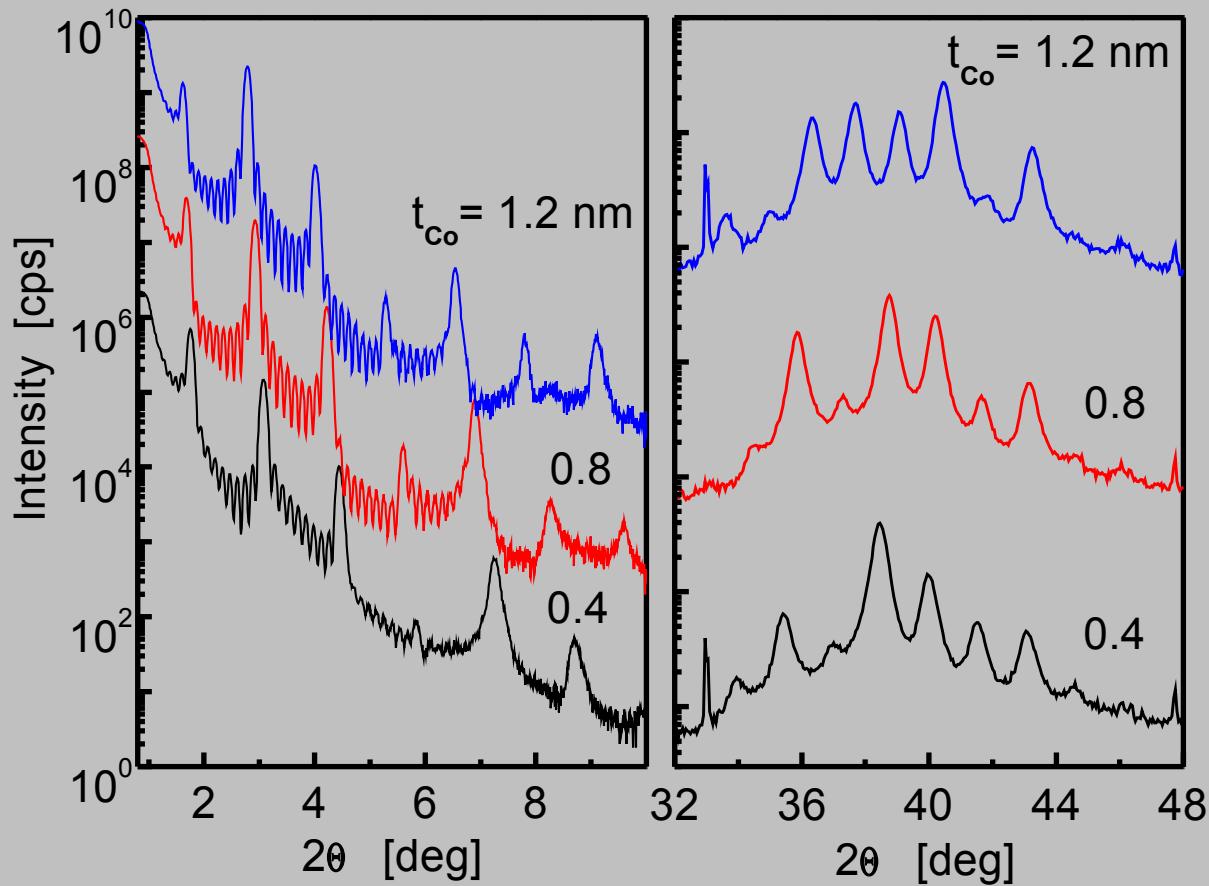
$$t_{Co} = 0.2-1.5 \text{ nm}$$

$$t_{Au} = 1.5-3 \text{ nm}$$

$$t_{NiFe} = 0.5-4 \text{ nm}$$

$(Ni_{80}Fe_{20})$

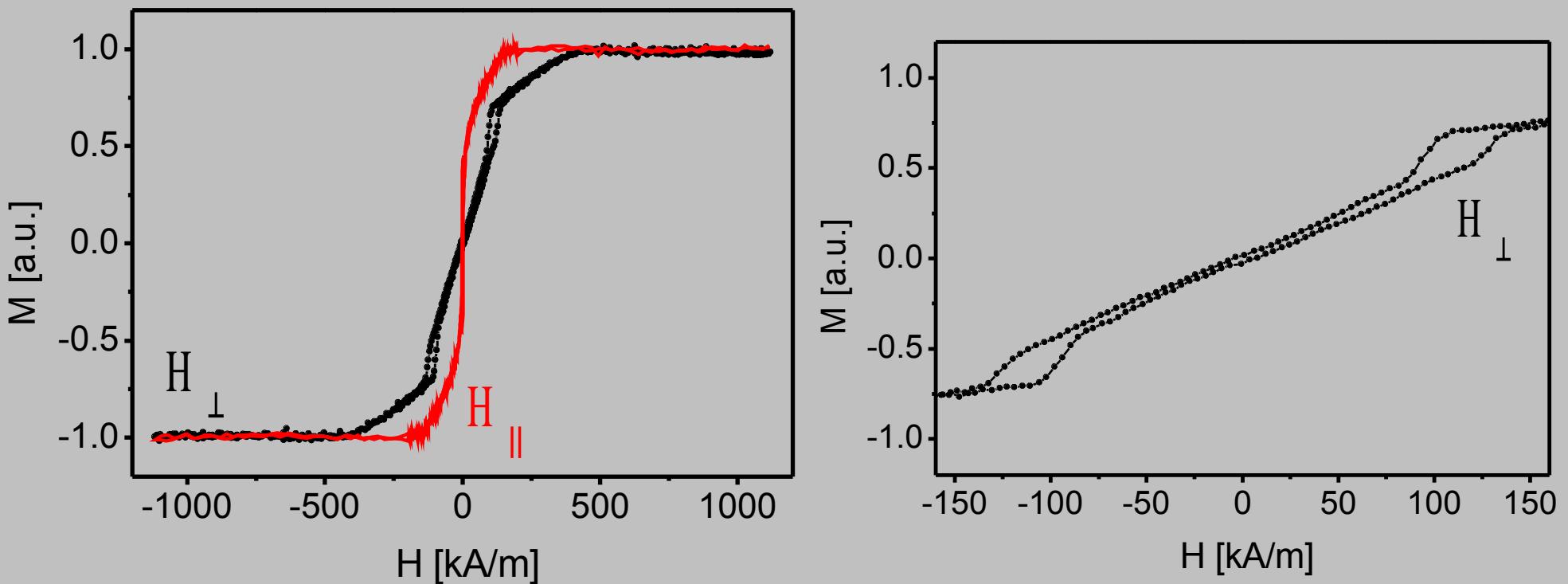
# Introducing $[NiFe/Au/Co/Au]_N$ – the structure



$[Ni_{80}Fe_{20}(2 \text{ nm})/Au(1.9 \text{ nm})/Co(t_{Co})/Au(1.9 \text{ nm})]_{10}$

Cu K $\alpha$

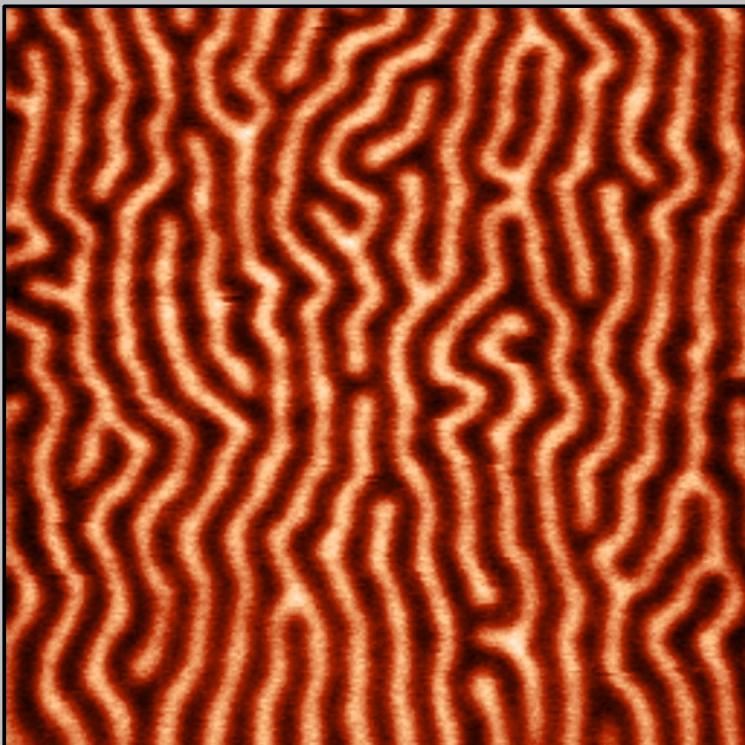
# Introducing $[NiFe/Au/Co/Au]_N$ - magnetic properties



- Small field range of hysteresis with field applied perpendicularly is characteristic of systems with **stripe domains**.
- In both field configurations NiFe and Co layers reverse quasi independently.

$[Ni_{80}Fe_{20}(2 \text{ nm})/Au(1.9 \text{ nm})/Co(0.8 \text{ nm})/Au(1.9 \text{ nm})]_{10}$

# Introducing [NiFe/Au/Co/Au]<sub>N</sub> – magnetic properties



AC demagnetized sample  
 $5 \times 5 \mu\text{m}^2$

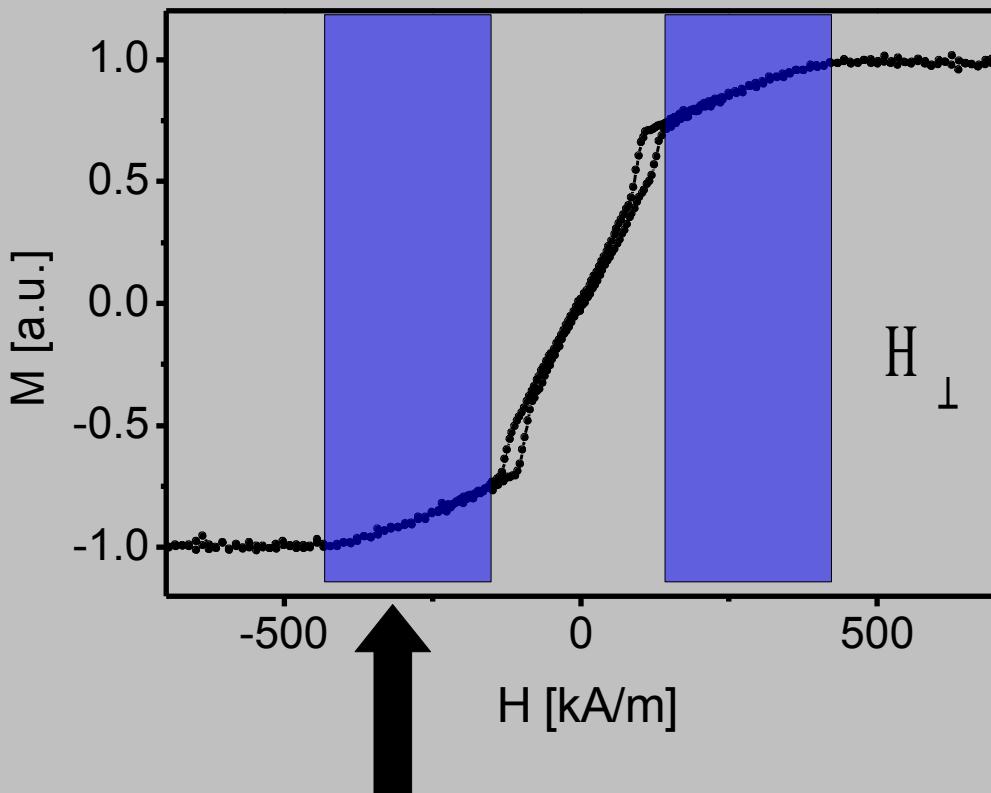
period 400-1000 nm

- MFM measurements prove the presence of the **stripe domains** which are characteristic for systems with perpendicular anisotropy.
- A period of the domain structure depends strongly on the thickness of Co and Au layers.

$[\text{Ni}_{80}\text{Fe}_{20}*(2 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(1.2 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

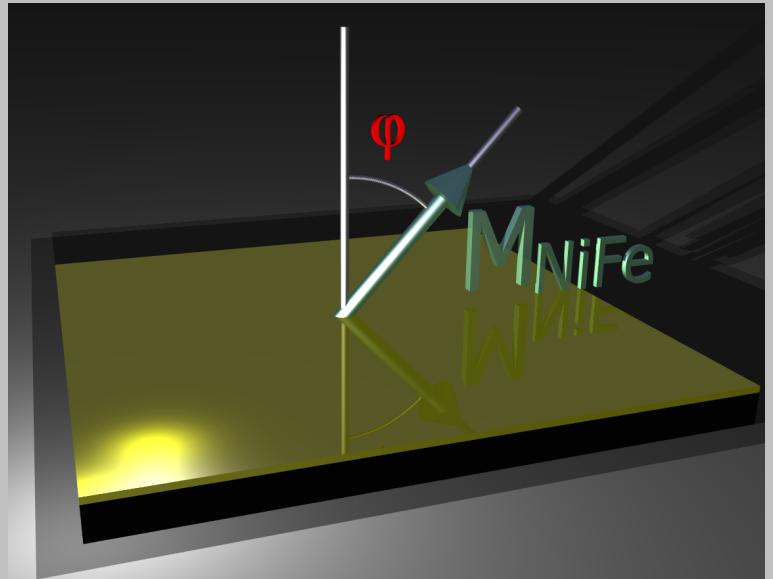
\* with  $^{57}\text{Fe}$

# Introducing $[NiFe/Au/Co/Au]_N$ - magnetic properties



Reversal of NiFe only

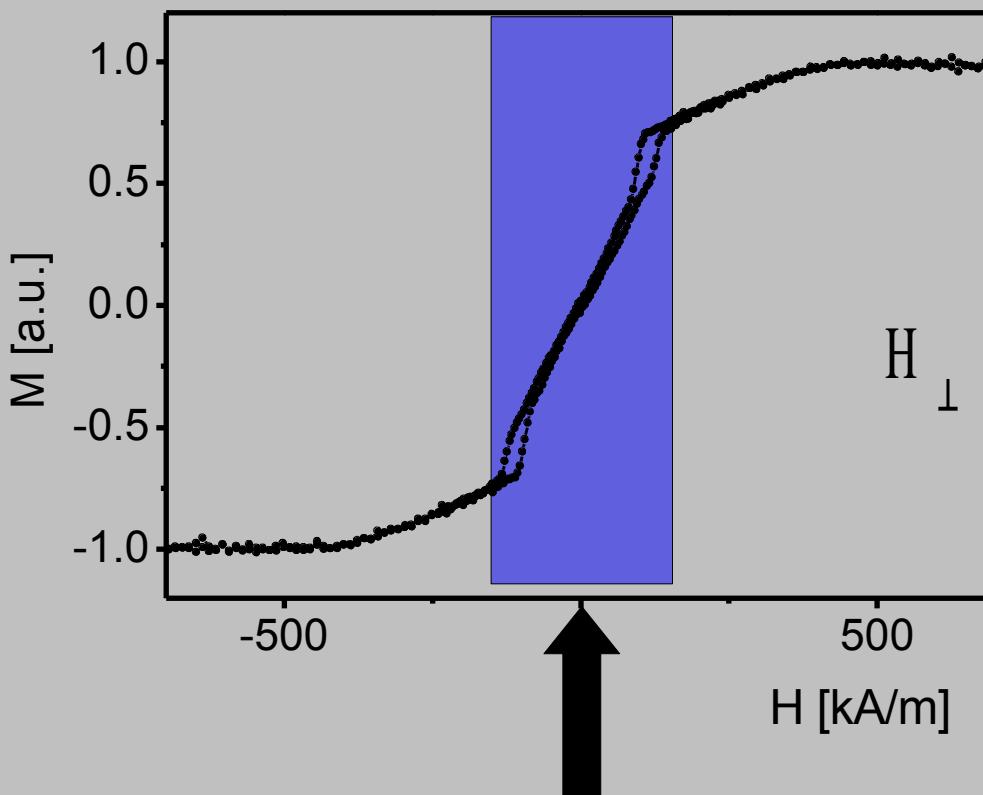
$$K_u = \frac{1}{2} \mu_0 (M_S^{NiFe})^2$$



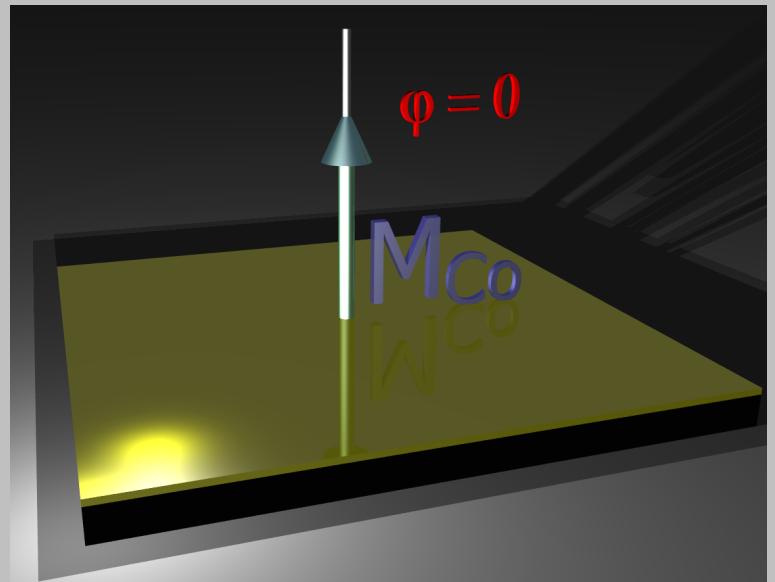
Shape anisotropy:

$$\cos(\varphi) = \frac{H}{M_S}$$

# Introducing $[NiFe/Au/Co/Au]_N$ - magnetic properties

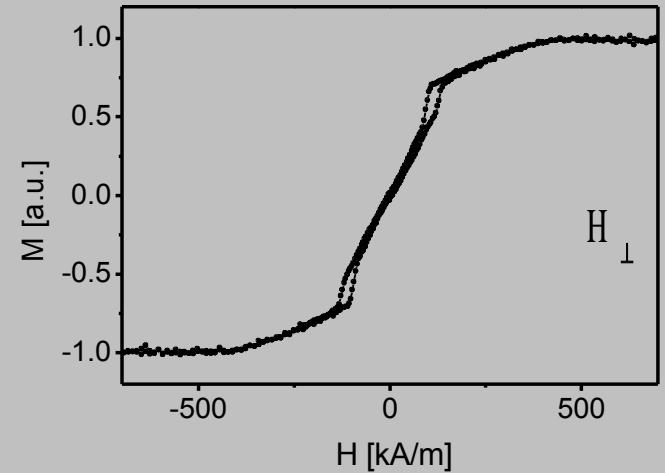
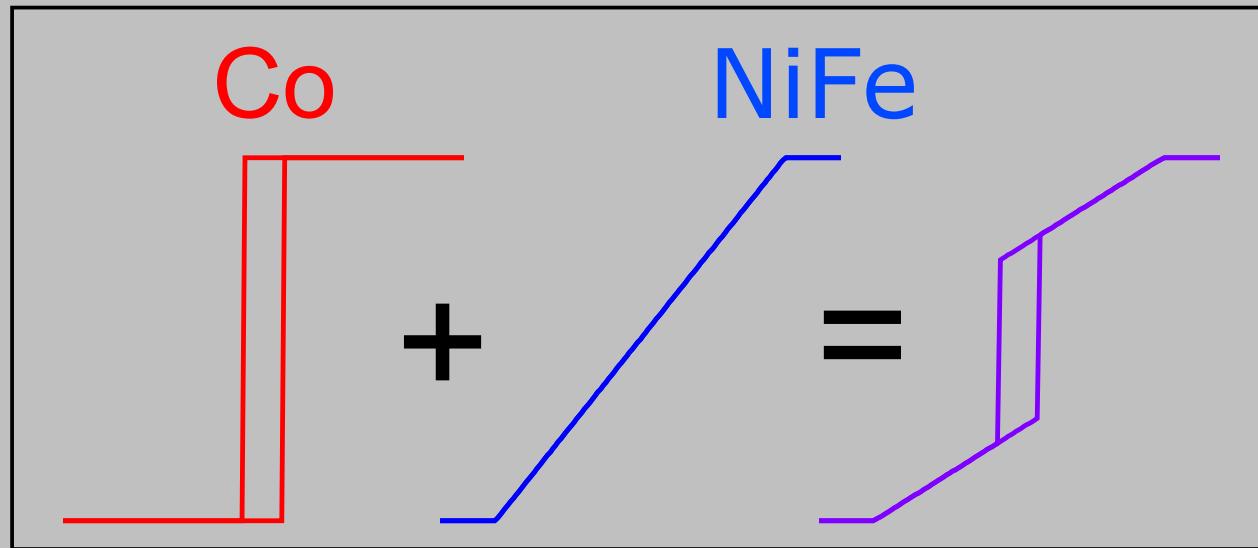


Simultaneous reversal  
of  
NiFe and Co



An easy axis of the Co layers  
is perpendicular to surface of  
multilayer.

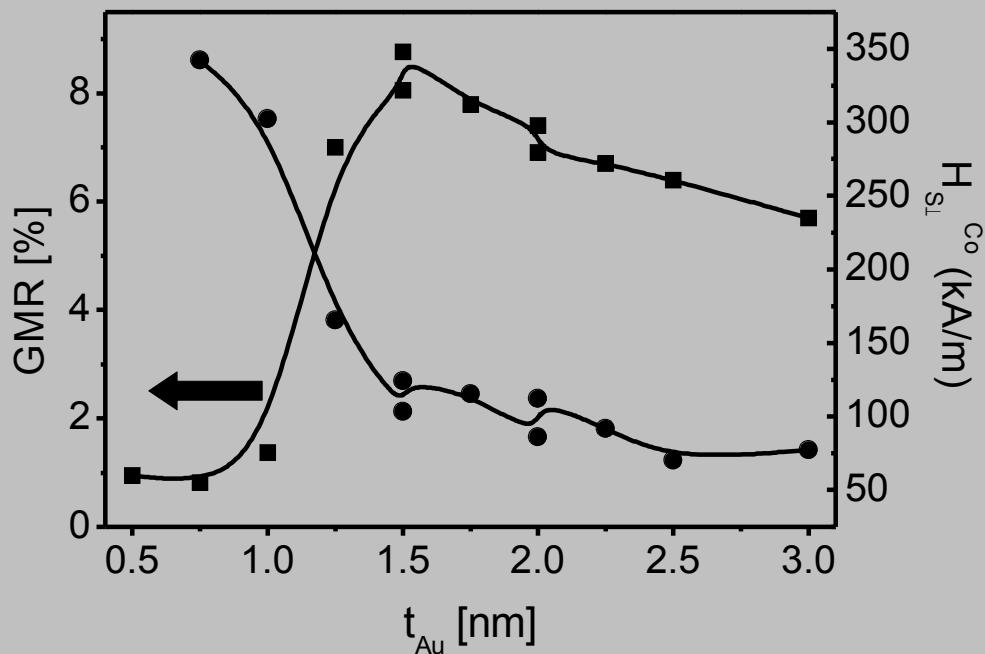
# Introducing $[NiFe/Au/Co/Au]_N$ – magnetic properties



In the first approximation Co and NiFe layers can be thought of as uncoupled.

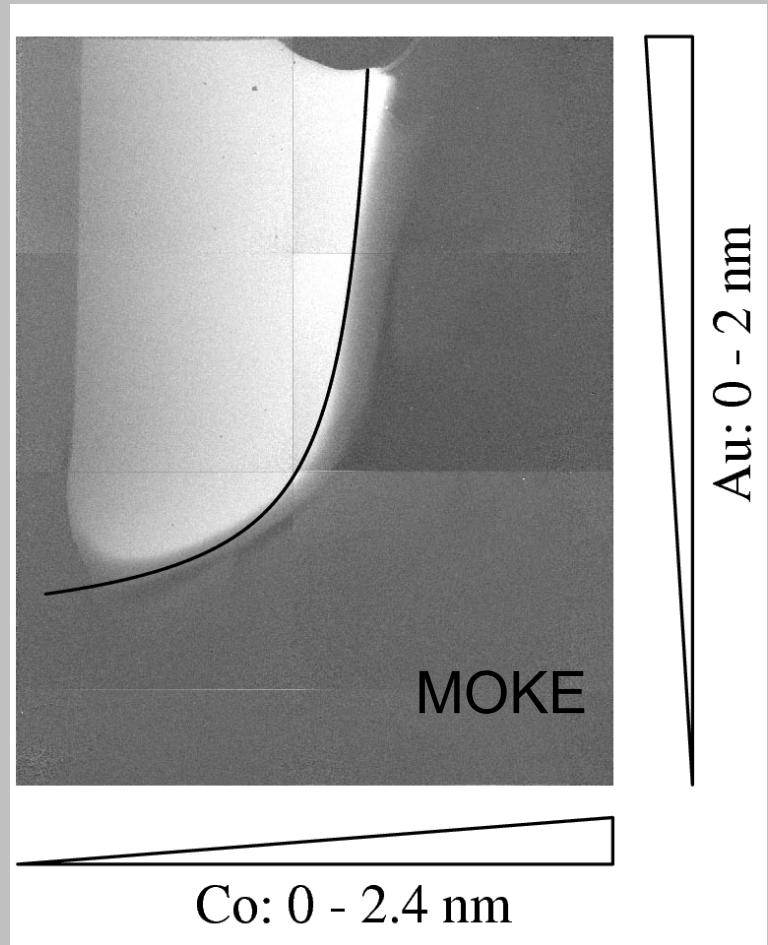
$M(H)$  dependence of the  $NiFe/Au/Co$  structure is then an arithmetic sum of the  $M(H)$  dependencies of Co and NiFe layers.

# Introducing $[NiFe/Au/Co/Au]_N$ - direct NiFe-Co coupling



$[Ni_{80}Fe_{20}(2 \text{ nm})/Au(t_{Au})/Co(0.6 \text{ nm})/Au(t_{Au})]_{15}$

For small  $t_{Au}$  ferromagnetic bridges (pinholes) lead to the direct coupling between Co and NiFe layers.



**White - magnetic moments perpendicular to the sample plane**

# Giant Magnetoresistance (GMR)- Nobel 2007 (Fert, Grünberg)

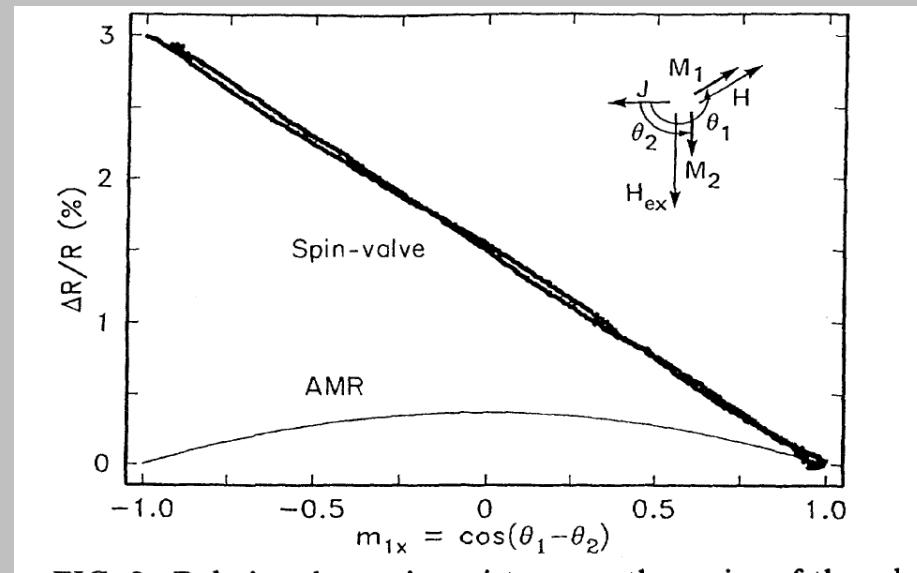
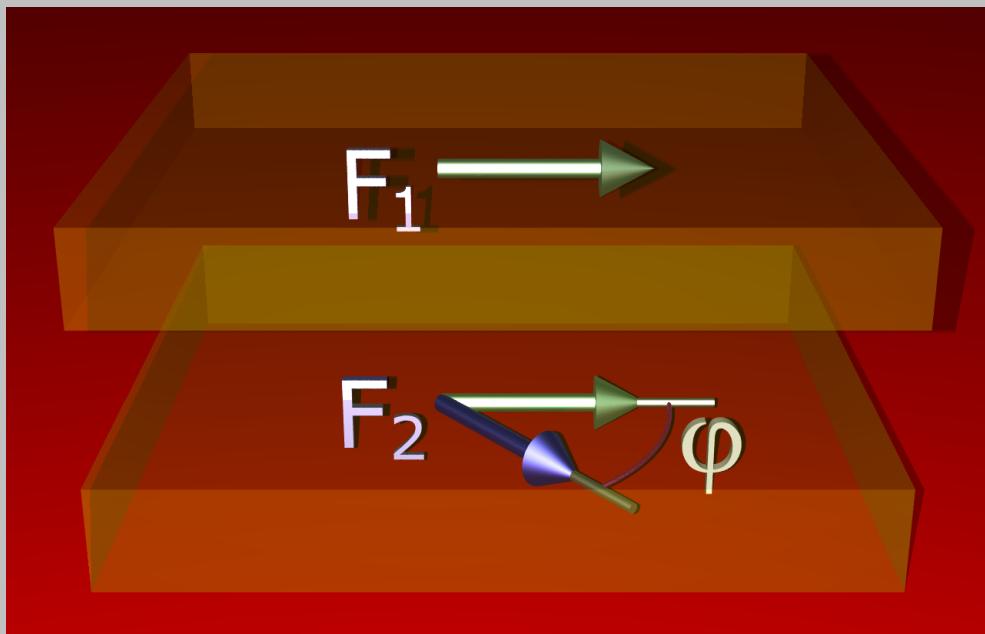
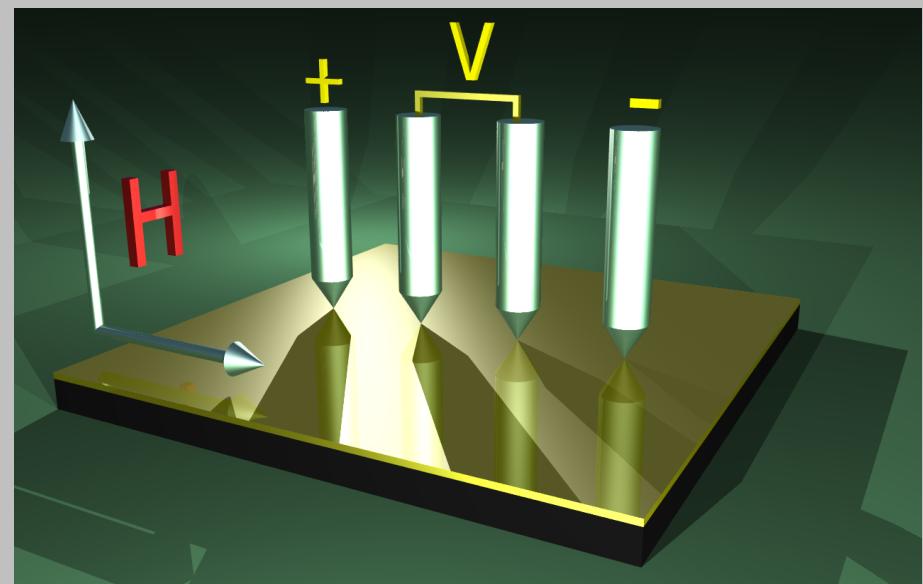
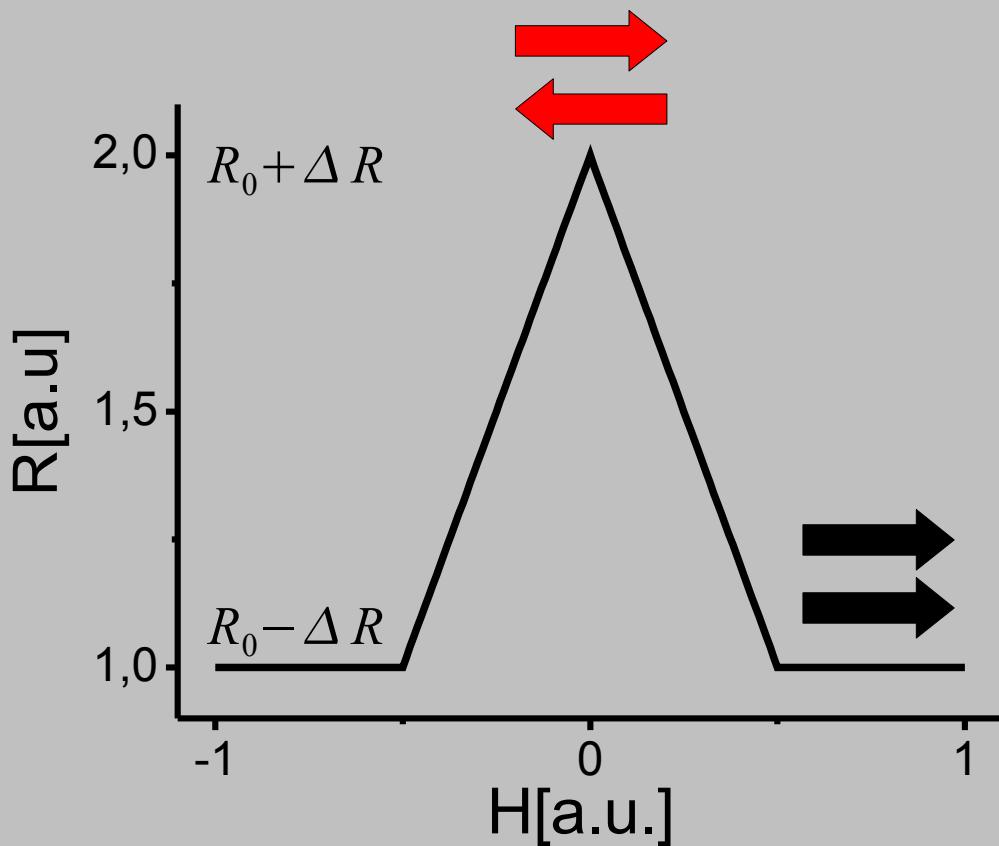


FIG. 2. Relative change in resistance vs the cosine of the relative angle between the magnetizations of the two NiFe layers of Si/(60-Å NiFe)/(26-Å Cu)/(30-Å NiFe)/(60-Å FeMn)/(20-Å Ag). Inset shows the orientation of the current  $J$ , exchange field  $H_{ex}$ , applied field  $H$ , and magnetizations  $M_1$  and  $M_2$ .

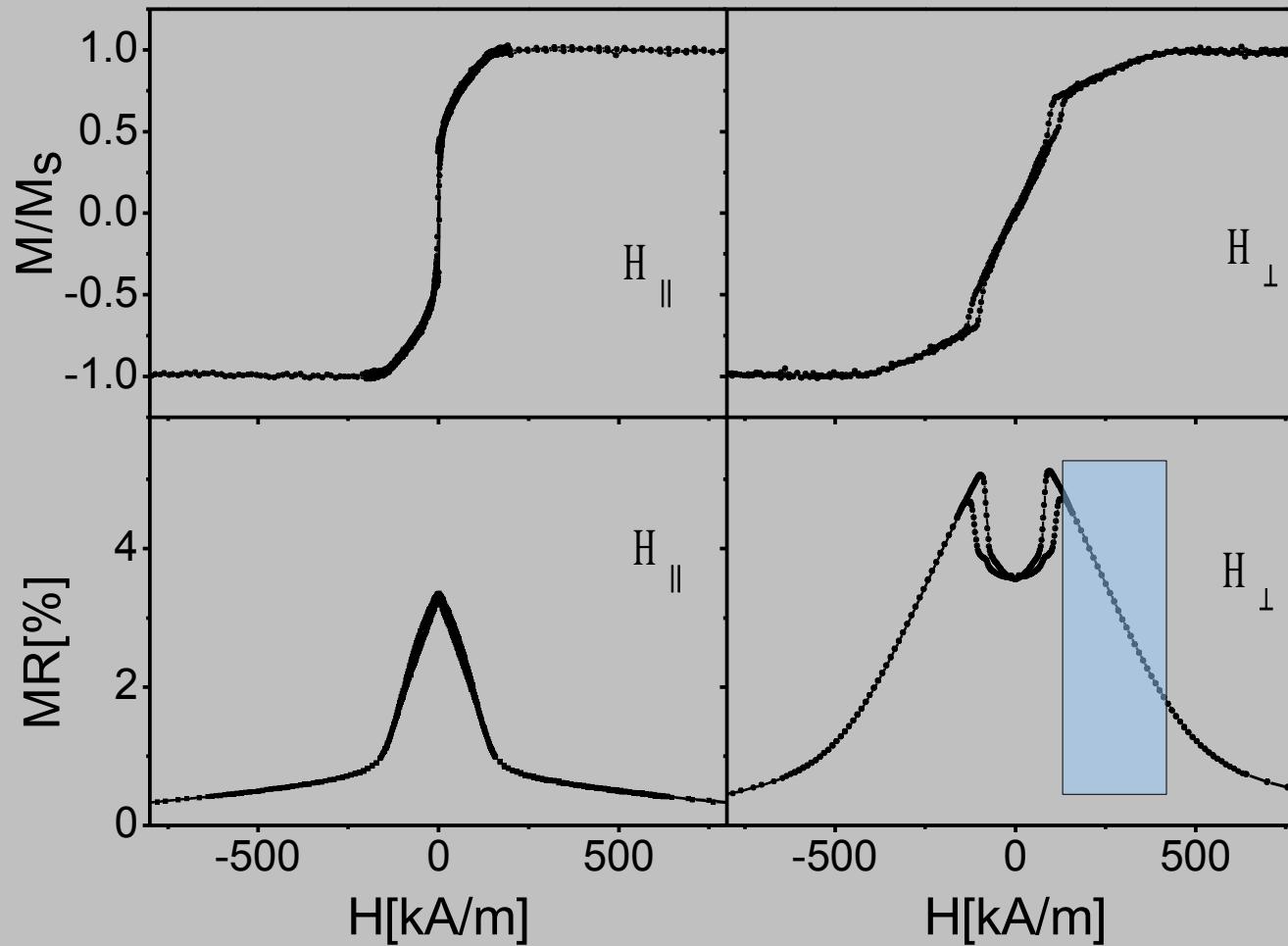
$$\Delta R \propto \cos(\varphi)$$

# Giant Magnetoresistance (GMR)- Nobel 2007 (Fert, Grünberg)



$$R = R_0 - \Delta R \cos(\varphi)$$

# Giant Magnetoresistance in NiFe/Au/Co/Au



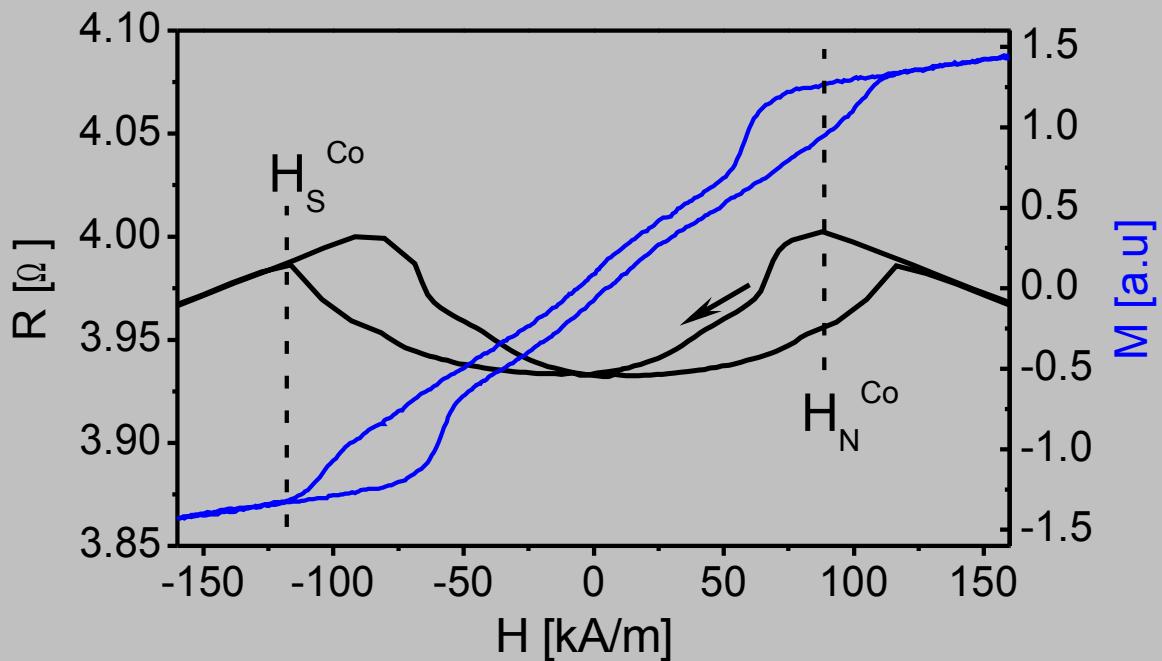
Broad linearity range  
in  $R(H)$  dependence:

- magnetic layer magnetized along hard axis

- no hysteresis in linear range

$[\text{Ni}_{80}\text{Fe}_{20}(2 \text{ nm})/\text{Au}(1.9 \text{ nm})/\text{Co}(1 \text{ nm})/\text{Au}(1.9 \text{ nm})]_{10}$

# Giant Magnetoresistance in NiFe/Au/Co/Au

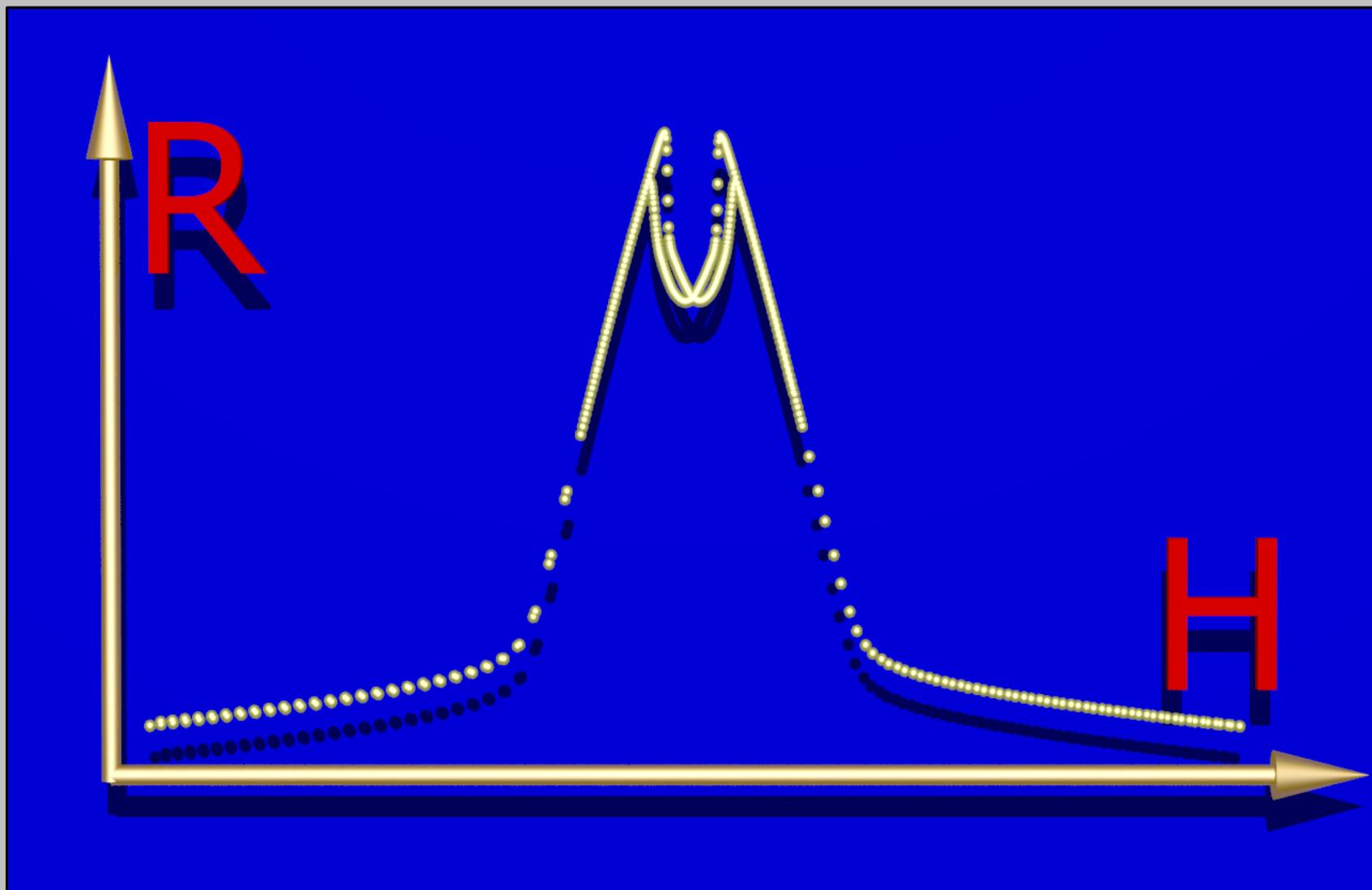


There is a local minimum of resistance in the  $R(H)$  dependence.

The nucleation field (creation of the domain structure) and the annihilation field (saturation of Co layers) are visible both in  $R(H)$  and  $M(H)$  dependencies.

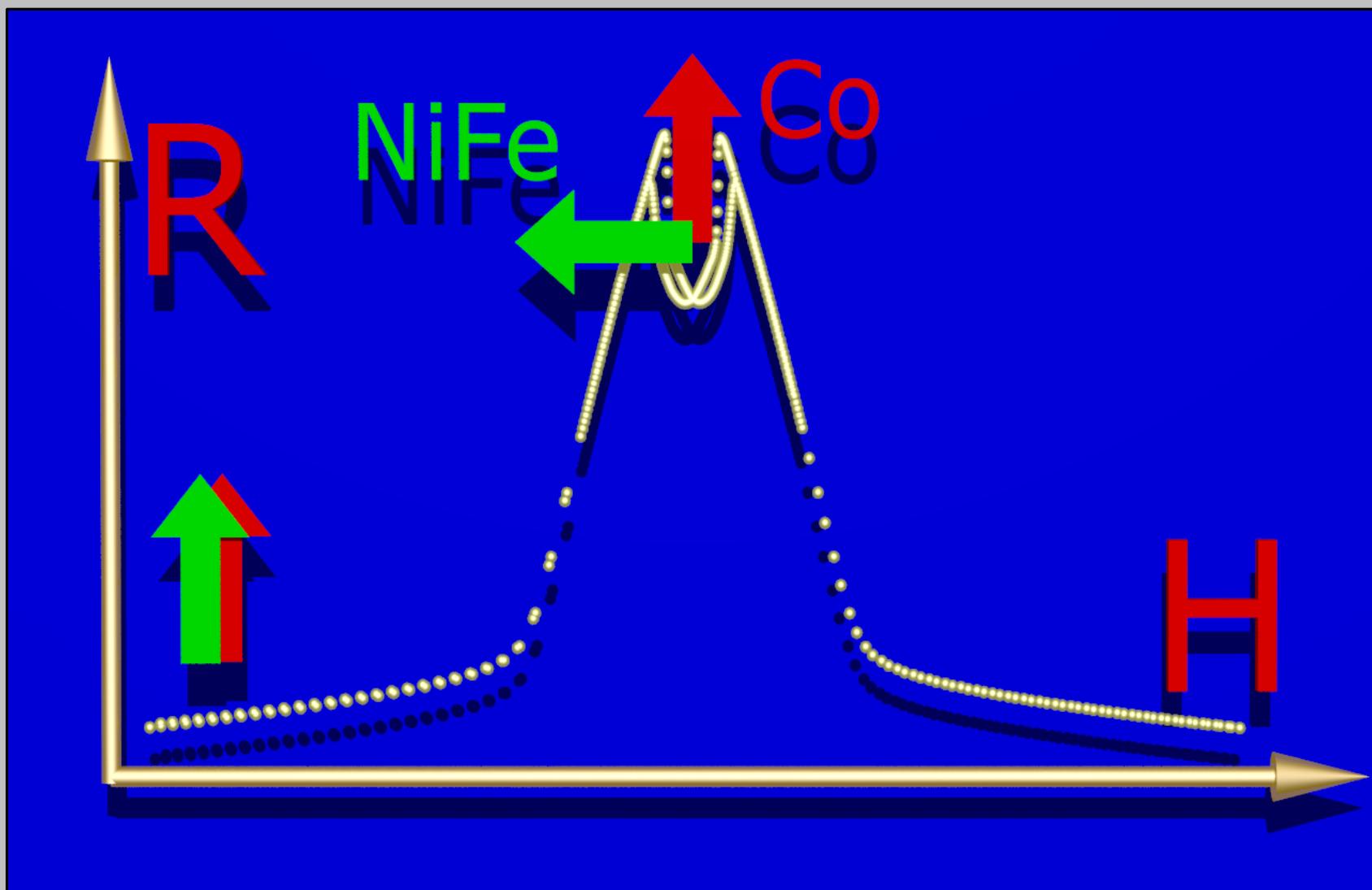
# Giant Magnetoresistance in NiFe/Au/Co/Au

Explaining the  $R(H)$  dependence



# Giant Magnetoresistance in NiFe/Au/Co/Au

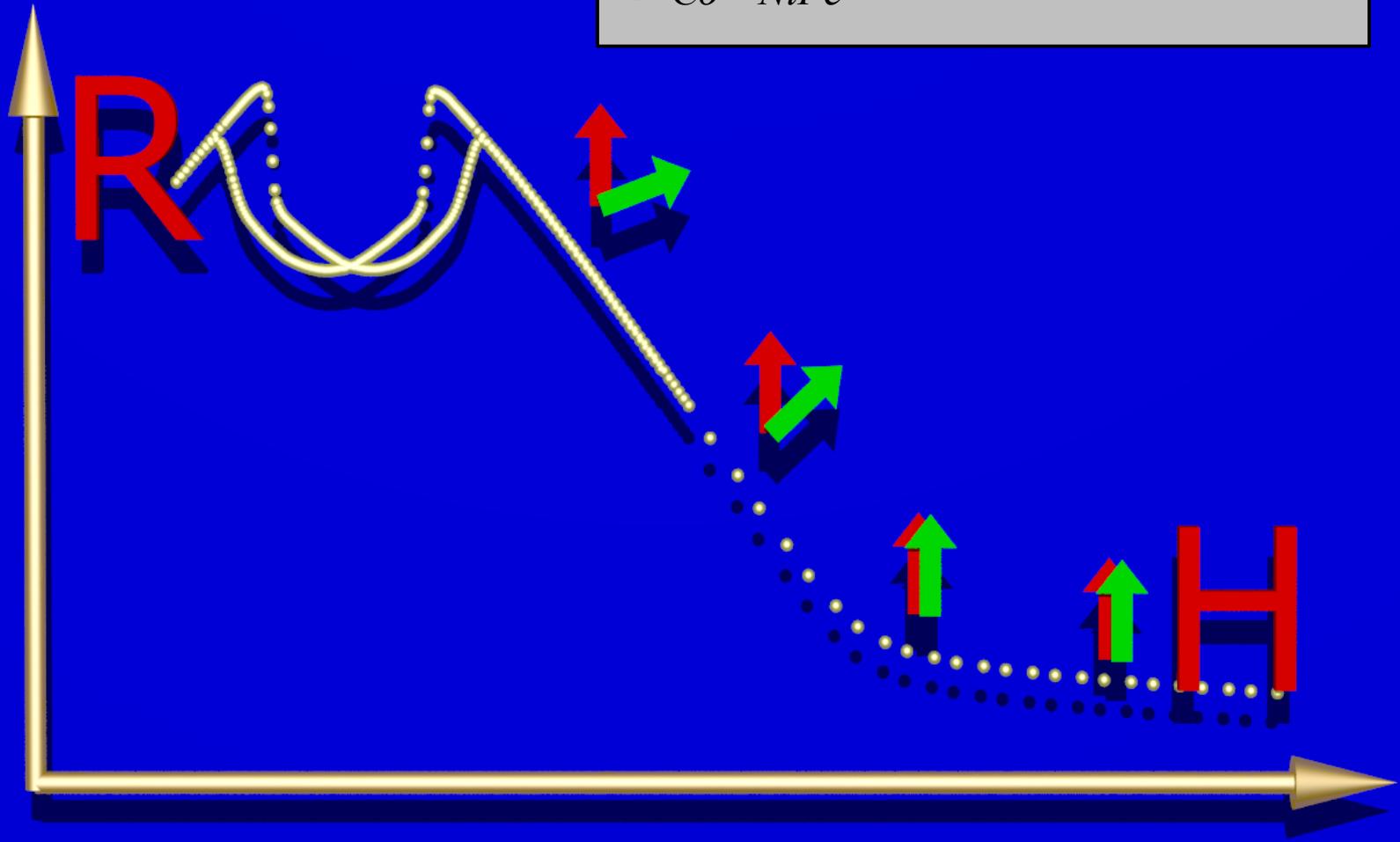
Explaining the R(H) dependence



# Giant Magnetoresistance in NiFe/Au/Co/Au

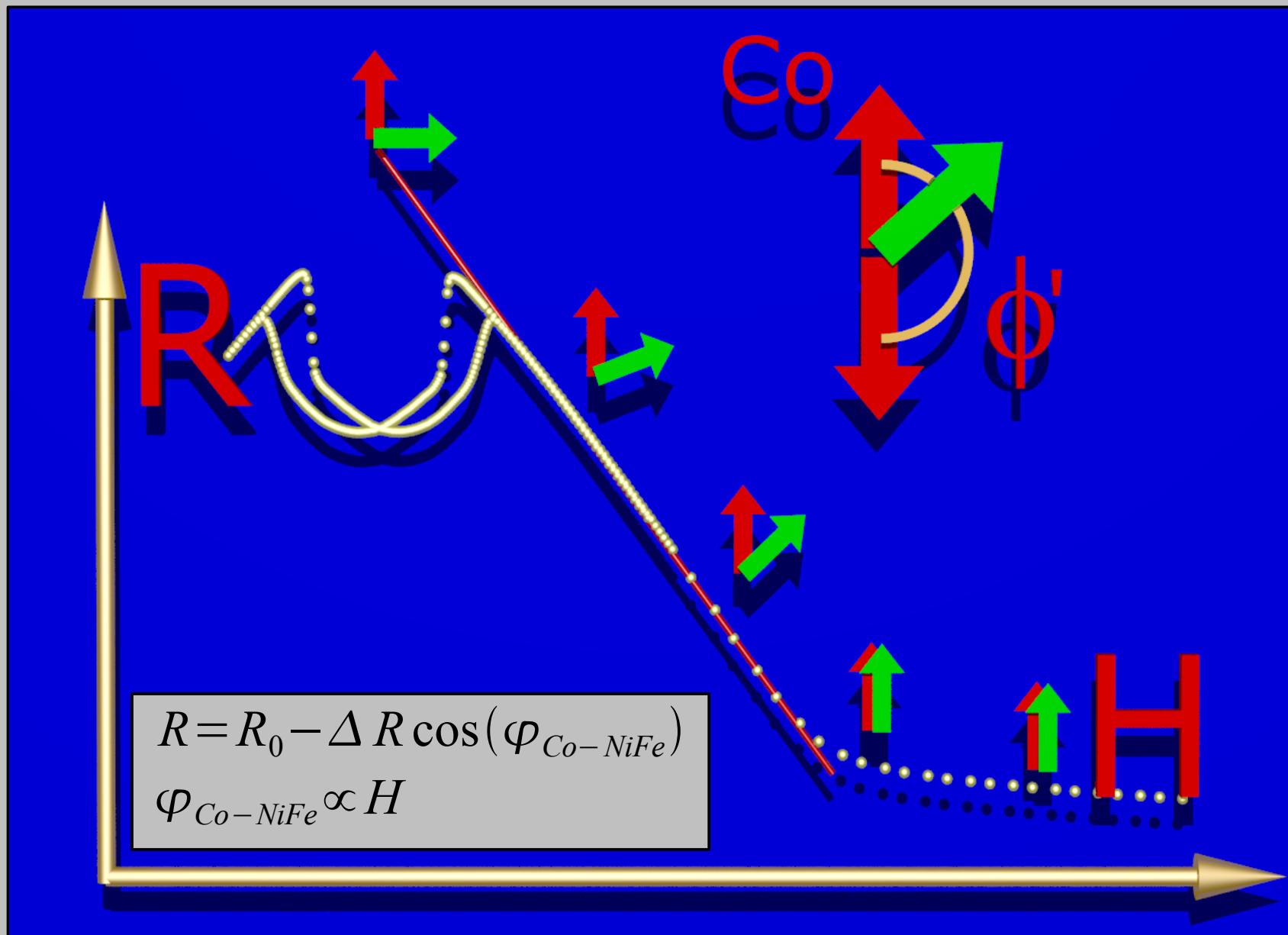
## Explaining the R(H) dependence

$$R = R_0 - \Delta R \cos(\varphi_{Co-NiFe})$$
$$\varphi_{Co-NiFe} \propto H$$

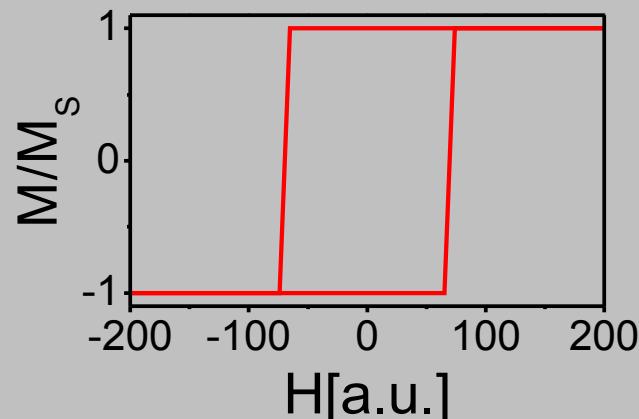


# Giant Magnetoresistance in NiFe/Au/Co/Au

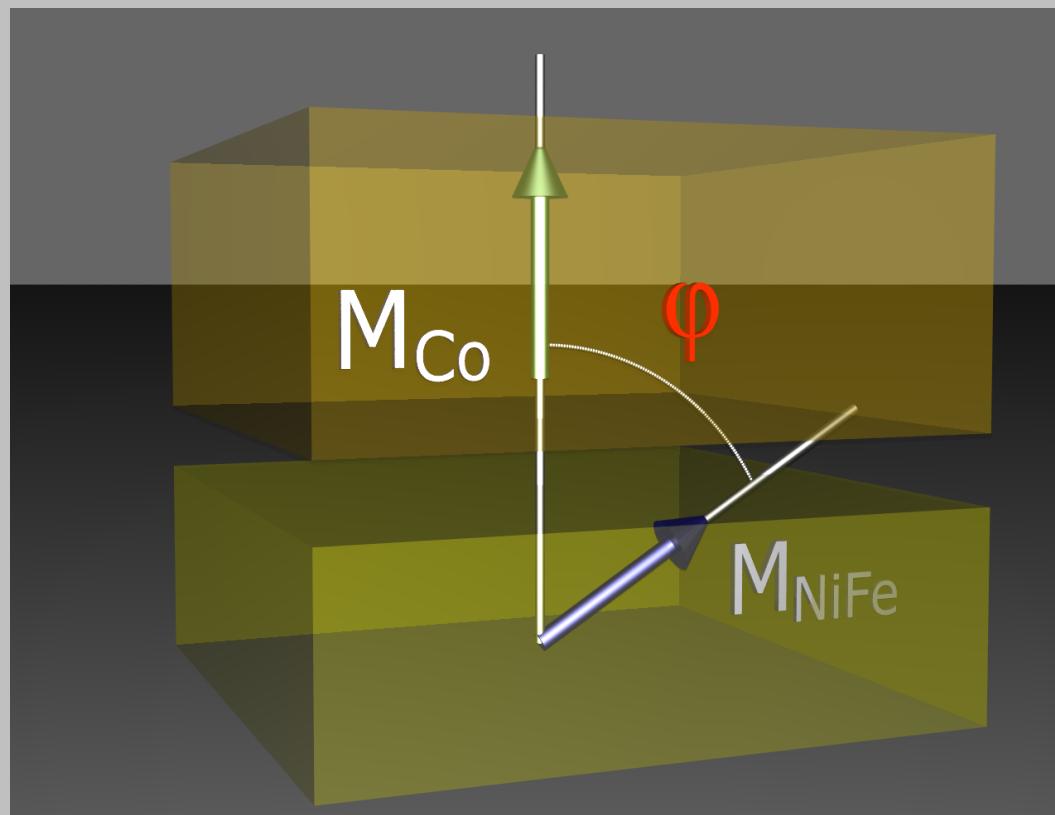
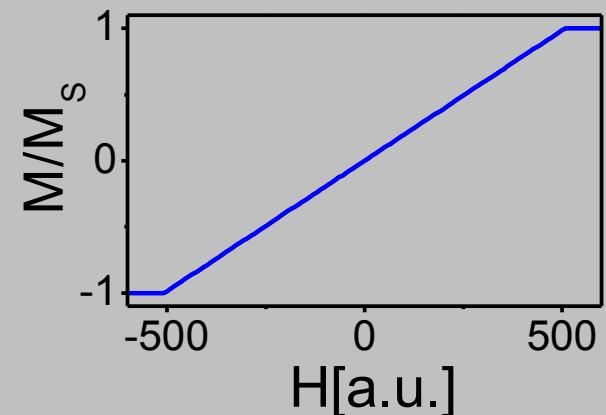
## Explaining the R(H) dependence



# Giant Magnetoresistance in NiFe/Au/Co/Au-model



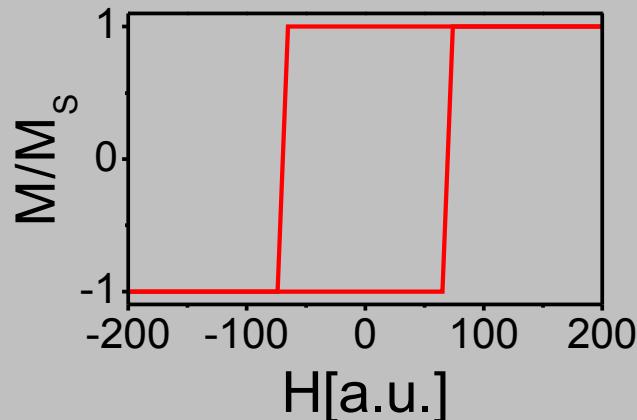
$$\Delta R \propto \cos(\varphi)$$



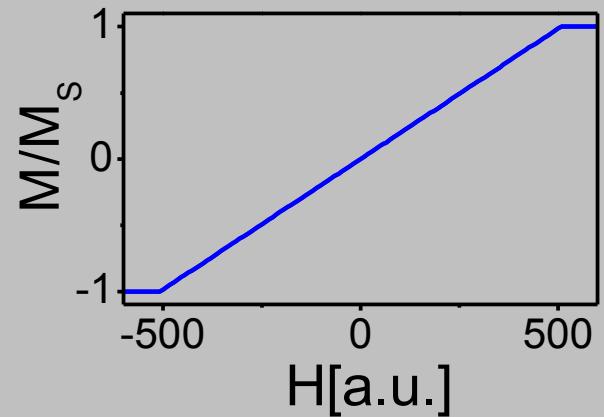
Model assumptions:

- magnetic moments in Co layers are perpendicular to the plane of the sample
- NiFe layer: easy-plane shape anizotropy
- magnetic field applied perpendicularly

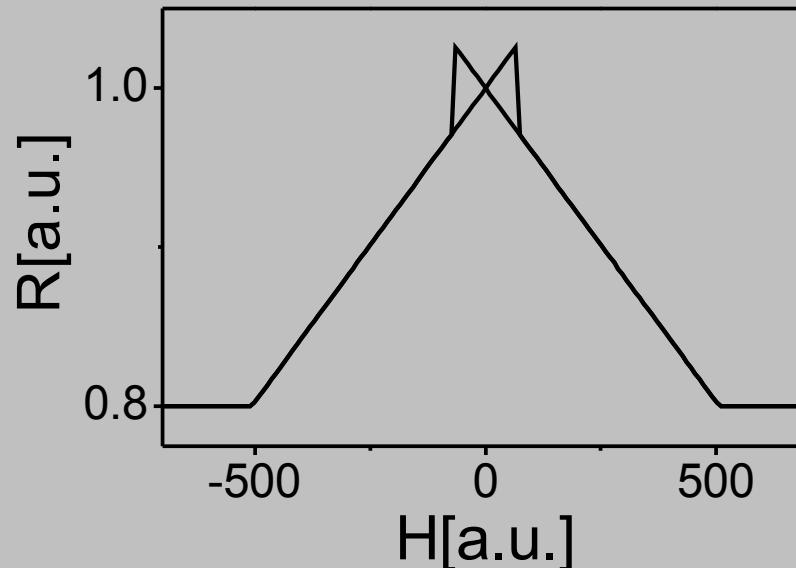
# Giant Magnetoresistance in NiFe/Au/Co/Au-model



$$\Delta R \propto \cos(\varphi)$$

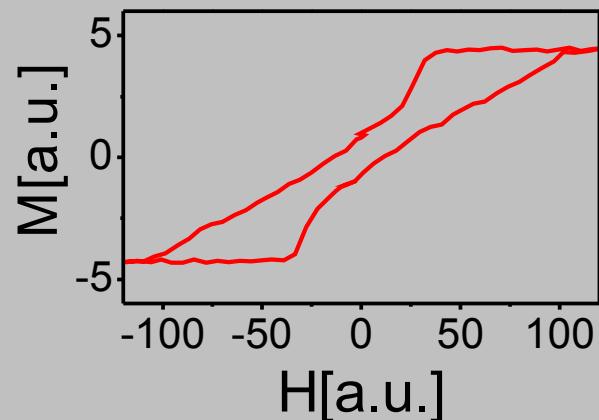


$$R(H) = a(R_0 - \Delta R \cdot \cos(\varphi)) + (1-a)(R_0 + \Delta R \cdot \cos(\varphi))$$

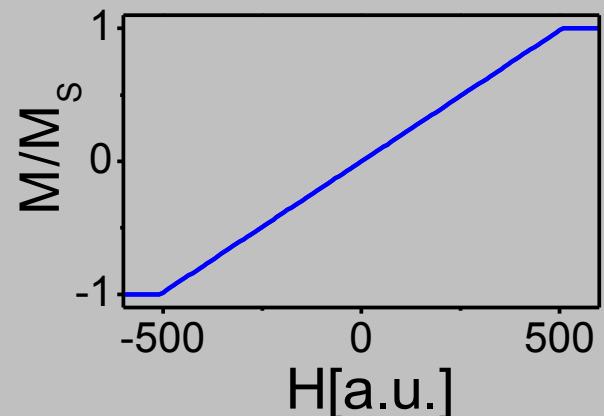


$$a(H) = \frac{1}{2} \left( \frac{M^{Co}(H)}{M_S^{Co}} + 1 \right)$$

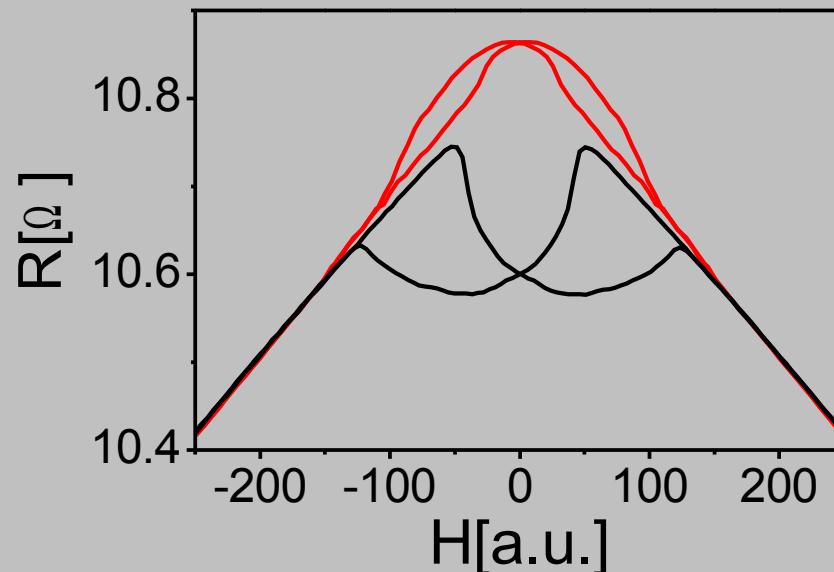
# Giant Magnetoresistance in NiFe/Au/Co/Au-model



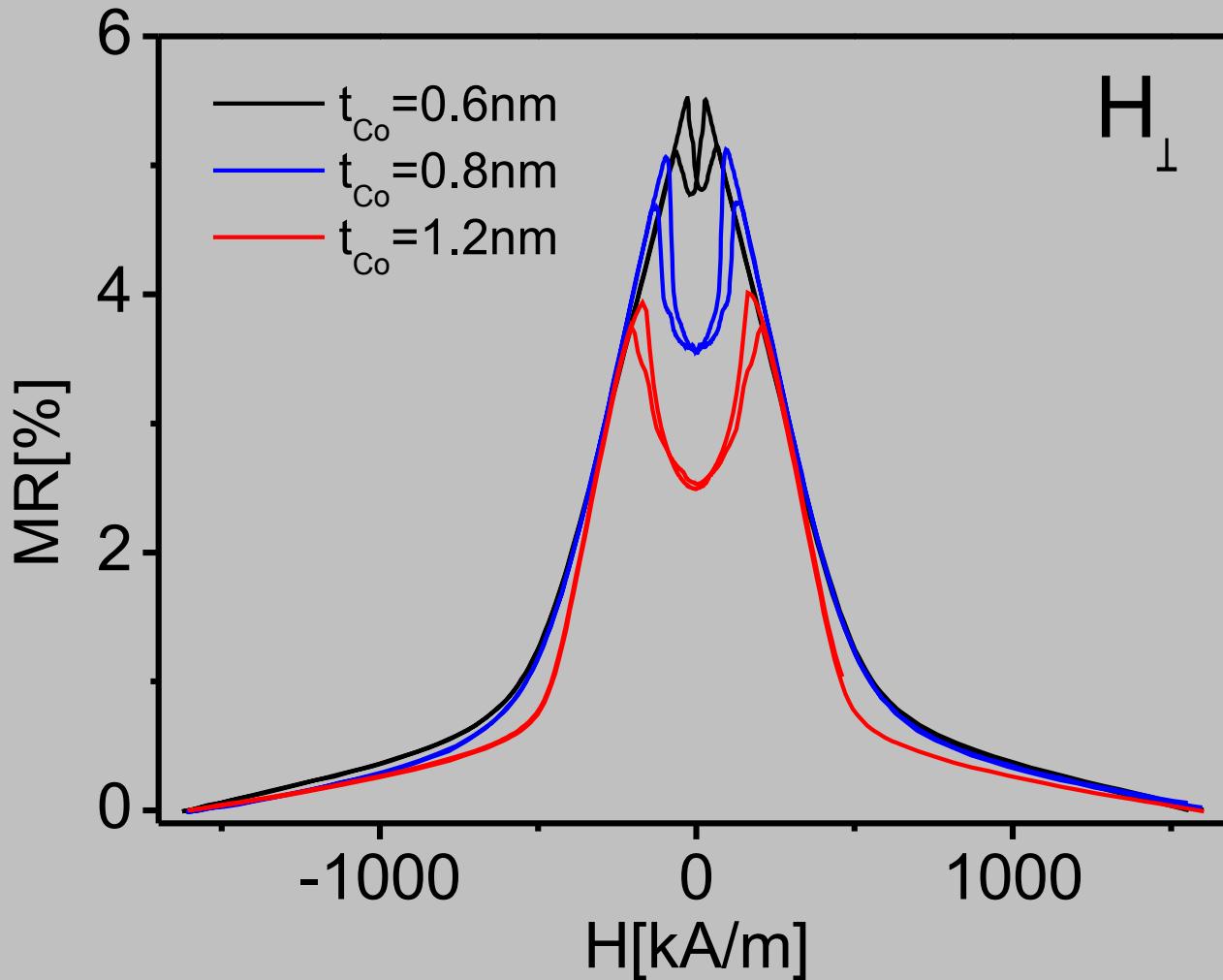
$$\Delta R \propto \cos(\varphi)$$



The independent reversal of Co in NiFe does not result in the local minimum of resistance



# Giant Magnetoresistance in NiFe/Au/Co/Au

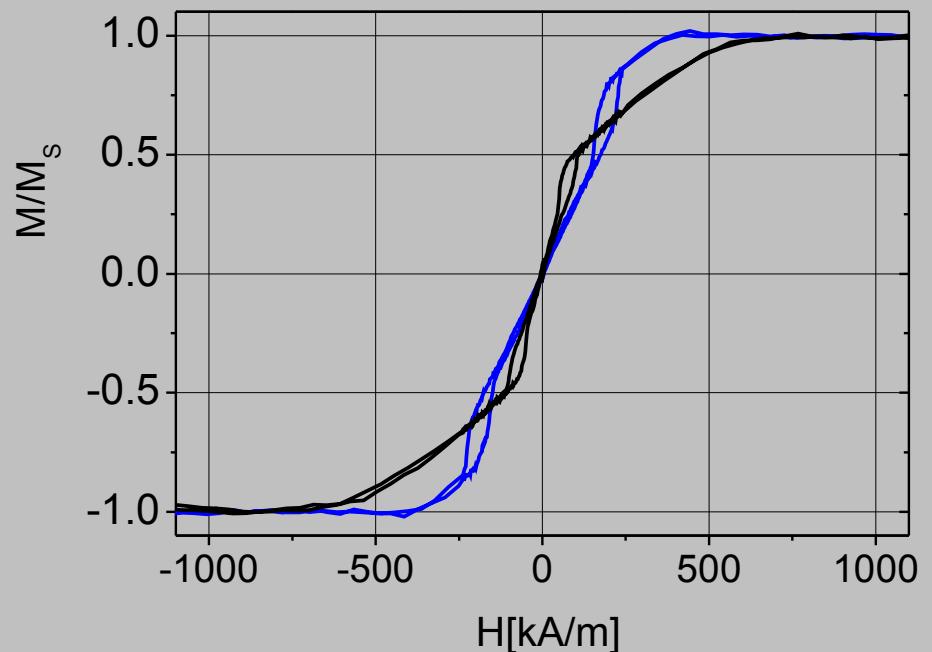
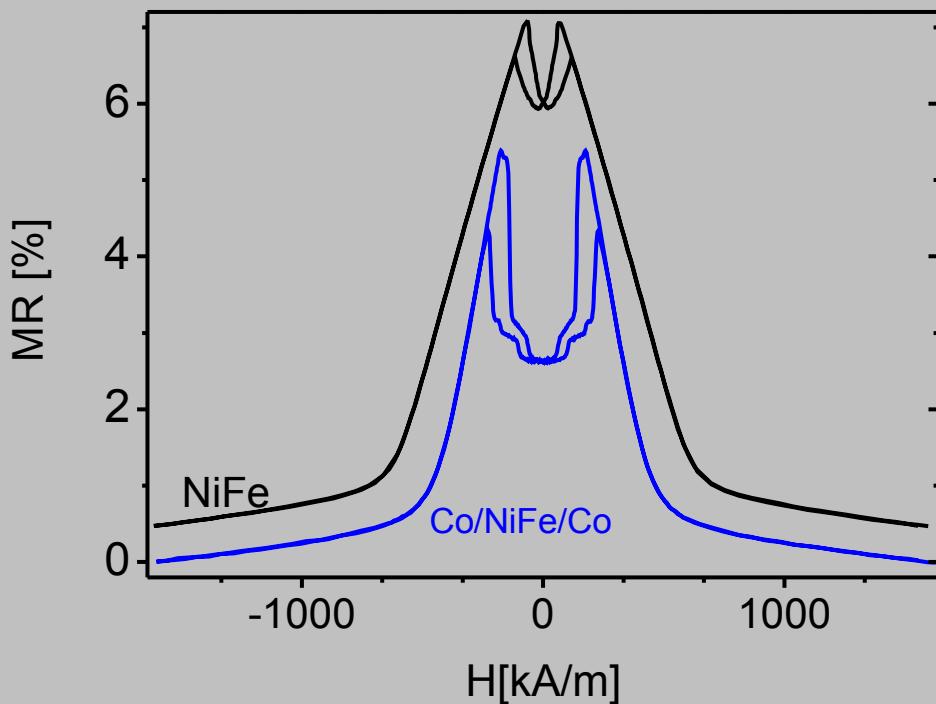


A relative "depth" of resistance minimum is a strong function of Co layers thickness.

$[\text{Ni}_{80}\text{Fe}_{20}(2 \text{ nm})/\text{Au}(1.9 \text{ nm})/\text{Co}(\mathbf{t}_{\text{Co}})/\text{Au}(1.9 \text{ nm})]_{10}$

# Modification of the structure of easy-plane anisotropy layer

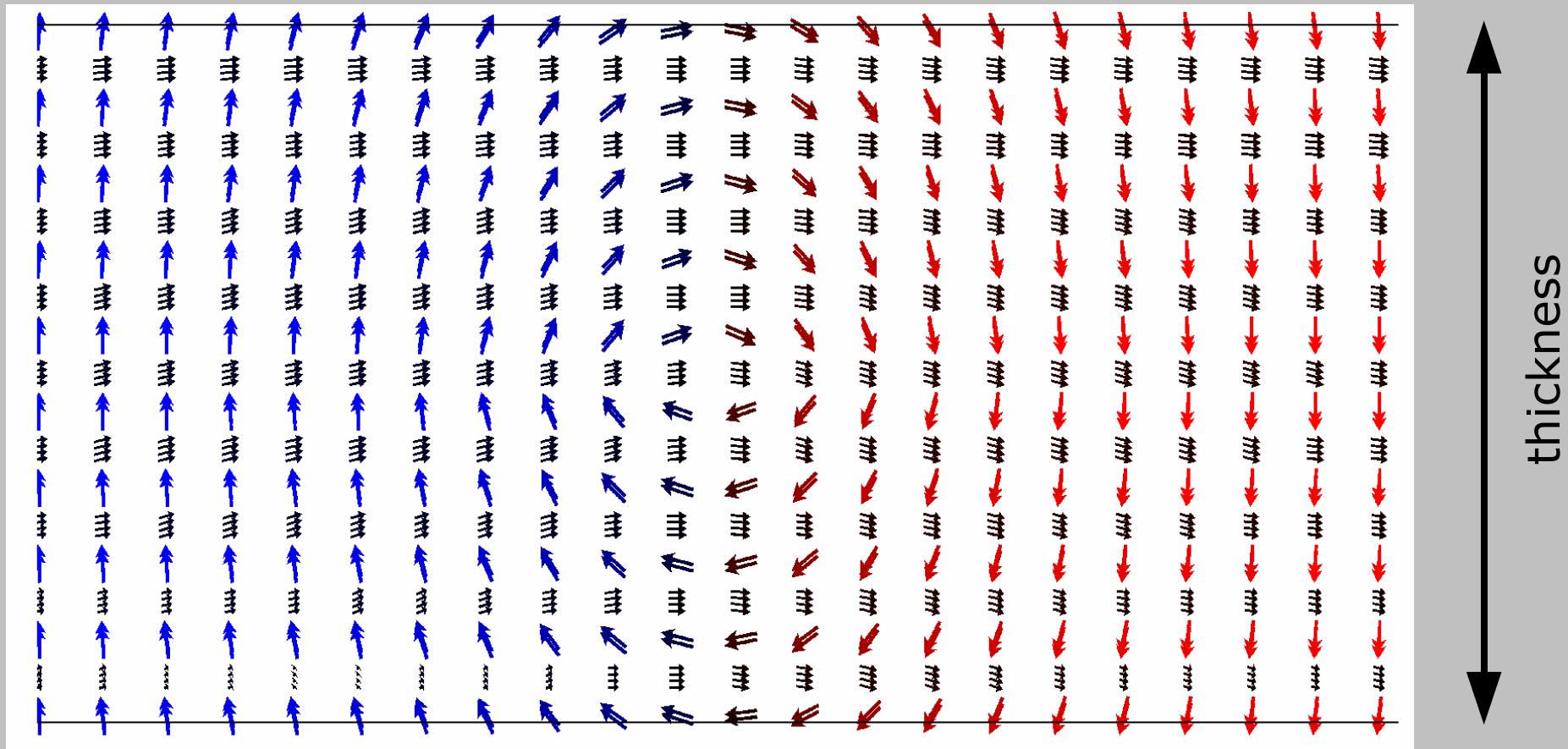
$[\text{Co}_1/\text{Ni}_{80}\text{Fe}_{20}/\text{Co}_1/\text{Au}/\text{Co}/\text{Au}]_N$



NiFe layer was replaced by a Co/NiFe/Co hybrid trilayer – this leads to the decrease of the effective easy-plane anisotropy.

The local minima of resistance are more pronounced.

# The domain structure of NiFe/Au/Co/Au - simulation\*

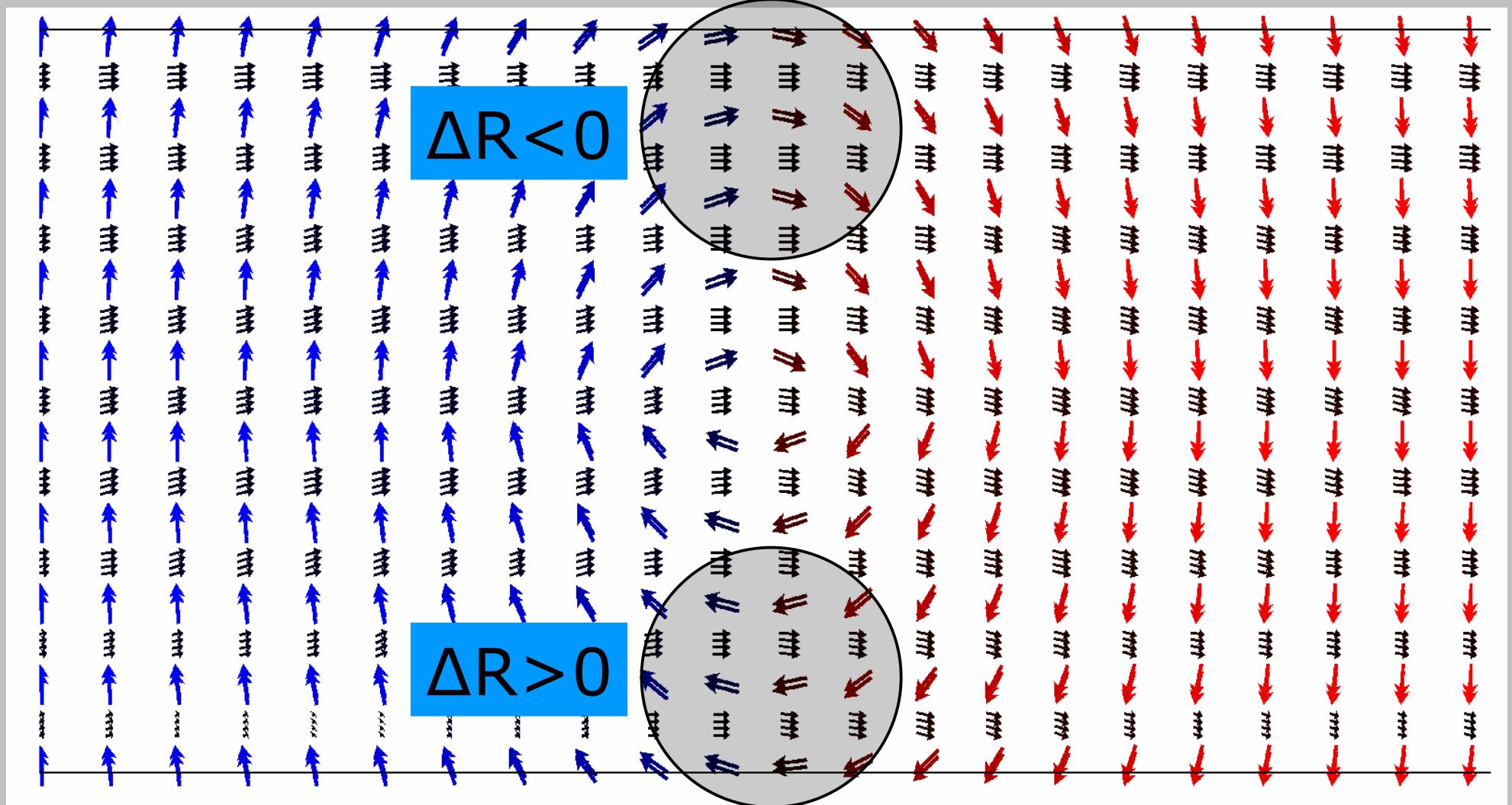


[Co(1nm)/Au(1.5nm)/Ni<sub>80</sub>Fe<sub>20</sub>(2 nm)/Au(1.5 nm)]<sub>9</sub>/Co(1nm)

$H=0$

\*Simulation with free oommf package from NIST; (1×1  $\mu\text{m}^2$ )×55nm;  
Co domains 200 nm wide;  $\alpha=0.5$ ; regular mesh with cell size of  
(5×0.5×50nm<sup>3</sup>); stiffness: Co: 30e-12 J/m, NiFe: 13e-12 J/m

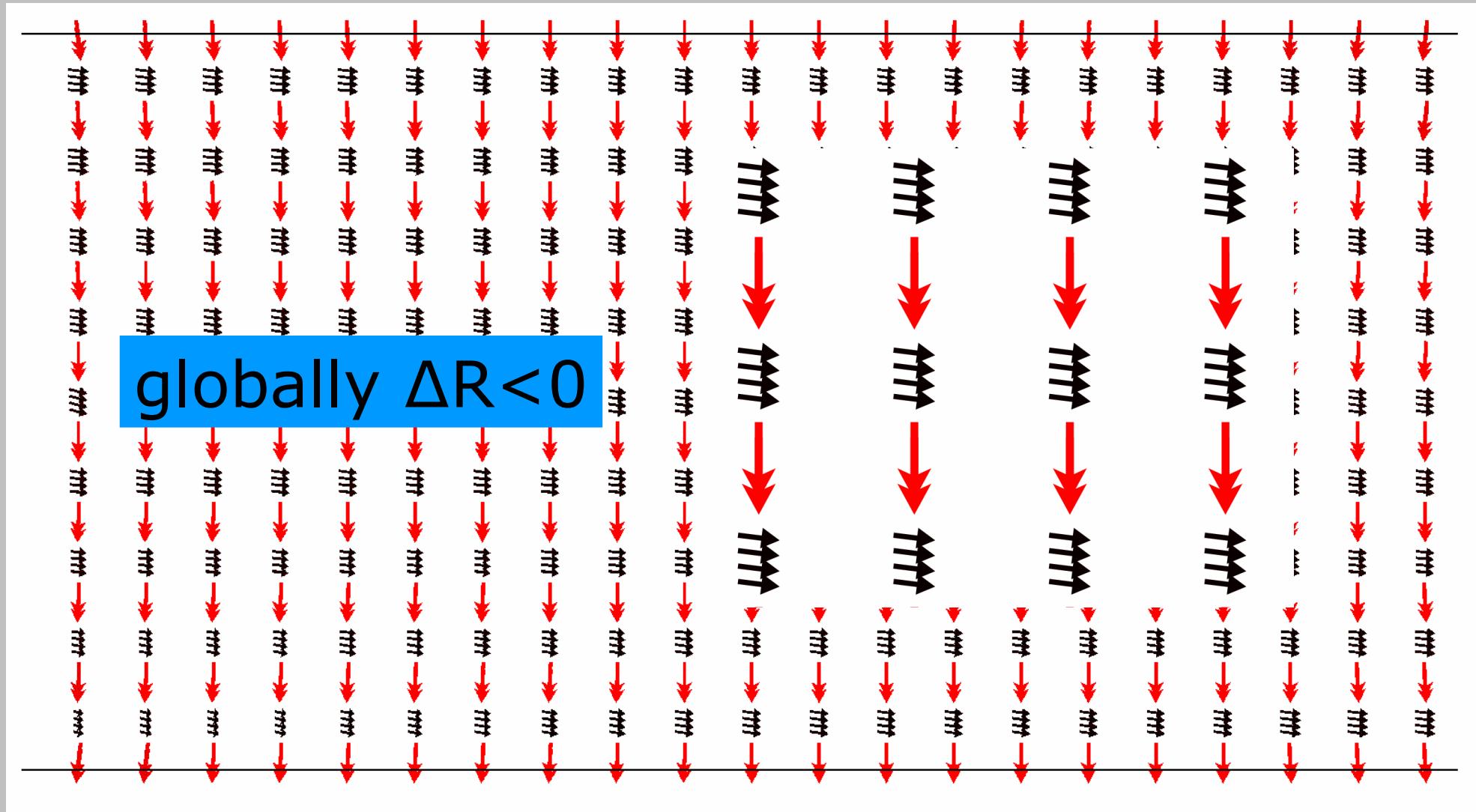
# The domain structure of NiFe/Au/Co/Au - simulation



$$\Delta R \propto \cos(\varphi)$$

$$H=0$$

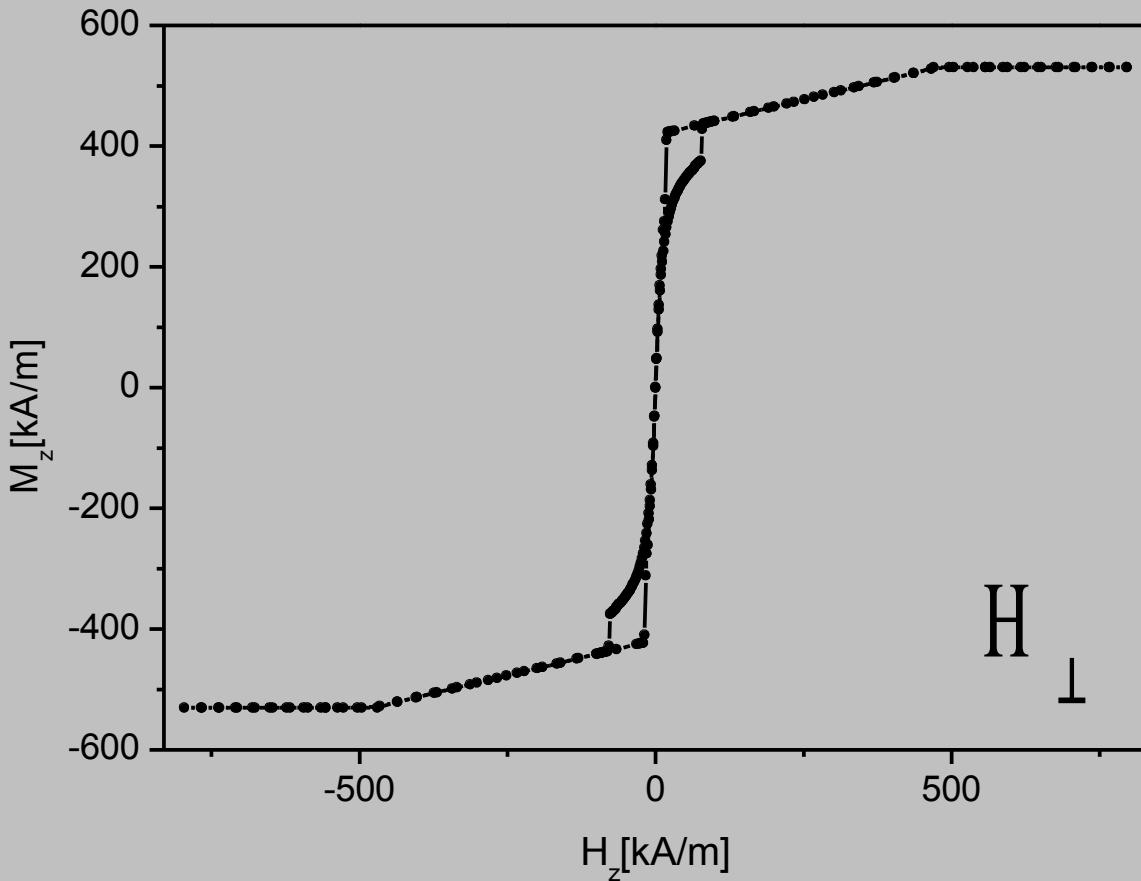
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$$\Delta R \propto \cos(\varphi)$$

$$H=0$$

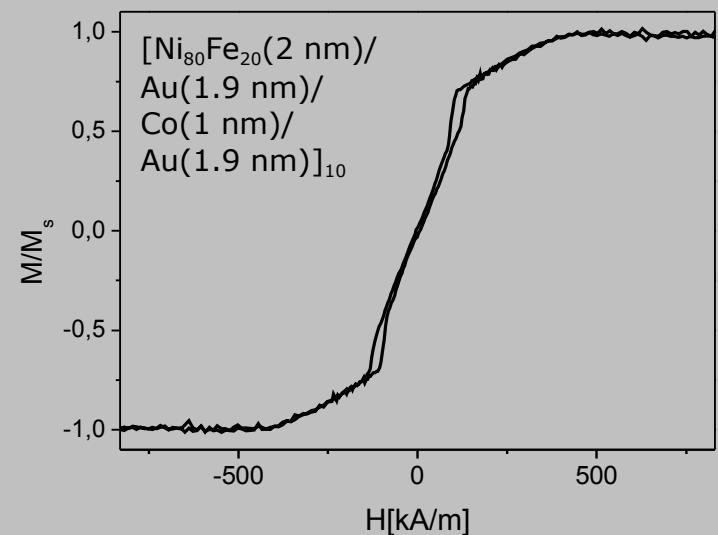
# The GMR of NiFe/Au/Co/Au - simulation\*



Co-perpendicular anisotropy

NiFe-shape anisotropy

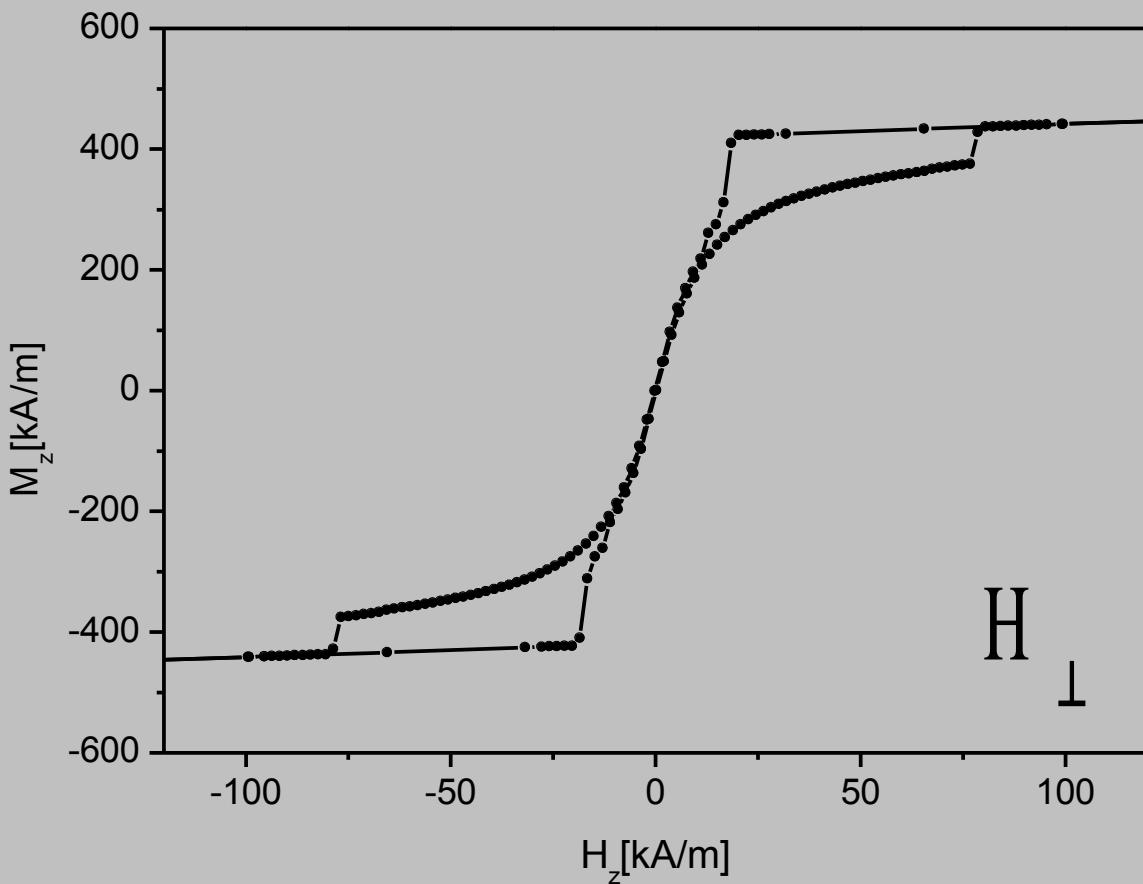
measurement:



[Co(1nm)/spacer(1nm)/NiFe(1nm)/spacer(1nm)]<sub>4</sub>/Co(1nm)

\*Simulation with free oommf package from NIST (M.J. Donahue and D.G. Porter);  
 $\alpha=0.5$ ; regular mesh with cell size of (5×20000×1nm<sup>3</sup>);  
stiffness: Co: 30e-12 J/m, NiFe: 13e-12 J/m

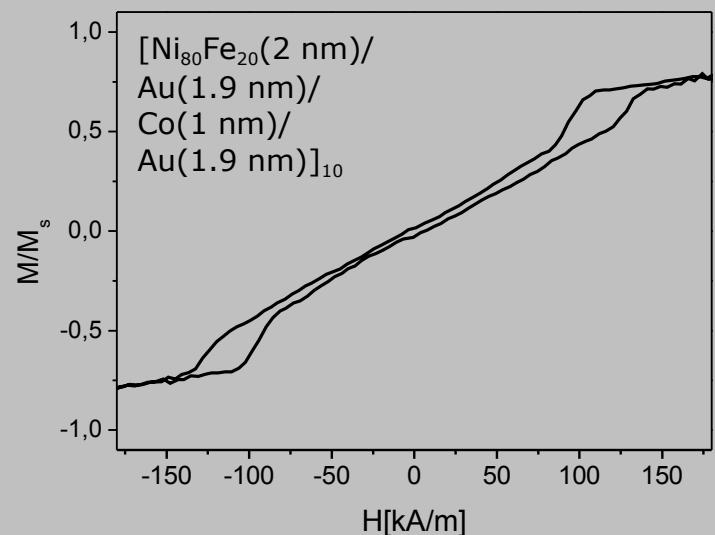
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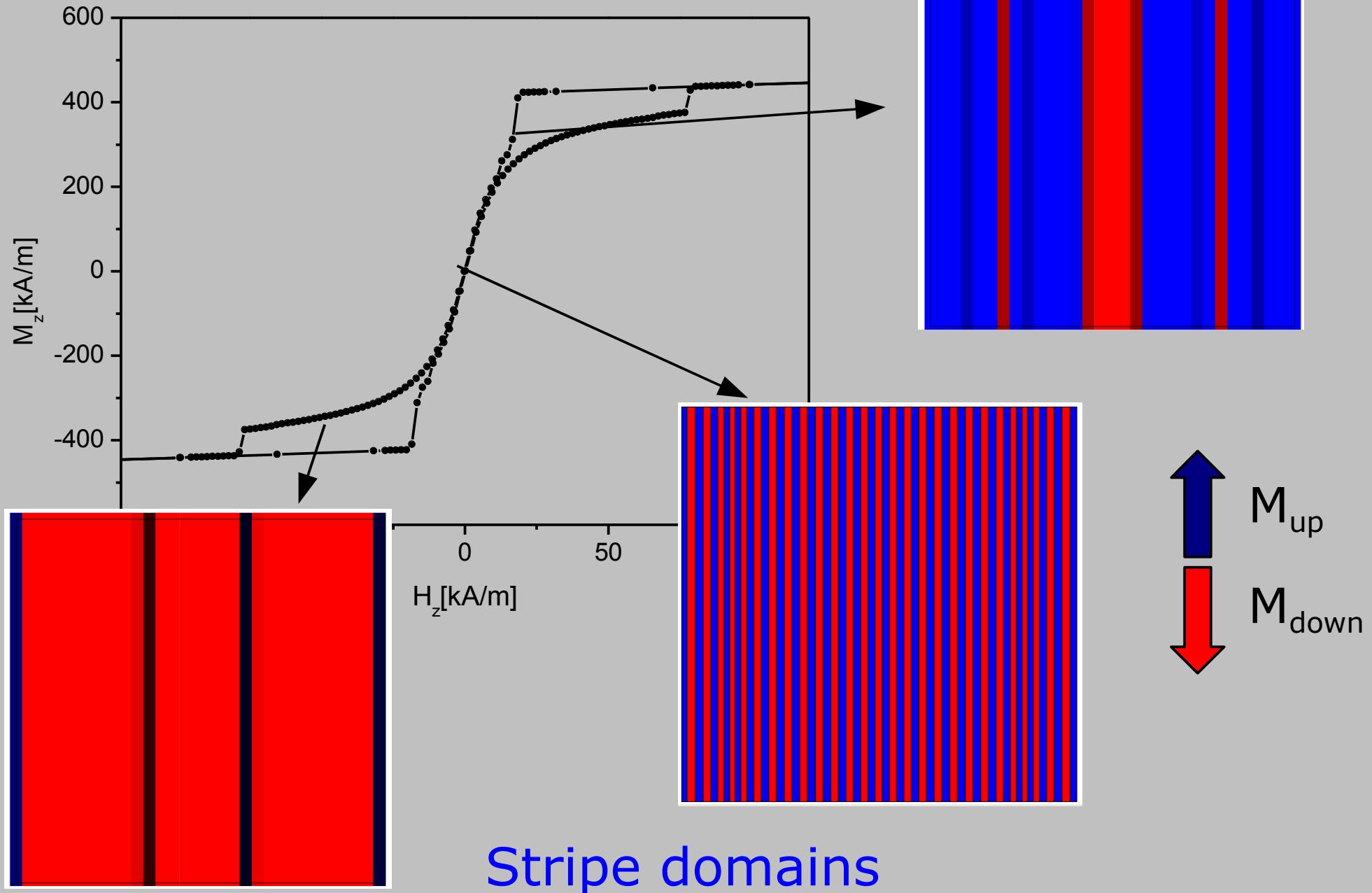
measurement:



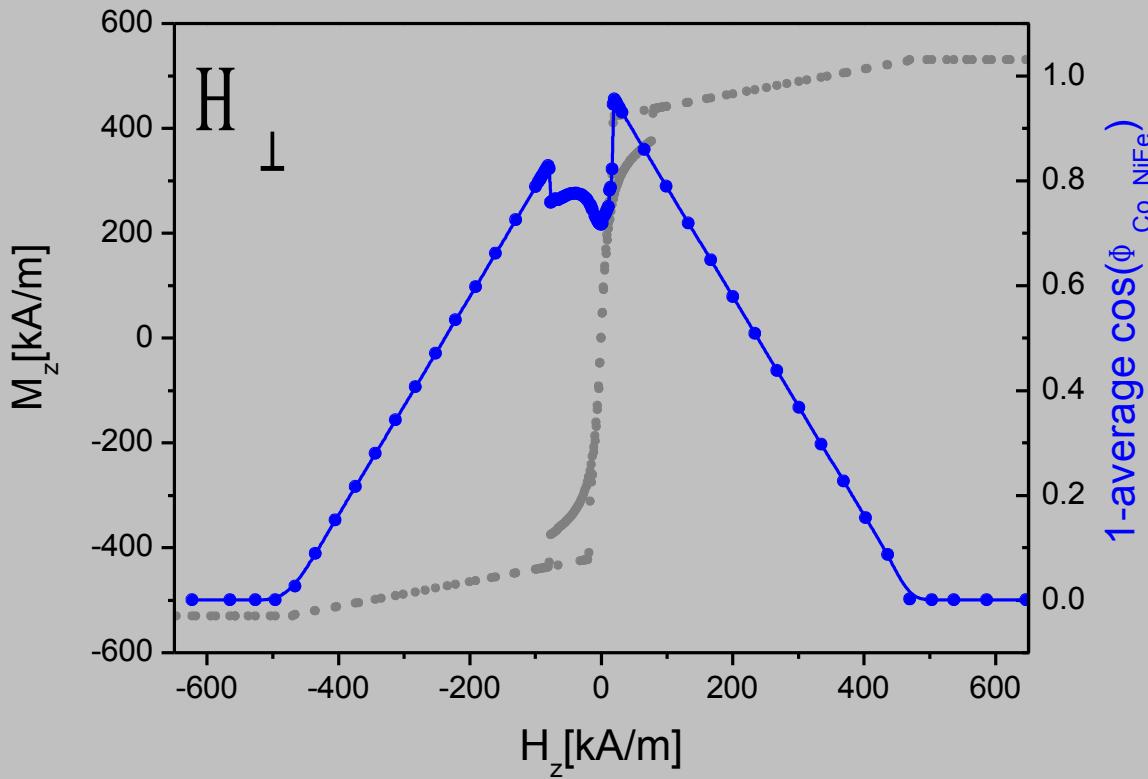
$[Co(1\text{ nm})/\text{spacer}(1\text{ nm})/\text{NiFe}(1\text{ nm})/\text{spacer}(1\text{ nm})]_4/\text{Co}(1\text{ nm})$

No attempts were made to exactly mirror the  $M(H)$  dependence ,i.e. , nucleation and annihilation fields.

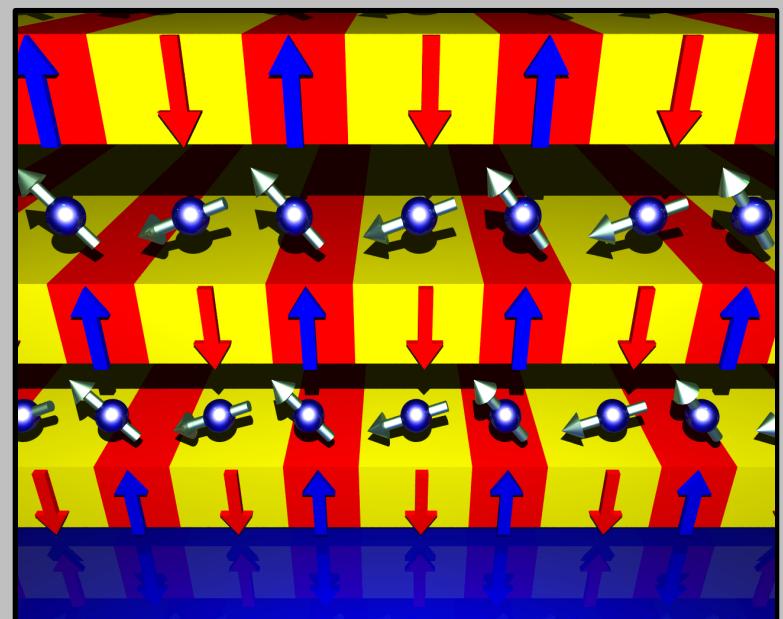
# The GMR of NiFe/Au/Co/Au - simulation



# The GMR of NiFe/Au/Co/Au - simulation

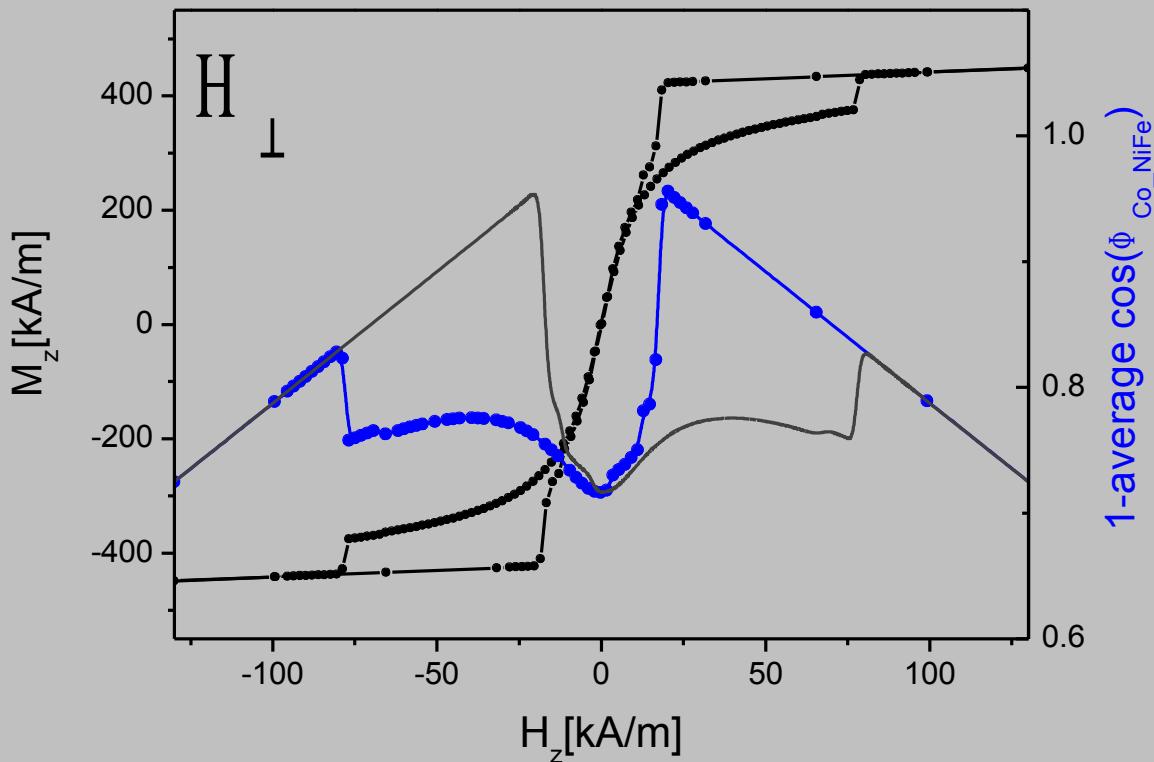


The GMR( $H$ ) dependence was calculated as proportional to **an average cosine** of the angle between magnetic moment direction of juxtaposed NiFe and Co cells.

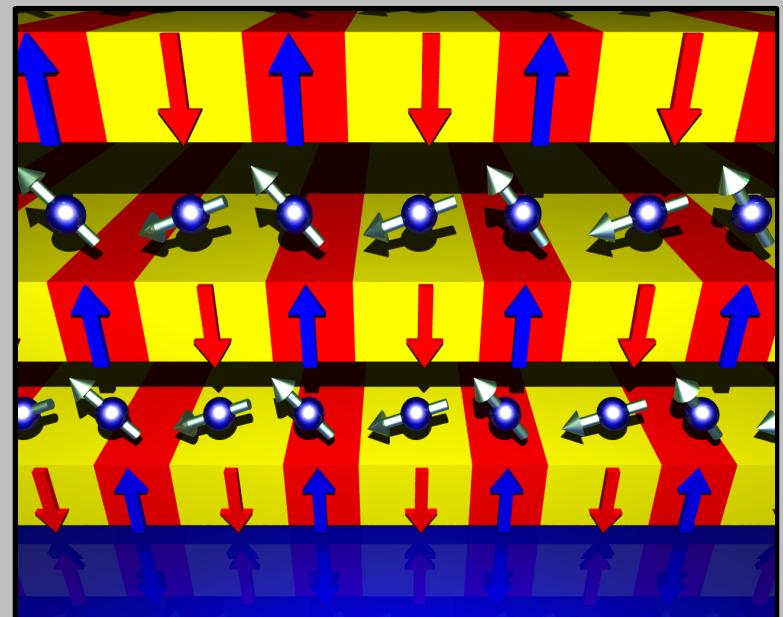


Giant magnetoresistance dependencies of NiFe/Au/Co/Au multilayers can be approximated from micromagnetic simulations.

# The GMR of NiFe/Au/Co/Au - simulation

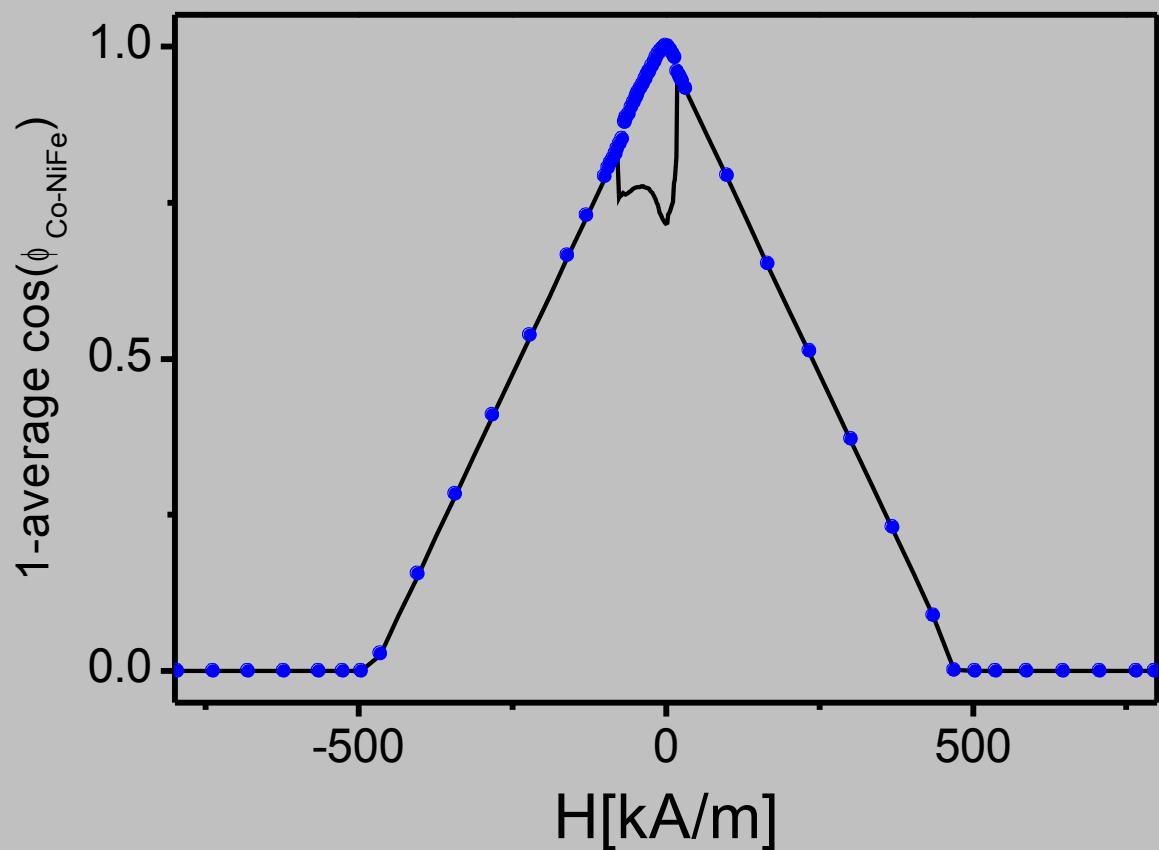


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Giant magnetoresistance dependencies of NiFe/Au/Co/Au multilayers can be approximated from micromagnetic simulations.

# The GMR of NiFe/Au/Co/Au - simulation

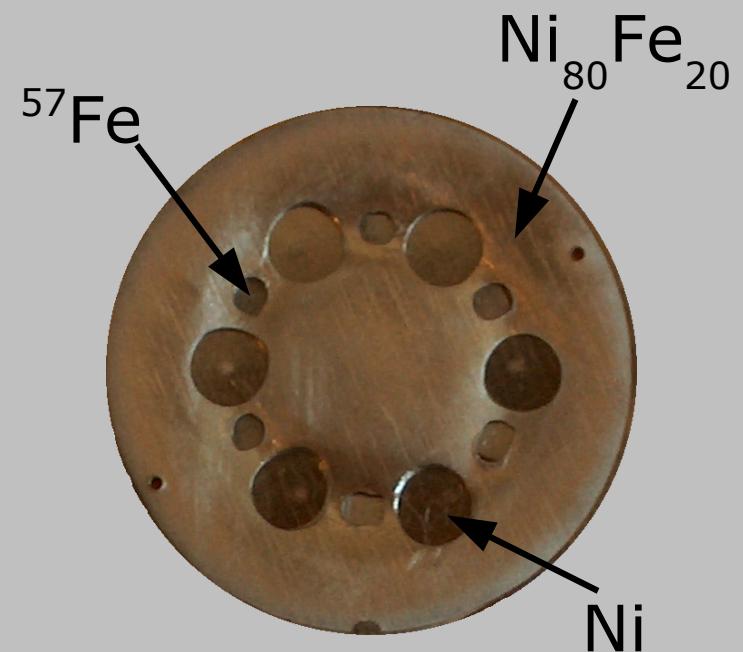
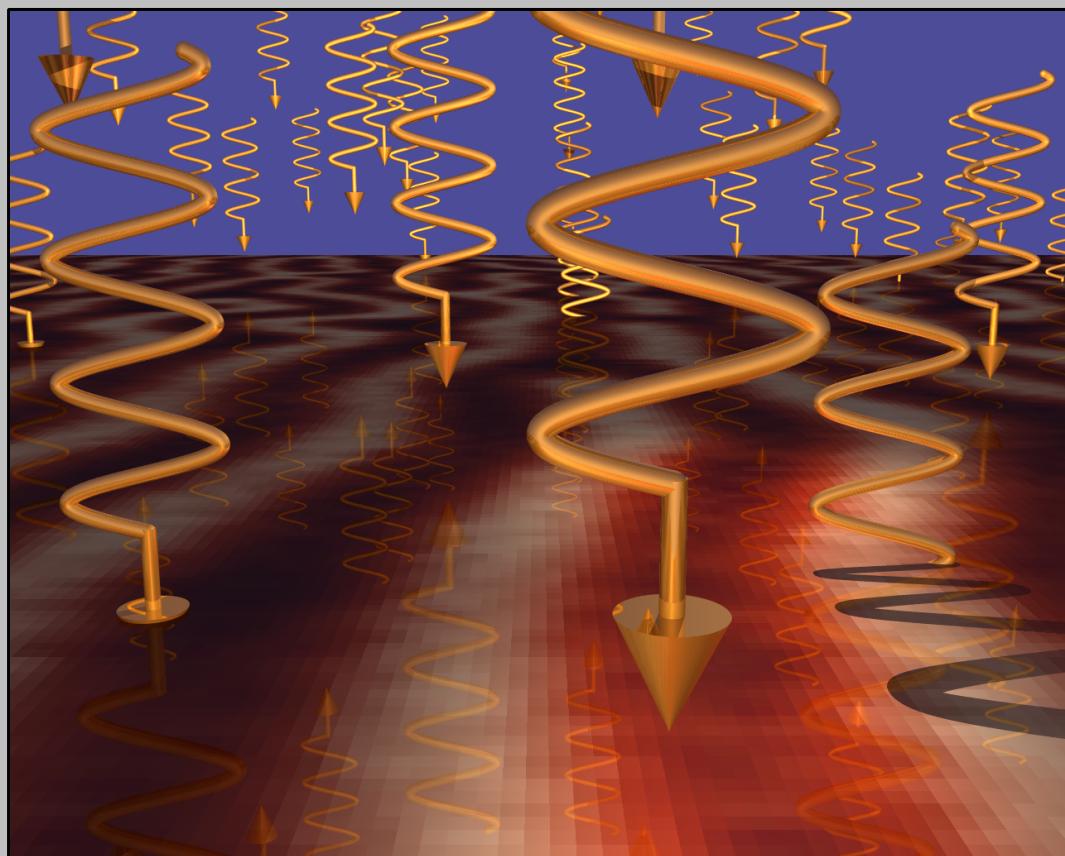


The GMR( $H$ ) dependence was calculated as proportional to **an average cosine** of the angle between magnetic moment direction of juxtaposed NiFe and Co cells.

••••• – no coupling

Without magnetostatic coupling between Co and NiFe layers there are no local minima of resistance.

# Mössbauer spectroscopy



Conversion electron Mössbauer spectroscopy (CEMS)

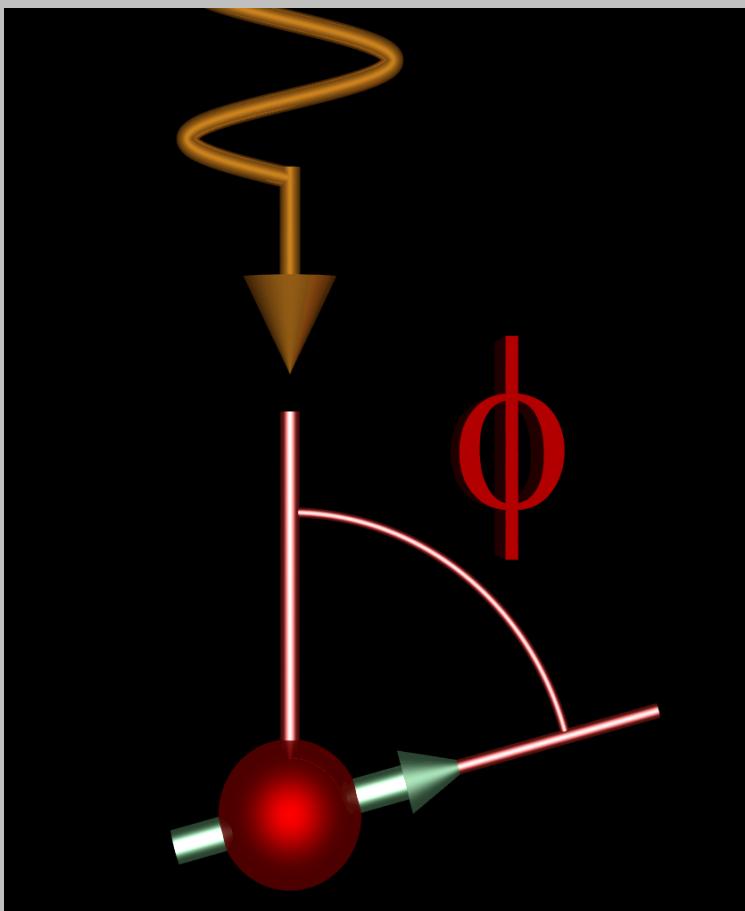
$^{57}\text{Co}$  source

$^{57}\text{Fe}$  95.3 at.%

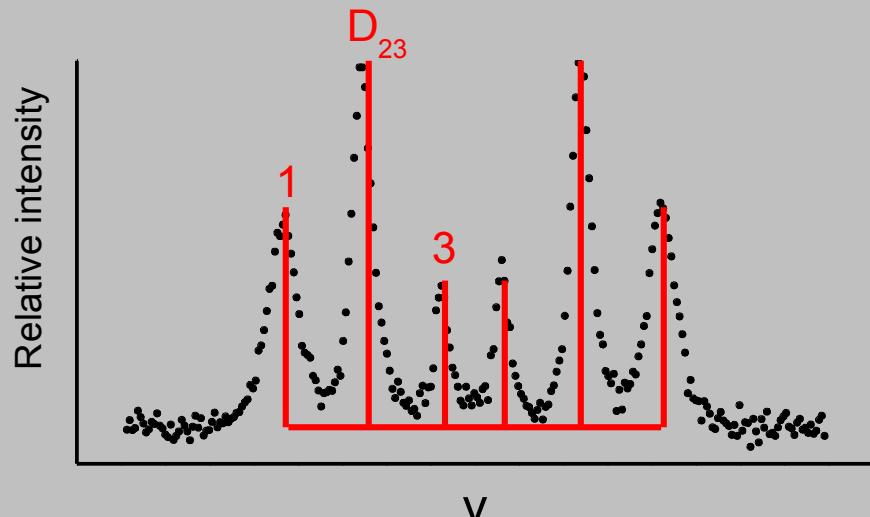


$[\text{Ni}_{80}\text{Fe}_{20}/\text{Au}/\text{Co}/\text{Au}]_{10}$

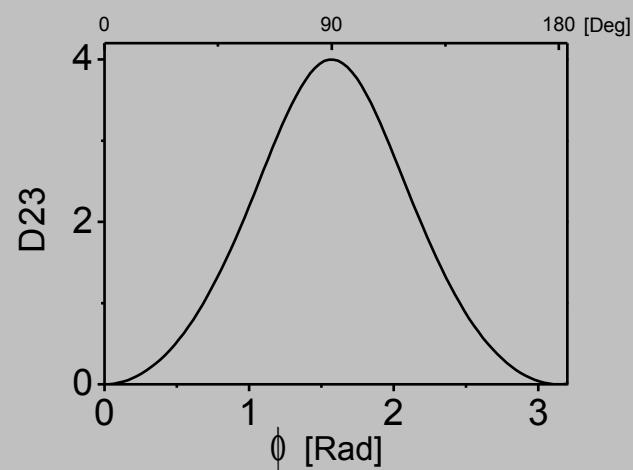
# Mössbauer spectroscopy



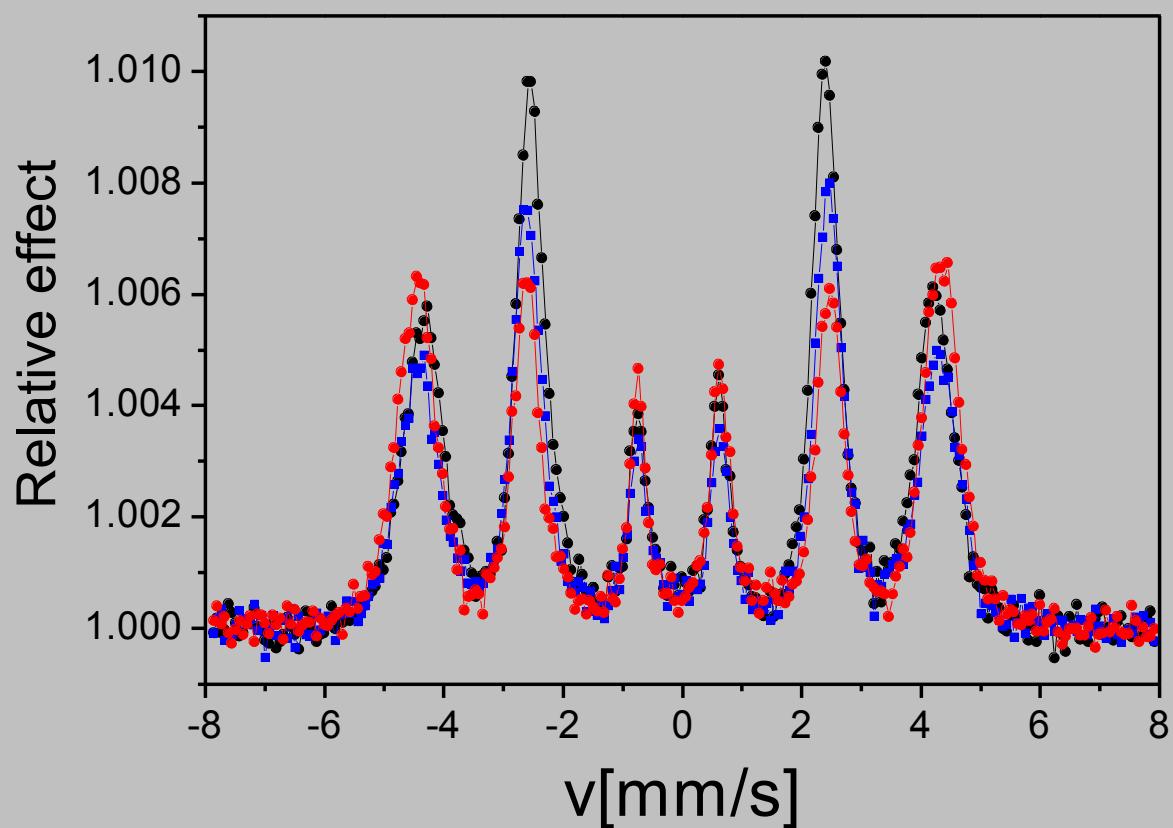
$$D_{23} = \frac{4 \sin^2(\phi)}{(1 + \cos^2(\phi))}$$



Relative intensities of the hyperfine lines vary with the angle  $\phi$  between the incident  $\gamma$ -ray and the magnetic moment.



# Mössbauer spectroscopy

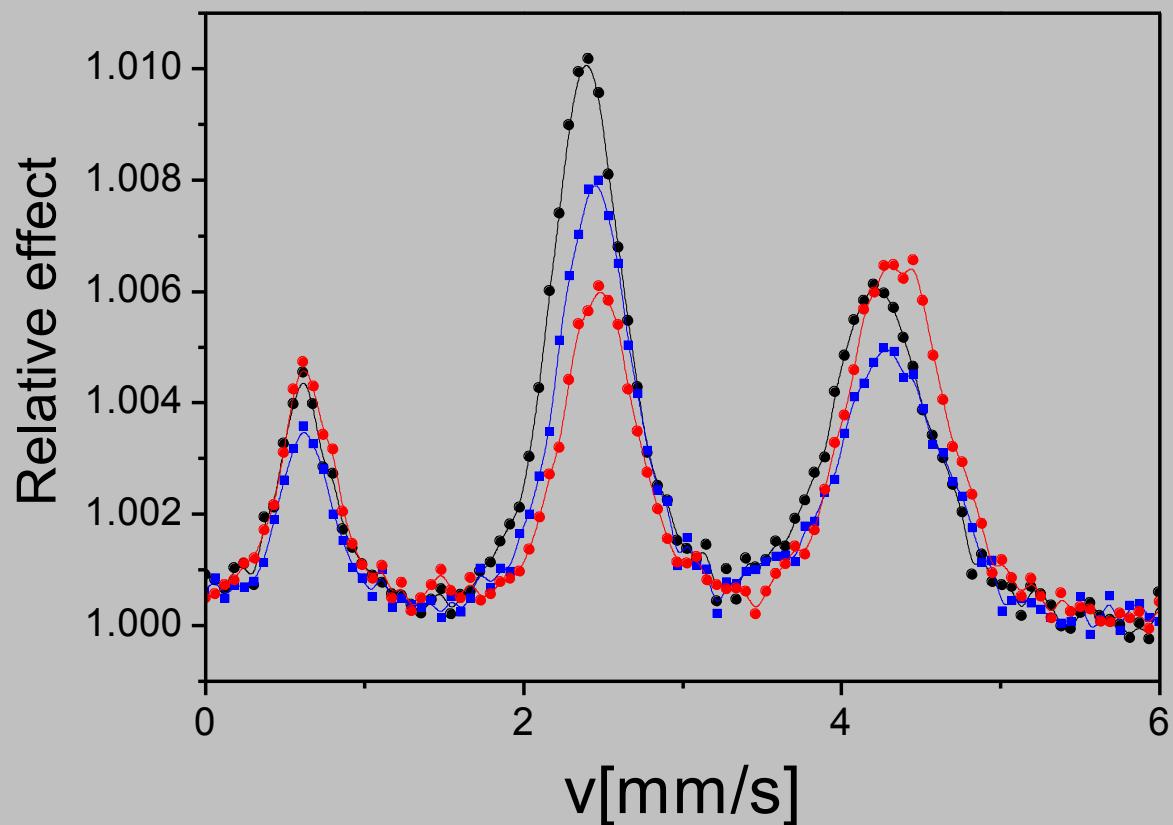


$[\text{Ni}_{80}\text{Fe}_{20}(3.2 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

$[\text{Ni}_{80}\text{Fe}_{20}(2.6 \text{ nm})/\text{Co}(0.6 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

$[\text{Co}(0.6 \text{ nm})/\text{Ni}_{80}\text{Fe}_{20}(2.6 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

# Mössbauer spectroscopy

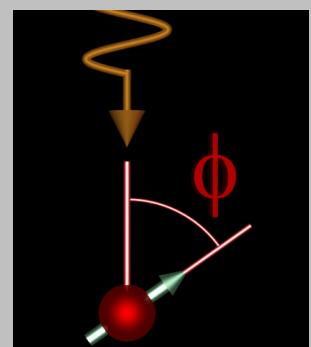
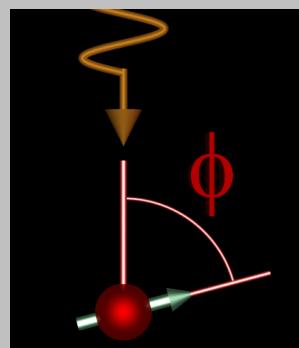
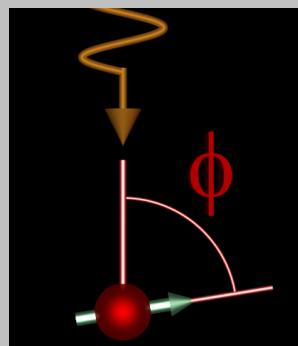
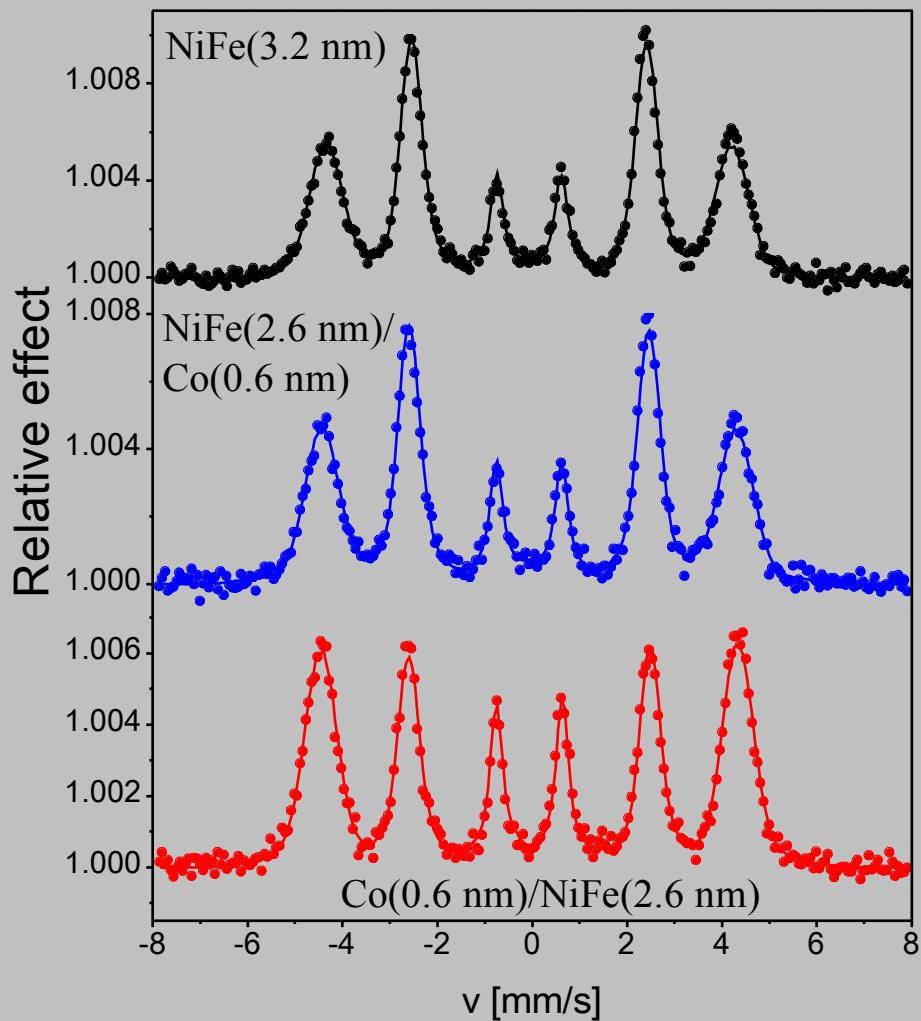


$[\text{Ni}_{80}\text{Fe}_{20}(3.2 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

$[\text{Ni}_{80}\text{Fe}_{20}(2.6 \text{ nm})/\text{Co}(0.6 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

$[\text{Co}(0.6 \text{ nm})/\text{Ni}_{80}\text{Fe}_{20}(2.6 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

# Mössbauer spectroscopy

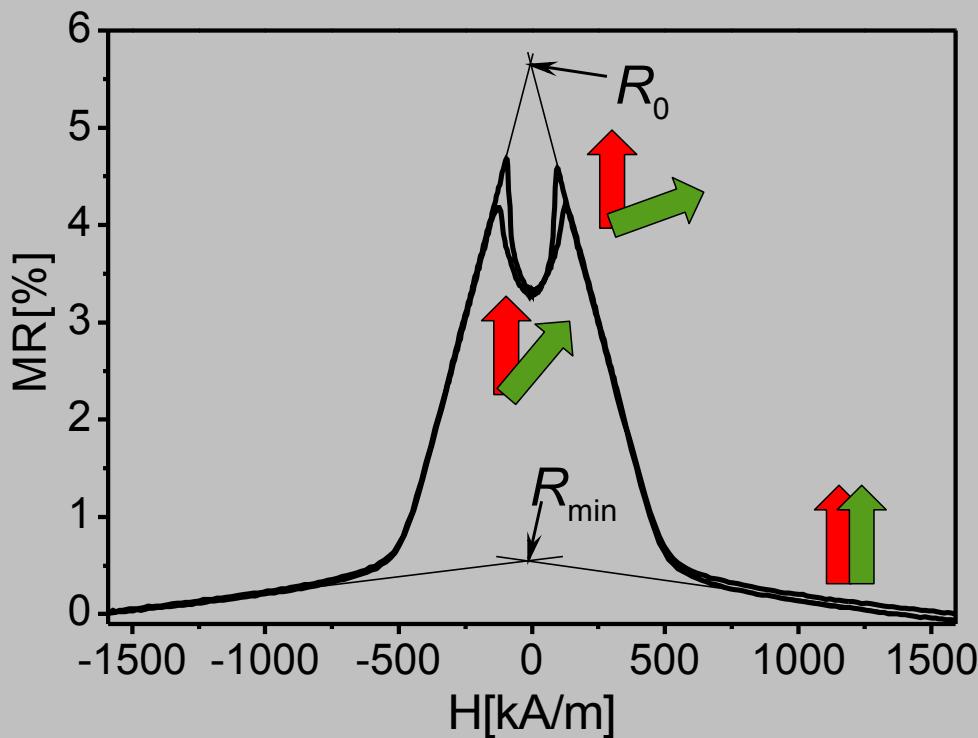


$[\text{Ni}_{80}\text{Fe}_{20}(3.2 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

$[\text{Ni}_{80}\text{Fe}_{20}(2.6 \text{ nm})/\text{Co}(0.6 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

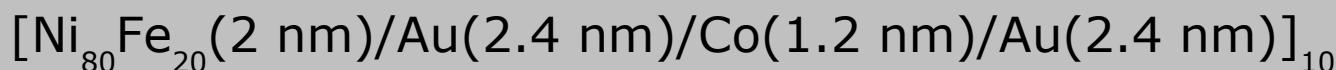
$[\text{Co}(0.6 \text{ nm})/\text{Ni}_{80}\text{Fe}_{20}(2.6 \text{ nm})/\text{Au}(2.4 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2.4 \text{ nm})]_{10}$

# Mössbauer spectroscopy

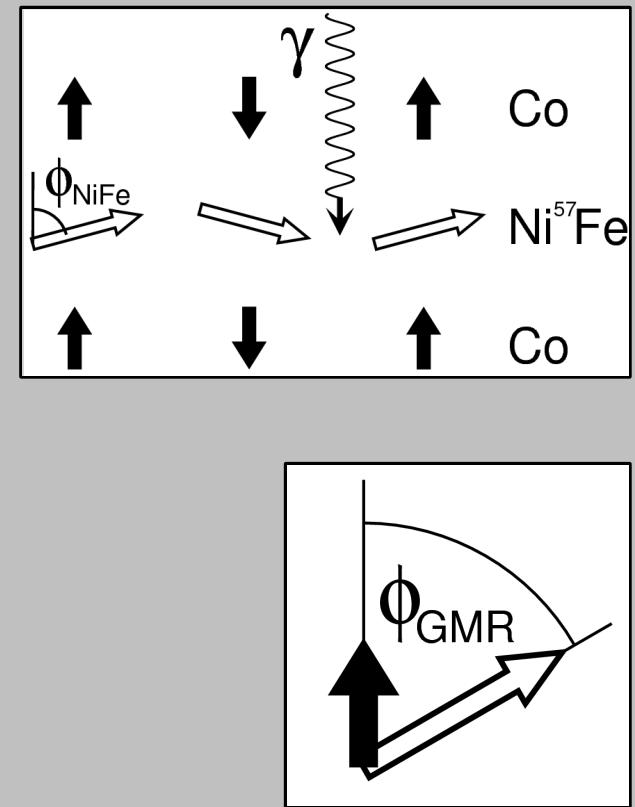
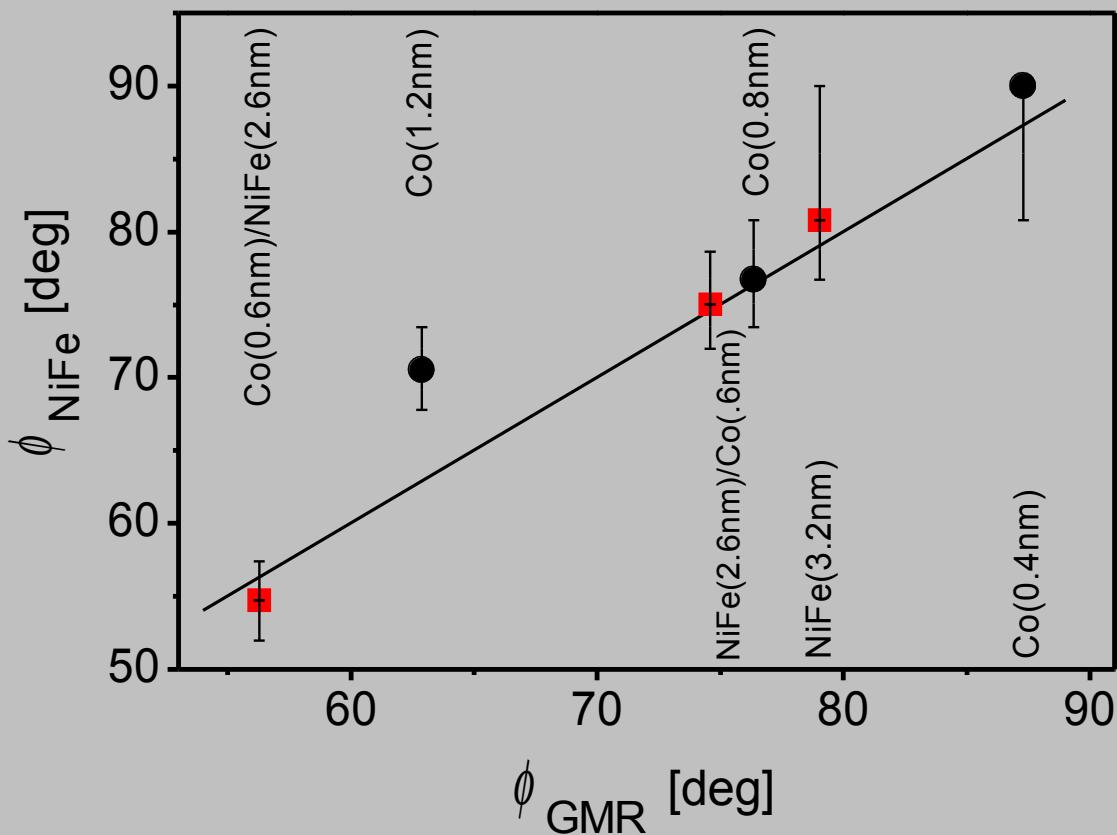


$$R = R_0 - (R_0 - R_{min}) \cos(\varphi_{Co-NiFe})$$

Resistance measurements allow the determination of the average cosine of the angle between magnetic moments of Co and NiFe layers.



# Mössbauer spectroscopy versus GMR

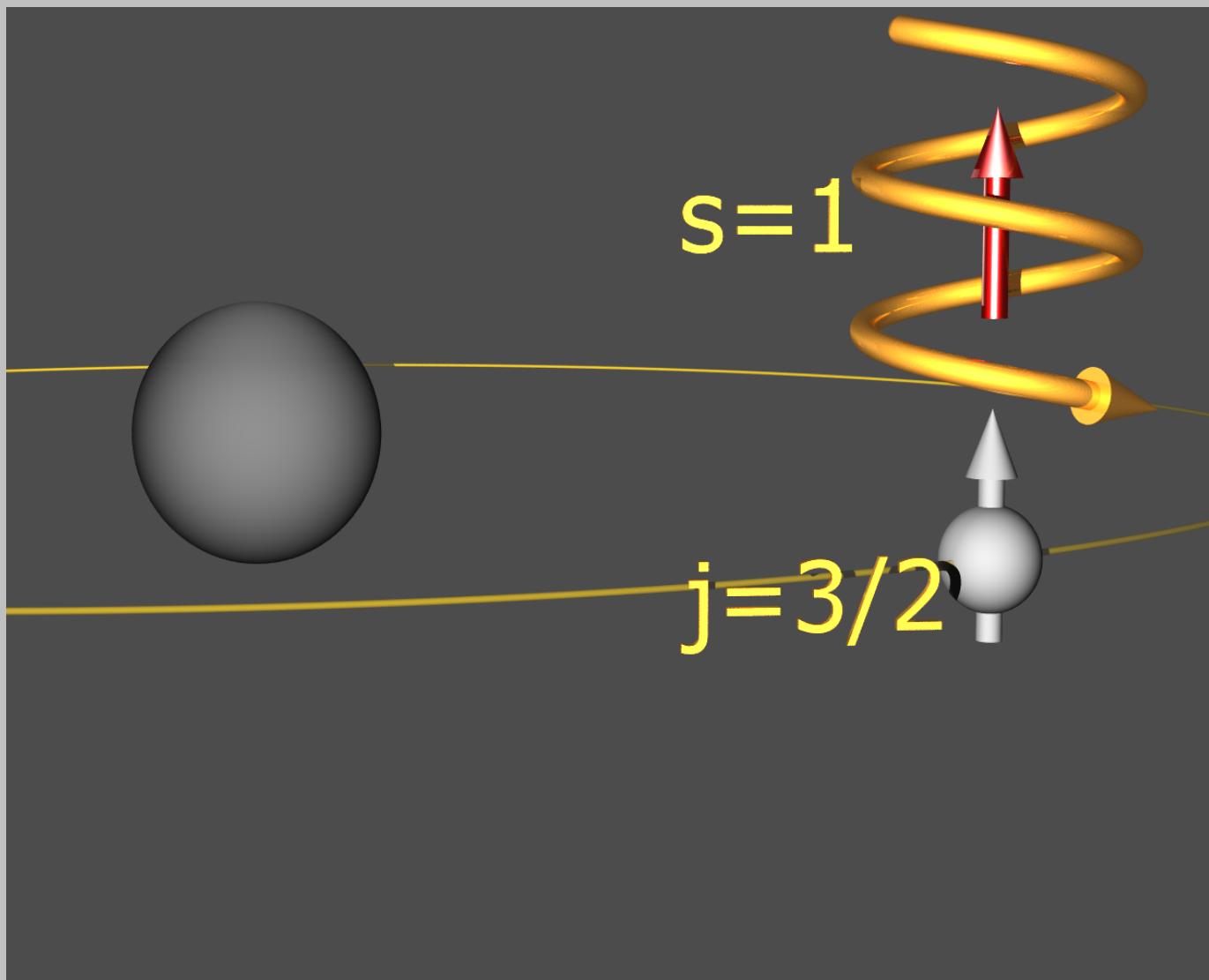


The magnetostatic fields of the Co domains cause the deflection of the magnetic moments of the NiFe layers. The deflection is stronger if the effective easy-plane anisotropy of NiFe layers is weaker.

[**X**/Au(2.4 nm)/Co(0.8 nm)/Au(2.4 nm)]<sub>10</sub>

[Ni<sub>80</sub>Fe<sub>20</sub>(2 nm)/Au(2.4 nm)/**Co**/Au(2.4 nm)]<sub>10</sub>

# Soft x-ray resonant magnetic scattering (SXRMS)



Circularly  
polarized light

$\lambda \approx 1.4 \text{ nm}$

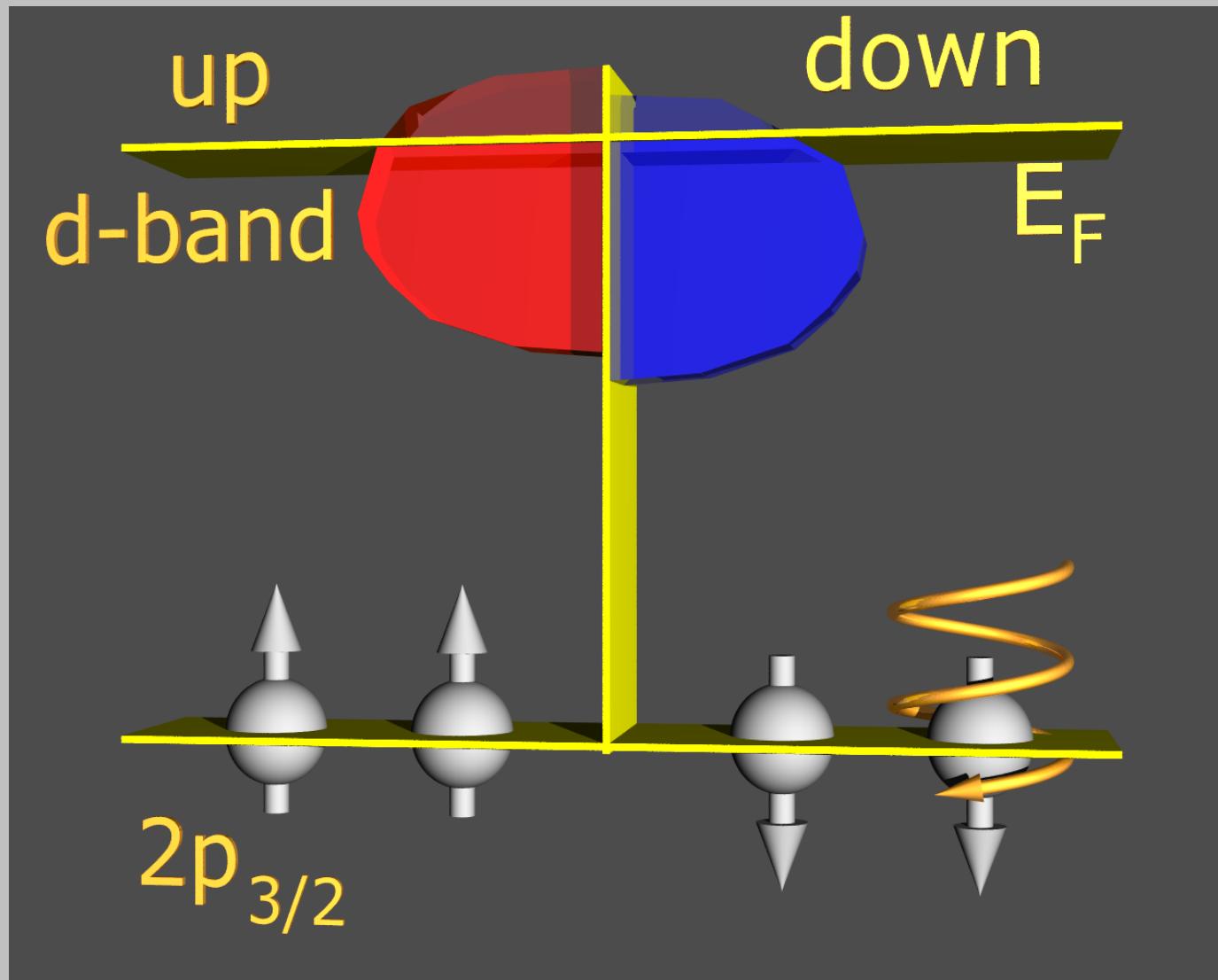
interaction with  
core electrons

photon energy  
tuned to  
absorption edge  
⇒

**elemental  
selectivity**

ALICE diffractometer at the undulator beamline UE56/2-PGM2  
at BESSY II (Berlin)

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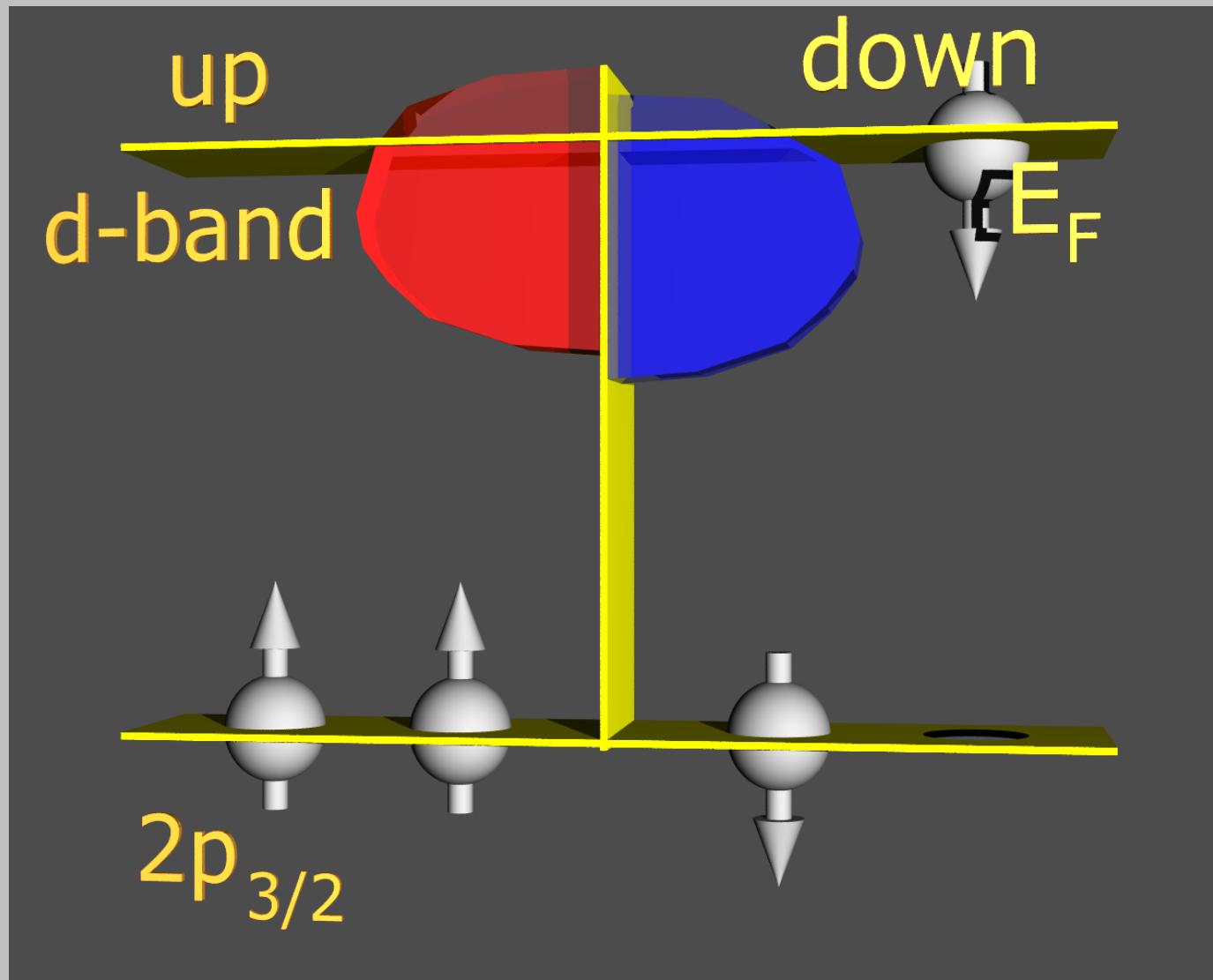
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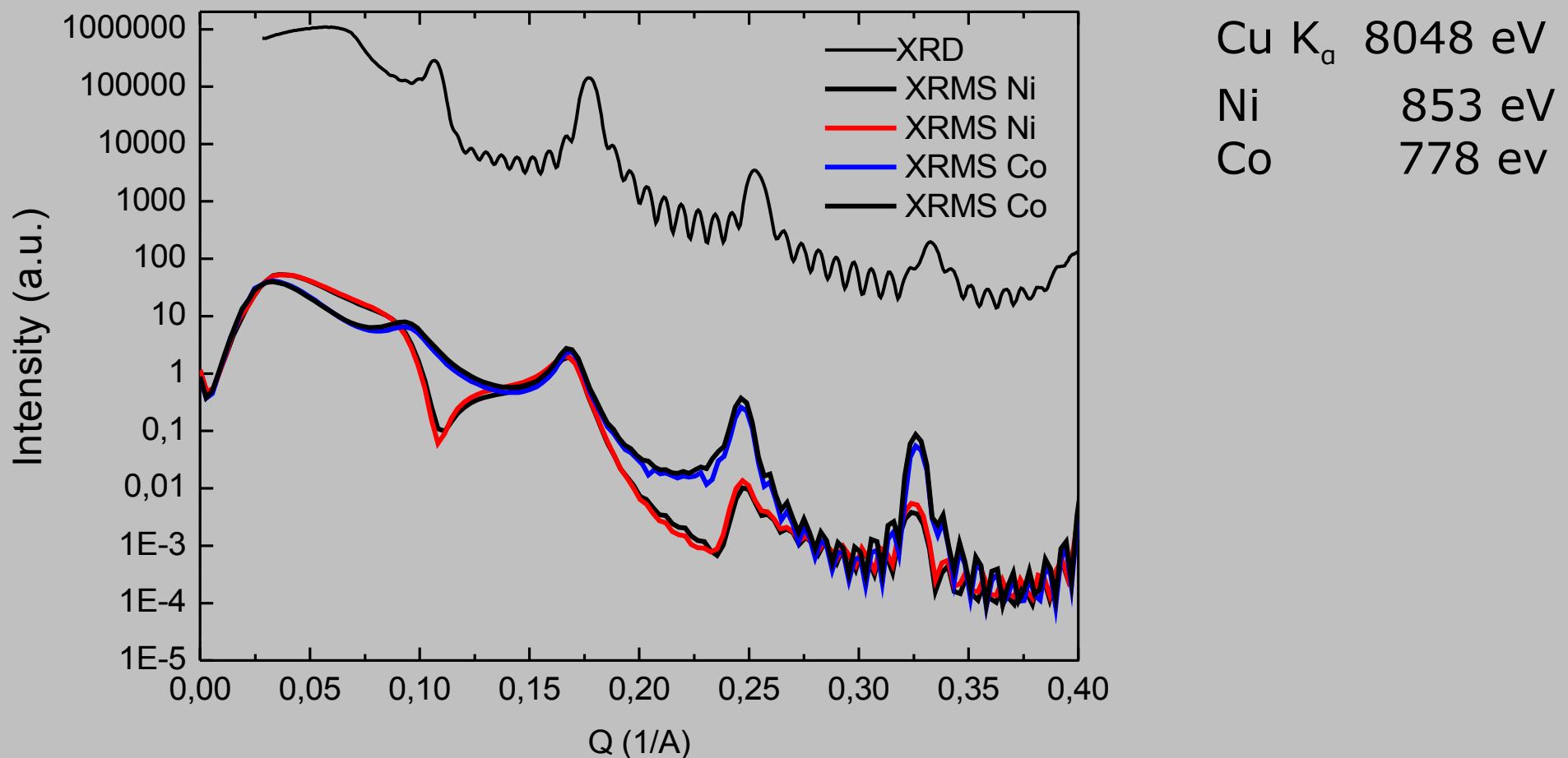
interaction with  
core electrons

photon energy  
tuned to  
absorption edge

;elemental  
selectivity

In ferromagnetic metals an imbalance in empty spin-up and spin-down states exists and valence shell can act as a "**spin detector**".

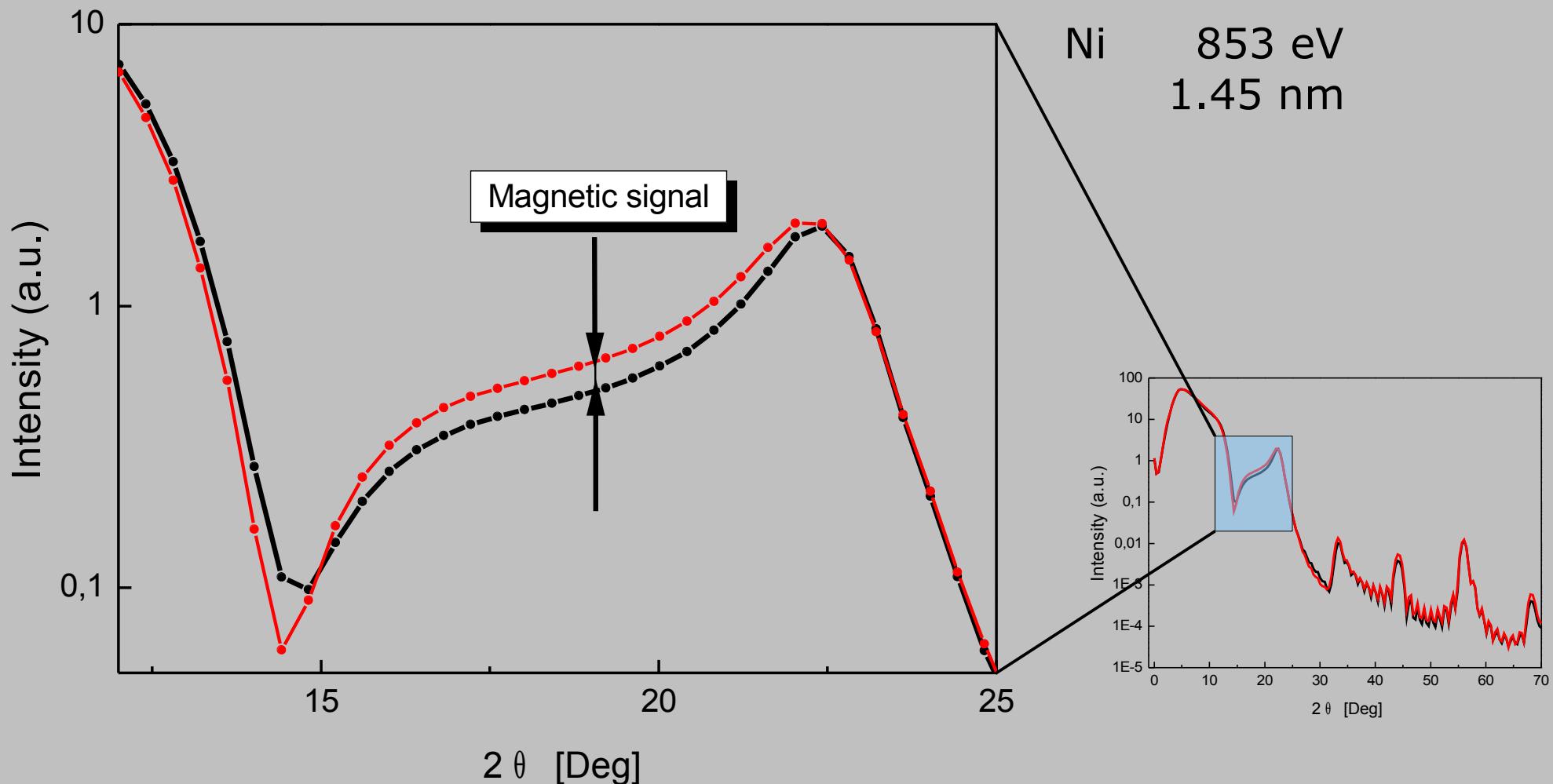
# Soft x-ray resonant magnetic scattering (SXRMS)



$[\text{Ni}_{80}\text{Fe}_{20}(2 \text{ nm})/\text{Au}(2 \text{ nm})/\text{Co}(0.8 \text{ nm})/\text{Au}(2 \text{ nm})]_{10}$

ALICE diffractometer at the undulator beamline UE56/2-PGM2  
at BESSY II (Berlin)

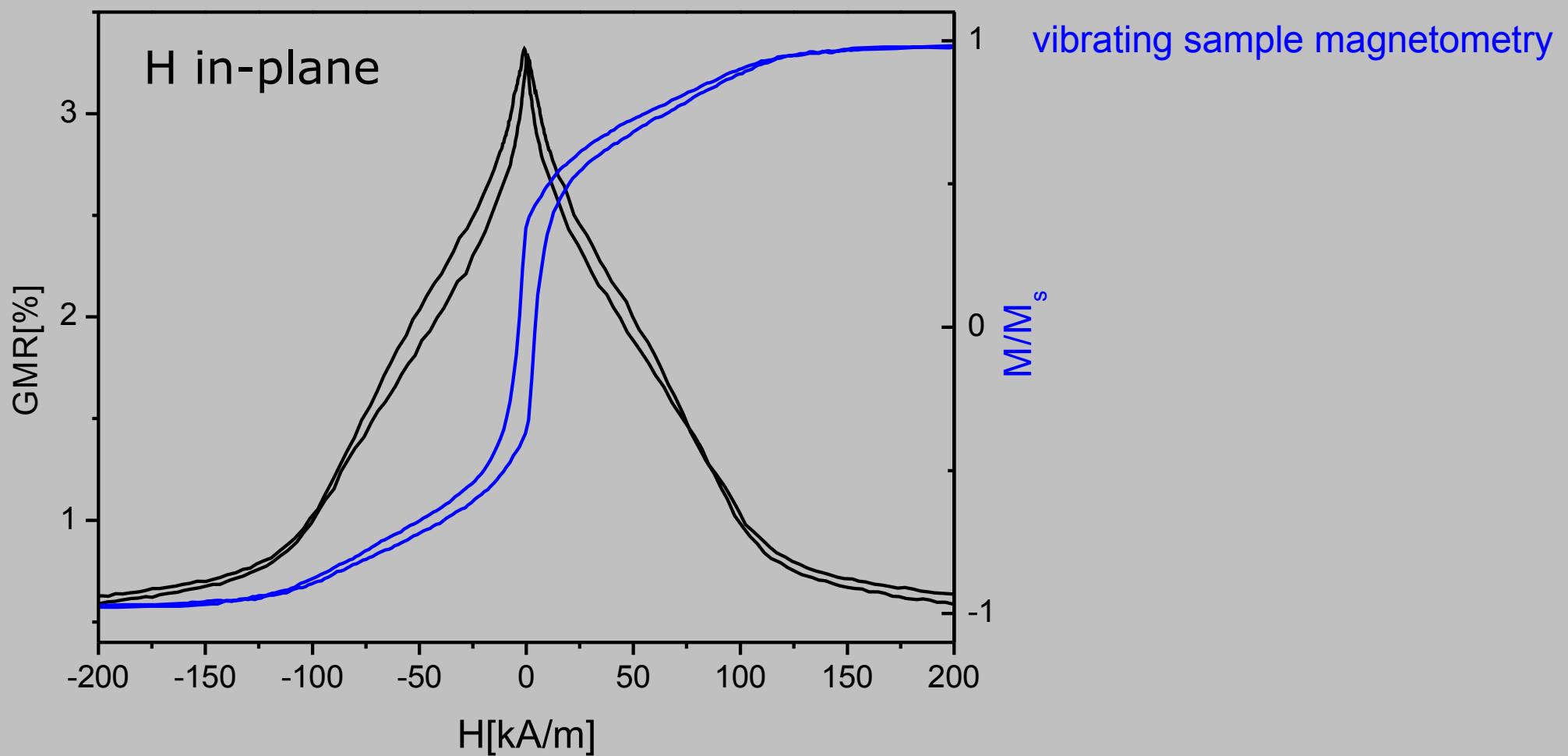
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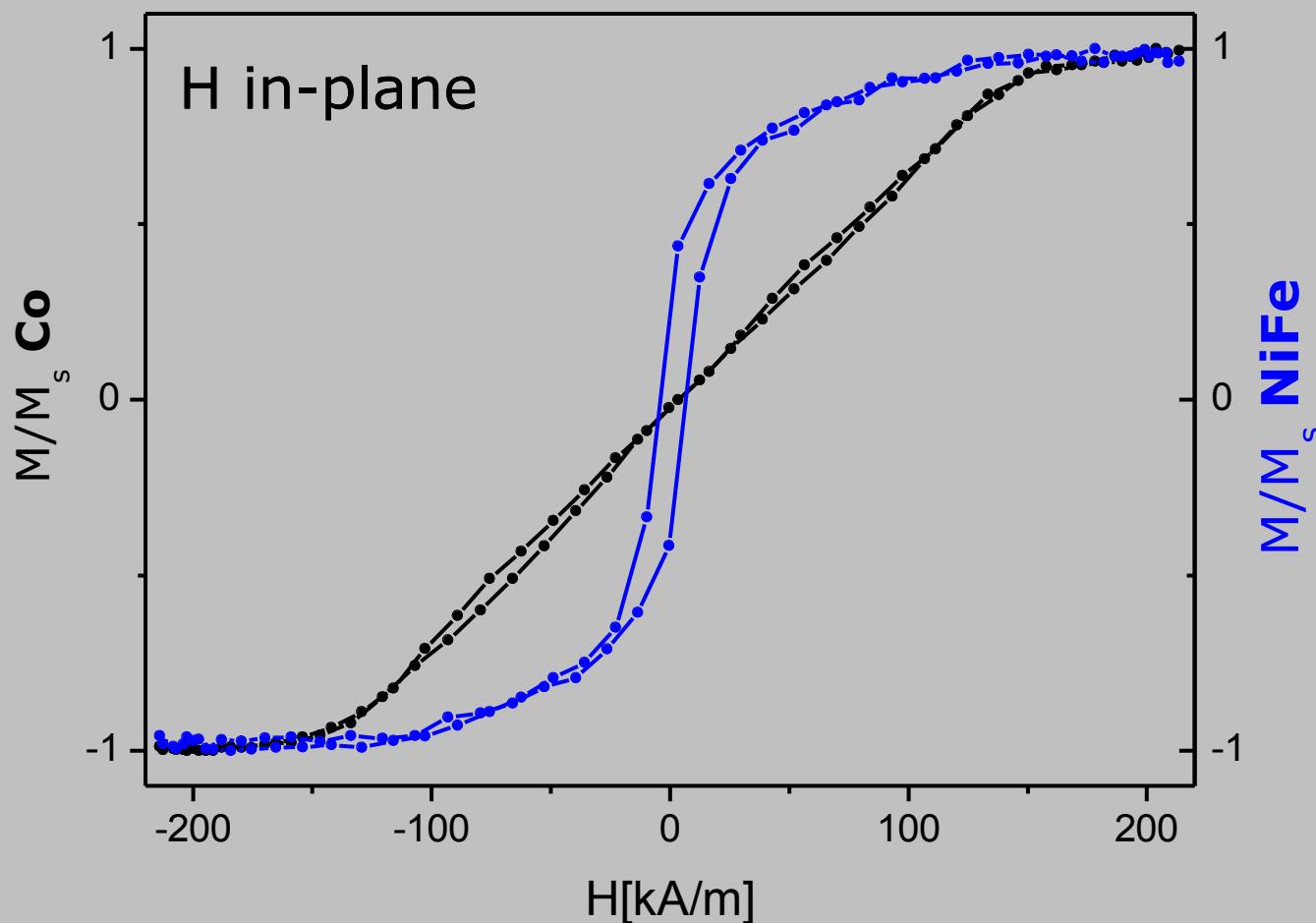
# Soft x-ray resonant magnetic scattering (SXRMS)



$$R(H) \leftrightarrow M(H)$$

$[\text{Ni}_{80}\text{Fe}_{20}(2\text{ nm})/\text{Au}(2\text{ nm})/\text{Co}(1.1\text{ nm})/\text{Au}(2\text{ nm})]_{10}$

# Soft x-ray resonant magnetic scattering (SXRMS)



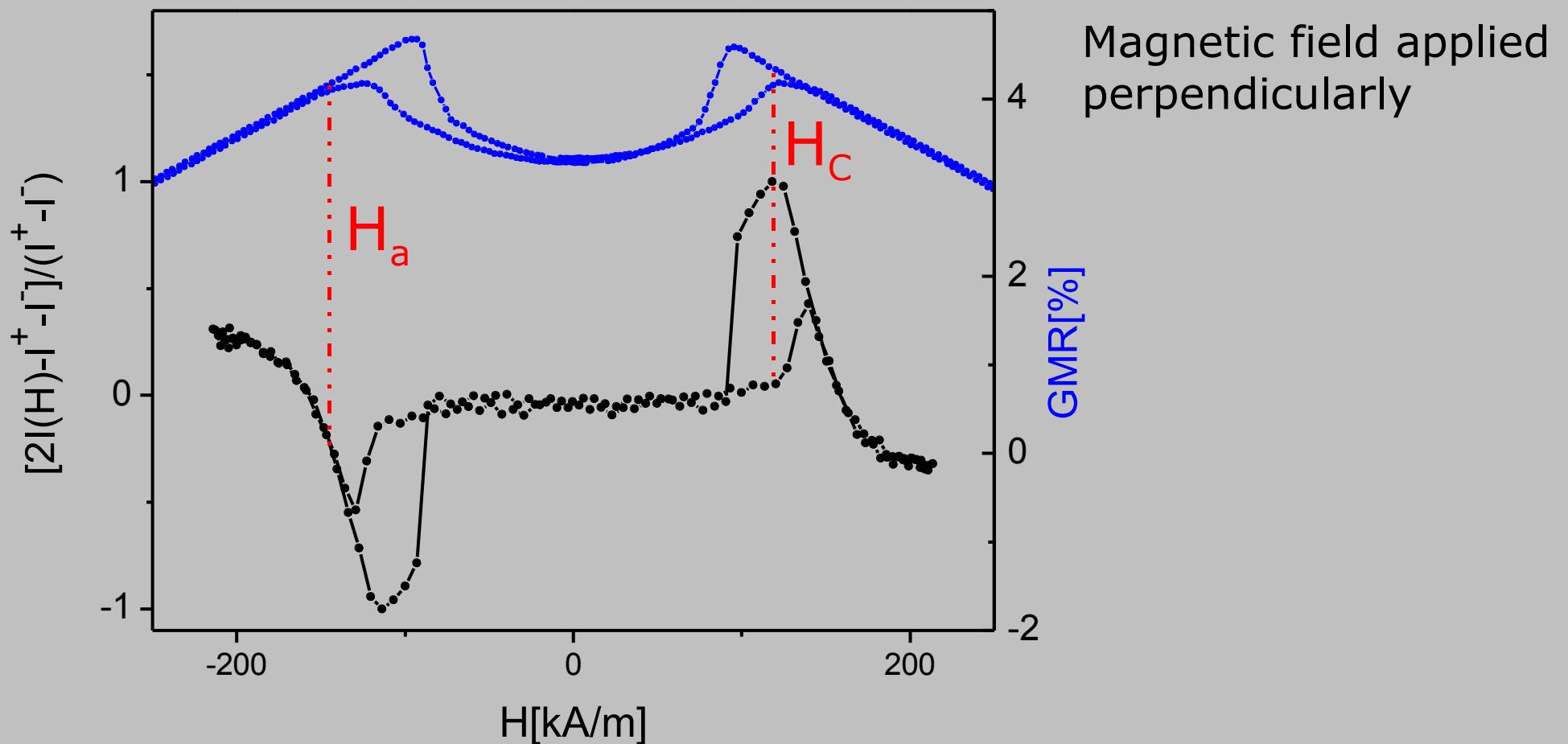
XRMS allows independent measurement of  $M(H)$  dependence of **Co** and **NiFe** layers.

$\Theta=8.5\text{Deg}$   
 $E=853\text{eV (Ni L}_3\text{)}$   
 $E=778\text{eV (Co L}_3\text{)}$

$$M/M_s \propto [2I(H) - I^+ - I^-] / (I^+ - I^-)$$

$[\mathbf{Ni}_{80}\mathbf{Fe}_{20}(2\text{ nm})/\mathbf{Au}(2\text{ nm})/\mathbf{Co}(1.1\text{ nm})/\mathbf{Au}(2\text{ nm})]_{10}$

# Soft x-ray resonant magnetic scattering (SXRMS)



SXRMS signal from **NiFe layers** shows fields characteristic for Co layers reversal:  
-creation of the stripe domain structure ( $H_c$ )  
-annihilation field of domain structure ( $H_a$ )

$[\text{Ni}_{80}\text{Fe}_{20}(2\text{ nm})/\text{Au}(2\text{ nm})/\text{Co}(1.1\text{ nm})/\text{Au}(2\text{ nm})]_{10}$

# Conclusions

- Element specific measurements of magnetization in NiFe/Au/Co/Au multilayers confirm the influence of stray fields on the magnetization processes
- Micromagnetic simulations approximate main features of  $R(H)$  dependencies

It should be pointed out that  $R_{\min}$  does not correspond to the lowest resistance shown in figure 3. This is due to the fact that the superparamagnetic precipitates, at interfaces, contribute to resistance changes up to higher fields [21] determined by the Langevin function. This effect was taken into account by determining  $R_{\min}$  via linear extrapolation of high-field resistance change (3). The anisotropic magnetoresistance (AMR), which samples never exceeds 0.5% and is less than or equal to 10% of the total resistance change, is neglected in our work [6]. The hysteresis visible in figure 3, in  $\pm 150$  kA m<sup>-1</sup>, originates from the stripe domain structure of the  $M(H)$  dependence of Co layers that are magnetized in their easy axis direction [6]. It should be emphasized that the determination of average  $\phi_{\text{GMR}}$  from GMR is an approximation, since the resistance of the system is not necessarily the sum of the resistances of its components, determined by the local angle between the magnetic moments of NiFe and Co bilayers.

Figure 4 illustrates the main finding of the present work: it shows the dependence of the average angle between the magnetic moments of NiFe and Co bilayers on the sequence of the layers.

# Thank you for your attention

the local Co moment direction clearly in figure 4. As expected [6], the resistance decreases because the stray fields of the Co layers (dots in figure 3) decrease. The stray fields  $H_{\perp}$  are of the same order as the  $\phi_{\text{NiFe}}(t_{\text{Co}})$  dependence. The NiFe films are magnetized due to shape anisotropy and, consequently, i.e. the  $M(H_{\perp})$  dependence of effective easy-plane anisotropy  $\phi_{\text{NiFe}}$  decreases with increasing  $X$  bilayers, Co and NiFe bilayers, between them, behave like the effective anisotropy of a single layer. As mentioned in the discussion, the influence of the perpendicular magnetic field on the sequence of the layers is more pronounced for the NiFe/Co bilayers than for the Co/NiFe bilayers. The attempts were made to find an independent way to explain the observed effect.