

Quantum electronics

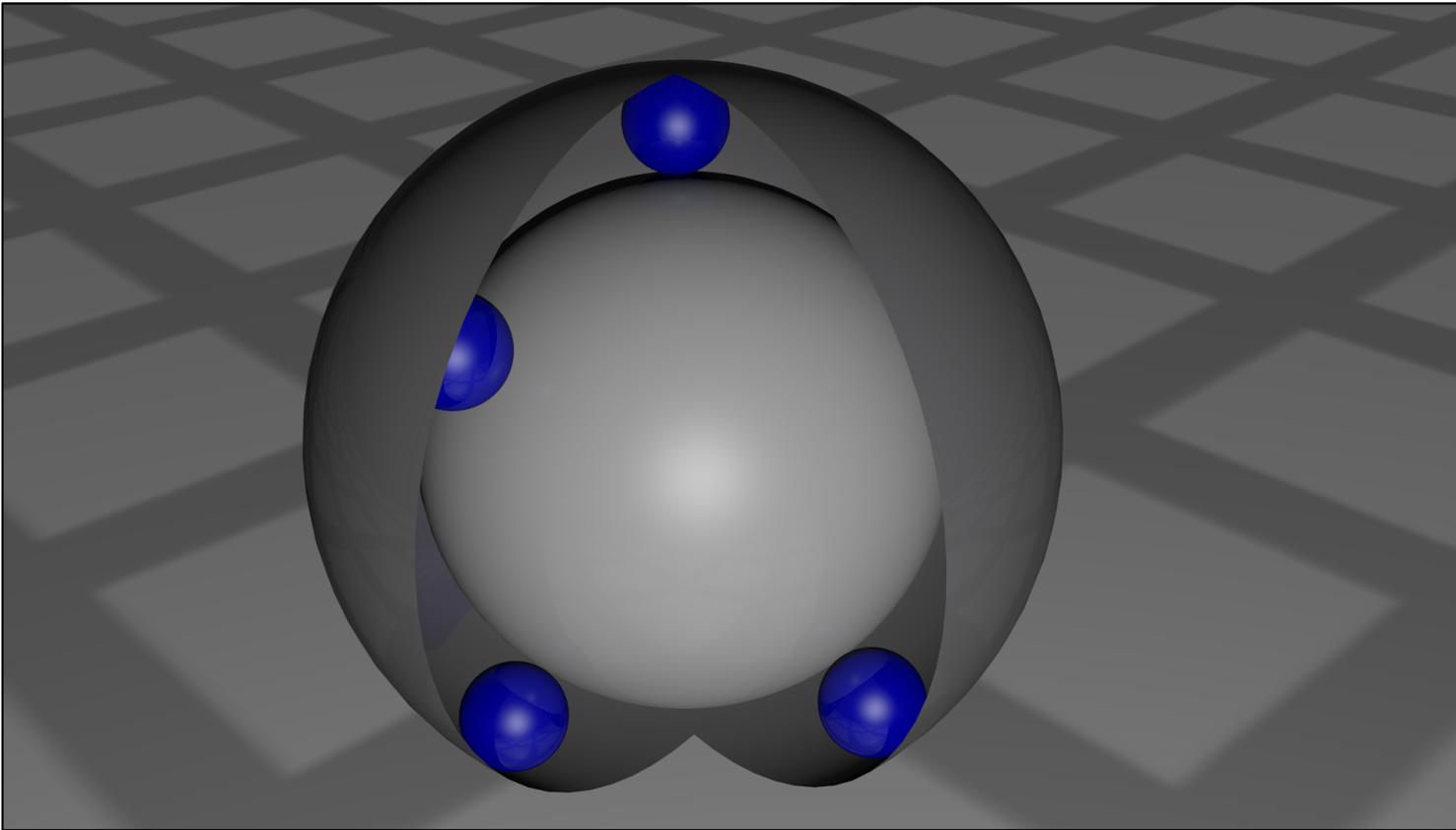
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Geometry of solid state

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- Constraints and order
- Symmetry properties of crystals

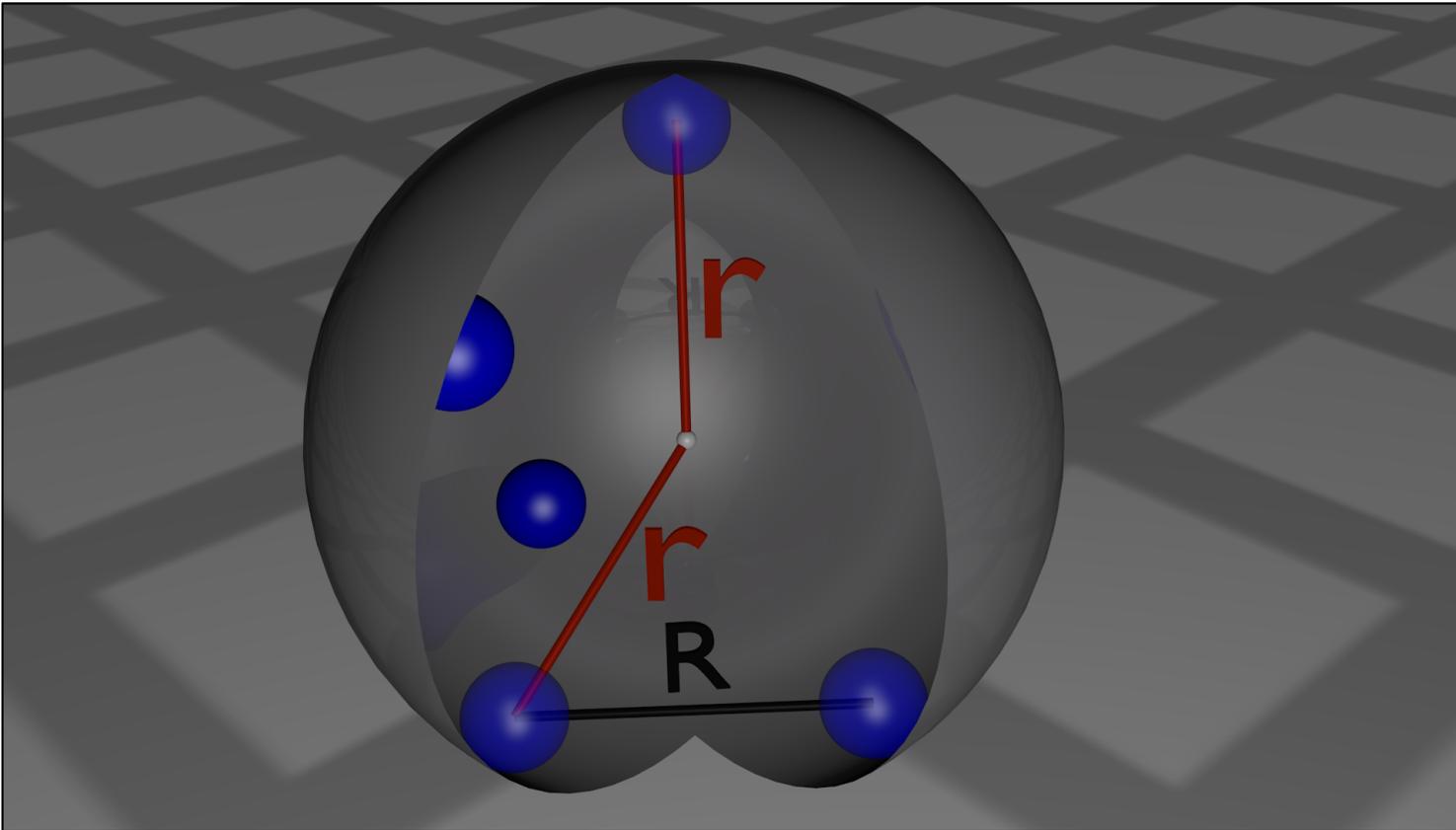
- Minimum electrostatic energy configuration for n electrons constrained to a surface of a sphere
- **Electrons** interact (repel each other) according to Coulomb's law



$$F \propto \frac{1}{R^2}$$

- The problem is related to the early (“pre quantum”) investigations of electrons in atoms [1]

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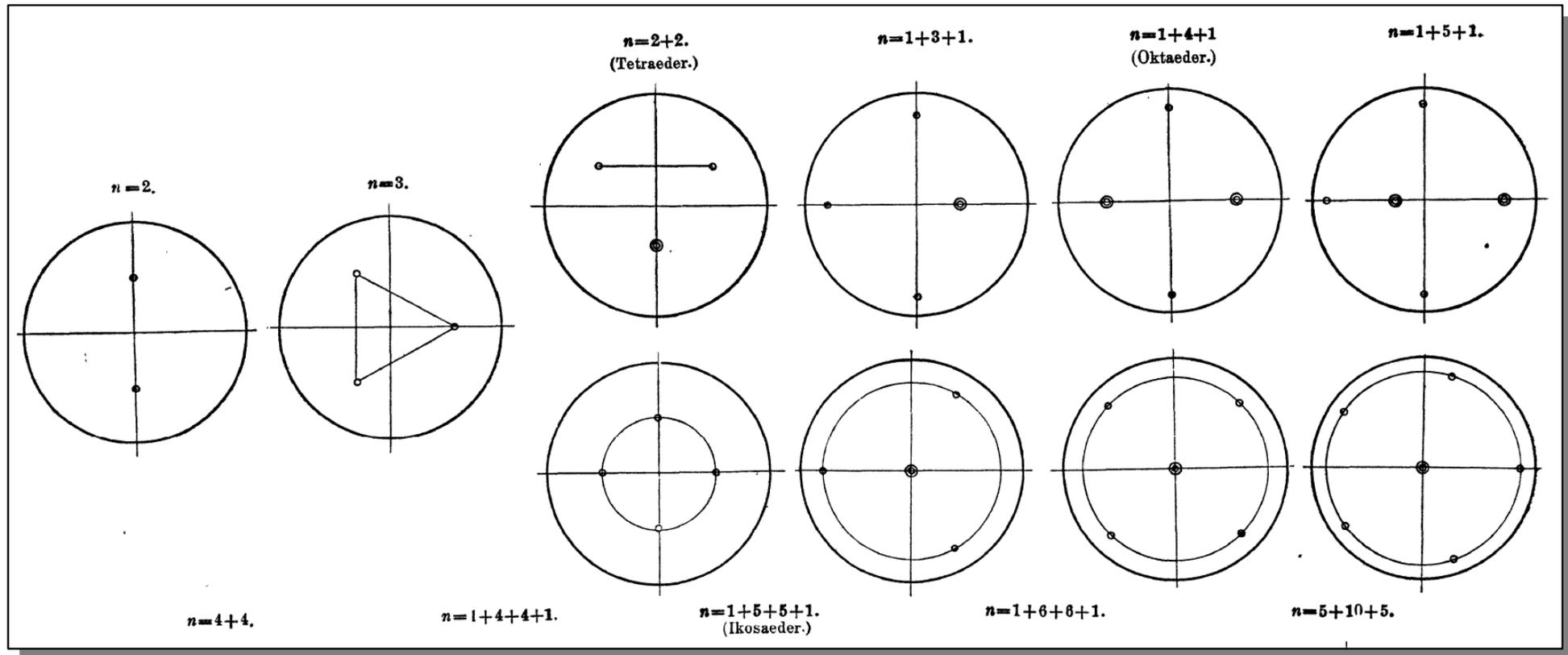
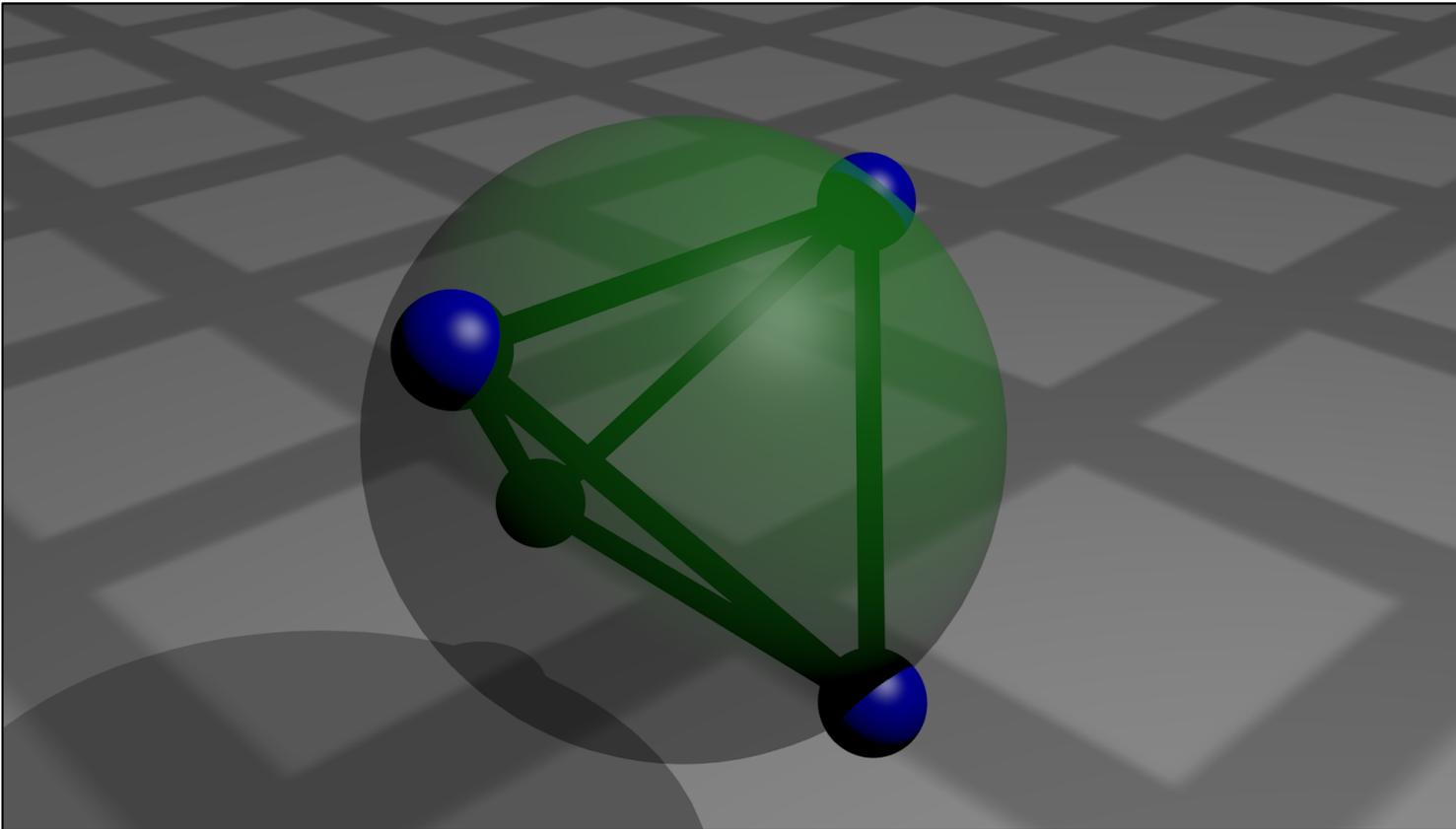


Image source: L. Föppl, Journal für die reine und angewandte Mathematik 141, 251 (1912) [6]

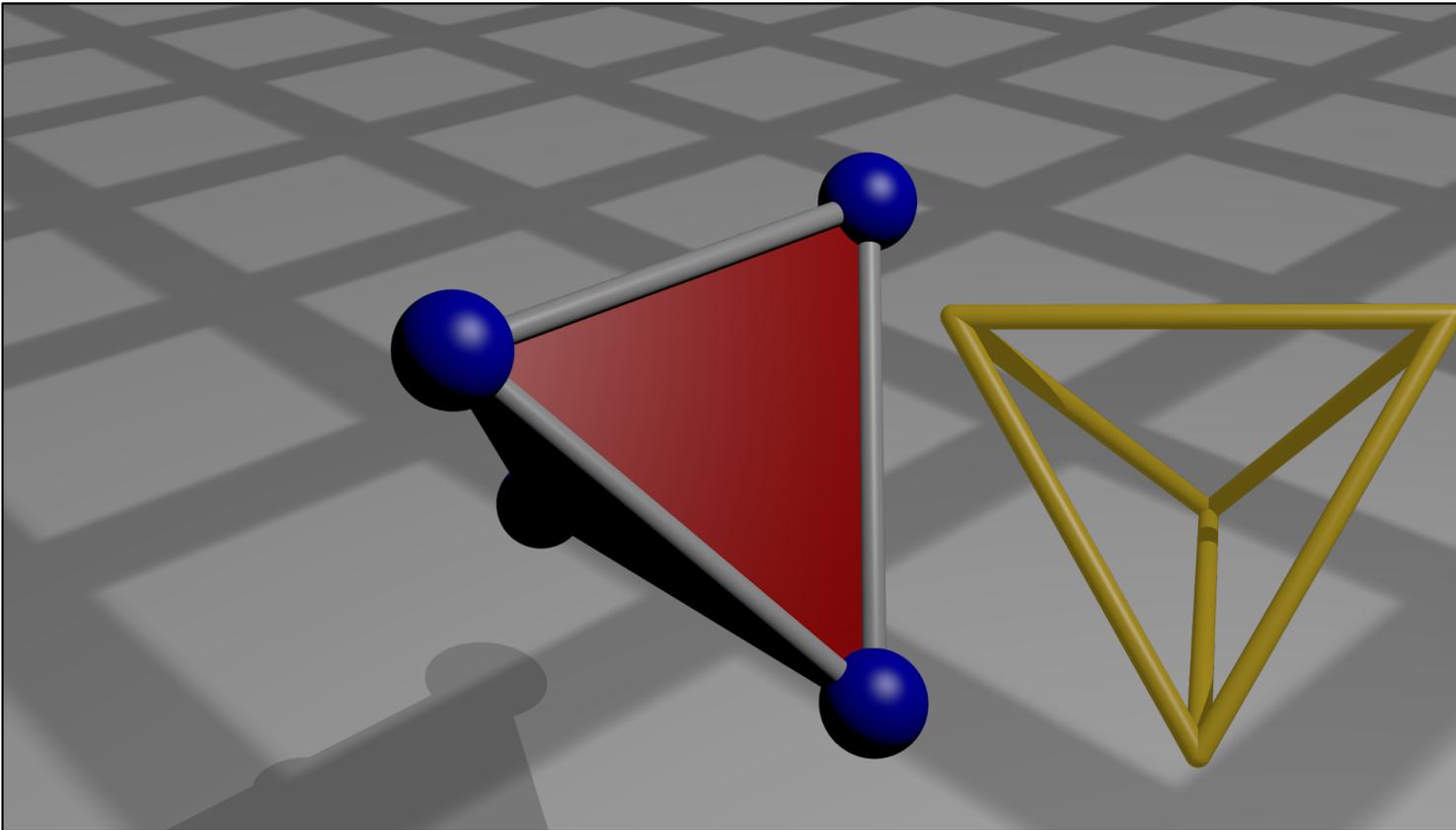
- The drawing shows stable configurations of electrons as determined by L. Föppl [1]

- Minimum electrostatic energy configuration for n electrons constrained to a surface of a sphere
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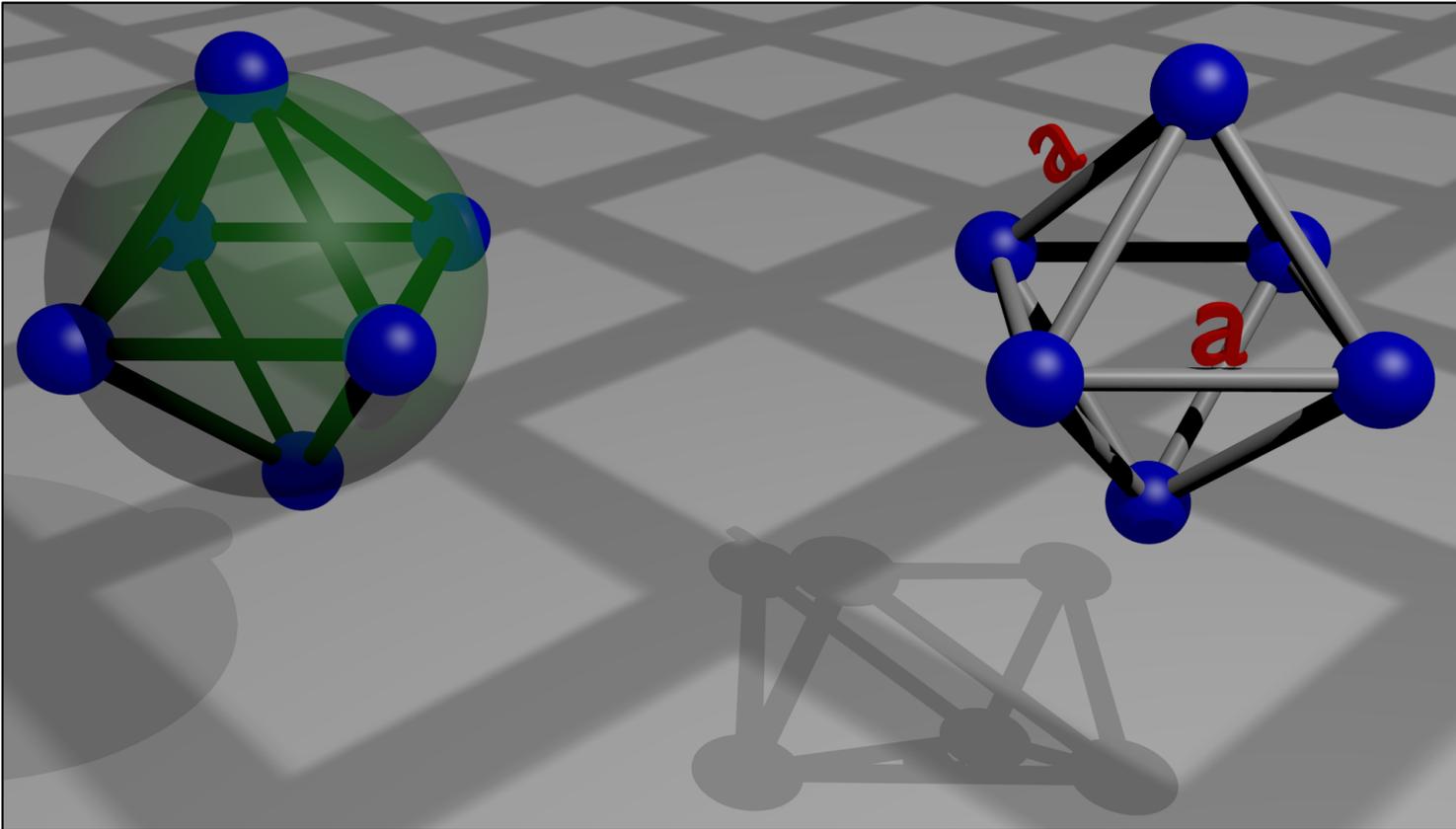
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- Minimum electrostatic energy configuration for n electrons constrained to a surface of a sphere
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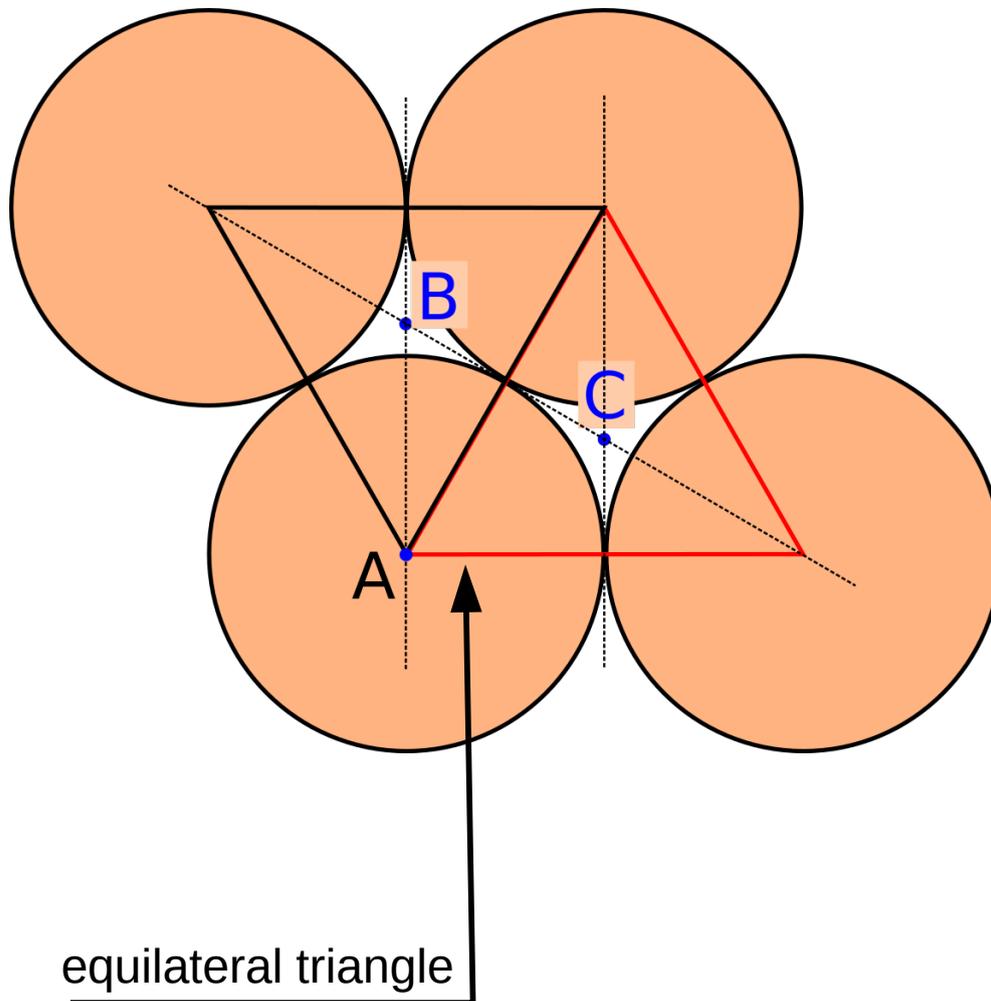
$$F \propto \frac{1}{R^2}$$

- Case of 6 electrons

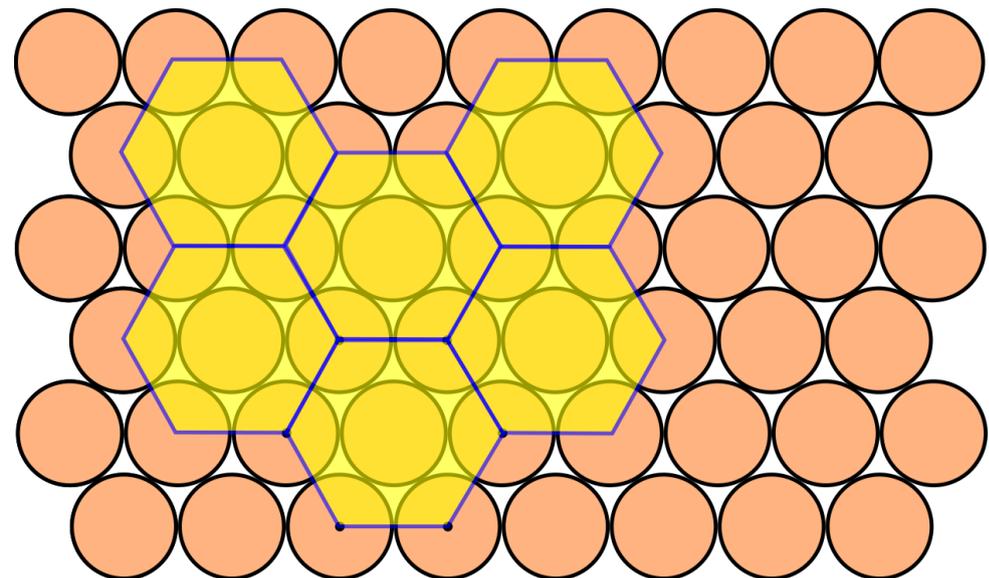


$$F \propto \frac{1}{R^2}$$

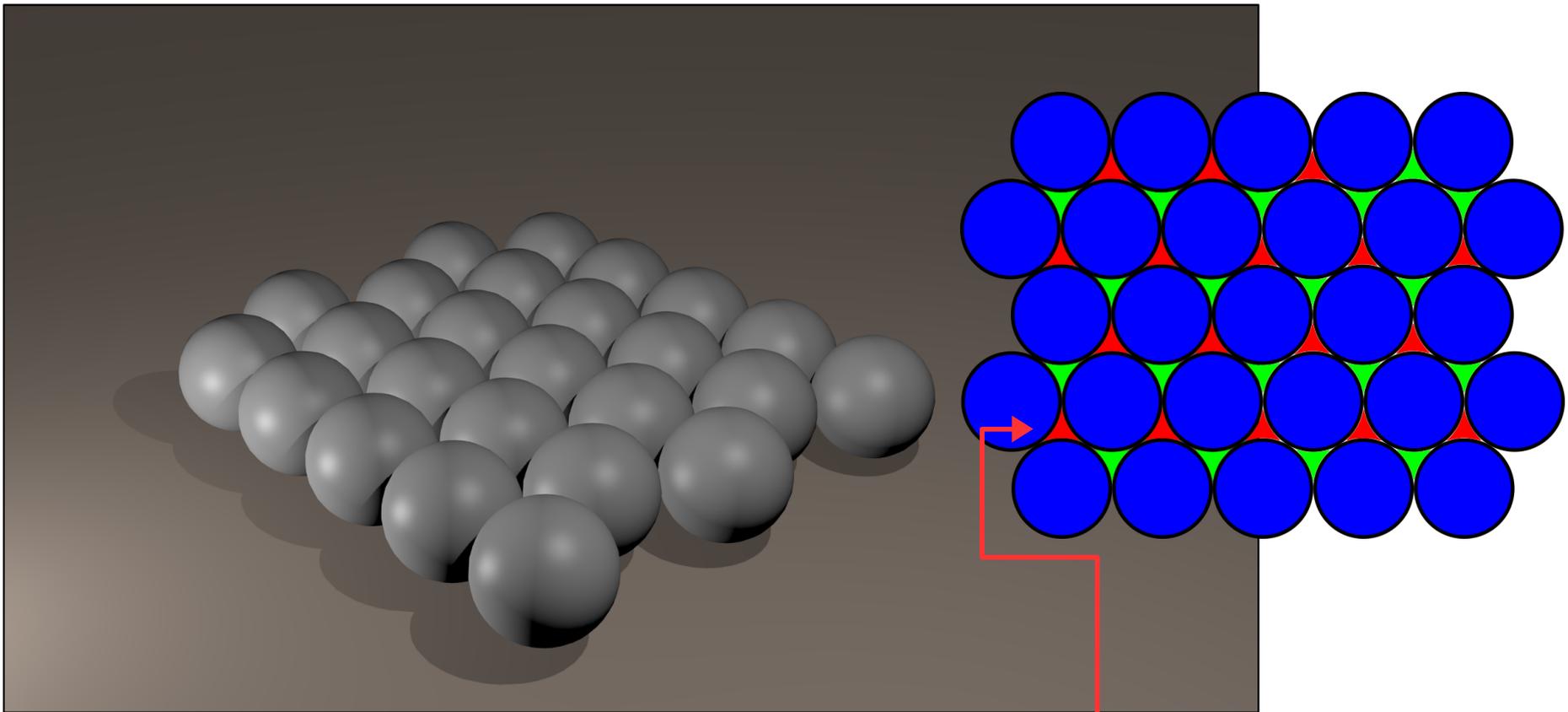
- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- First try to do this in 2D:



Hexagonal structure

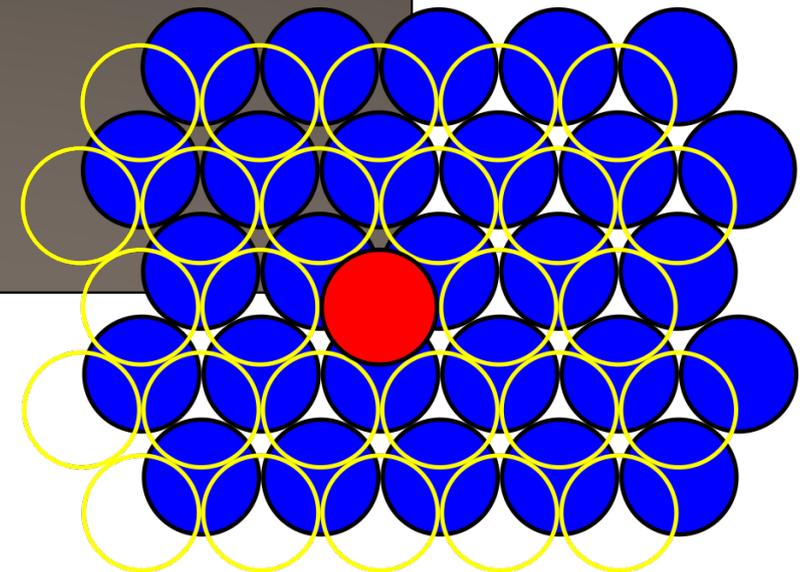
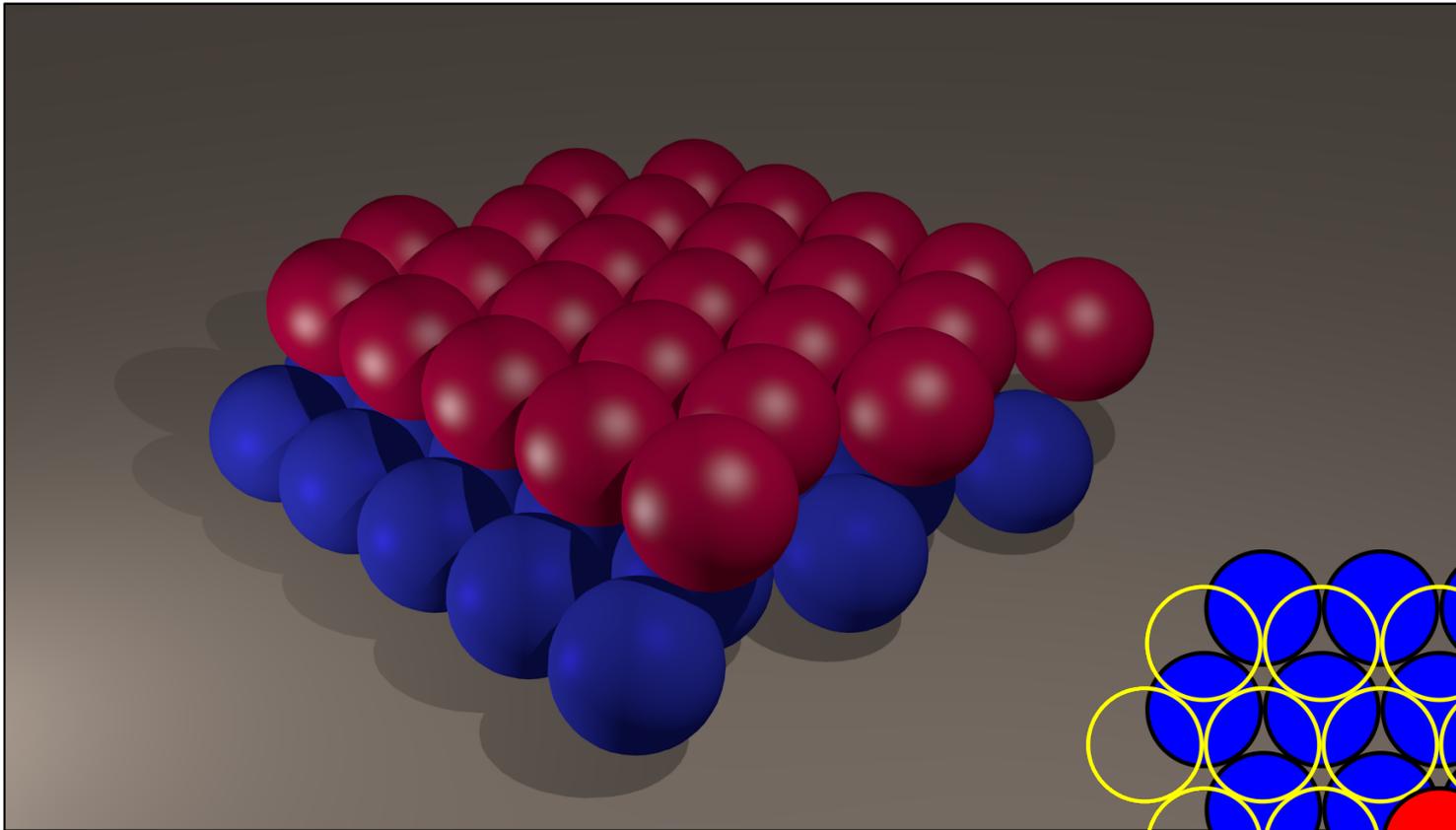


- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- And in 3D:



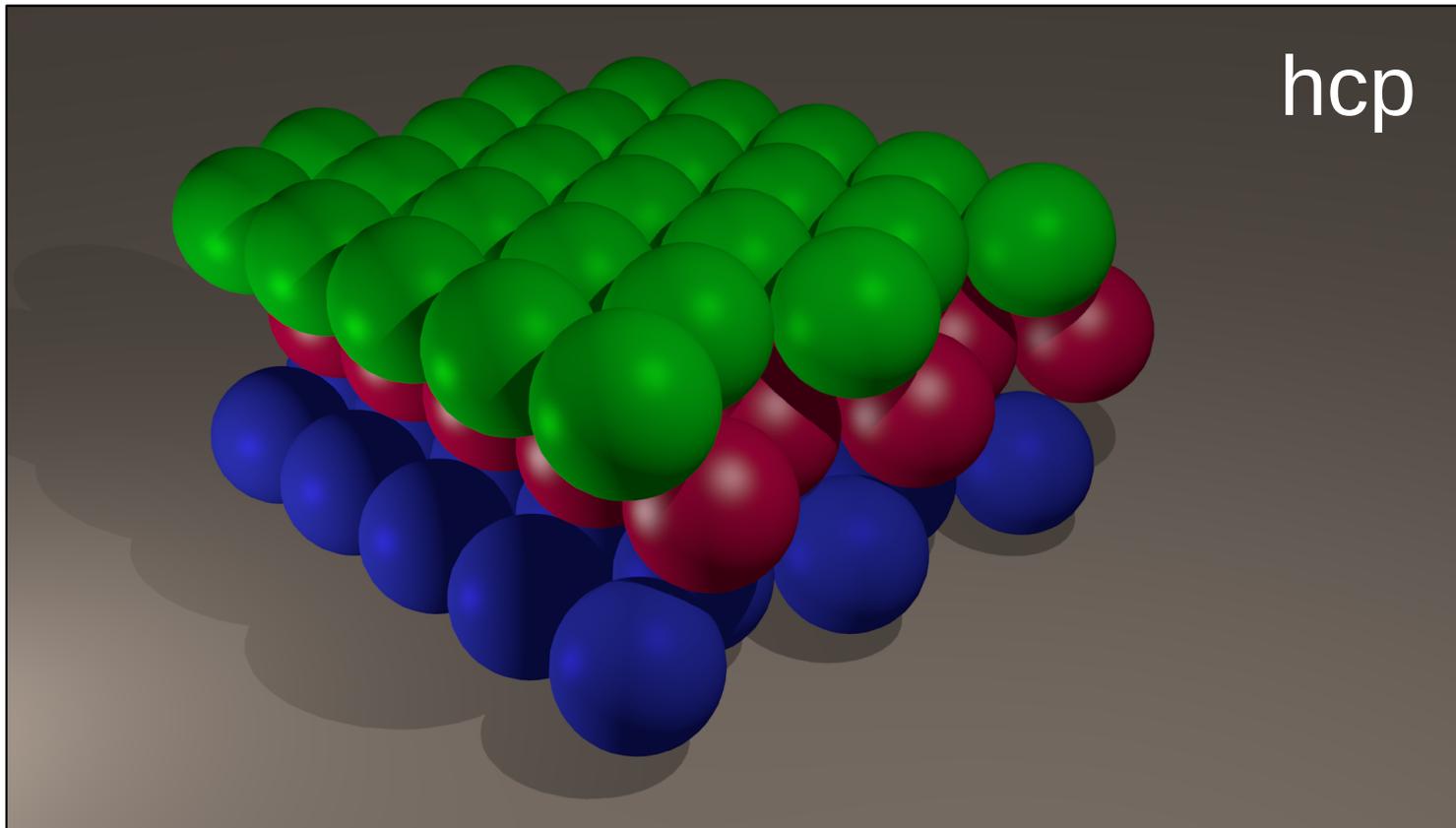
- Single layer of spheres has three times as many interstitials (**B** and **C**) as there are spheres (see schematic on previous page)

- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- And in 3D:



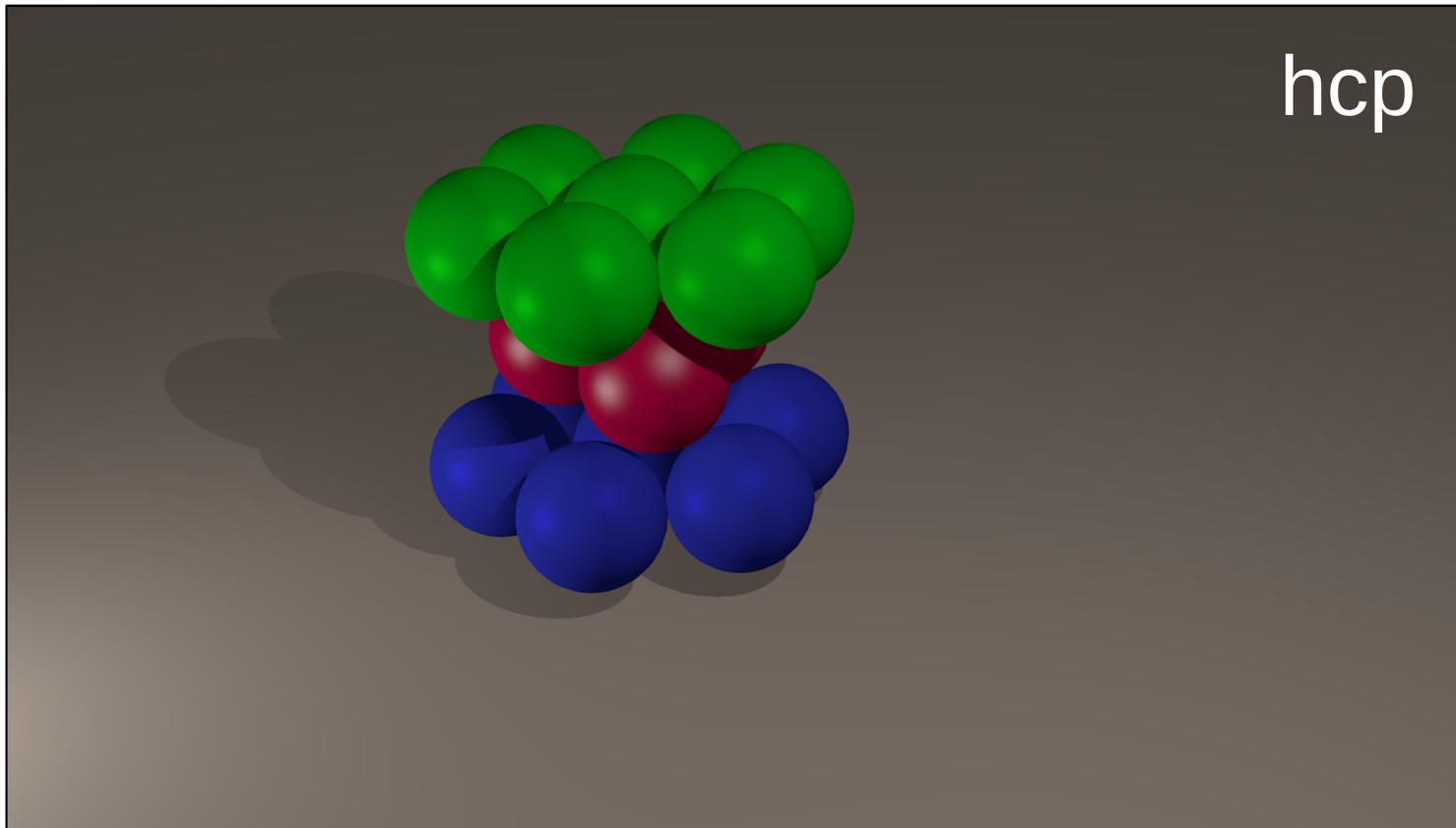
- The next, identical, layer of spheres can be put onto B or C sites

- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- And in 3D:



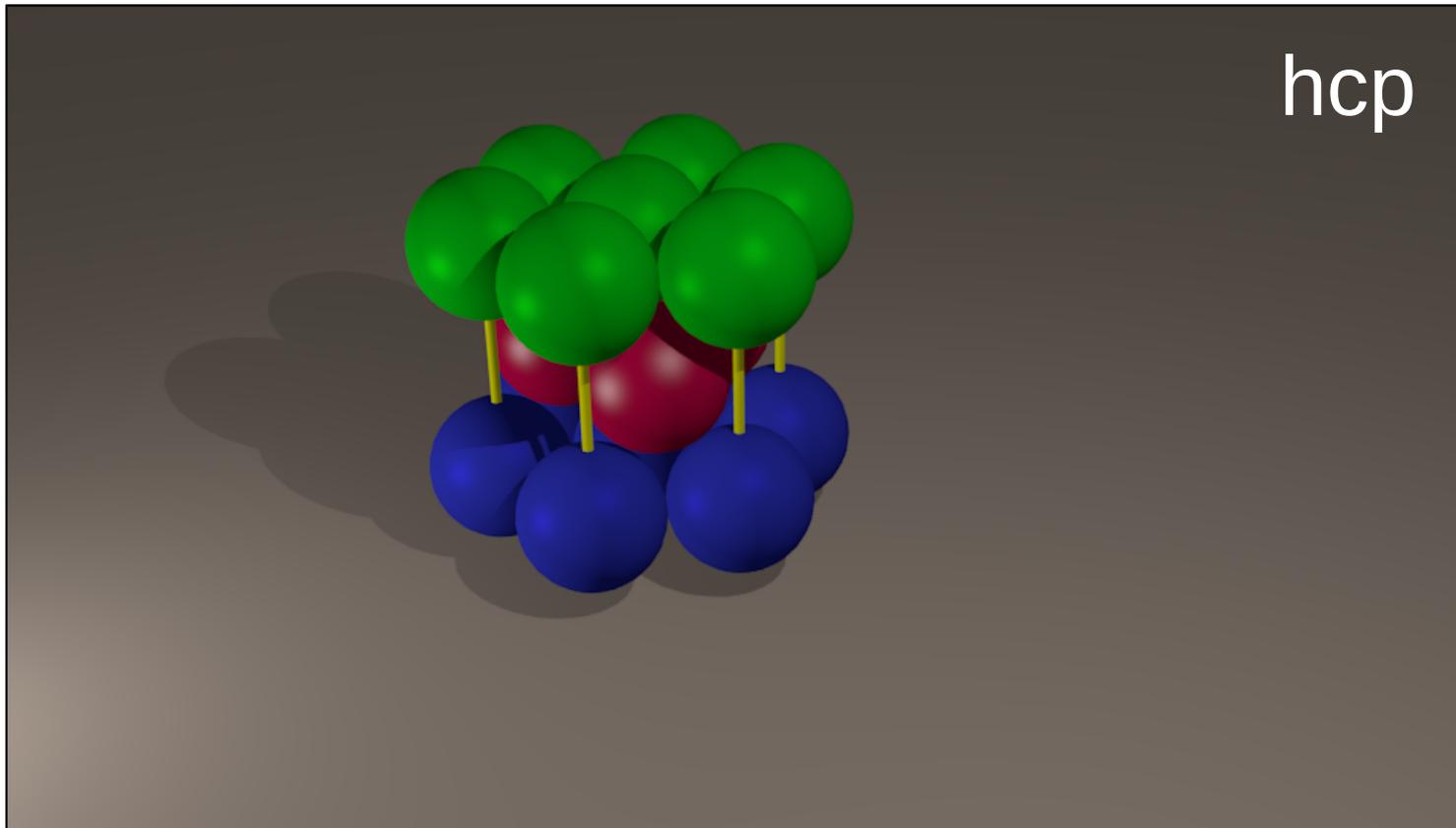
- The third layer of spheres can be put onto B or C sites of the second layer
- For ABAB... sequence the structure is called hexagonal close packed (hcp)
- For ABCABC... sequence the structure is cubic close packed or face centered cubic (see later)

- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- And in 3D:



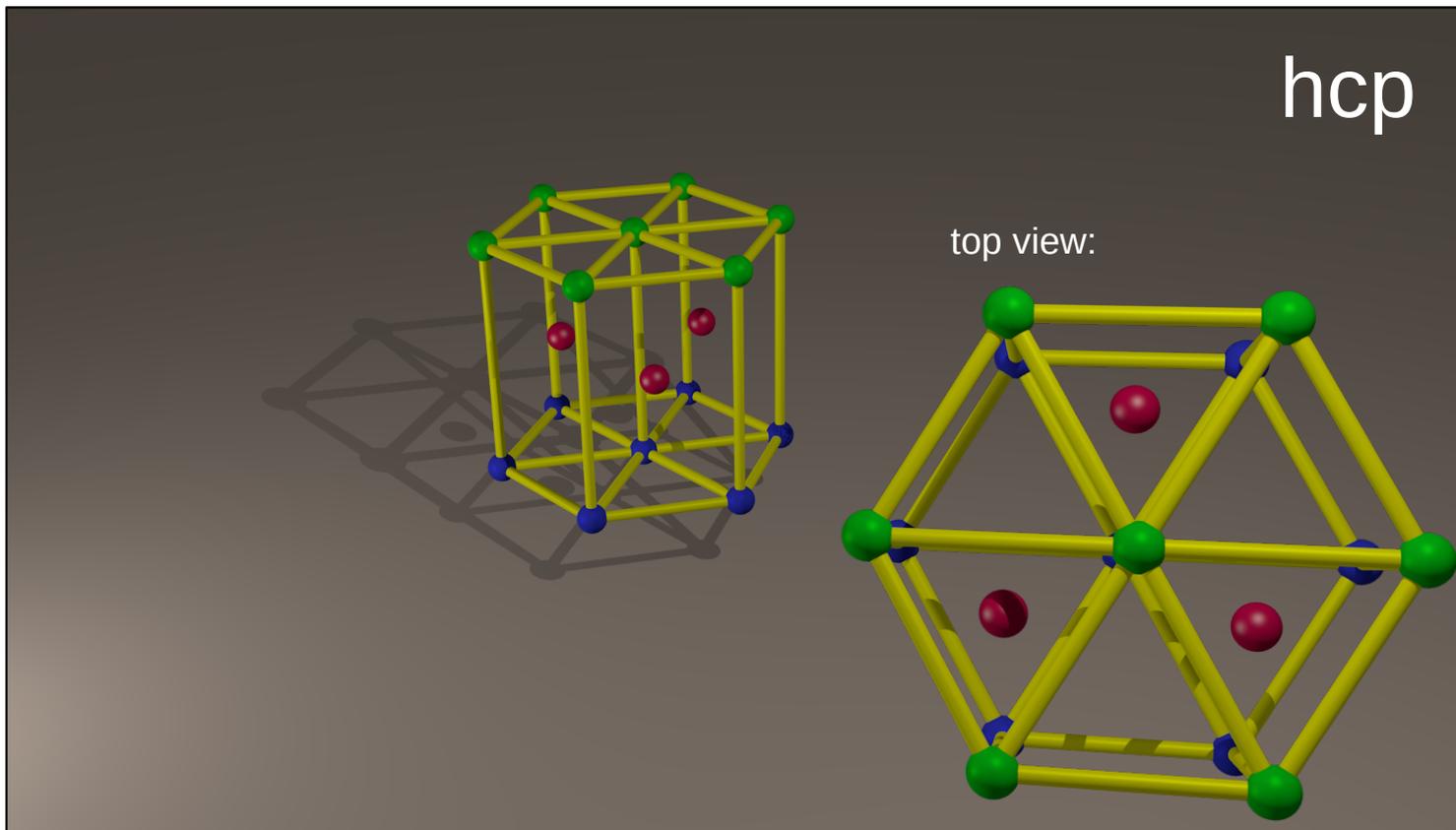
- The third layer of spheres

- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- And in 3D:



- We add the lines connecting the centers of the spheres

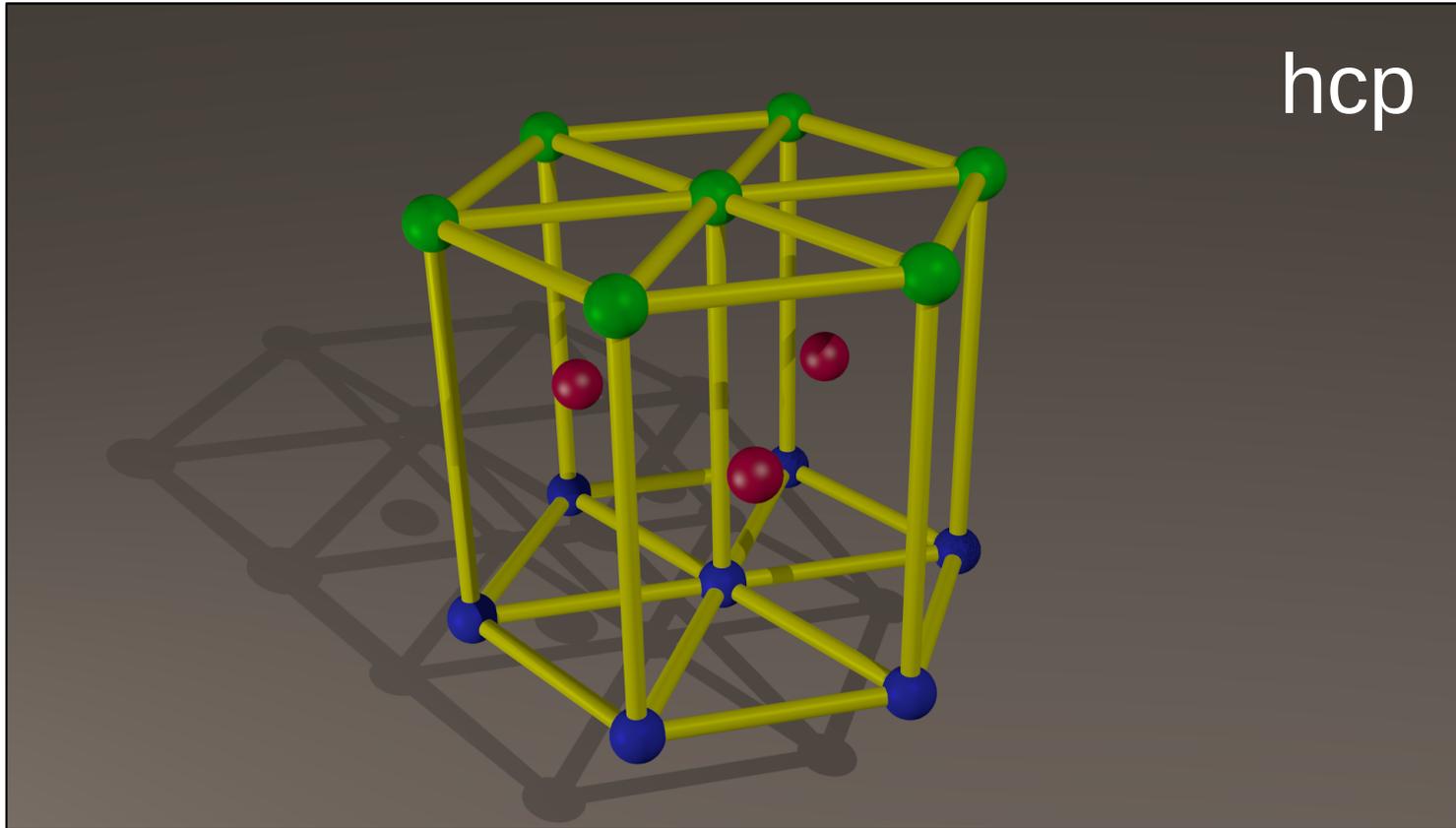
- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- And in 3D:



Notes that the spheres are deflated but the distances between them are unchanged

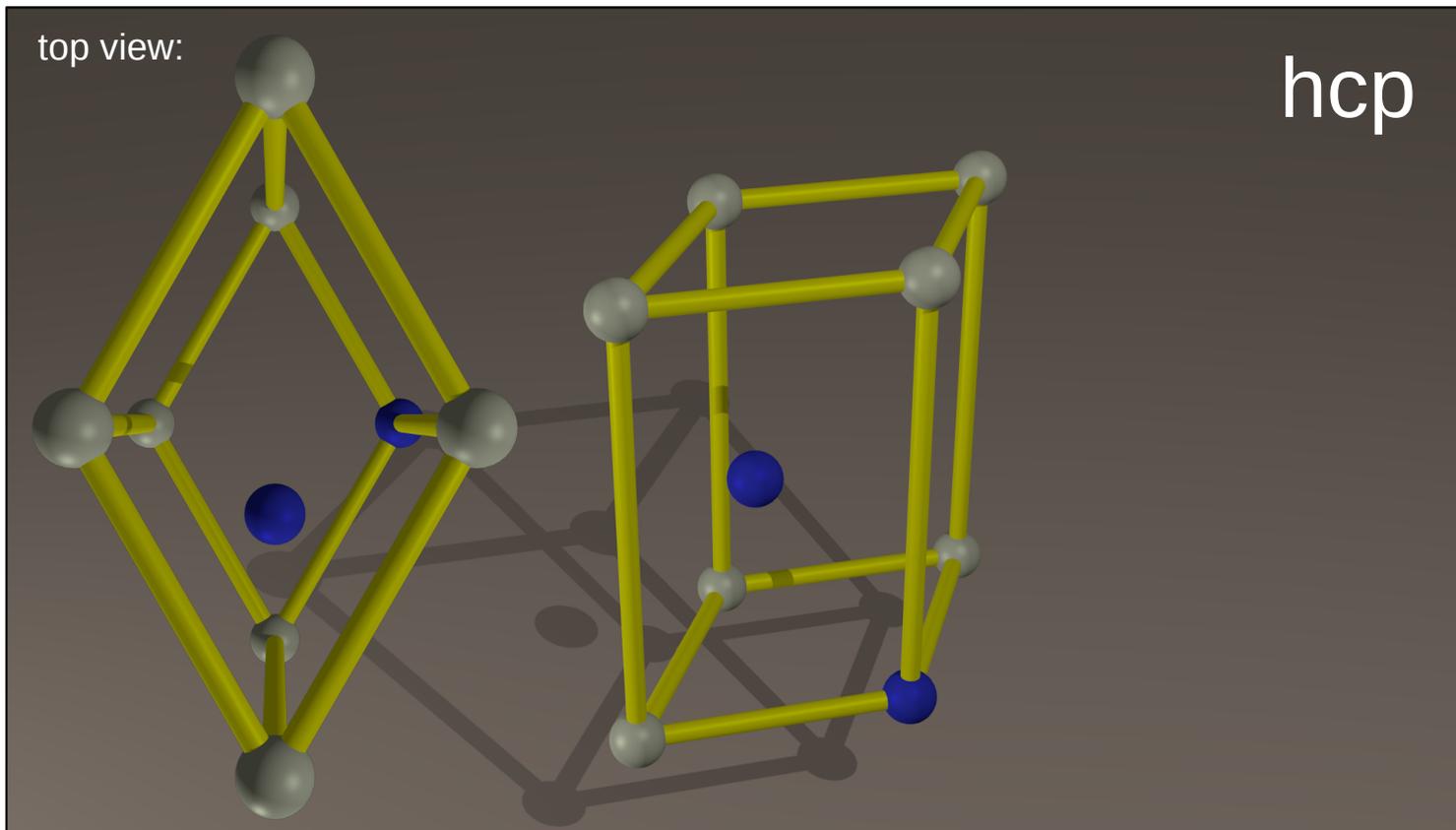
- The spheres shrunk

- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- And in 3D:



Notes that the spheres are deflated but the distances between them are unchanged

- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- And in 3D:

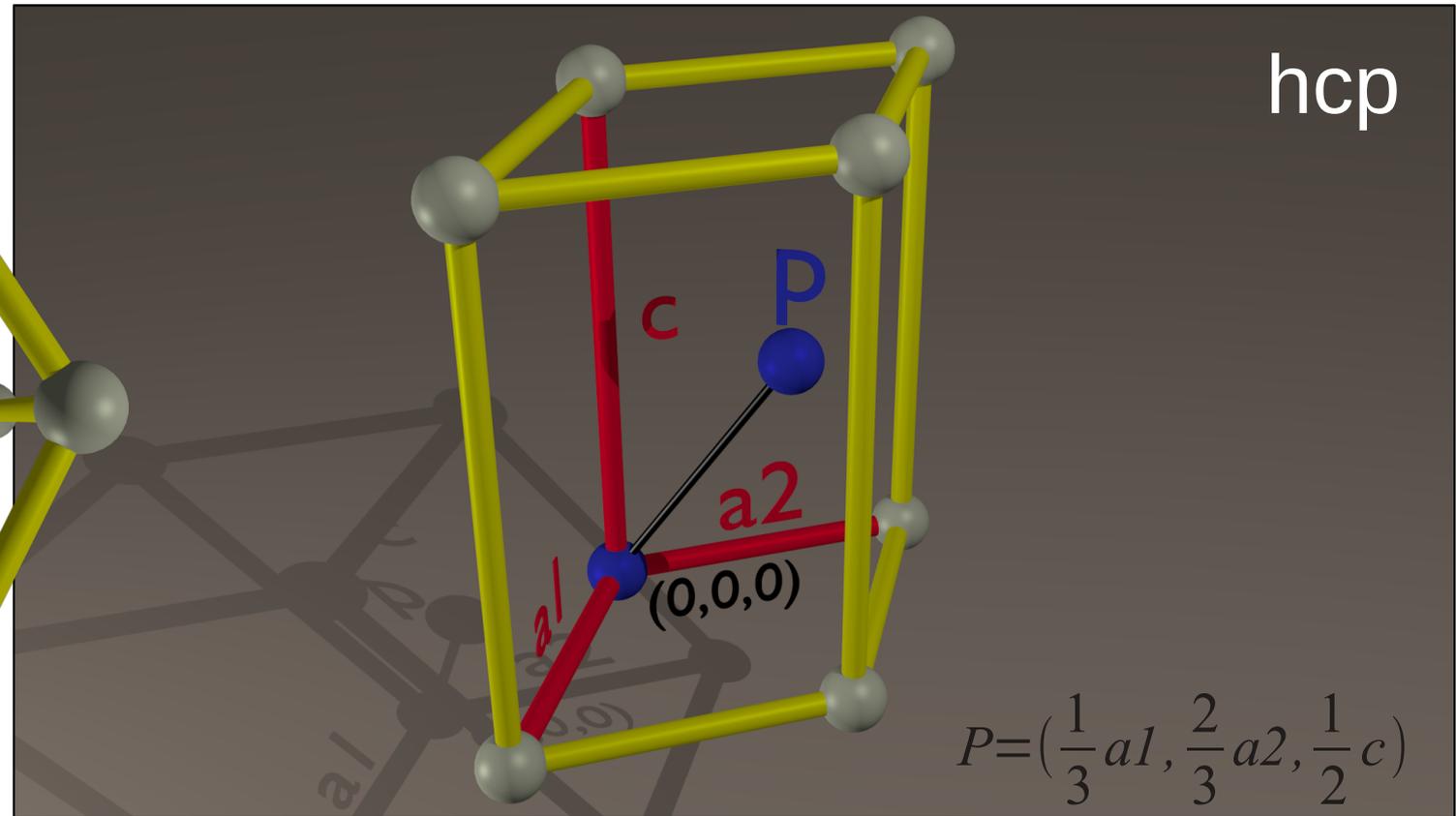
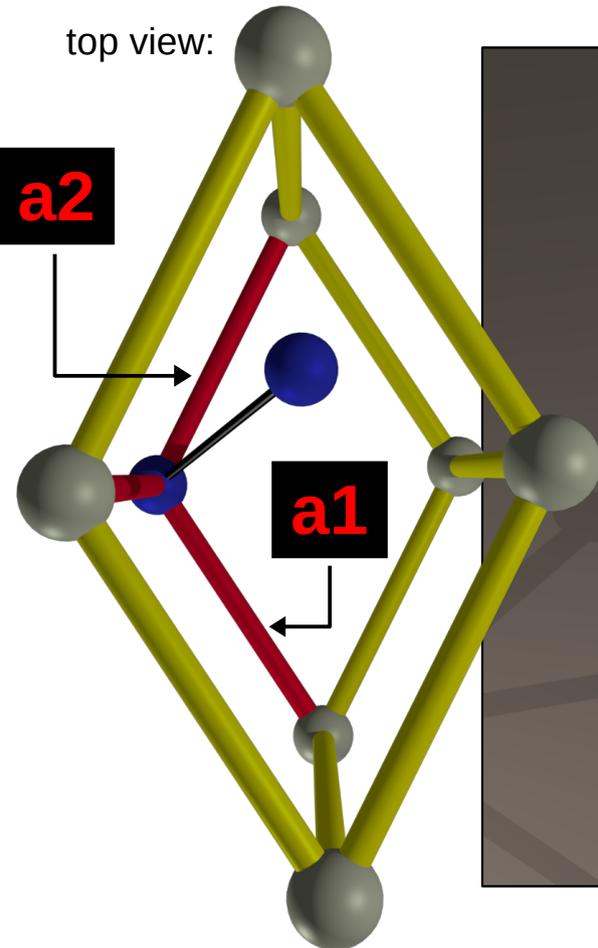


Notes that the spheres are deflated but the distances between them are unchanged

- **Two blue** spheres form the base of *hcp* structure – see Bravais's lattice (later on)

- How to arrange hard spheres to fill the space most effectively (to have as little empty space between them as possible)?
- And in 3D:

top view:

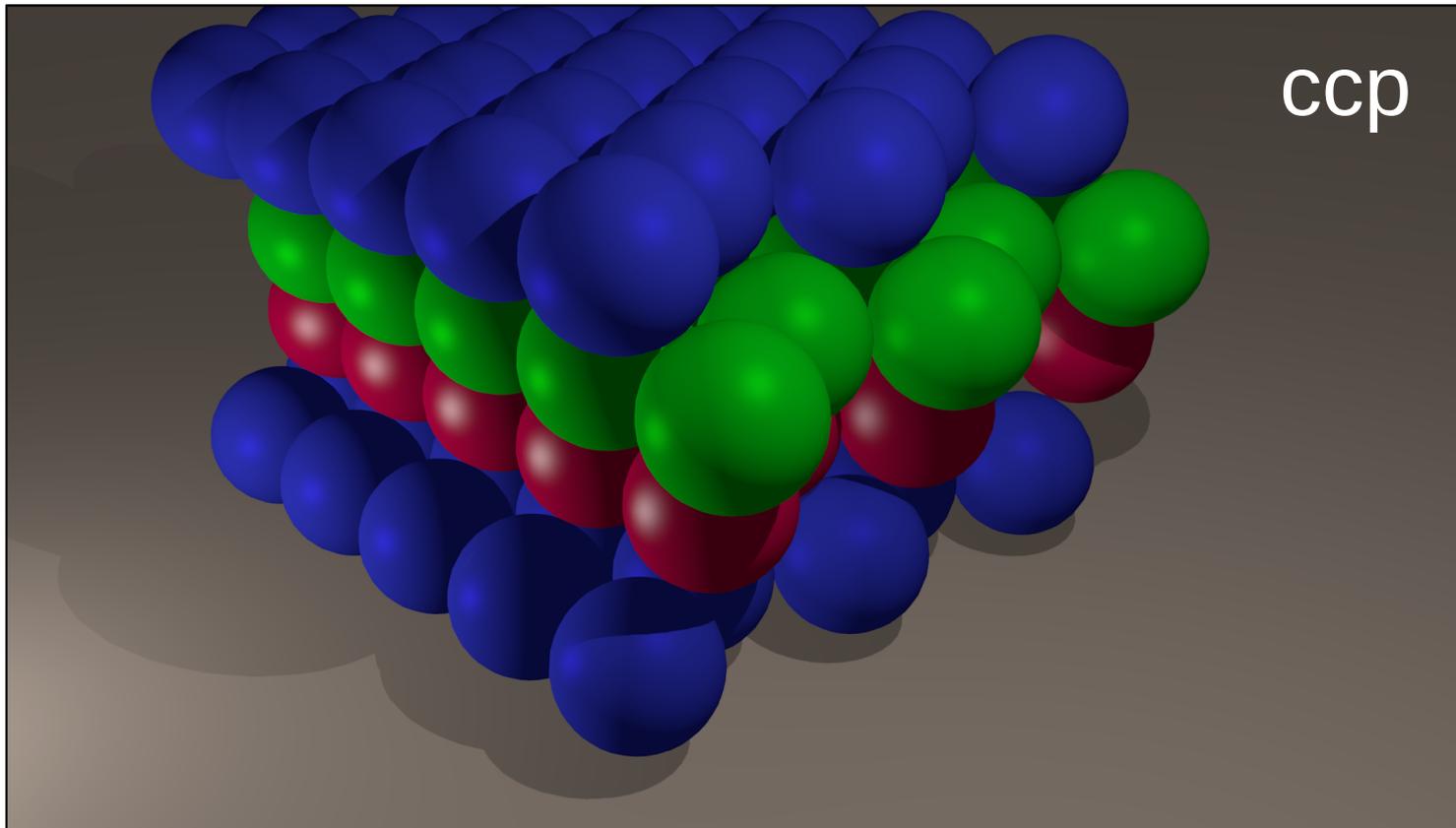


Notes that the spheres are deflated but the distances between them are unchanged

- Edges of the cell: $a_1 = a_2 = 2 \cdot r$ r – sphere radius

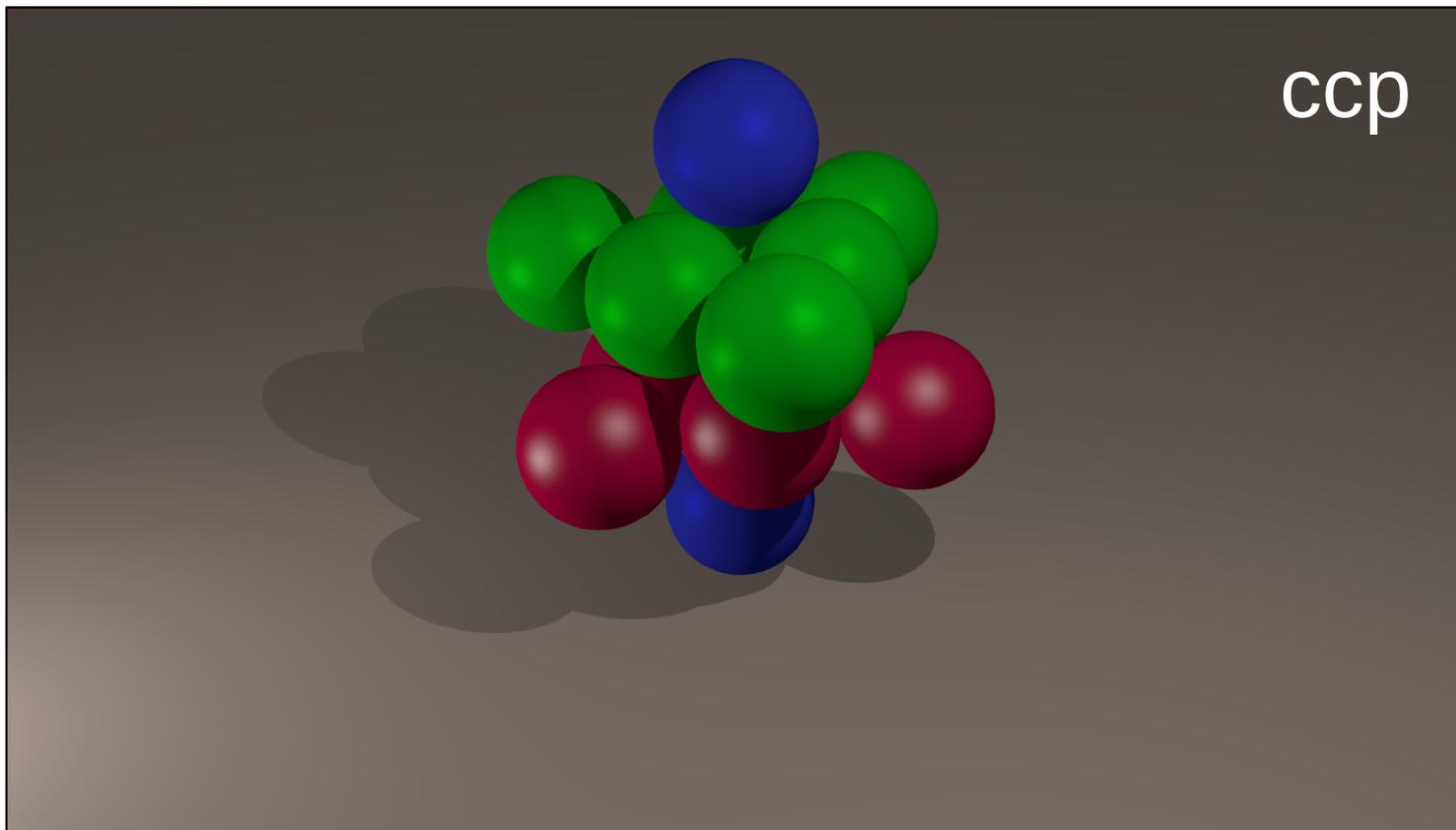
$$c = 2 \frac{\sqrt{6}}{3} \cdot a_1 \approx 1.63 \cdot a_1$$

- Cubic close packing [2]



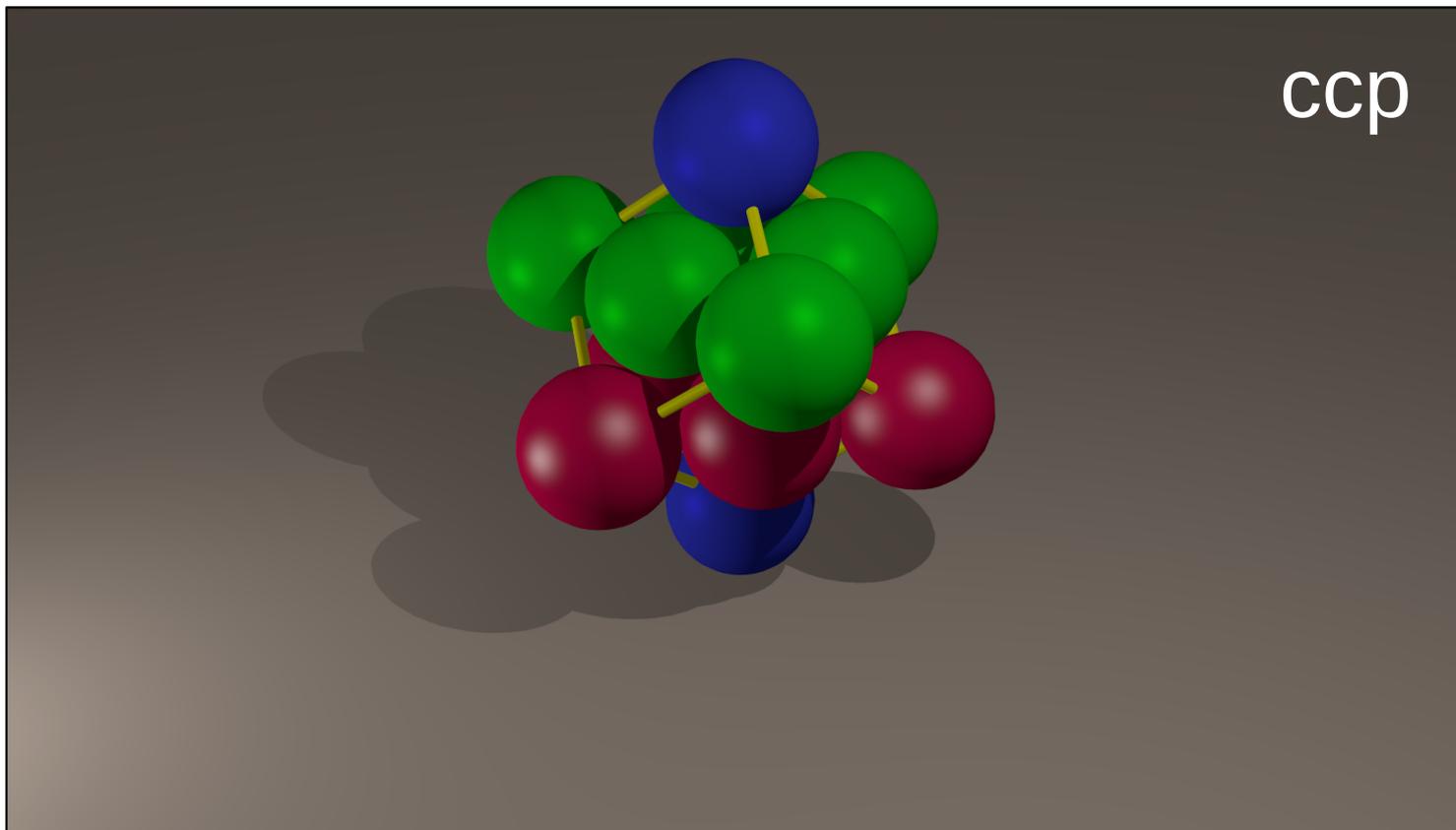
- ABCABC... stacking
- Note that any stacking (without repetition) gives close-packing – so called stacking faults [2]

- Cubic close packing [2]



- ABCABC... stacking

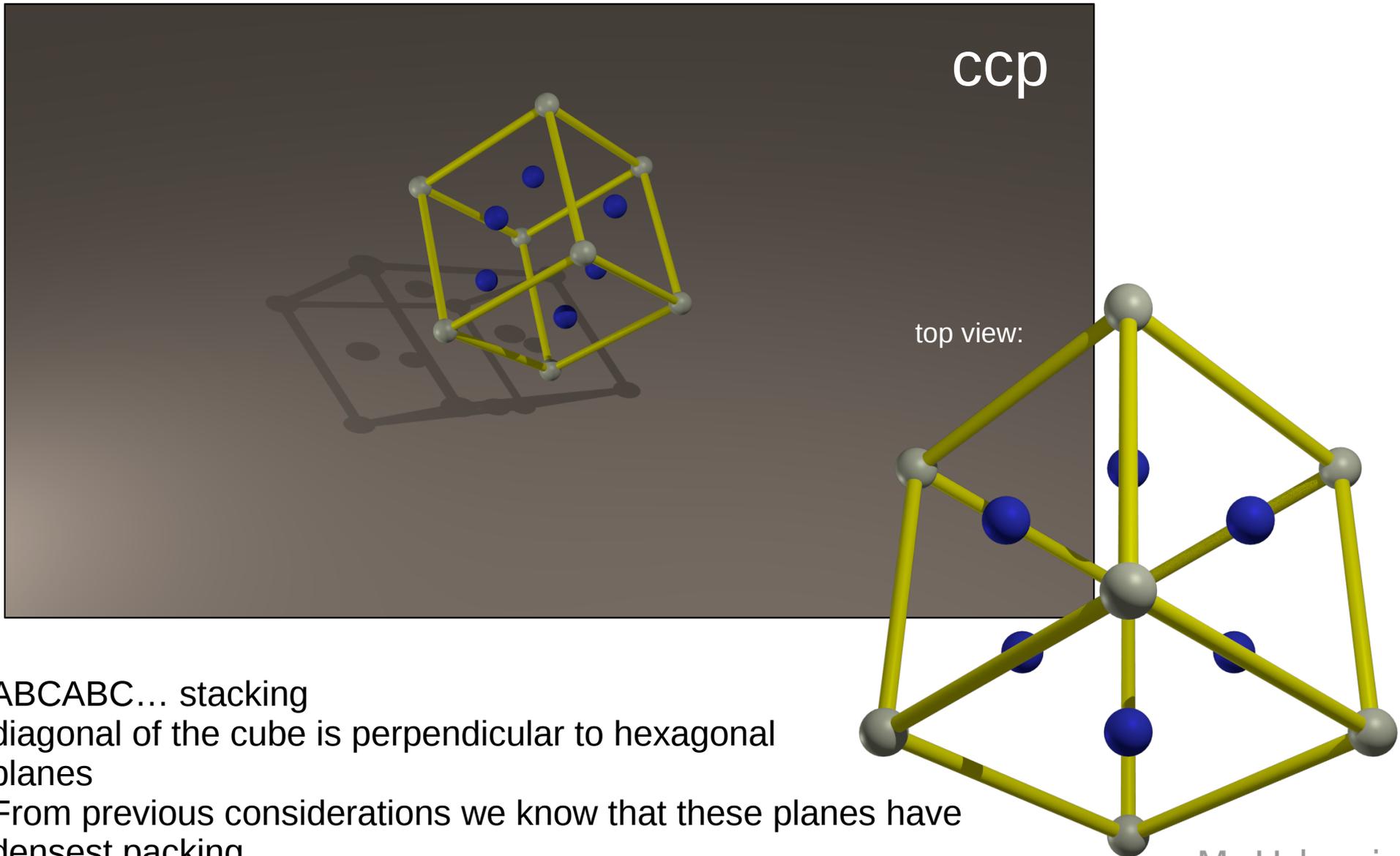
- Cubic close packing [2]



- ABCABC... stacking
- We add the lines connecting the centers of the spheres

- Cubic close packing [2]

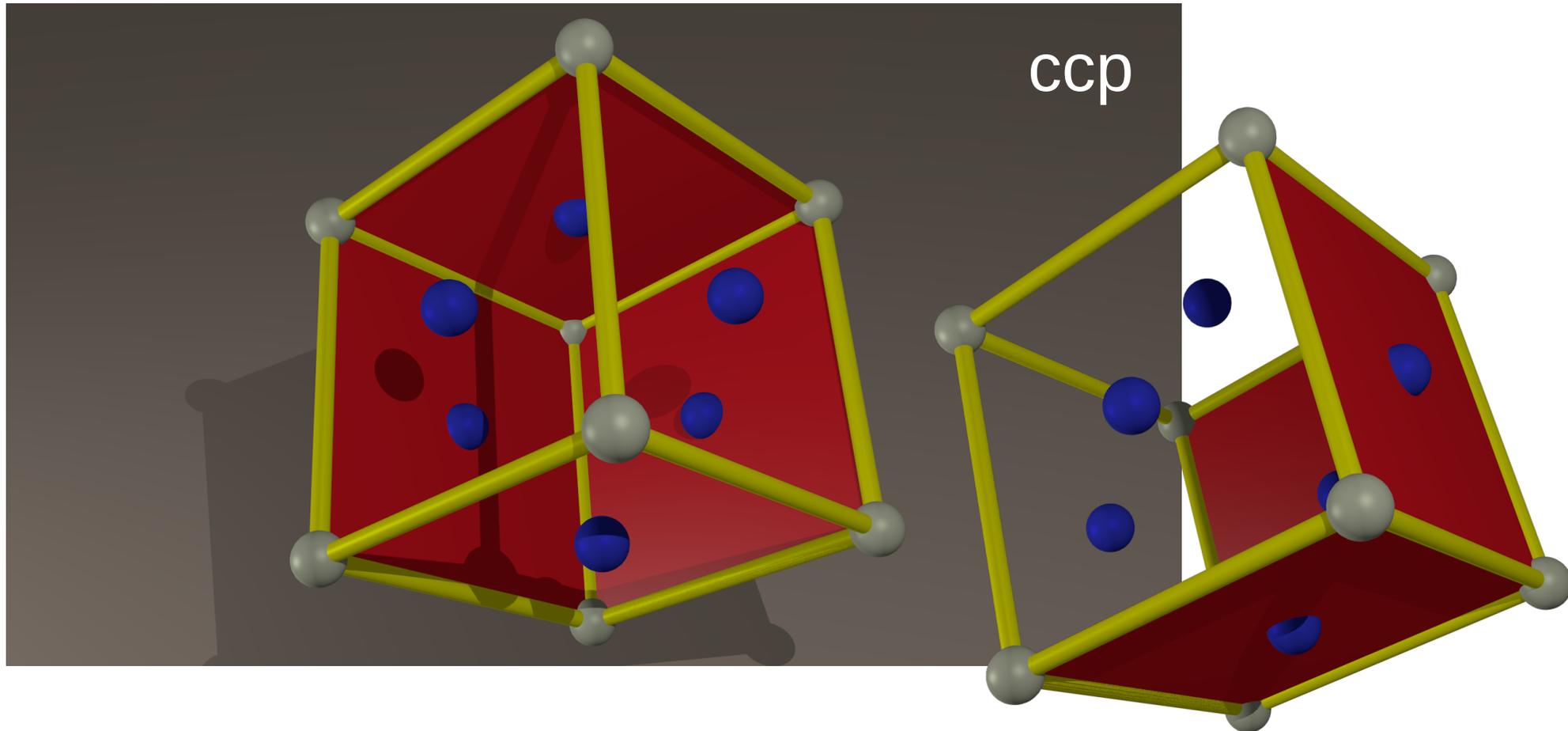
Notes that the spheres are deflated but the distances between them are unchanged



- ABCABC... stacking
- diagonal of the cube is perpendicular to hexagonal planes
- From previous considerations we know that these planes have densest packing

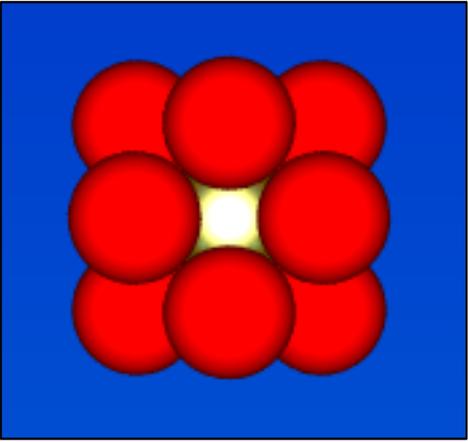
- Cubic close packing [2] = **face-centred cubic**

Notes that the spheres are deflated but the distances between them are unchanged

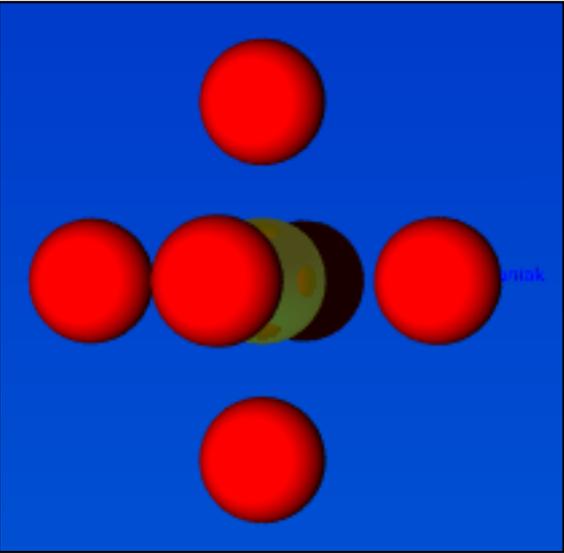


- ABCABC... stacking
- Center of each face of the cube is occupied by a **sphere** (atom)
- The cube contains 4 atoms ($8 \cdot \frac{1}{8}$ corner atom + $6 \cdot \frac{1}{2}$ face atom)

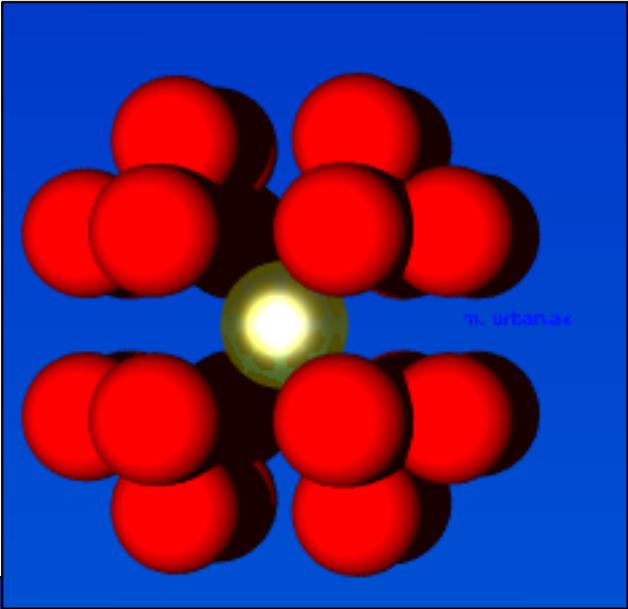
- Coordination layers in **fcc** structure – number of nearest neighbors, second nearest neighbors (snn),...



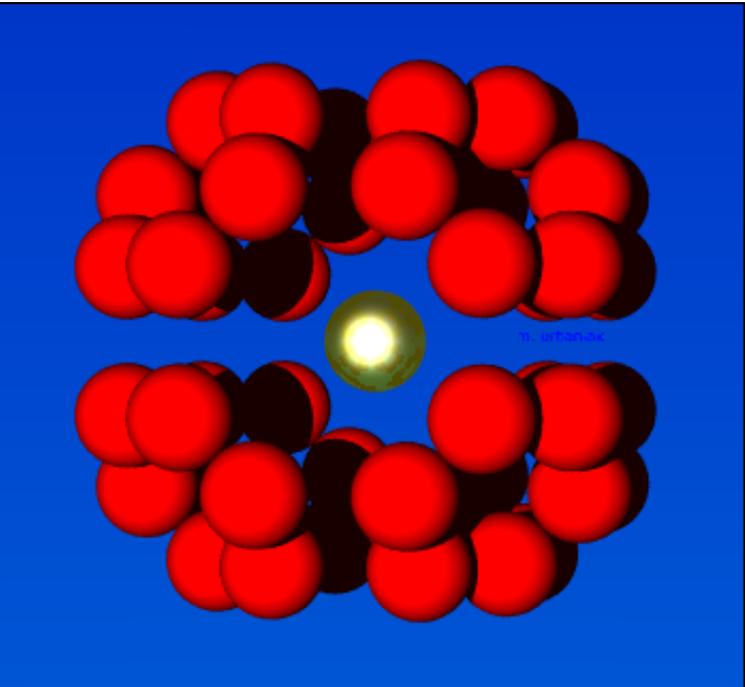
first layer



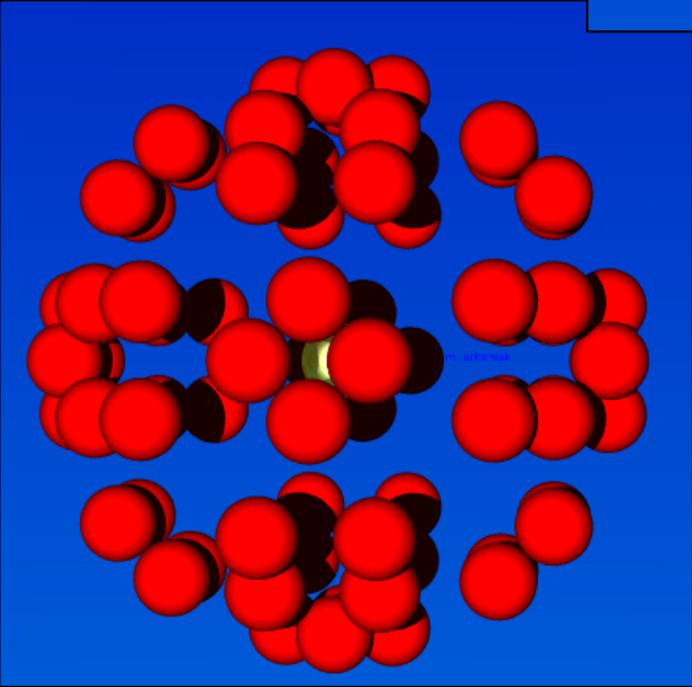
second layer



third layer



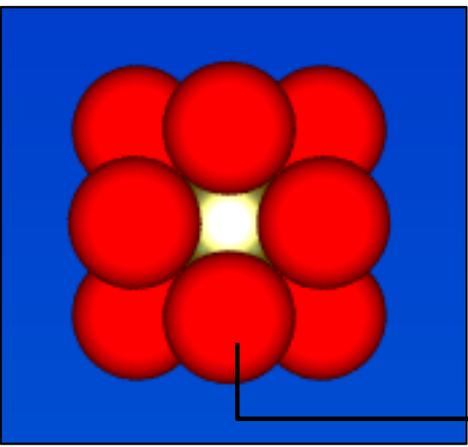
7th layer – 24 neighbors



13th layer – 72 neighbors

Close packing of equal spheres

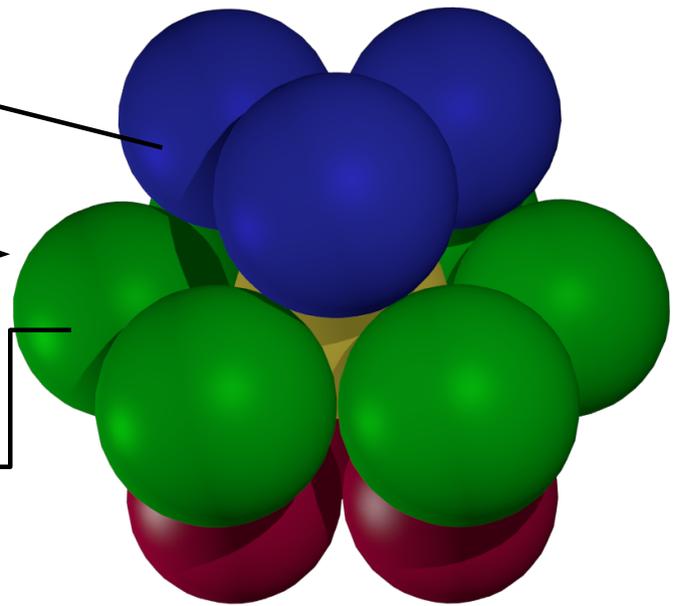
- Coordination layers in **fcc** structure – number of nearest neighbors, second nearest neighbors (snn),...



3 neighbors in **lower** and **upper** layers

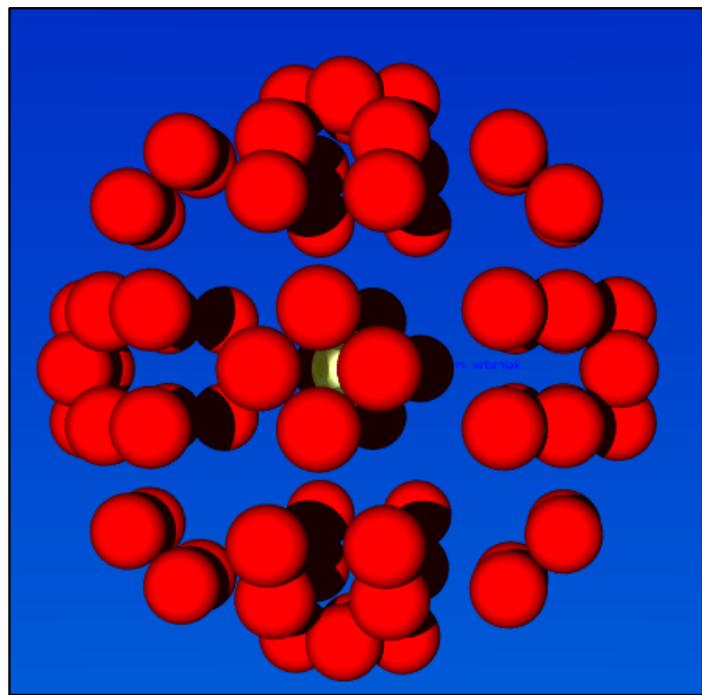
alternative view (stacking) →

6 neighbors in "own" close-packed layer



first layer – 12 neighbors

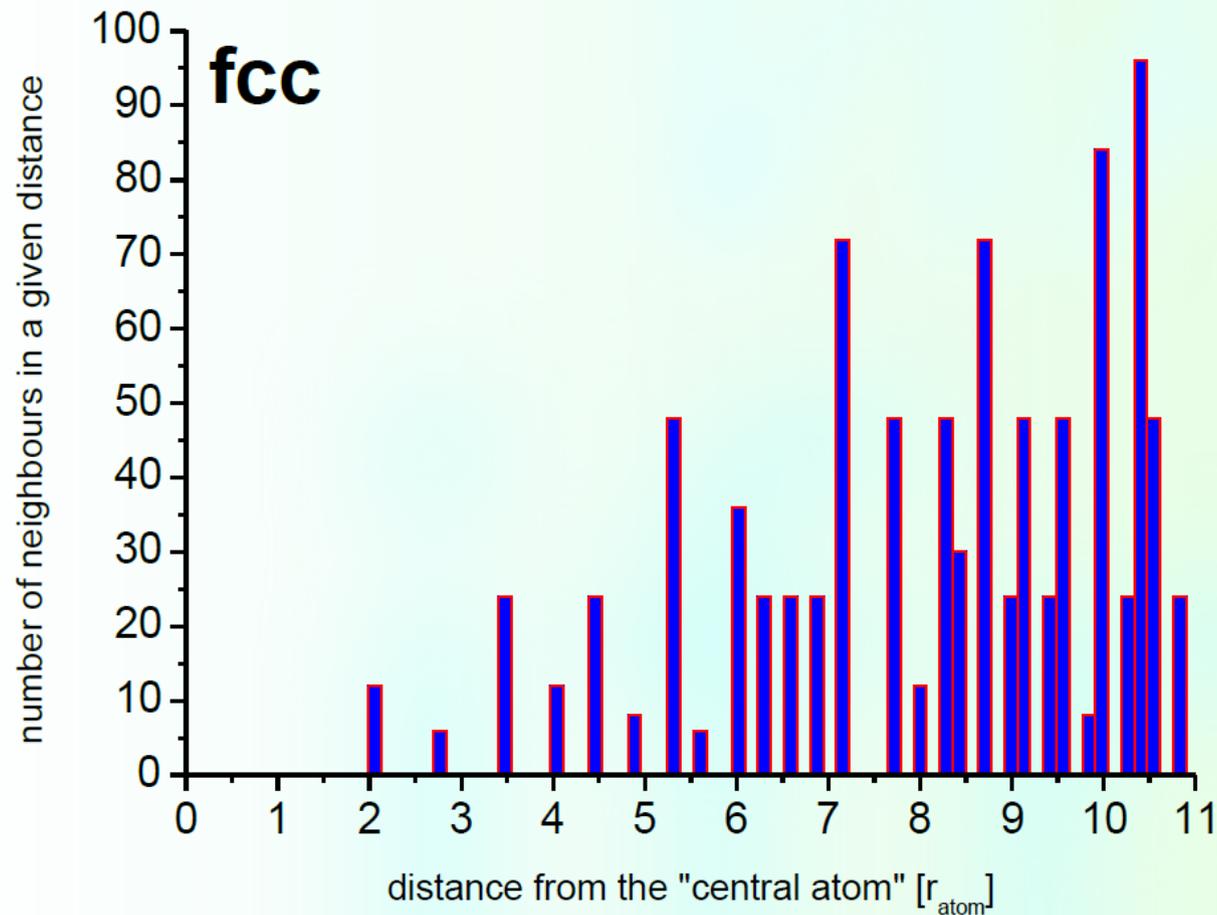
One is usually not interested in neighbors farther than in the second layer*



13th layer – 72 neighbors

*structural investigations may be exception

- Coordination layers in **fcc** structure – number of nearest neighbors, second nearest neighbors (snn),...



Fcc structure:

nn 12

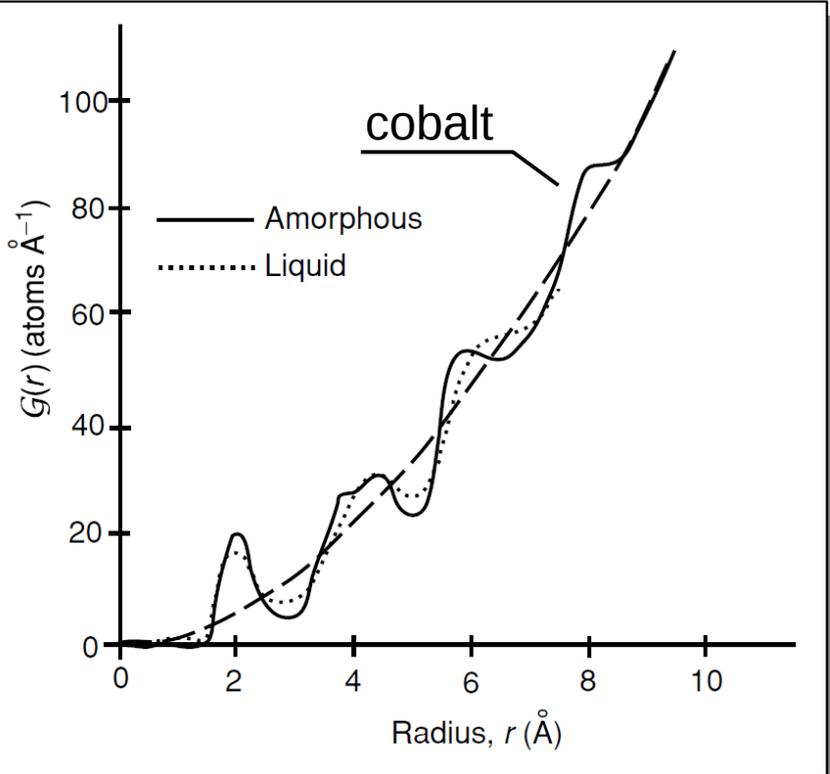
nnn 6

snnn 24

....

Close packing of equal spheres

- Amorphous substances – radial distribution function



- Amorphous solids have no crystal lattice*
- The metallic amorphous alloys are usually produced by quenching (very fast cooling) from liquid or vapor

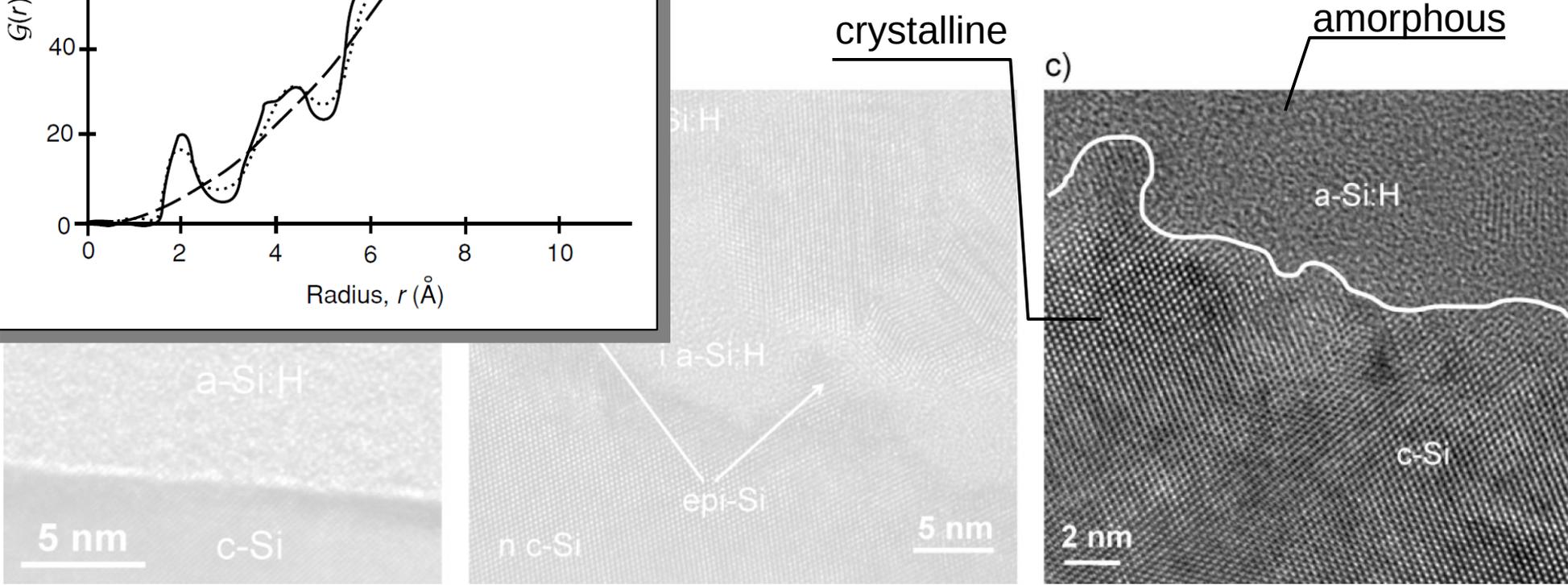


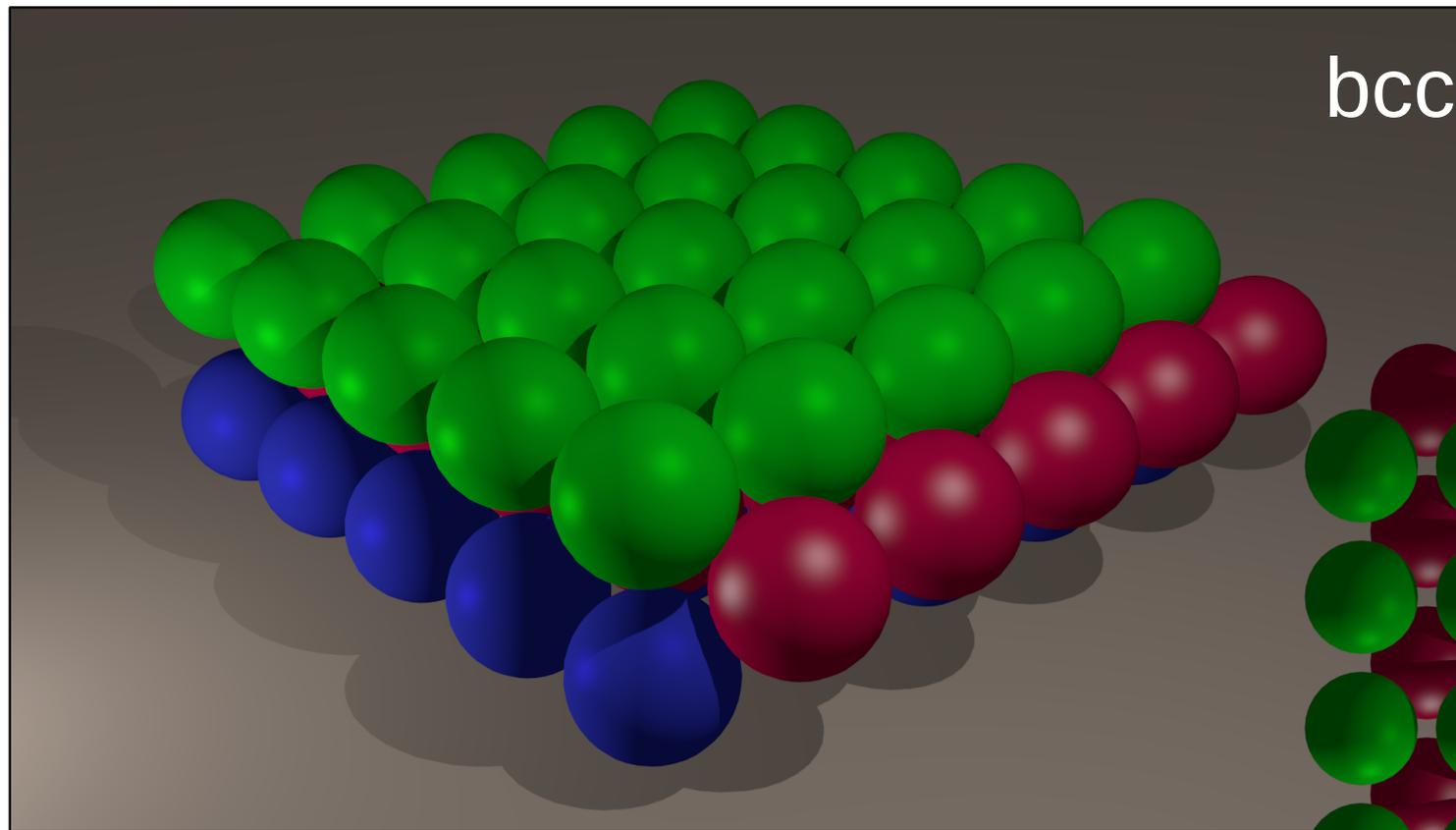
Figure 8 HR-TEM micrographs showing (a) abrupt flat crystallographic c-Si/a-Si:H/ μ c-Si:H interfaces of Si HJ solar cells, (b) an epitaxially connected *i* a-Si:H interface passivation layer in the pyramidal groove of a textured Si HJ solar cell, and (c) a rough a-Si:H/c-Si interface, also of a textured Si HJ solar cell.

Image source: J. M. D. COEY, Magnetism and Magnetic Materials, Cambridge University Press 2009 [6]

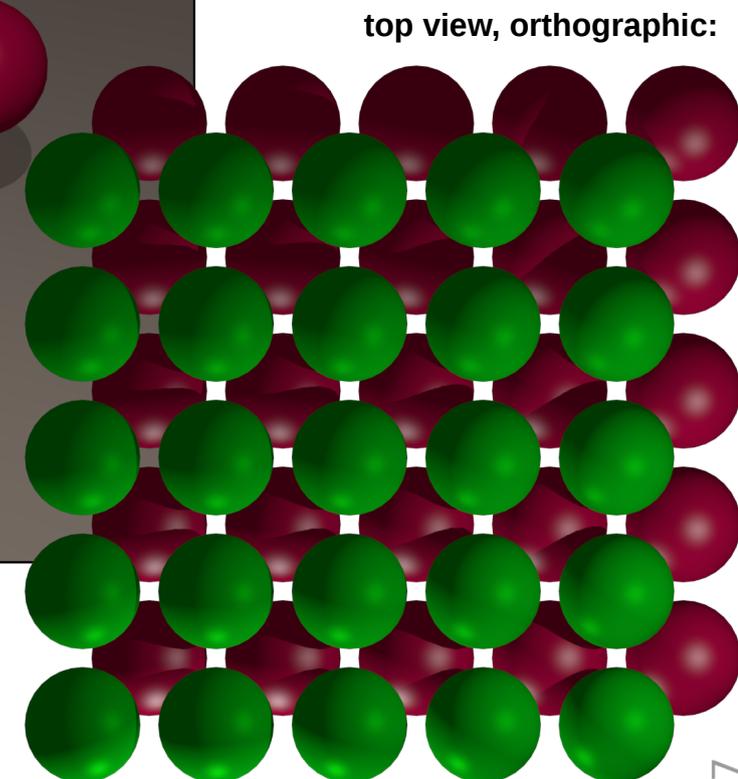
Image source: S. Olibet et al., Phys. Status Solidi A, 1–6 (2010) / DOI 10.1002/pssa.200982845

*see later in this lecture

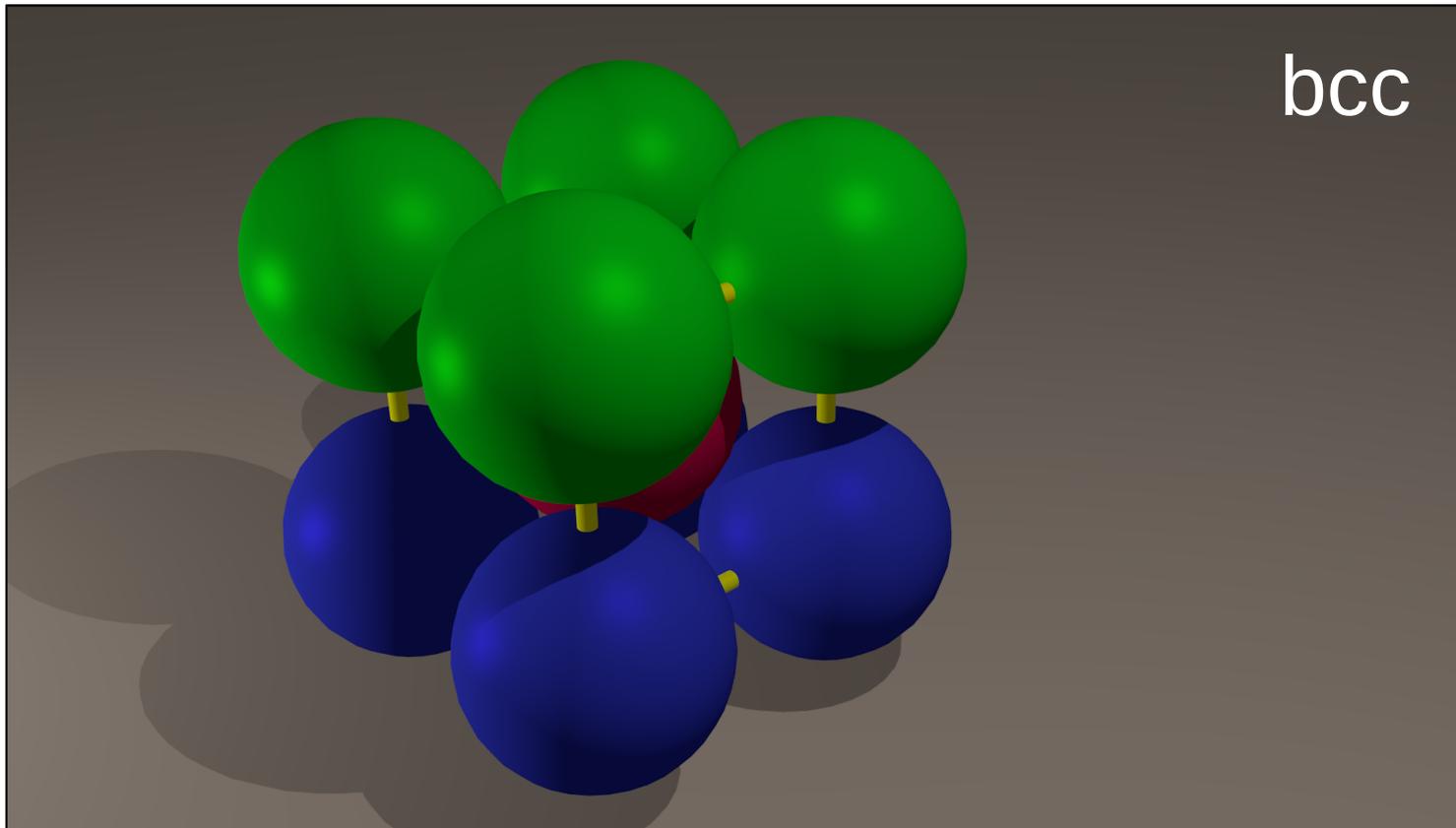
- Body centred cubic (bcc)



- The **bcc** structure is **not close-packed!**

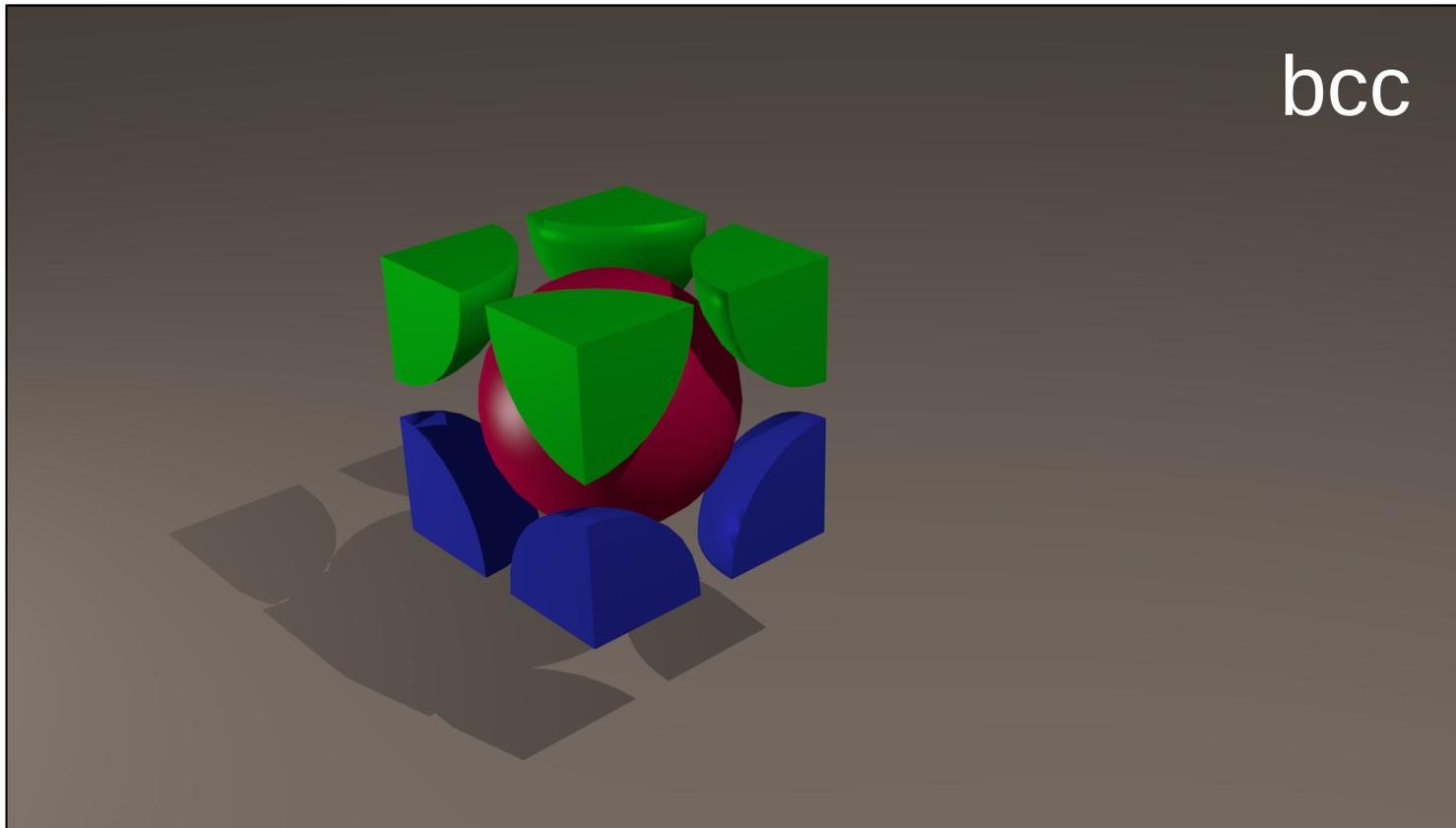


- Body centred cubic (bcc)



- The **bcc** structure is **not close-packed!**
- Unit cell contains two atoms

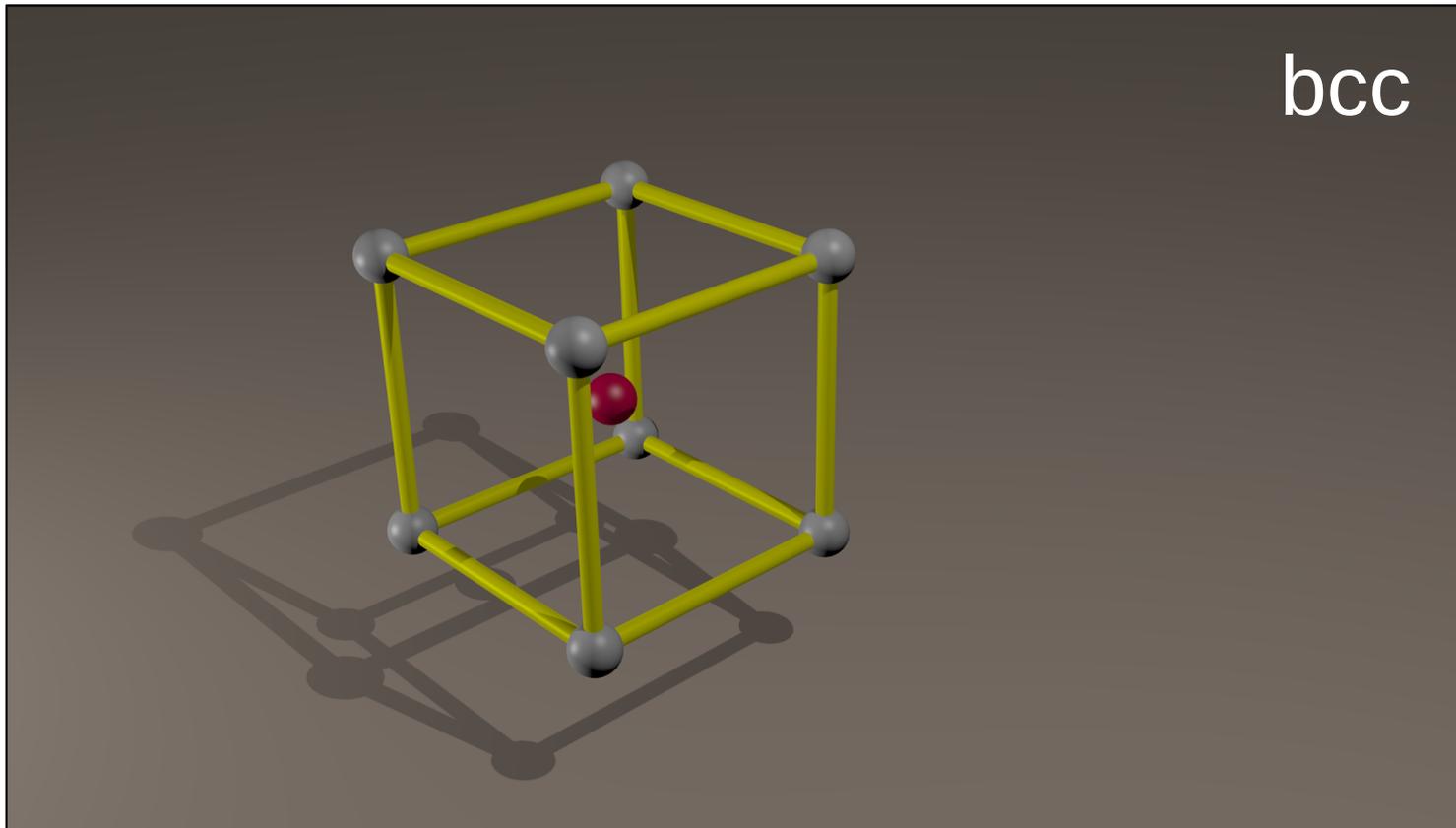
- Body centred cubic (bcc)



- The **bcc** structure is **not close-packed!**
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- Body centred cubic (bcc)

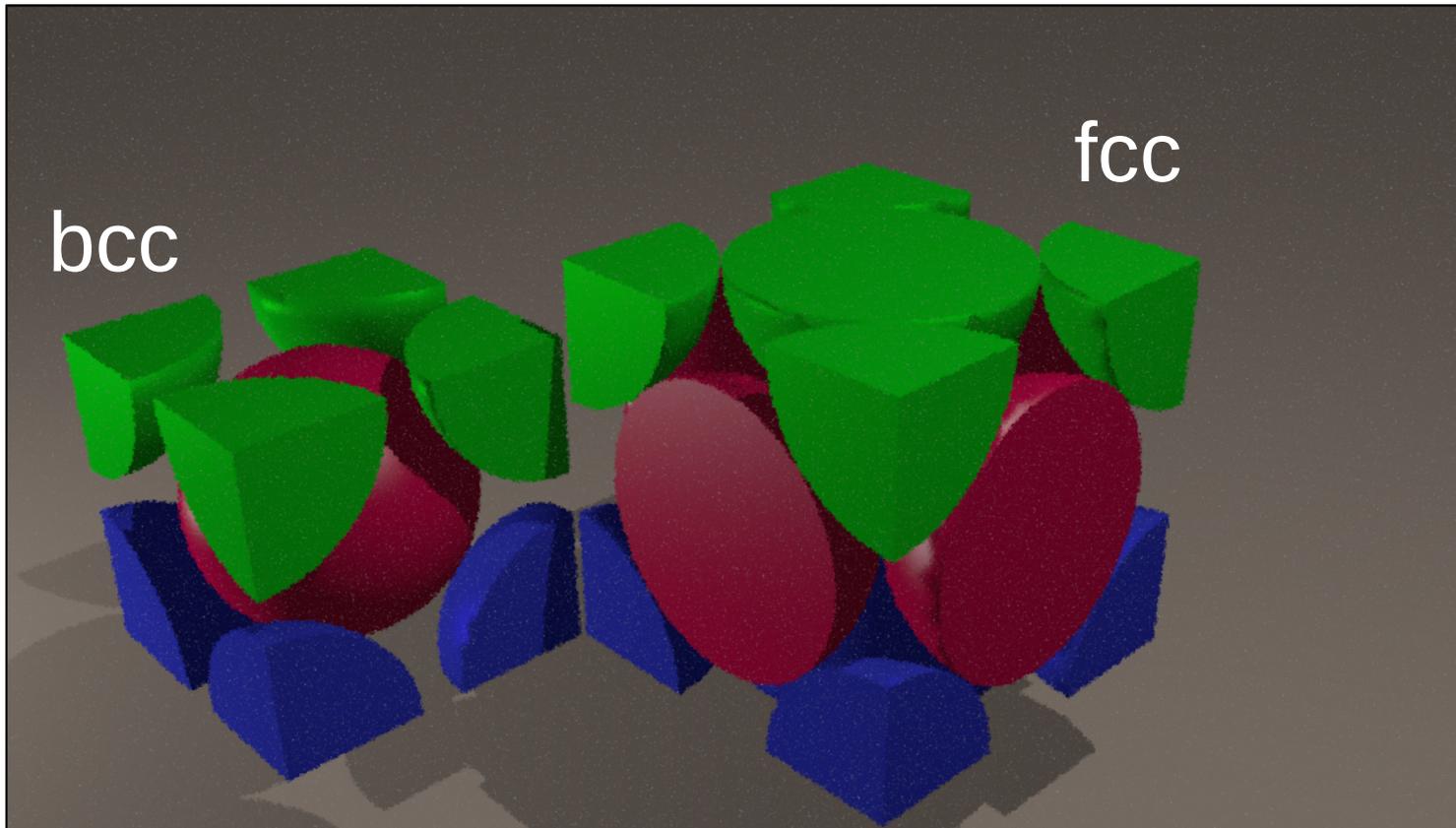
Notes that the spheres are deflated but the distances between them are unchanged



- The **bcc** structure is **not close-packed!**
- Unit cell contains two atoms

- For same size of spheres **bcc** cell is smaller than **fcc** cell

Both cubes in the image are drawn in the same scale



$$bcc : a = \frac{4r}{\sqrt{3}}$$

$$fcc : a = \frac{4r}{\sqrt{2}}$$

$$\frac{a_{fcc}}{a_{bcc}} = \frac{\sqrt{3}}{\sqrt{2}} \approx 1.23$$

- For same size of spheres **bcc** cell is smaller than **fcc** cell

| bcc structure | fcc structure | |
|-------------------------------------------------------------------------------------------------------------------------------------------|---------------------------------------------------------------------------------------------------------------------------------------------|--------------------------------------------------------------------|
| 2 atoms in a unit cell | 4 atoms in a unit cell | $bcc : a = \frac{4r}{\sqrt{3}}$ |
| Fraction of space occupied by the spheres: $f_{packing} = \frac{2(4/3 \pi r^3)}{(4r/\sqrt{3})^3} = \frac{\sqrt{3}\pi}{8} \approx 0.68$ | Fraction of space occupied by the spheres: $f_{packing} = \frac{4(4/3 \pi r^3)}{(4r/\sqrt{2})^3} = \frac{\sqrt{8}\pi}{12} \approx 0.741$ | $fcc : a = \frac{4r}{\sqrt{2}}$ |
| Coordination number*: 8 | Coordination number*: 12 | $\frac{a_{fcc}}{a_{bcc}} = \frac{\sqrt{3}}{\sqrt{2}} \approx 1.23$ |

*number of nearest neighbors

Close packing of equal spheres

- In real crystal structures the ratio between the lengths of cell edges a (i.e., ion/sphere diameter) to a spacing between successive close-packed layers always differs from the predictions of purely geometrical model described previously [2]

Table 1

| Material | h/a | Material | h/a |
|----------|--------|----------|-------|
| Cd | 0.943 | AgI | 0.815 |
| Zn | 0.928 | BeO | 0.815 |
| He | 0.8165 | CdSe | 0.815 |
| Co | 0.814 | ZnO | 0.800 |
| Mg | 0.812 | AlN | 0.800 |
| Sc | 0.797 | CdS | 0.810 |

For ideal close-packed structures the (stacking faults can be present) h/a ratio is approx. 0.8165

- In fabrication of thin films one can obtain, using molecular beam epitaxy (MBE), structures which are ideal over hundreds of interatomic distances

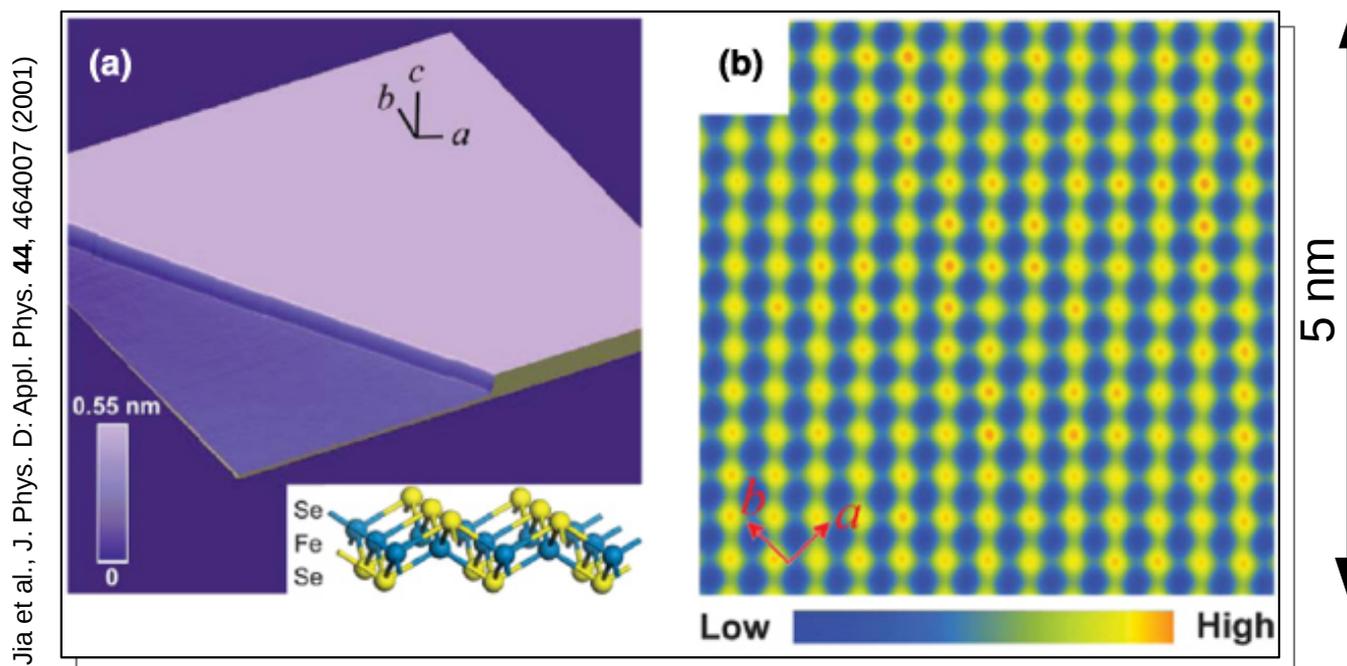


Figure 10. STM characterization of the as-grown FeSe films. (a) Topographic image (2.5 V, 0.1 nA, $200 \times 200 \text{ nm}^2$) of a FeSe film (about 30 unit cells thick). The step height is 5.5 \AA . Inset shows the crystal structure. (b) Atomic resolution STM topography (10 mV, 0.1 nA, $5 \times 5 \text{ nm}^2$) of the FeSe film. The bright spots correspond to the Se atoms in the top layer. a and b correspond to either of the Fe–Fe bond directions. The same convention is used for a - and b -axes throughout the paper. (c) Temperature dependence of differential conductance spectra (setpoint: 10 mV, 0.1 nA). (d) Schematic of the unfolded Brillouin zone and the Fermi surface (green ellipse). The nodal lines for $\cos k_x \cos k_y$ and $(\cos k_x + \cos k_y)$ gap functions are indicated by black and red dashed lines, respectively. The size of all pockets is exaggerated for clarity. The black arrow shows the direction of nesting (from [73]).

- “The STM topographic images (figures 10(a) and (b)) revealed atomically flat and defect-free Se-terminated (0 0 1) surfaces with large terraces.” [3]
- layer-by-layer growth on SiC(0 0 0 1) substrate

Metals – crystal structures

Metallic bonds – bonds between atoms of low electronegativity that contain **low number of electrons in outer shell** [12]:

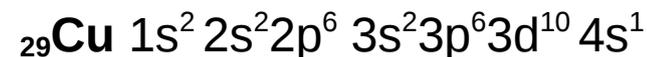
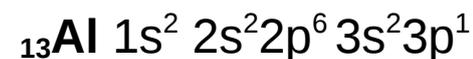
- the metallic bond is not directional
- the bond is weaker than covalent bond
- metal may be viewed as a set of positive ions immersed in electronic cloud

Characteristic features of metal structures [12]:

- they form high symmetry structures, regular or hexagonal
- due to the dense packing or packing close to dense they have high density**
- high coordination numbers (about 12) and relatively high bonding energies result in relatively high melting temperatures
- the metals are malleable***

Metals that are most important for electronics:

- Al, Cu, Au
- Fe, Co, Ni



* the power of an atom to attract electrons to itself [13]. The higher the electronegativity the higher the tendency to attract electrons.

**in fact in every period of the Mendeleev's periodic table the metals are the elements with the highest density (bor is a half-metal) [14]

***able to be hammered or pressed into shape without breaking or cracking (google dictionary)

Derivative structures

- More complicated structures can be obtained from simpler ones by operations removing some symmetries present in the initial structure – the idea introduced by M. Buerger [4]
- One dimensional example*:

Parent structure



Substitution derivative



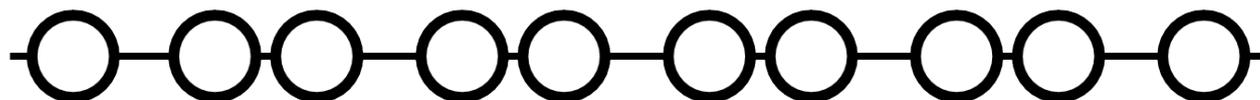
Omission derivative



Addition derivative



Distortion derivative



The translational periodicity of the derivatives is higher than that of the parent



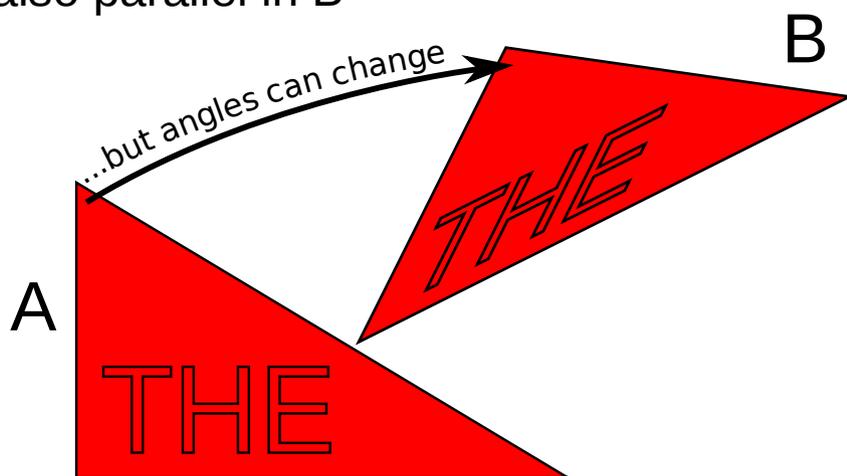
superstructures

*taken from crystallography lectures by Prof. Bernhardt Wensch – MIT, USA, 2005 [4]

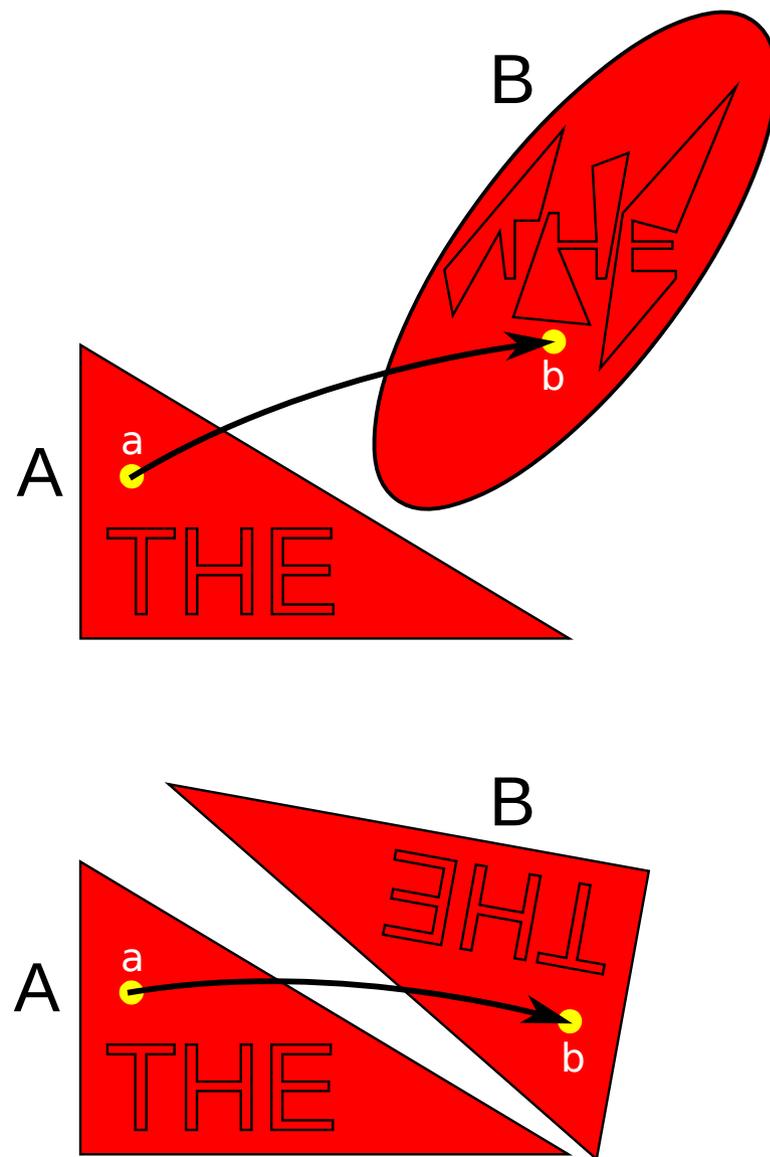
Some facts about crystal symmetries*

- **A mapping** of a set A into a set B is a relation such that for each element $a \in A$ there is a *unique* element $b \in B$ which is assigned to a . The element b is called the image of a .

- **An affine mapping** – lines parallel in A are also parallel in B



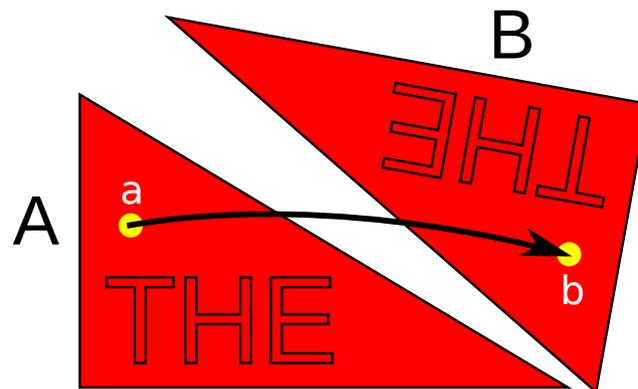
- **An isometry** - affine mapping that leaves all distances between points (and thus angles between lines) invariant



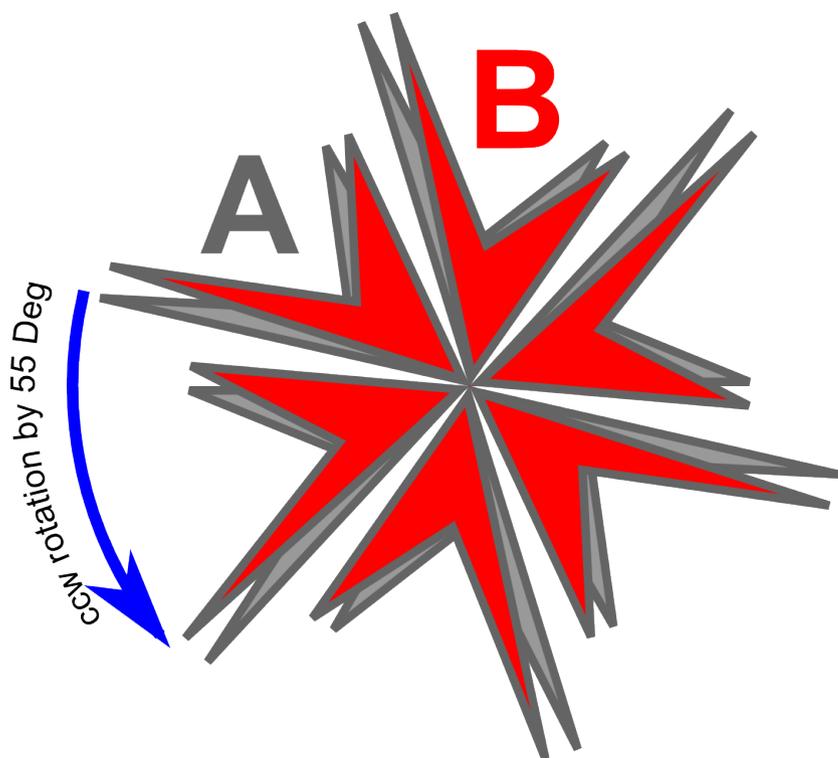
*taken mostly from H. Wondratschek, *Matrices, Mappings and Crystallographic Symmetry*, 2002 [5]

Some facts about crystal symmetries*

- **An isometry** - affine mapping that leaves all distances between points (and thus angles between lines) invariant



- **Symmetry operations** of an object – “isometries which map the object onto itself such that the mapped object cannot be distinguished from the object in the original state” [5]

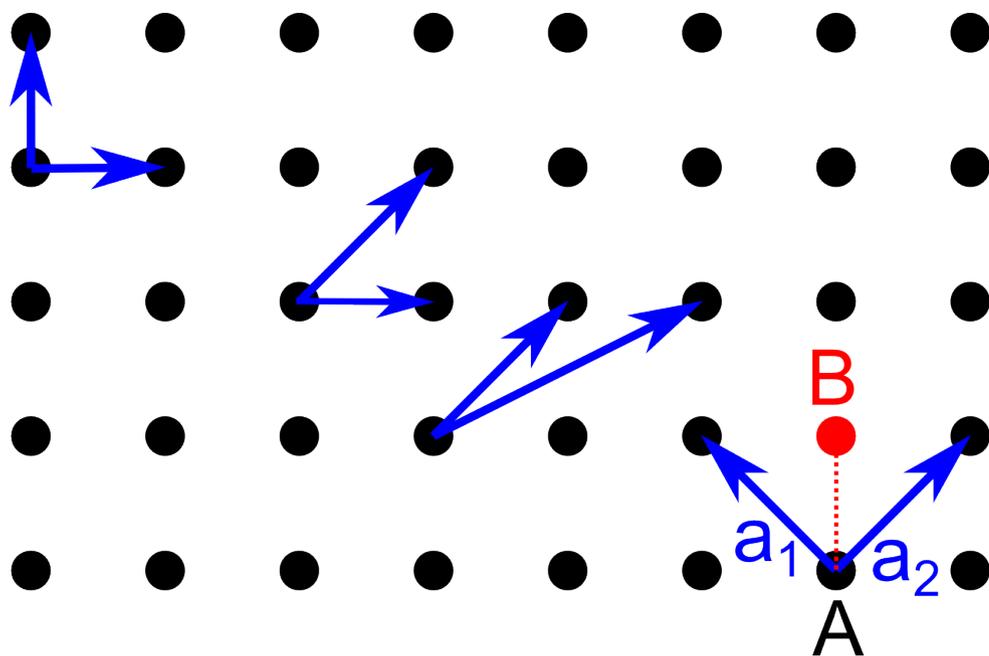


Rotation by 60 Deg (cw or ccw) brings A into itself (**B=A**)

*taken mostly from H. Wondratschek, *Matrices, Mappings and Crystallographic Symmetry*, 2002 [5]

Primitive (lattice) basis*

- **crystal patterns** - idealization of real crystals in the physical space by 3-dimensional periodic sets of points representing, e. g., the centres of the atoms of the crystal [5].
- The set of all translation vectors belonging to symmetry translations of a crystal pattern is called the **vector lattice** of the crystal pattern (and of the real crystal). Its vectors are called **lattice vectors**.
- A basis of three (two in 3-dimensional patterns) linearly independent lattice vectors is called a *lattice basis*. If all lattice vectors are integer linear combinations of the basis vectors, then the basis is called a **primitive basis**.

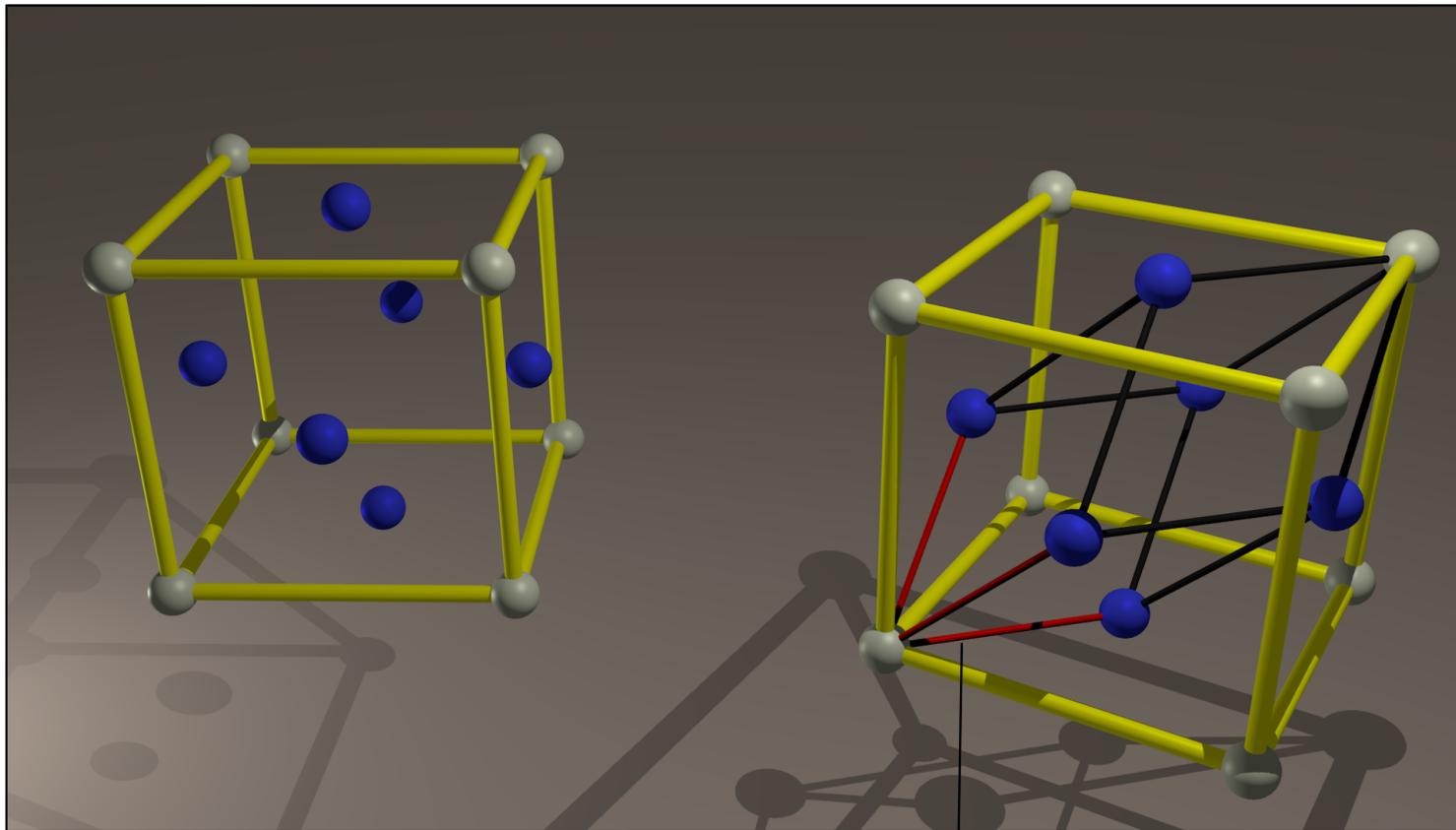


For each lattice there exists an infinite number of primitive bases

Basis (a_1, a_2) is not a primitive basis: one can not reach **B** from **A** by translation with integer multiplies of a_1 and a_2

Primitive (lattice) basis*

- Example of primitive basis for 3-dimensional lattice – **fcc** structure

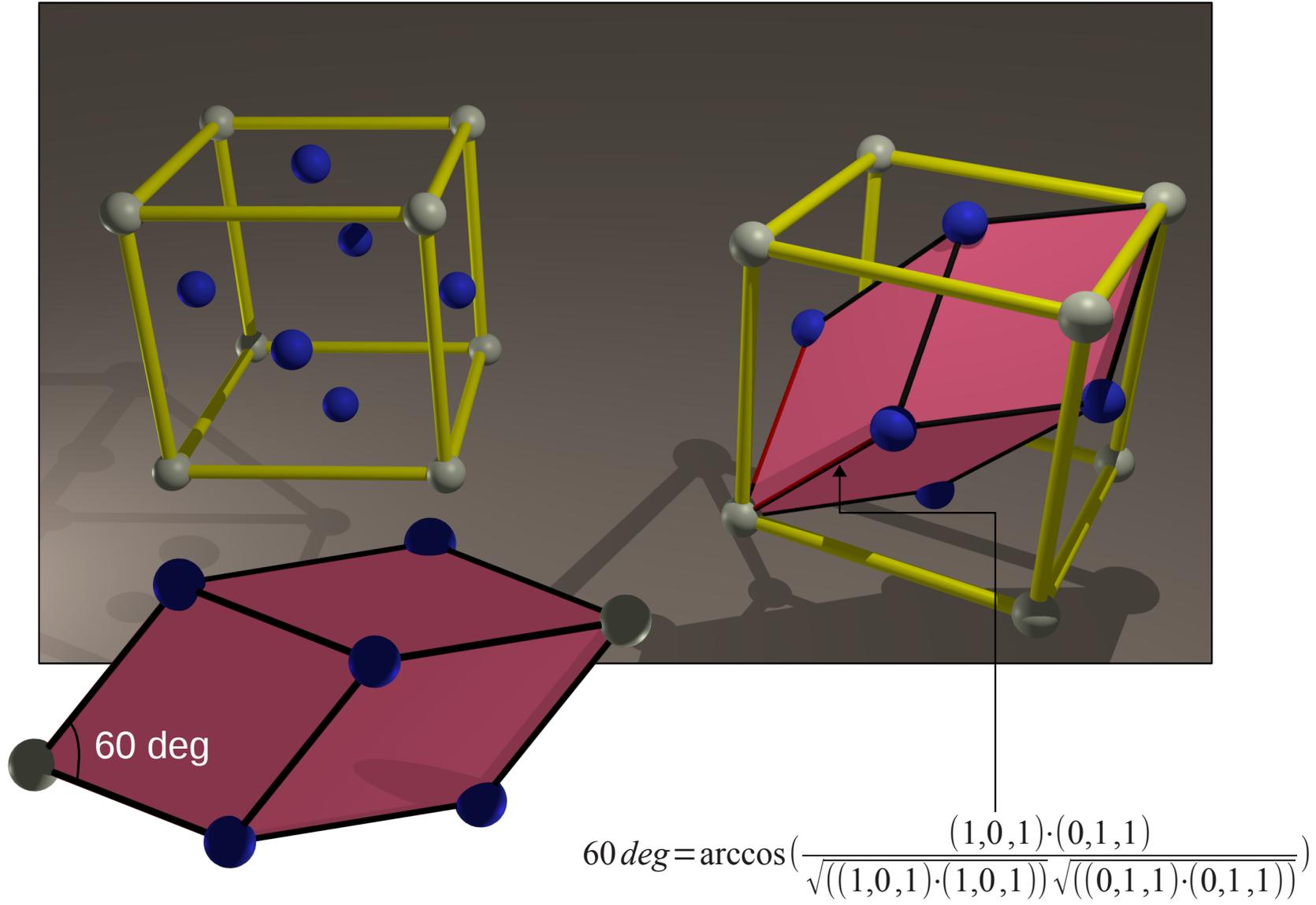


primitive basis (red rods)

*taken mostly from H. Wondratschek, *Matrices, Mappings and Crystallographic Symmetry*, 2002 [5]

Primitive (lattice) basis*

- Example of primitive basis for 3-dimensional lattice – **fcc** structure



*taken mostly from H. Wondratschek, *Matrices, Mappings and Crystallographic Symmetry*, 2002 [5]

Bravais lattice:

- an **infinite** arrangements of points in space (“A **lattice** is an infinite array of points in space, in which each point has identical surroundings to all others” [8])
- the lattice can be generated from one point using the translation operators:

$$t_{n_1, n_2, n_3} = n_1 \vec{a} + n_2 \vec{b} + n_3 \vec{c}, \quad n_1, n_2, n_3 - \text{integers}$$

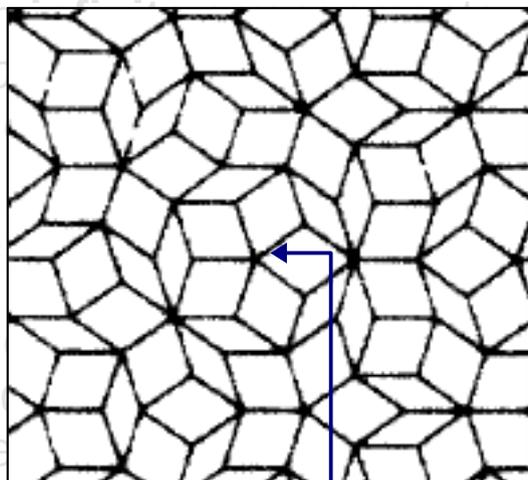
- the lattice points are given by the end points of the vectors \mathbf{t} (vectors $\mathbf{a}, \mathbf{b}, \mathbf{c}$ are linearly independent*)
- vectors $\mathbf{a}, \mathbf{b}, \mathbf{c}$ have the same origin; no matter which lattice point is chosen as an origin, the array always looks the same when viewed from it [8]
- forming the parallelepiped from $\mathbf{a}, \mathbf{b}, \mathbf{c}$ vectors one obtains a **unit cell**
- unit cells fill the entire space
- The only rotations that are symmetry elements of crystals are n-fold rotations with $n=1, 2, 3, 4, 6$ (proper or improper) – it is not possible to fill the space with uniformly shaped unit cells of other symmetries (without gaps and overlapping)
- One can fill the space with cells of other rotational symmetries using two or more kinds of cells – **quasi-crystals** [8,9]

*it is not possible to get zero vector summing integer multiples of $\mathbf{a}, \mathbf{b}, \mathbf{c}$

Bravais lattice

Bravais lattice:

- an infinite array of points in space
- each point has identical surroundings
- the array is invariant under translation from one point to another
- the array is invariant under rotation by the end point of a vector \mathbf{a} or \mathbf{b} or \mathbf{c}
- the array is invariant under reflection through the origin; no matter what the origin, the array always looks the same when viewed from the same direction
- forming the parallelepiped from $\mathbf{a}, \mathbf{b}, \mathbf{c}$ vectors
- unit cells fill the entire space
- The only rotations that are symmetry elements are those for which $n=1, 2, 3, 4, 6$ (proper or improper) – it is not possible to fill space with identical shaped unit cells of other symmetries (with



local 5-fold symmetry

- One can fill the space with cells of other rotational symmetries using two or more kinds of cells – **quasi-crystals** [8,9]

Penrose tiling example – 5 fold and 10-fold symmetry

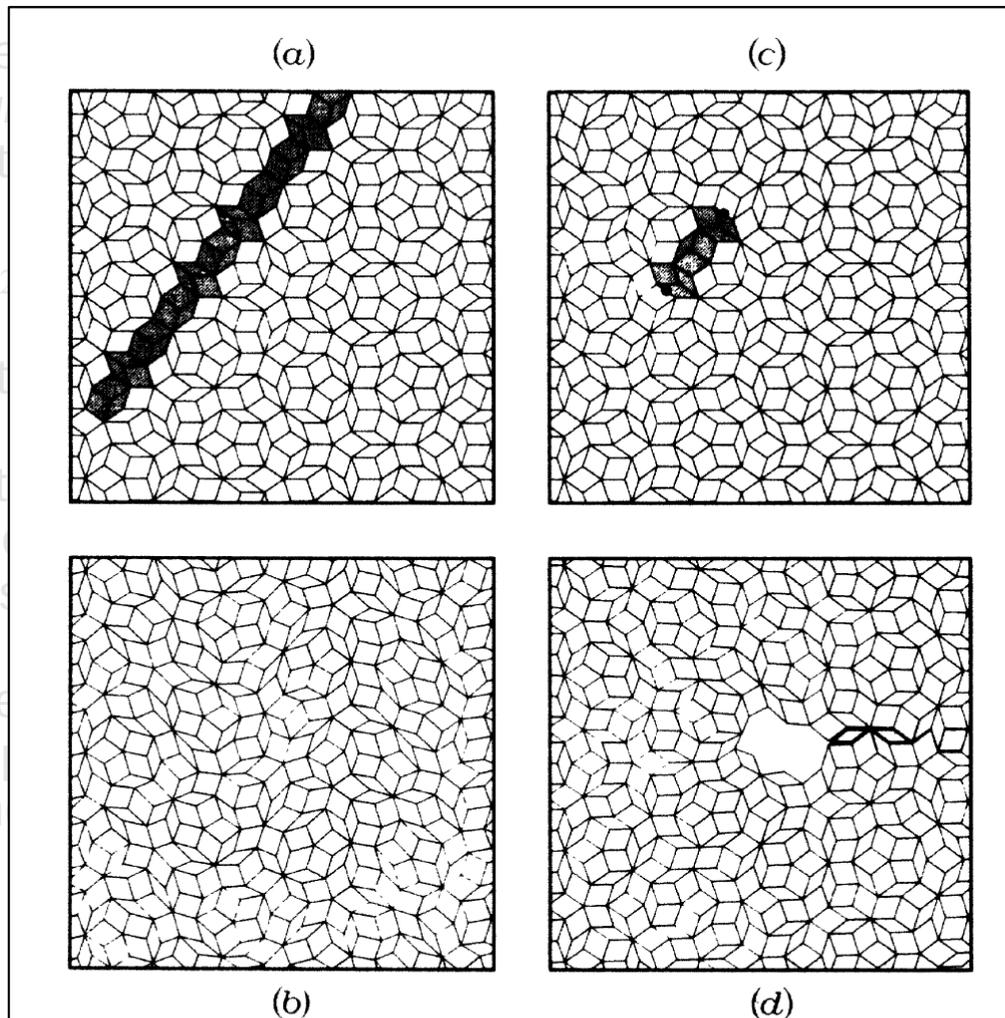
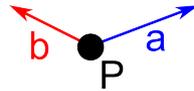


FIG. 3. Penrose tilings. (a) A portion of a perfect Penrose tiling. The shaded unit cells compose a segment of a “worm.” (b) A distortion of the tiling of (a) corresponding to variations in the phonon degree of freedom u_i . The unit-cell shapes are distorted, but their arrangement is the same as in (a). (c) A tiling

Image source : J. E. S. Socolar, T. C. Lubensky, P. J. Steinhardt, Phys. Rev. B **34**, 3345 (1986) [10]

*it is not possible to get zero vector summing integer multiples of $\mathbf{a}, \mathbf{b}, \mathbf{c}$

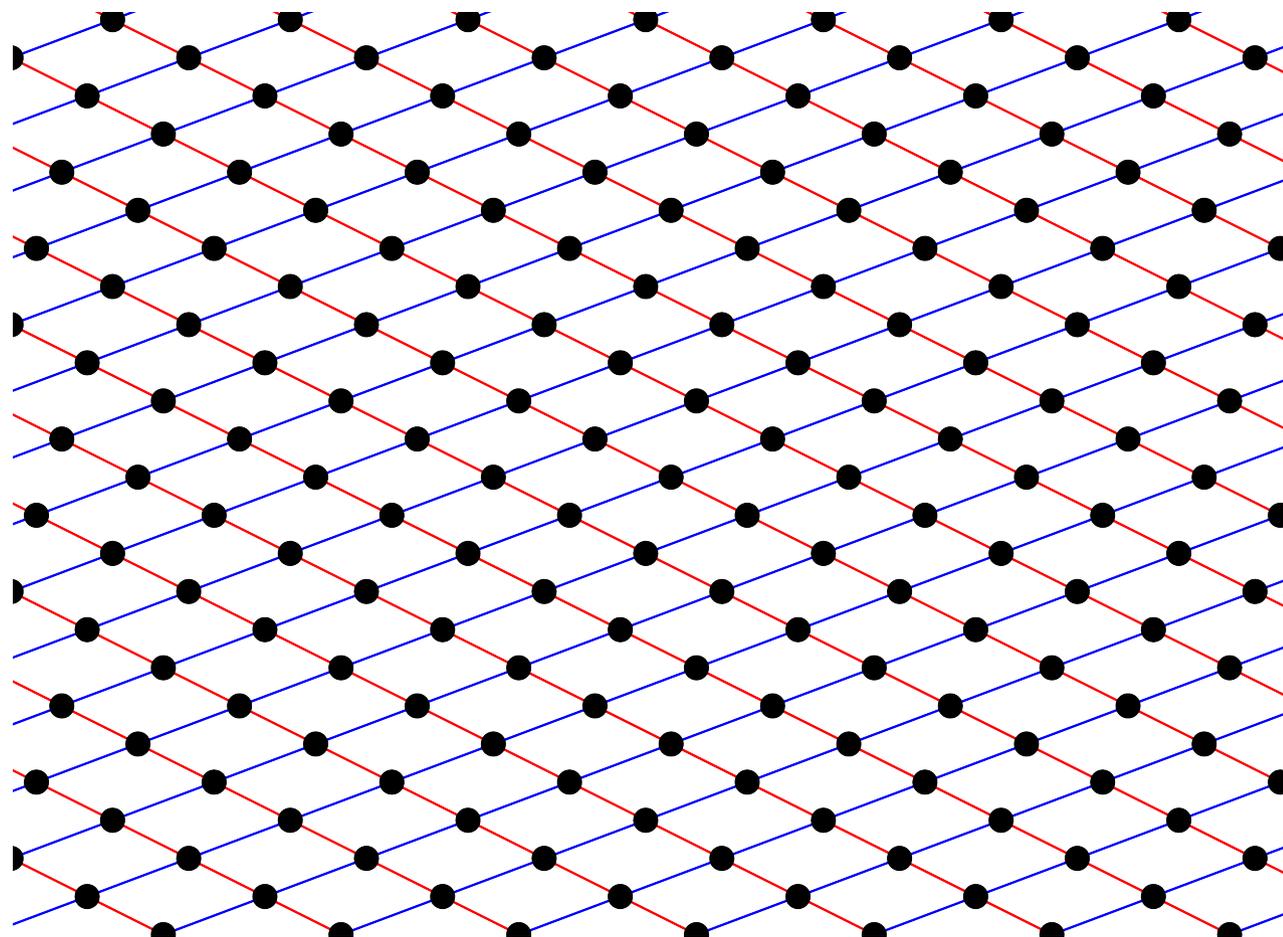
- There are 14 Bravais lattices in 3D
- There are 5 Bravais type lattices in 2D: oblique, rectangular, centered rectangular, hexagonal and square



We start with an arbitrary point P in a plane and arbitrary vectors \mathbf{a} and \mathbf{b}

Bravais lattice - 2D examples

- There are 14 Bravais lattices in 3D
- There are 5 Bravais type lattices in 2D: oblique, rectangular, centered rectangular, hexagonal and square



For arbitrary base vectors identity** may be the only symmetry of the structure

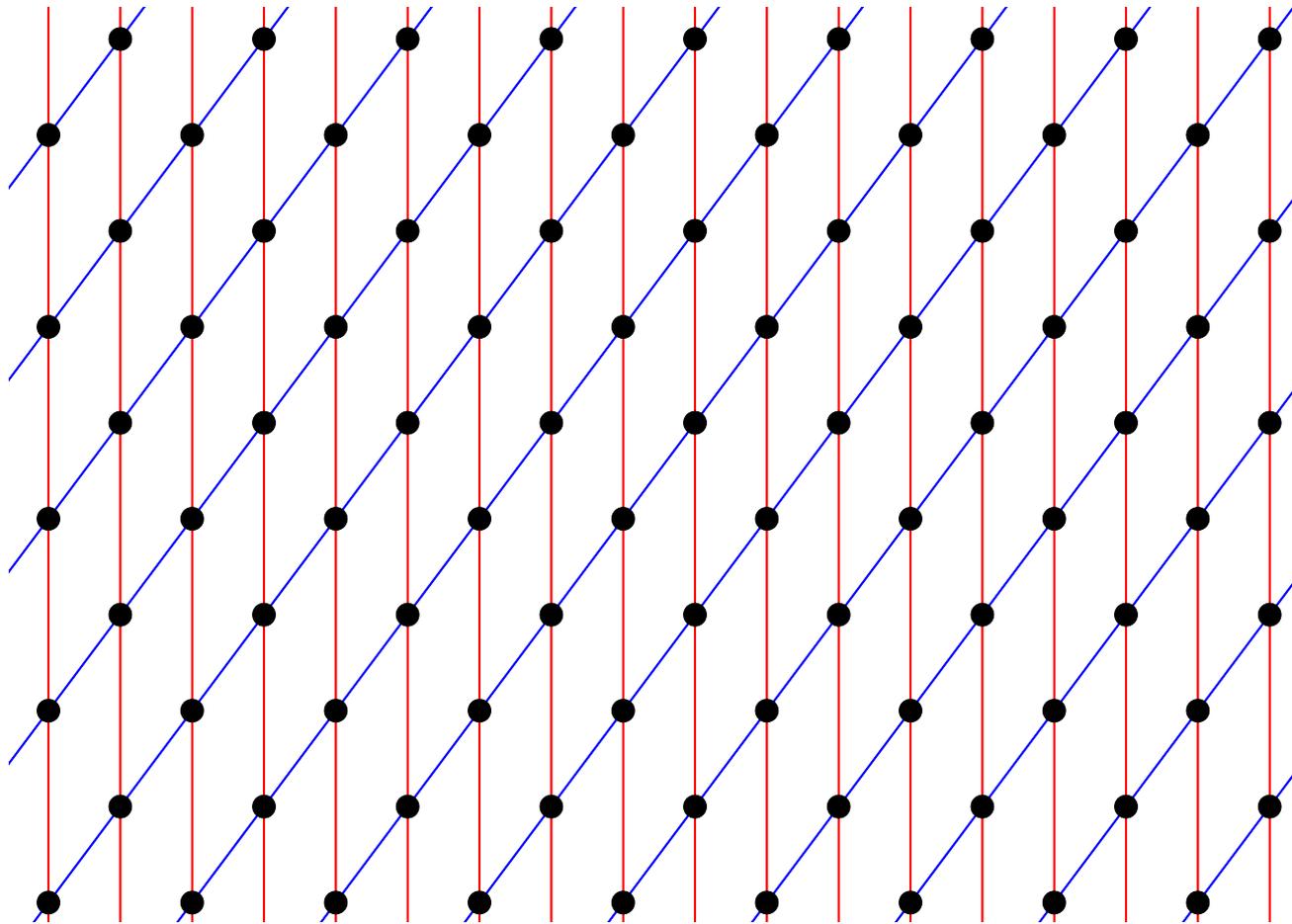
We start with an arbitrary point P in a plane and arbitrary vectors **a** and **b**
 ...and “orbit“* with translations $t_{n_1, n_2, n_3} = n_1 \vec{a} + n_2 \vec{b} + n_3 \vec{c}$

*The infinite set of symmetry-related points is known as a **crystallographic orbit** [8]

**denoted usually with 1 or E

Bravais lattice - 2D examples

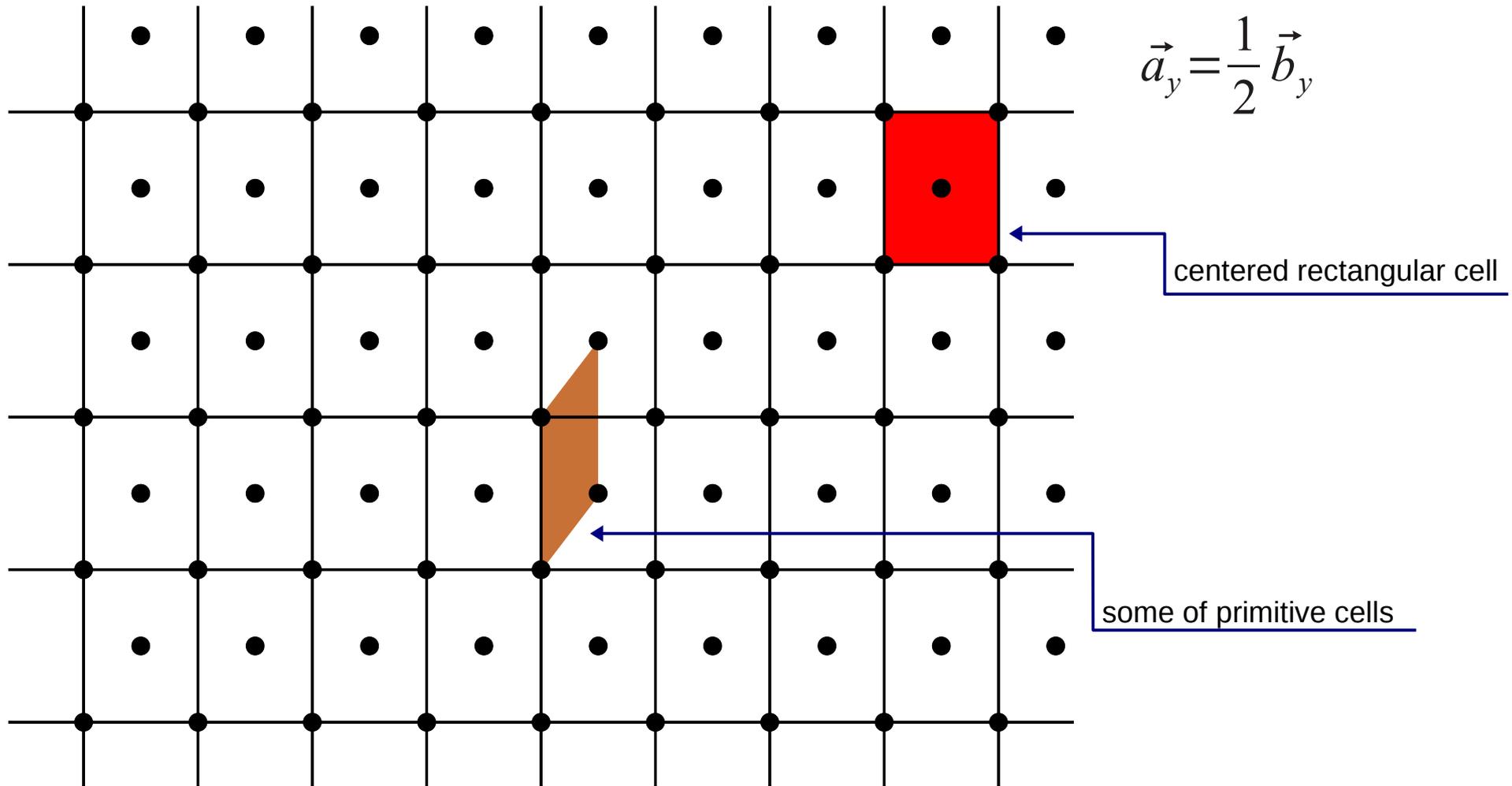
- There are 14 Bravais lattices in 3D
- There are 5 Bravais type lattices in 2D: oblique, rectangular, centered rectangular, hexagonal and square



$$\vec{a}_y = \frac{1}{2} \vec{b}_y$$

For some special relations between vectors **a** and **b** it is customary to use unit cell that is different from the primitive cell [11]

- There are 14 Bravais lattices in 3D
- There are 5 Bravais type lattices in 2D: oblique, rectangular, centered rectangular, hexagonal and square

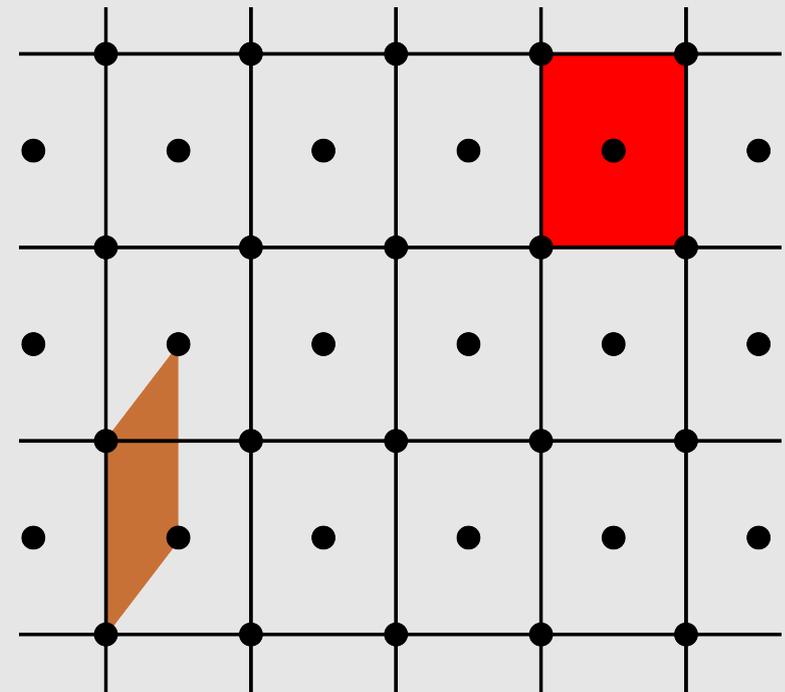


For some special relations between vectors **a** and **b** it is customary to use unit cell that is different from the primitive cell [11]

The choice of the Bravais unit cell for a given basis is made using the following conditions* [11]:

1. The symmetry of the unit cell should correspond to the symmetry of the crystal (lattice). Edges of the cell should be lattice vectors
2. The unit cell should contain maximum possible number of right angles (between edges) or equal angles and equal edges
3. The unit cell should have minimum volume

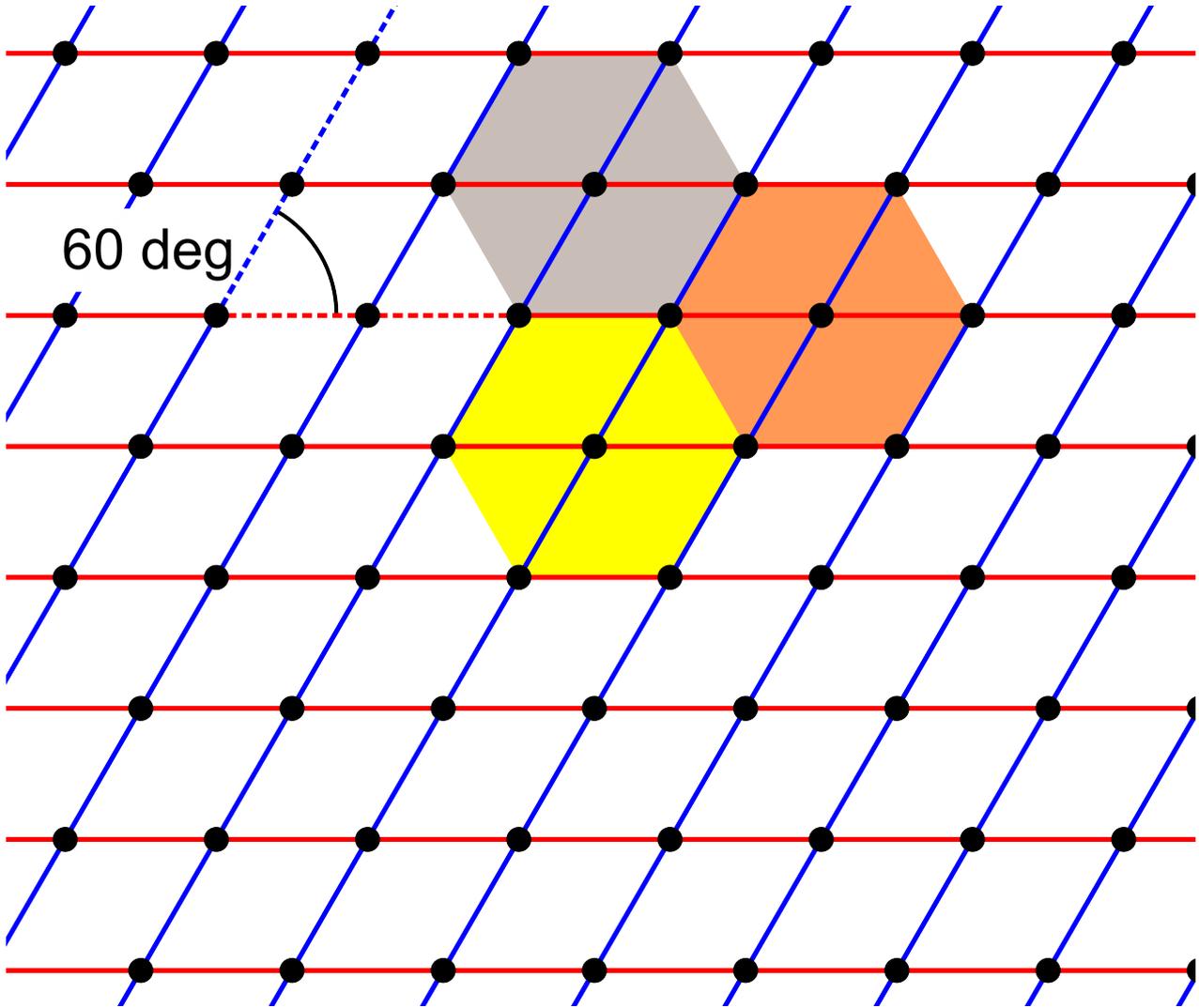
- The primitive cell to the right has minimum volume
- The cell has no right angles
- The cell (*its symmetry operations*) has no planes of symmetry present in the symmetry of the lattice



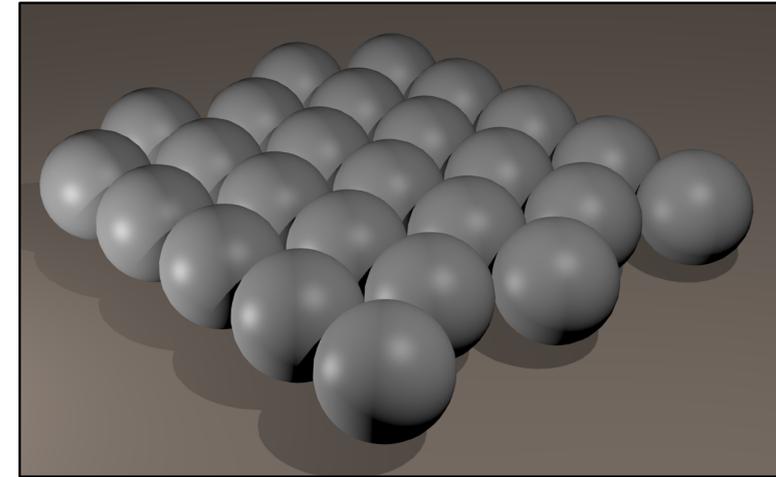
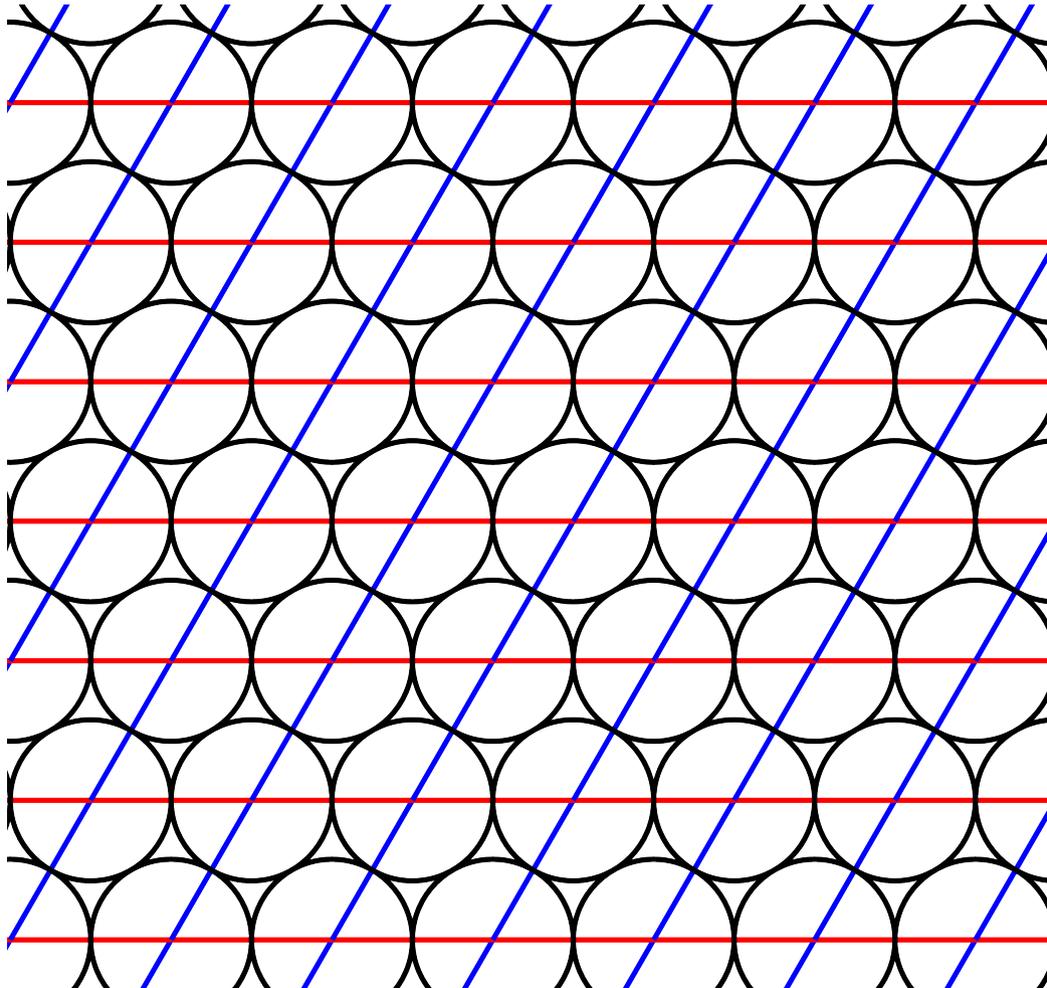
*the conditions should be fulfilled consecutively: the first condition is more important than the second

Bravais lattice - 2D examples

For special relation between basis vectors **a** and **b** one (equal length, in-between angle of 60 deg) one obtains lattice with 6-fold rotational symmetry consistent with close packing of spheres in a plane (*hexagonal lattice*)



For special relation between basis vectors **a** and **b** one (equal length, in-between angle of 60 deg) one obtains lattice with 6-fold rotational symmetry consistent with close packing of spheres in a plane (**hexagonal lattice**)



primitive base centered body centered face centered

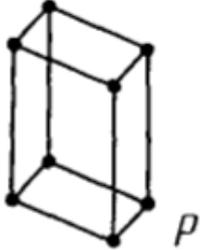
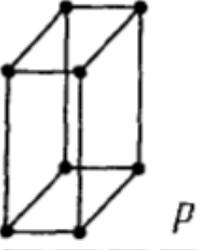
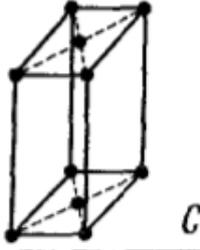
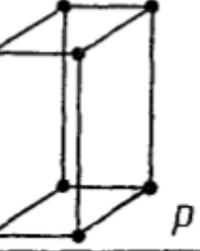
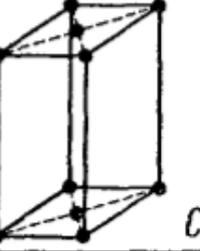
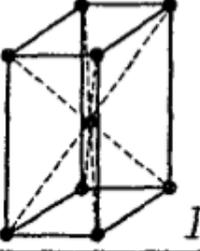
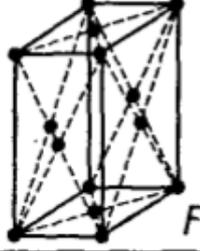
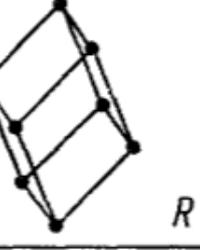
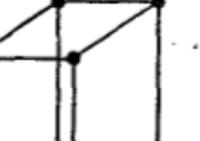
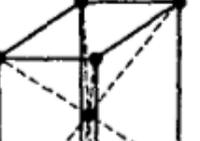
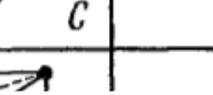
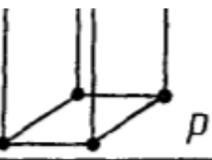
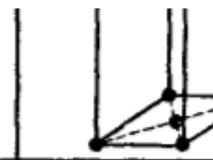
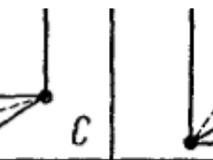
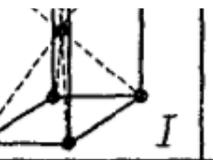
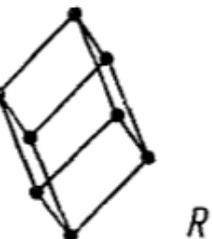
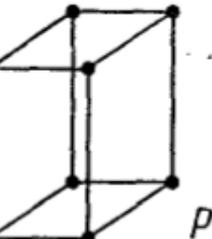
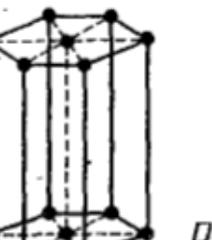
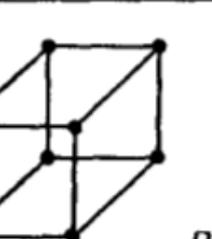
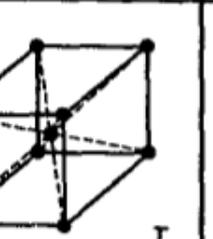
| Сингония | Тип решетки | | | |
|------------------------------------------------------|----------------------------------------------------------------------------------------------|----------------------------------------------------------------------------------------------|-----------------------------------------------------------------------------------------------|-----------------------------------------------------------------------------------------------|
| | Примитивная | Базоцентрированная | Объемно-центрированная | Гранецентрированная |
| Триклинная triclinic |  <i>P</i> | | | |
| Моноклинная monoclinic |  <i>P</i> |  <i>C</i> | | |
| Ромбическая orthorombic |  <i>P</i> |  <i>C</i> |  <i>I</i> |  <i>F</i> |
| Тригональная (ромбоэдрическая) trigonal |  <i>R</i> | | | |
| Тетрагональная |  | |  | |

Image source : М.П. Шаскольская, Кристаллография, Москва, "Высшая Школа", 1984, Fig.92

Image source : М.П. Шаскольская, Кристаллография, Москва, "Высшая Школа", 1984, Fig.92

Bravais lattices in 3D

| |  primitive |  base centered |  body centered |  face centered |
|--------------------------------------------------------------------|------------------------------------------------------------------------------------------|----------------------------------------------------------------------------------------------|-----------------------------------------------------------------------------------------------|------------------------------------------------------------------------------------------------|
| <i>Ромбическая</i> orthorombic |  P |  C |  I |  F |
| <i>Тригональная (ромбо- эдрическая)</i> trigonal |  R | | | |
| <i>Тетраго- нальная</i> tetragonal |  P | |  I | |
| <i>Гексаго- нальная</i> hexagonal |  P | | | |
| <i>Кубическая</i> cubic |  P | |  I |  F |

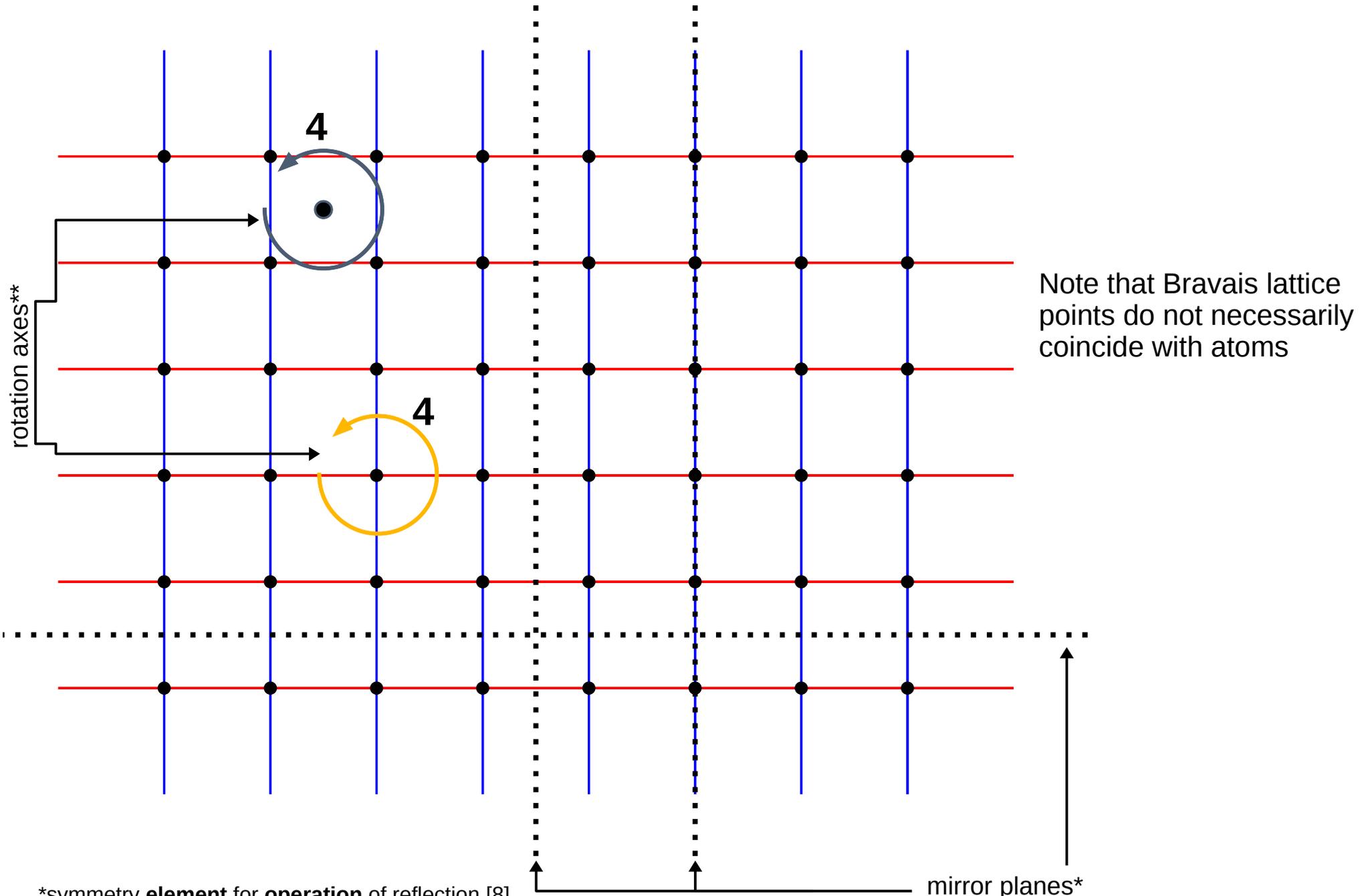
92.
14 решеток Бравэ

- **Point group** – all the symmetry operations that leave one point fixed; this point is called the origin [8].
- “the point group which has the symmetry of the lattice is called the **holohedral point group**, and as such it possesses the largest number of symmetry operations” [8]
- There are several methods to develop (find) point groups; one of them is to:
 1. start with five point groups defined by rotations 1, 2, 3, 4 ,and 6*
 2. add 2-fold rotations perpendicular to these axes
 3. add reflections perpendicular to, or containing the cyclic axis
 4. substitute improper** for proper rotation
- One arrives then at 32 **crystallographic point groups** (allowed in normal crystals – filling all space)
- Point groups characterize macroscopic crystals
- *“most of the macroscopic symmetry aspects of the physical properties of solids are related to the point group, as given by the so-called Neumann principle” [8]*

*in *international notation* the *n* symbol denotes symmetry axis (*cyclic axis of symmetry*) with rotation by $2\pi/n$ radians

**rotation followed with inversion; symbol \bar{n}

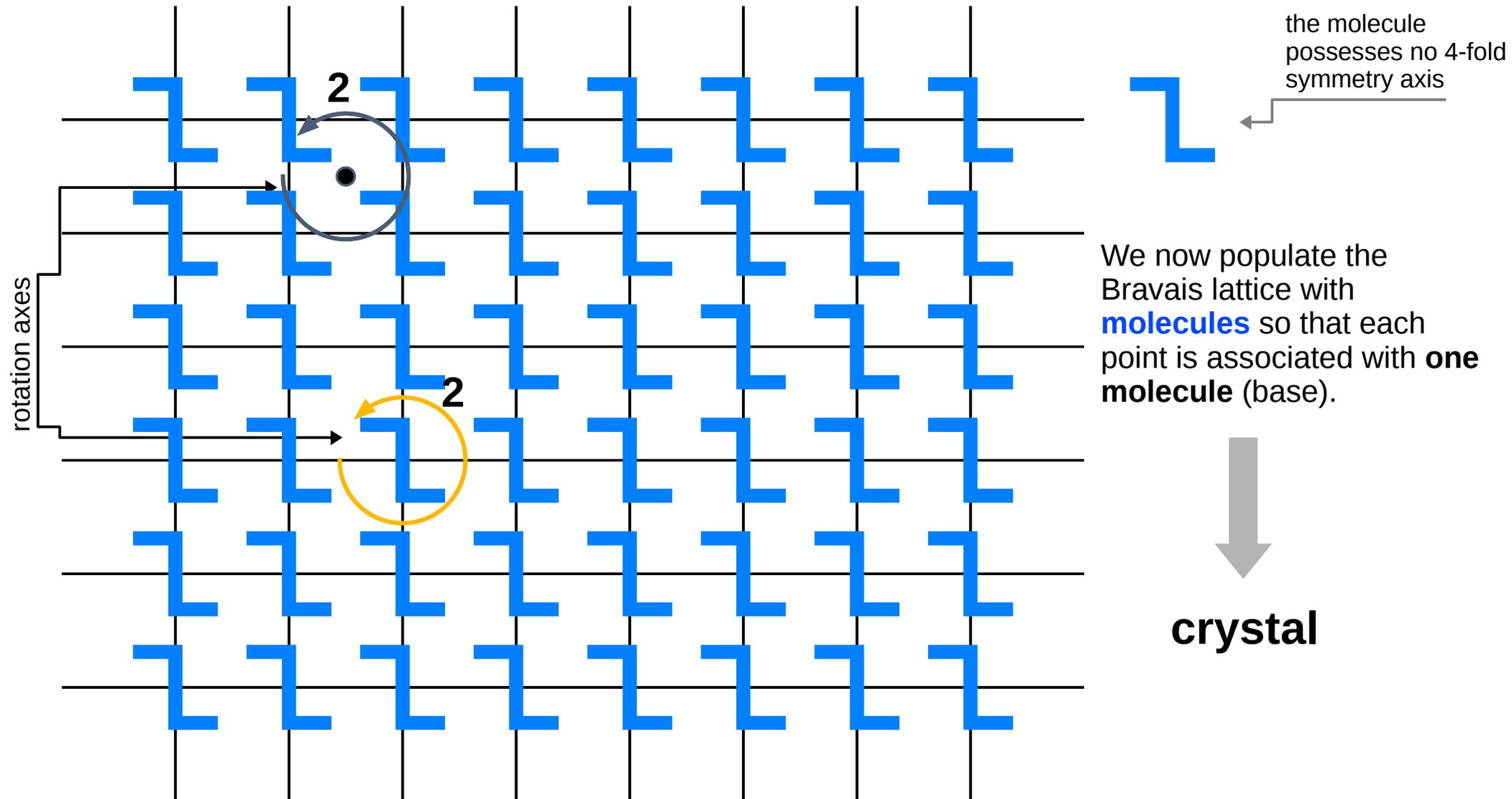
Consider some 2D Bravais lattice*** (rectangular, equal length of basis vectors)



*symmetry element for operation of reflection [8]

it is customary to talk about counter-clockwise rotations, * 2D lattice is called a net

Consider some 2D Bravais lattice (rectangular, equal length of basis vectors)



The new structure (crystal) has **no mirror planes** and only 2-fold axes of rotation (previously 4-fold). **The translational symmetry is preserved.**

- **Space group** - the set of geometrical symmetry operations that take a three-dimensional periodic object (a crystal) into itself [8]
- The operations belonging to the space group must form a **group** in the mathematical sense.
- Space groups are obtained combining translational symmetry (Bravais lattice) with point symmetry (point groups) together with two additional symmetry operations:
 - glide reflection
 - screw rotation
- Both these operations may involve translation τ smaller than a primitive lattice translation
- Simple combination of 32 point groups with 14 Bravais lattice gives 73 out of 230 space groups (so called *symmorphic groups***)

Group [16]:

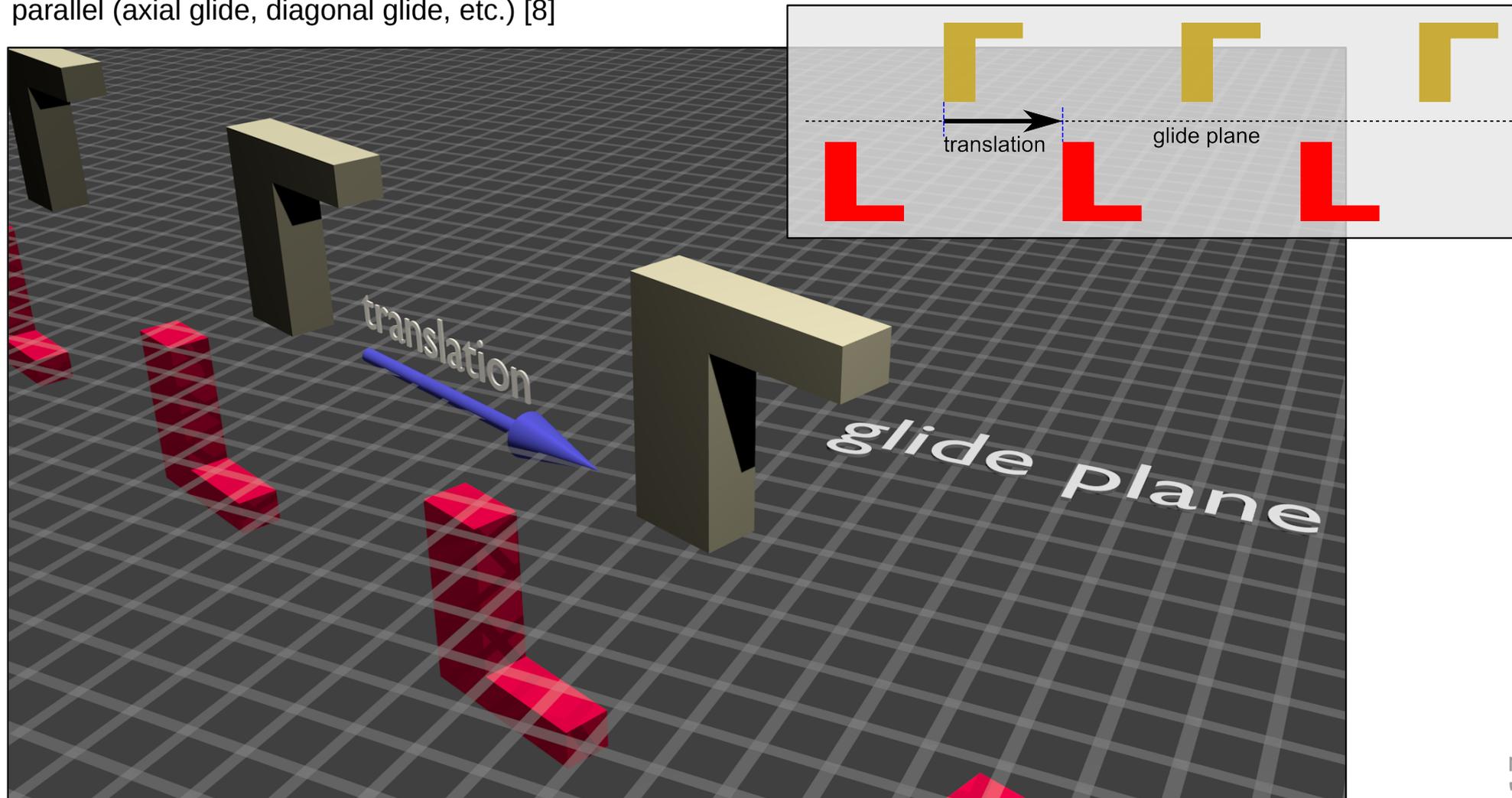
- the product of any two elements in the group must be an element of the group (square of each of them too)
- one element of the group must commute with all others and leave them unchanged (identity element)
- the associative law of multiplication holds:
 $A(BC)=(AB)C$
- every element must have a reciprocal within the group

*taken largely from G. Burns and A.M. Glazer, *Space Groups for Solid State Scientists* [8]

**screw and glide operations can be symmetry operations but they can be obtained by combining other operations

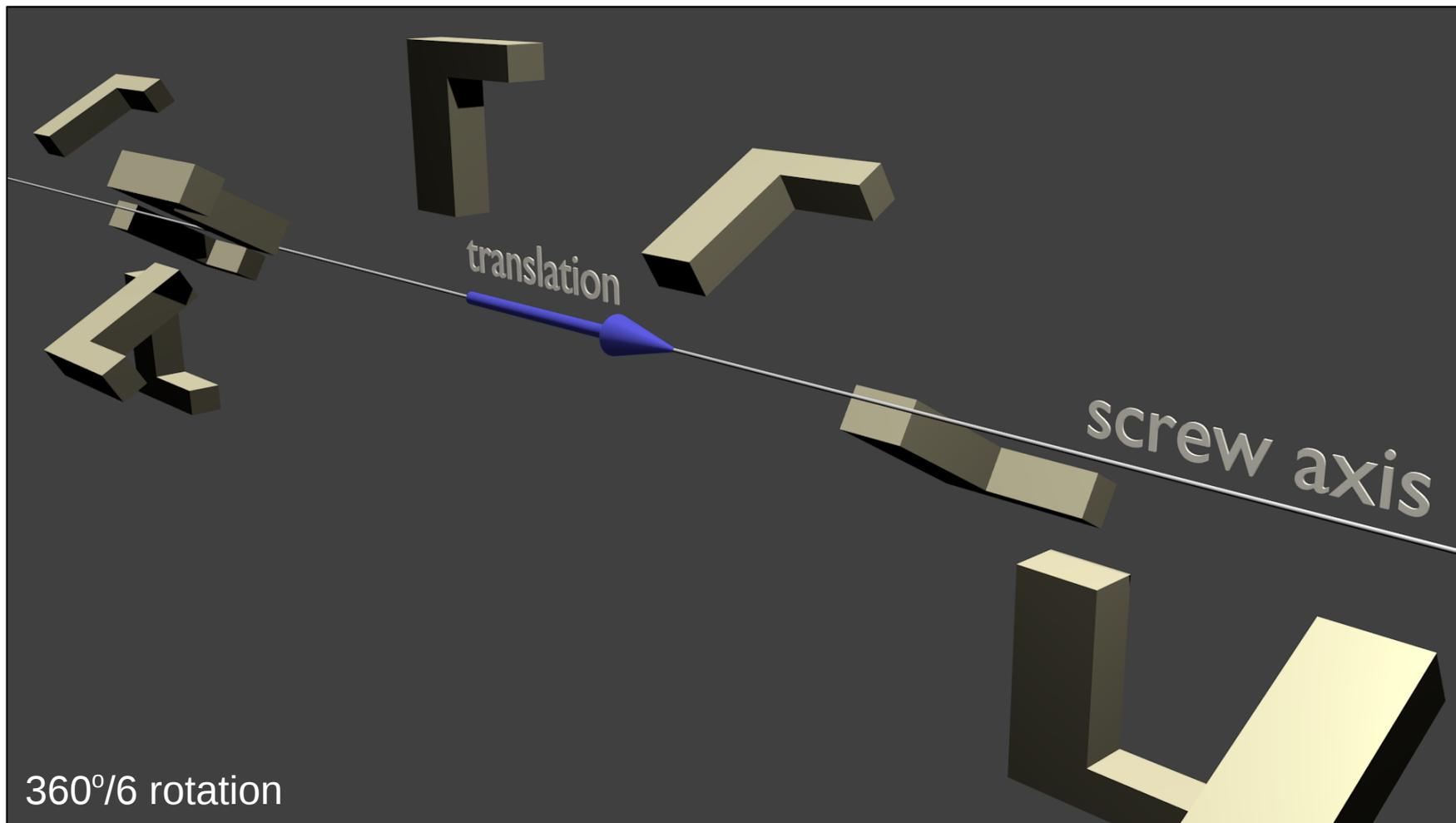
- Space groups are obtained combining translational symmetry (Bravais lattice) with point symmetry (point groups) together with two additional symmetry operations:
 - glide reflection:** reflection through glide plane followed by translation
 - screw rotation

Note that translations can have various orientations relative to basis vectors, i.e., not necessarily parallel (axial glide, diagonal glide, etc.) [8]



- Space groups are obtained combining translational symmetry (Bravais lattice) with point symmetry (point groups) together with two additional symmetry operations:
 - glide reflection: reflection through glide plane followed by translation
 - screw rotation**: rotation by $360^\circ/n$ followed by a translation

Note that "..., the amount of translation is $(1/n)$ th of one or more unit repeat distances" [8] ← Performing n screw rotations results in multiples of unit vector translation



References

1. L. Föppl, *Journal für die reine und angewandte Mathematik* **141**, 251 (1912)
2. P. Krishna and D. Pandey, *Close-Packed Structures*, 1981 by the International Union of Crystallography, ISBN 0 906449 08 1
3. Jin-Feng Jia, Xucun Ma, Xi Chen, T Sakurai and Qi-Kun Xue, *J. Phys. D: Appl. Phys.* **44**, 464007 (2001)
4. B. Wuensch, *3.60 Symmetry, Structure, and Tensor Properties of Materials*, MIT, Boston, USA, Fall 2005
5. H. Wondratschek, *Matrices, Mappings and Crystallographic Symmetry*, International Union of Crystallography 2002
6. J. M. D. COEY, *Magnetism and Magnetic Materials*, Cambridge University Press 2009
7. S. Olibet, E. Vallat-Sauvain, L. Fesquet, Ch. Monachon, A. Hessler-Wyser, J. Damon-Lacoste, S. De Wolf, and Ch. Ballif, *Phys. Status Solidi A*, 1–6 (2010) / DOI 10.1002/pssa.200982845
8. G. Burns and A.M. Glazer, *Space Groups for Solid State Scientists*, Elsevier ????
9. R. Lifshitz, *Symmetry of Quasicrystals*, www
10. J. E. S. Socolar, T. C. Lubensky, P. J. Steinhardt, *Phys. Rev. B* **34**, 3345 (1986)
11. М.П. Шаскольская, *Кристаллография*, Москва, Высшая Школа, 1984
12. M. van Meerssche, J. Feneau-Dupont, *Krytalografia i chemia strukturalna*, Warszawa, PWN 1984
13. International Union of Pure and Applied Chemistry, *Compendium of Chemical Terminology - Gold Book*, Version 2.3.3, 2014
14. Z. Dobkowska, K.M. Pazdro, *Szkolny Poradnik Chemiczny*, Warszawa, Wydawnictwa Szkolne I Pedagogiczne 1986
15. M. Główka *et al.*, *Słownik Terminów Krytalograficznych*, 2003 (angielsko-polski) Łódź, 2003
16. F.A. Cotton, *Chemical Applications of Group Theory*, Interscience Publishers, 1964