Quantum electronics

Magnetic anisotropy

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- Magnetocrystalline anisotropy
- Shape anisotropy
- Surface anisotropy
- Stress anisotropy

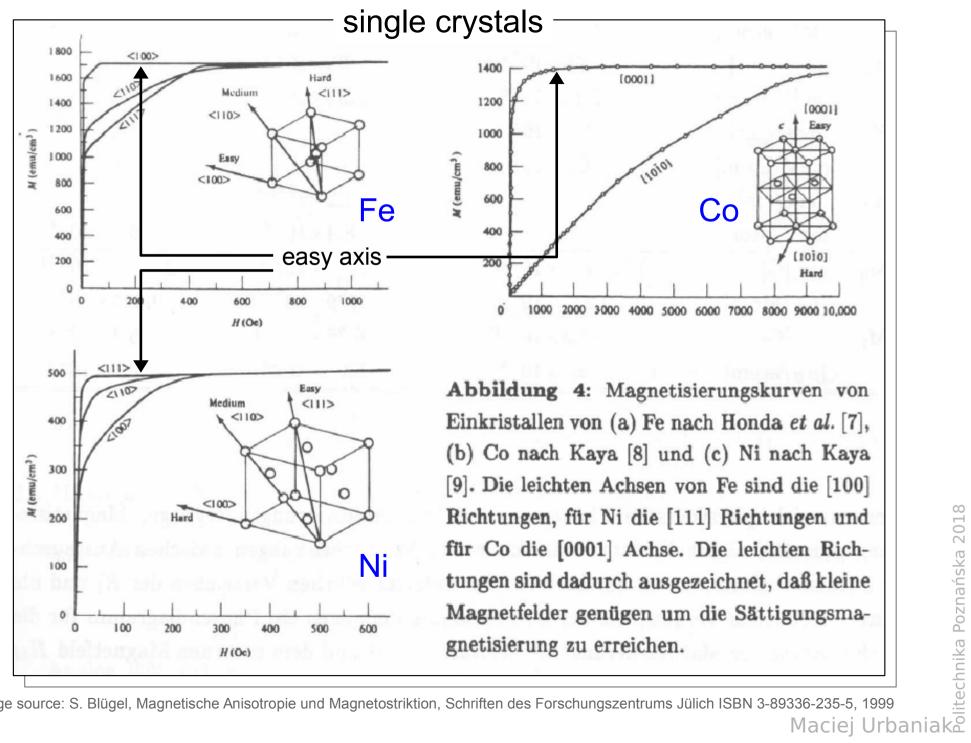
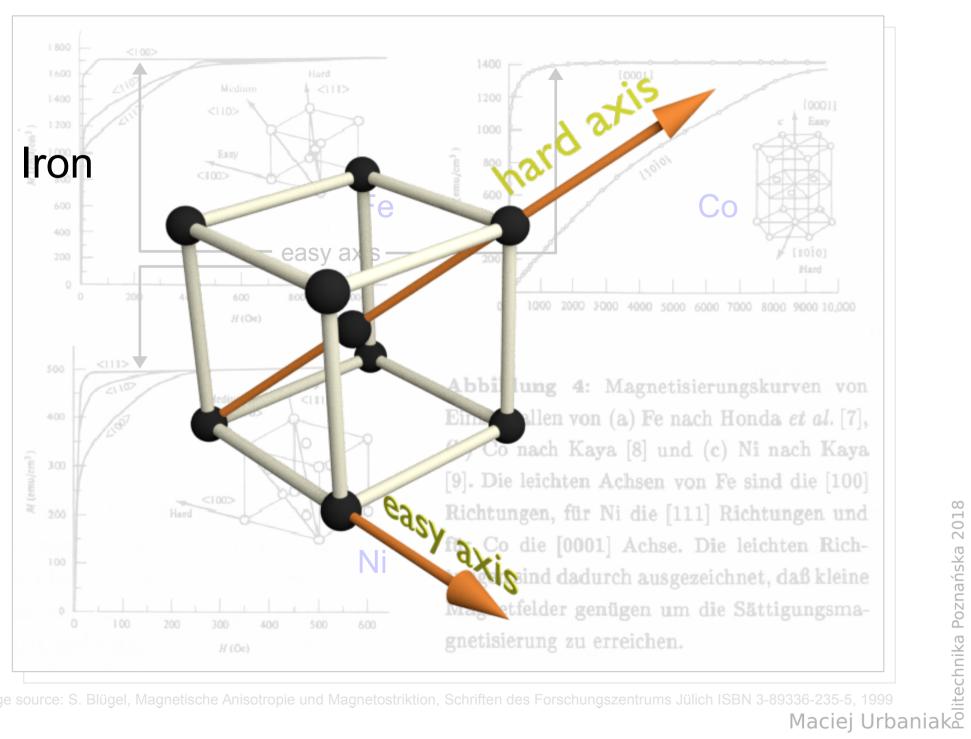
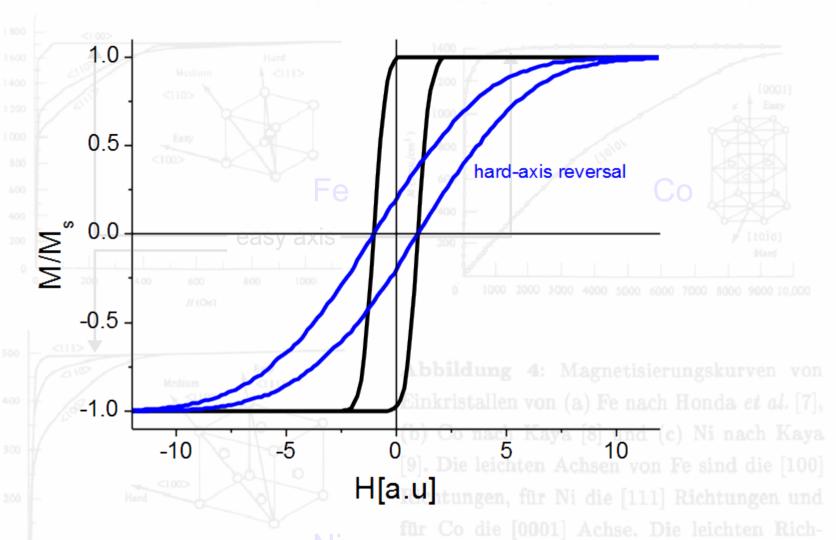


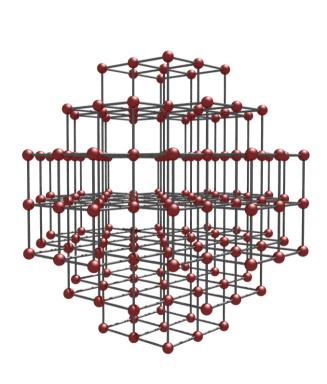
image source: S. Blügel, Magnetische Anisotropie und Magnetostriktion, Schriften des Forschungszentrums Jülich ISBN 3-89336-235-5, 1999

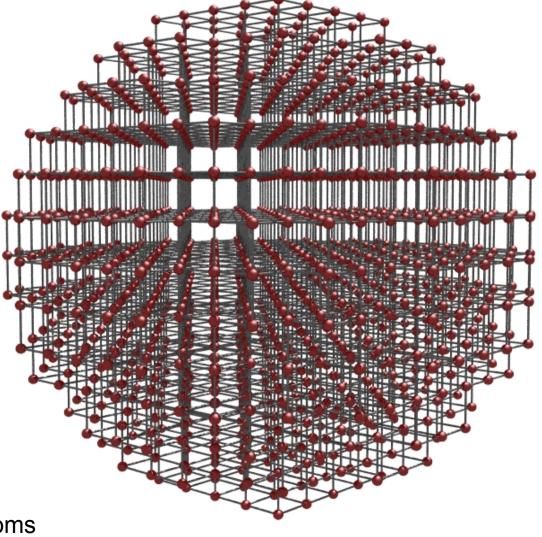




- •hard-axis reversal is characterized by higher field needed to saturate the sample
- •the easy-axis reversal is usually characterized by higher hysteresis losses

- •In case of large sphere (containing many atoms) the shape of the sample does not introduce additional anisotropy
- •In small clusters the magnetization reversal is complicated by the reduction of symmetry (and the increased relative contribution of surface atoms)





In Fe sphere of radius **1µm** the surface atoms constitute roughly 0.04% of all atoms

litechnika Poznańska 2018

- •In case of large sphere (containing many atoms) the shape of the sample does not introduce additional anisotropy
- •In small clusters the magnetization reversal is complicated by the reduction of symmetry (and the increased relative contribution of surface atoms)

sphere-like - no breaking of crystal symmetry for high r 6.4 nm 9.7 nm Co Fe (b) FIG. 6. (a) high-resolution transmission electron microscopy (HRTEM) observation of a cobalt cluster along a [110] direction. Interplanar distances $d_{(111)} = 2.04 \pm 0.02 \text{ Å}$, $d_{(200)} = 1.77 \pm 0.02 \text{ Å}$

(HRTEM) observation of a cobalt cluster along a [110] d Interplanar distances $d_{(111)} = 2.04 \pm 0.02$ Å, $d_{(200)} = 1.77 \pm 0.04$ and angles correspond to bulk fcc cobalt. Moreover facet sistent with the Wulff theorem (Ref. 27). (b) HRTEM of an iron cluster along a [110] direction. The interplanation $d_{(110)} = 2.01 \pm 0.03$ Å corresponds to the bulk bcc iron on is also consistent with the Wulff theorem (Ref. 27).

Interplanar distances $d_{(111)} = 2.04 \pm 0.02$ Å, $d_{(200)} = 1.77 \pm 0.02$ Å and angles correspond to bulk fcc cobalt. Moreover faceting is consistent with the Wulff theorem (Ref. 27). (b) HRTEM observation of an iron cluster along a [110] direction. The interplanar distance $d_{(110)} = 2.01 \pm 0.03$ Å corresponds to the bulk bcc iron one. Faceting

- Co atoms deposited by molecular beam epitaxy on Pt(111) surface
- Coverage less than 0.03 ML
- "The XMCD signal (Fig. 1C) is the difference between the XAS* spectra recorded for parallel and antiparallel alignment of the photon helicity with the applied field B. Fields of up to **7 T** were used to magnetize the sample at

angles 0° and 70° with respect to the surface

normal."

- The presence of Pt surface induces very high magnetic anisotropy of 9.3±1.6 meV/atom
- In SmCo₅ magnets the anisotropy is 0.3 meV/Co atom

It is about 6% of a latent heat of melting (243 kJ/kg [14]) of bulk Co

isolated Co adatoms

very high saturation field

P. Gambardella et al., Science **300**, 1130 (2003)

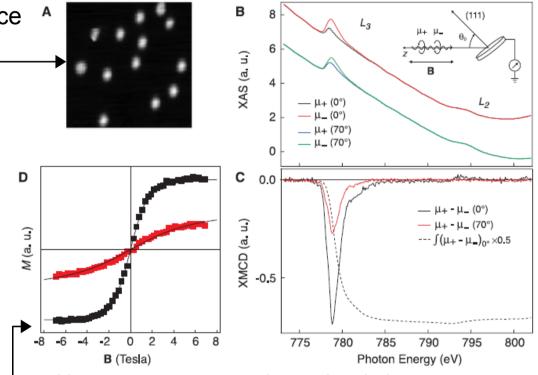


Fig. 1. (A) STM image of isolated Co adatoms (bright dots) on Pt(111). The Co coverage is 0.010 ML, and the image size is 85 Å by 85 Å. (B) L_{2.3} XAS spectra of isolated Co adatoms (0.010 ML) at $T = 5.5 \pm 0.5$ K, B = 7 T taken with parallel (μ_{\perp}) and antiparallel (μ_{\perp}) alignment of light helicity with respect to B at $\theta_0 = 0^{\circ}$, 70° relative to the surface normal (inset). The spectra at 70° have been normalized to the ($\mu_+^{"} + \mu_-$) L_3 intensity at 0° to eliminate the dependence of the electron yield on the sample orientation. (C) XMCD spectra ($\mu_+ - \mu_-$) obtained for the $\theta_0 = 0^\circ$ and 70° magnetization directions. The dashed line is the integrated XMCD at $\theta_0 = 0^\circ$. (D) Magnetization curves at $\theta_0 = 0^\circ$ (black squares) and 70° (red squares) measured at T = 5.5 K. The points represent the peak of the L_3 XMCD intensity at 778.6 eV divided by the pre-edge intensity at 775 eV as a function of B. The difference between the $\theta_{0}=0^{\circ}$ and 70° curves was checked for consistency with the XAS-normalized XMCD spectra. The solid lines are fits to the data according to Eq. 3.

- •The electron is orbiting the nucleus of the +Ze charge*
- •Looking at the nucleus from electron we have the magnetic field due to the motion of the nucleus. The energy of electron in that field is

$$\vec{E} = -g \mu_B m_s B$$

Correspondingly every electronic state splits into two (with two orientations of the spin). We assume that an electron is orbiting the nucleus in xy plane and that its instantaneous velocity is along x-direction.

The electric field of the nucleus at the place of an electron is along y-direction then

$$E_{y} = \frac{1}{4\pi\epsilon_{0}} \frac{Ze}{r^{2}}$$

From special relativity theory (A. Einstein) for the components of the magnetic field in the electron reference frame we have [16]

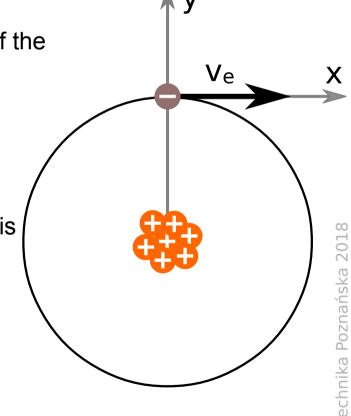
$$B_x^{\text{el(ectron)}} = B_x = 0$$

$$B_{y}^{\text{el}} = \frac{1}{\sqrt{1 - v^{2}/c^{2}}} \left(B_{y} + \frac{v_{e}}{c^{2}} E_{z} \right) = 0 \qquad B_{z}^{\text{el}} = \frac{1}{\sqrt{1 - v^{2}/c^{2}}} \left(B_{z} - \frac{v_{e}}{c^{2}} E_{y} \right) \approx -\frac{v_{e}}{c^{2}} E_{y}$$

Electron feels then the magnetic field that is oriented along z-axis Further, the field seen by the electron can be written as

$$B^{\text{el}} = \frac{1}{c^2} (\vec{E} \times \vec{v}) = \frac{1}{mc^2} (\vec{E} \times \vec{p})$$

$$\vec{p} = \frac{m_0 \vec{v}}{\sqrt{1 - v^2/c^2}}$$



$$B^{\text{el}} = \frac{1}{c^2} (\vec{E} \times \vec{v}) = \frac{1}{m c^2} (\vec{E} \times \vec{p})$$

Inserting the above calculated magnetic field into the expression for energy yields

$$\Delta E_{spin-orbit} = -g \mu_B m_s B = -g \mu_B m_s \frac{1}{m c^2} (\vec{E} \times \vec{p}) = -g \mu_B m_s \frac{1}{m c^2} \left[\left[\frac{1}{4 \pi \epsilon_0} \frac{Ze}{r^2} \right] \frac{\vec{r}}{r} \times \vec{p} \right]$$

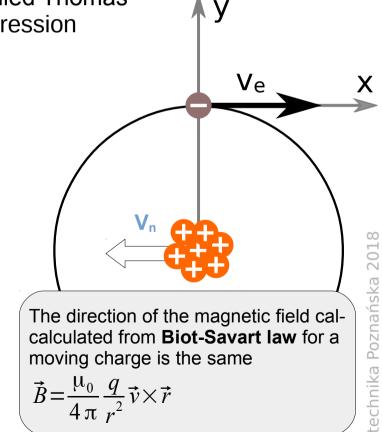
this give the appropriate direction of electric field

$$\Delta E_{spin-orbit} = -g \mu_B m_s \frac{1}{mc^2} \left(\left[\frac{1}{4\pi\epsilon_0} \frac{Ze}{r^3} \right] \vec{r} \times \vec{p} \right) = -g \mu_B m_s \frac{1}{mc^2} \left(\left[\frac{1}{4\pi\epsilon_0} \frac{Ze}{r^3} \right] \vec{L} \right)$$
 angular momentum

More exact calculations require taking into account the so called Thomas precession* – this leads to factor ½ which leads to "final" expression

$$\Delta E_{spin-orbit} = -\frac{g}{2} \mu_B \frac{1}{mc^2} \frac{1}{4\pi \epsilon_0} \frac{Ze}{r^3} \vec{L}$$

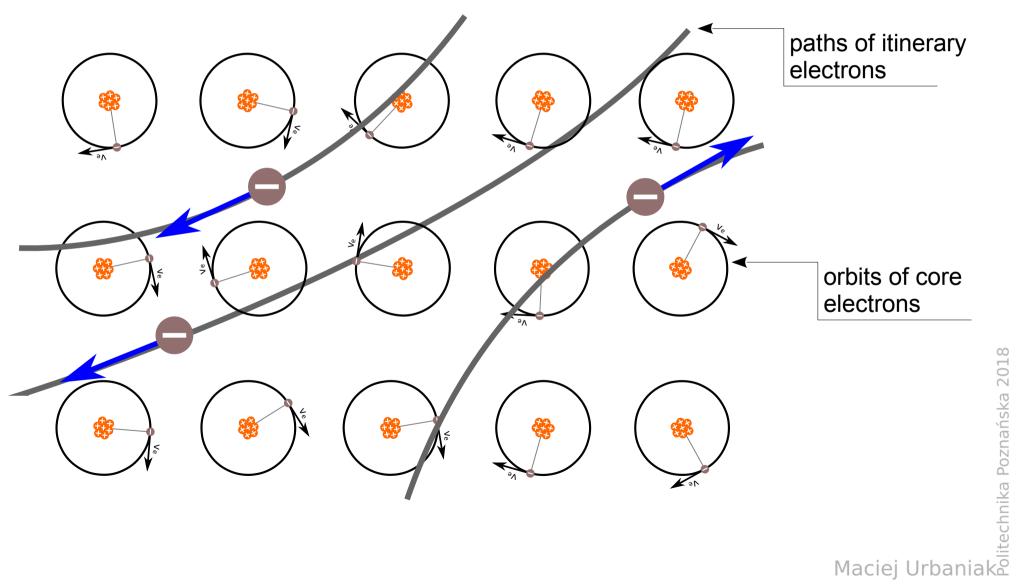
- note that spin-orbit coupling is proportional to Z
- ...and to orbital moment of an electron



calculated from Biot-Savart law for a moving charge is the same

$$\vec{B} = \frac{\mu_0}{4\pi} \frac{q}{r^2} \vec{v} \times \vec{r}$$

- •In a crystal conducting electrons move in the average electric field of the atom cores and other electrons
- •The core electrons that remain in the vicinity of the nucleus experience strong electric fields and, provided that the orbit is not centrosymmetric, they experience strong spin-orbit coupling (please see movies at https://staff.aist.go.jp/v.zayets/spin3 32 SpinOrbit.html)



- •In a crystal conducting electrons move in the average electric field of the atom cores and other electrons
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- •The magnetic fields due the core electron movement can be huge [15]:

Linear velocity of an electron rotating around a nucleus is ~2.1*106 m/s The electric field experienced by an electron in the vicinity of nucleus (calculated for 1s orbital of hydrogen atom) is roughly 5*1011 V/m The effective magnetic field of the spin-orbit interaction is about 12 T

For comparison we [15] estimate the effective field in devices in which we try to influence the behavior of itinerant electrons applying external electric fields The maximal electron velocity (saturation velocity, maximal drift speed) ~1*107 m/s The maximal available electric field (limited by breakdown voltage of the materials), for GaAs or Si it is roughly 5*107 V/m

The effective magnetic field of the spin-orbit interaction is about 5*10-4 T (about the strength of the earth magnetic field)

image source [19] PHYSICAL REVIEW B **90**, 165108 (2014)

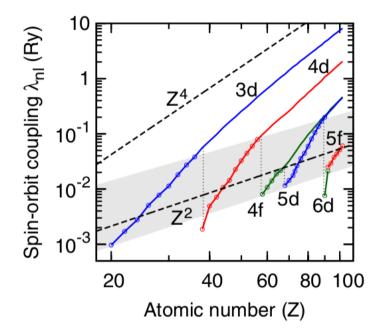


FIG. 1. (Color online) Dependence of the spin-orbit coupling strength λ_{nl} for atoms as a function of the atomic number Z. The calculated results of Herman and Skillman [24] using the Hartree-Fock method (colored lines) are compared to the hydrogenic Z^4 dependence, which is computed from Eq. (4) for the 3d series (upper dashed line). For the *outermost* electrons (indicated by the circles and the shaded area), which are the relevant electrons in the solid, the quantum numbers nl change with Z and the spin-orbit interaction increases much more slowly, following roughly the Landau-Lifshitz Z^2 scaling [lower dashed line, calculated from Eq. (5) with A = 0.10].

•The Z⁴ dependence for SO in central field (near core electrons) comes from perturbative correction [18]:

$$H_{SO}(r) = \frac{1}{2 m^2 c^2} \frac{\partial V(r)}{\partial r} \vec{L} \cdot \vec{S}$$

$$E_{nl} = E_{nl}^{0} + F(l, j) Z^{4}$$

•The Z² dependence is more relevant for solids

- •For all practical purposes the atomic magnetic moments of a macroscopic homogeneous magnetic *sphere* behave as if placed in infinite crystal of the *same shape*.
- A. Aharoni: "in ferromagnetism there is no physical meaning to the limit of an infinite *crystal without a surface*" [2]
- •We do not know a priori the dependence of the energy of the crystal on the orientation of magnetic moment of the sample.
- •It can be shown [1] that energy density related to the orientation of magnetic moment in a crystal structure can be expanded into power series of direction cosines relative to the crystal axes:

$$E_{crystal}(\vec{M}) = b_0 + \sum_{i=1,2,3} b_i \alpha_i + \sum_{i,j=1,2,3} b_{ij} \alpha_i \alpha_j + \sum_{i,j,k=1,2,3} b_{ijk} \alpha_i \alpha_j \alpha_k + \dots$$
 (1)

 $\alpha_{\rm 1}, \ \alpha_{\rm 2}, \ \alpha_{\rm 3}$ - direction cosines of magnetization

$$(\alpha_1, \alpha_2, \alpha_3) = (\sin(\theta)\cos(\phi), \sin(\theta)\sin(\phi), \cos(\theta))$$
 θ , ϕ - polar and azimuthal angles

•The experience shows that it is enough to use very limited number of expansion terms to describe the magnetic systems – the usual limit are sixth order anisotropy constants

•An example of the use of sixth order anisotropy constants for hysteresis description:

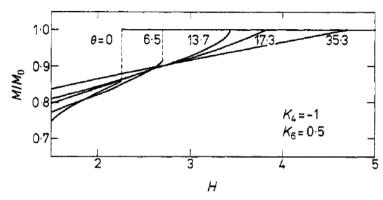


Figure 3. Magnetisation curves calculated for $\theta = 0^{\circ}$, 6.5° , 13.7° ($\simeq \theta_{\rm T}$), 17.3° and 35.3° [110]. The parameters K_4 and K_6 are phenomenological and not directly related to DyAl₂.

To see qualitatively that a sixth-order anisotropy term may indeed increase the discontinuity and the tricritical angle θ_T , consider the classical mean field energy

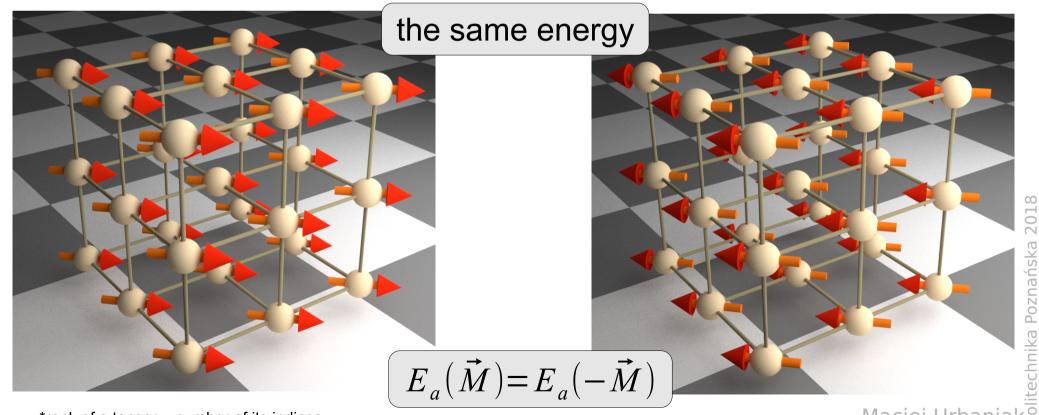
$$E = -H \cdot \frac{M}{M_0} + K_4 (M_x^4 M_y^4 + M_z^4) / M_0^4 + K_6 [M_x^6 + M_y^6 + M_z^6]$$
$$- \frac{15}{4} (M_x^4 M_y^2 + M_y^4 M_x^2 + M_z^2 M_y^4 + M_y^2 M_z^4 + M_z^2 M_x^4] / M_0^6$$

where K_4 and K_6 are phenomenological anisotropy constants. We have calculated magnetisation curves by minimising E with respect to M for various directions of H. In figure 3 are shown calculated magnetisation curves with $K_4 = -1$ and $K_6 = 0.5$. The discontinuity for $\theta = 0$ is 15% which corresponds to the situation for DyAl₂ at T = 20 K.

- •Intrinsic symmetries of the physical properties reduce the number of independent components of anisotropy tensors.
- •The energy of the system is the same for both opposite orientations of magnetic moment. From Eq. (1) we have:

$$\sum_{i=1,2,3} b_i \alpha_i = \sum_{i=1,2,3} b_i (-\alpha_i) \quad \text{for all } \alpha_i \ \Rightarrow \ b_1 = b_2 = b_3 = 0$$

•The magnetocrystalline anisotropy energy may not depend on odd powers of direction cosines α. Consequently all odd rank tensors in the expansion (1) are identically null [1].



The symmetry elements of any physical property of crystal must include all the symmetry elements of the point group* of the crystal.

•Consider a cubic crystal system with a 3-fold rotation axis [111] and the first nonvanishing anisotropy tensor (second rank):

$$b_{ij} = \begin{bmatrix} b_{11} & b_{12} & b_{13} \\ b_{21} & b_{22} & b_{23} \\ b_{31} & b_{32} & b_{33} \end{bmatrix}$$

Transpose of a matrix – switches rows and column indices

$$[M^T]_{ij} = M_{ji}$$

•The transformation matrix corresponding to that rotation is:

$$M = \begin{pmatrix} 0 & 0 & 1 \\ 1 & 0 & 0 \\ 0 & 1 & 0 \end{pmatrix}$$

Note: isometries (angles and distances preserved) of R³ space are described by square matrices for which an inverse of a matrix is equal to its transpose:

$$M^T = M^{-1}$$

and coordinates transform according to the following rule:

$$a'_i = \sum_j M_{ij} a_j$$

•Voigt's Principle:

The conditions of Neumann's principle are fulfilled if the physical property of the crystal is described by the tensor which is invariant under point symmetry operations which leave the crystal unchanged

•It follows that the physical property tensor must fulfill the condition $b=M^TbM$ for all symmetry operations of the point group.

^{*}A point group is a group of symmetry operations all of which leave at least on point unmoved.

•From Voigt's principle it follows for tensor b:

$$b=M^TbM$$

$$b_{ij} = \begin{bmatrix} 0 & 1 & 0 \\ 0 & 0 & 1 \\ 1 & 0 & 0 \end{bmatrix} \begin{bmatrix} b_{11} & b_{12} & b_{13} \\ b_{21} & b_{22} & b_{23} \\ b_{31} & b_{32} & b_{33} \end{bmatrix} \begin{bmatrix} 0 & 0 & 1 \\ 1 & 0 & 0 \\ 0 & 1 & 0 \end{bmatrix} = \begin{bmatrix} b_{22} & b_{23} & b_{21} \\ b_{32} & b_{33} & b_{31} \\ b_{12} & b_{13} & b_{11} \end{bmatrix}$$

rotation by 120Deg about [111] direction

effect of the rotation of the crystal on tensor b_{ij}

•Comparing the elements of both (identical) tensors we get:

•The invariance in respect the 120 Deg rotation leaves only **3 independent components**:

$$\begin{bmatrix} b_{ij} = \begin{bmatrix} a & c & b \\ b & a & c \\ c & b & a \end{bmatrix}$$

$$b=M^TbM$$

•We apply the same procedure again, but this time with other symmetry element of cubic crystal, namely 90Deg rotation around z-axis:

$$b_{ij} = \begin{bmatrix} 0 & 1 & 0 \\ -1 & 0 & 0 \\ 0 & 0 & 1 \end{bmatrix} \begin{bmatrix} a & c & b \\ b & a & c \\ c & b & a \end{bmatrix} \begin{bmatrix} 0 & -1 & 0 \\ 1 & 0 & 0 \\ 0 & 0 & 1 \end{bmatrix} = \begin{bmatrix} a & -b & c \\ -c & a & -b \\ b & -c & a \end{bmatrix}$$
 rotation by 90Deg about [001] direction effect of the rotation of the crystal on tensor b_{ij}

Comparing the elements of the first row of both (identical) tensors we get:

$$c=-b$$
, $b=c \Rightarrow b=c=0$

•It follows that the second rank tensor consistent with the above two symmetry operations possesses **one independent component**:

$$b_{ij} = \begin{bmatrix} a & 0 & 0 \\ 0 & a & 0 \\ 0 & 0 & a \end{bmatrix}$$

•Similar analysis can be performed for other tensors in the expansion (1):

$$E_{crystal}(\vec{M}) = b_0 + \sum_{i=1,2,3} b_i \alpha_i + \sum_{i,j=1,2,3} b_{ij} \alpha_i \alpha_j + \sum_{i,j,k=1,2,3} b_{ijk} \alpha_i \alpha_j \alpha_k + \dots$$

•Inserting tensor b into the third term of expansion (1) we get:

$$\sum_{i,\,j=1,2,3} b_{ij} \alpha_i \alpha_j = a(\alpha_1^2 + \alpha_2^2 + \alpha_3^2) = a \quad \text{- independent of the orientation of magnetic moment}$$

- •In cubic system there are no second order terms in the expansion of energy in directional cosines [1].
- •Using similar procedure we obtain the complete expression for the energy contribution related to the orientation of magnetic moment in cubic system [1]:

$$E_{\textit{crystal}}(\vec{M}\,,T) = K_0(T) + K_1(T)(\alpha_1^2 \alpha_2^2 + \alpha_2^2 \alpha_3^2 + \alpha_3^2 \alpha_1^2) + K_2(T)\alpha_1^2 \alpha_2^2 \alpha_3^2$$

- -the coefficients K_0 , K_1 ... are the linear combinations of tensor components b_{11} , b_{1111} , b_{11111} etc. [4].
- •For other crystal systems the similar procedure is employed to obtain the $E_{\text{crystal}}(M,T)$ expressions.
- •For hexagonal crystals the energy can be expressed as [1]:

$$E_{\mathit{crystal}}(\vec{M}\,,T)\!=\!K_{0}(T)\!+K_{1}(T)(\alpha_{1}^{2}\!+\alpha_{2}^{2})\!+K_{2}(T)(\alpha_{1}^{2}\!+\alpha_{2}^{2})^{\!2}\!+\ldots$$

which is usually expressed, using trigonometric identities, as:

$$E_{crystal}(\vec{M},T) = K_0(T) + K_1(T)\sin^2\theta + K_2(T)\sin^4\theta + \dots$$

$$= \frac{1}{a_1^2 + a_2^2 = \sin^2\theta \cos^2\phi + \sin^2\theta \sin^2\phi = \sin^2\theta}$$
(2)

- •Inserting tensor b into the third term of expansion (1) we get:
- $\sum_{i,\,j=1,2,3} b_{ij} \alpha_i \alpha_j = a(\alpha_1^2 + \alpha_2^2 + \alpha_3^2) = a \quad \text{- independent of the orientation of magnetic moment}$
- •In cubic system there are no second order terms in the expansion of energy in directional cosines [1].
- •Using similar procedure we obtain the complete expression for the energy contribution related to the orientation of magnetic moment in cubic system [1]:

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- -the coefficients K_0 , K_1 ... are the linear combinations of tensor components b_{11} , b_{1111} , b_{11111} etc. [4].
- •The terms of the type α_i^4 are omitted since because of the identity [4,5]:

$$2(\alpha_1^2 \alpha_2^2 + \alpha_2^2 \alpha_3^2 + \alpha_3^2 \alpha_1^2) + \alpha_1^4 + \alpha_2^4 + \alpha_3^4 = 1$$

they can be incorporated into K_0 , K_1 terms.

•The terms of the type α_i^6 can be similarly replaced by $\alpha_i^2 \alpha_j^2$ and $\alpha_1^2 \alpha_2^2 \alpha_3^2$ terms [6].

- •Number of independent components of the (second rank) tensor depends on the crystal symmetry
- •In crystals of cubic system there is one independent component of the tensor.
- •Hexagonal systems are characterized by two independent components of the second rank tensors.

Оптическая класси- фикация	Системы	Характернзующая симметрия *	Вид характеристиче- ской поверхности и ее ориентация	Число незави- симых коэф- фициен- тов	Тензор, приведенный к осям принятой ориентации **
Изотроп- ная среда	Куби- ческая	Четыре оси третьего по- рядка	Сфера	1	$\begin{bmatrix} S & 0 & 0 \\ 0 & S & 0 \\ 0 & 0 & S \end{bmatrix}$
Одно- осные кри- сталлы	Тетраго- нальная Гексаго- нальная Триго- нальная	Одна ось четвертого порядка Одна ось шестого порядка Одна ось третьего порядка	Поверхность вращения во- круг главной оси симметрии $x_3(z)$	2	$\begin{bmatrix} S_1 & 0 & 0 \\ 0 & S_1 & 0 \\ 0 & 0 & S_3 \end{bmatrix}$
	Орто- ромбиче- ская	Три взаимно перпендику- лярные оси второго по- рядка; осей высшего по- рядка нет	Произвольная поверхность второго порядка с осями $x_1, x_2, x_3,$ параллельными осям второго порядка x, y, z	3	$ \begin{bmatrix} S_1 & 0 & 0 \\ 0 & S_2 & 0 \\ 0 & 0 & S_3 \end{bmatrix} $
Дву- осные кри- сталлы	Моно- клинная	Одна ось вто- рого порядка	Произвольная поверхность второго порядка с одной осью x_2 , параллельной оси второго порядка у	4	$\begin{bmatrix} S_{11} & 0 & S_{81} \\ 0 & S_{22} & 0 \\ S_{81} & 0 & S_{88} \end{bmatrix}$
	Три- клинная	Центр симметрии или отсутствие симметрии	Произвольная поверхность второго порядка. Положение относительно кристаллографических осей не фиксировано	6	$\begin{bmatrix} S_{11} & S_{12} & S_{81} \\ S_{12} & S_{22} & S_{23} \\ S_{81} & S_{23} & S_{86} \end{bmatrix}$

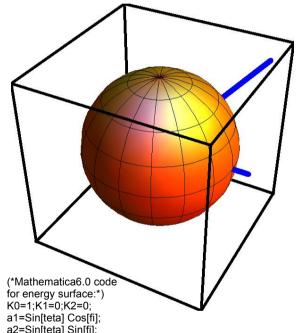
^{*} Оси симметрии могут быть поворотными или инверсионными; см. также стр. 335.

^{**} Ориентация осей x_1 , x_2 , x_3 поверхности второго порядка по отношению к кристаллографическим осям x, y, z и элементам симметрии указана в табл. 4. Добавочные замечания о выборе систем координат см. в приложении 3.

- Energy surface the distance from origin along the given direction is proportional to magnetocrystalline energy of the crystal with magnetization along that direction.
- •We start from the expression of the magnetocrystalline energy for cubic crystals:

$$E_{\mathit{crystal}}(\vec{M}\,,T) = K_0(T) + \,K_1(T) (\,\alpha_1^{\ 2}\alpha_2^{\ 2} + \,\alpha_2^{\ 2}\alpha_3^{\ 2} + \,\alpha_3^{\ 2}\alpha_1^{\ 2}) + \,K_2(T)\alpha_1^{\ 2}\alpha_2^{\ 2}\alpha_3^{\ 2} + \ldots$$

•For K_0 =1, K_1 =0 and K_2 =0 we have isotropic energy surface:

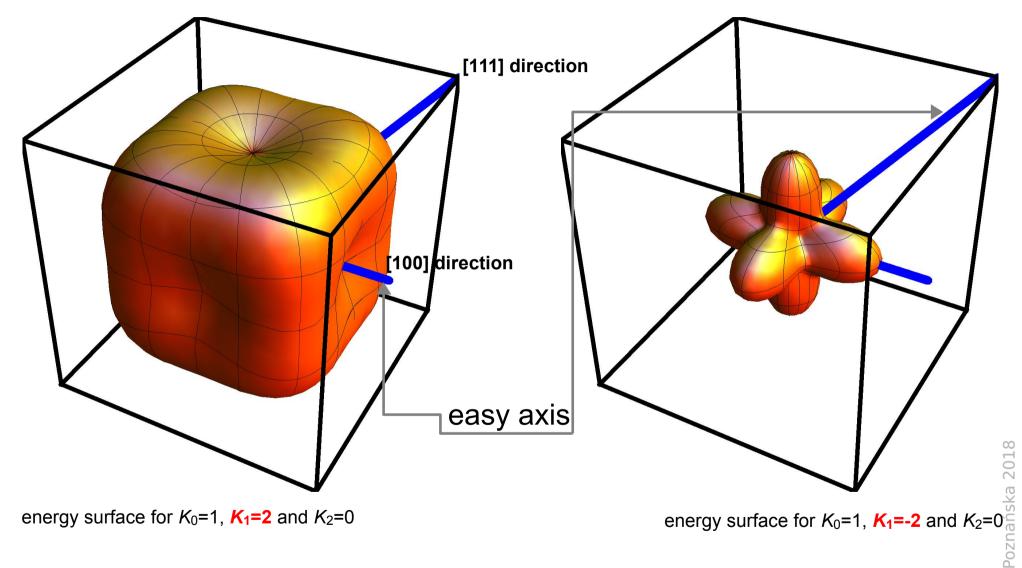


- Energy does not depend on the orientation of the magnetic moment
- •The magnetization reversal (hysteresis) itself does not depend on K_0 but to show the difference between the cases of $K_1>0$ and $K_1<0$ we need a reference level the surface of the sphere (r= K_0).

a2=Sin[teta] Sin[fi];
a3=Cos[teta];
Energy=K0+K1(a1^2 a2^2+a3^2 a2^2+a1^2 a3^2)+K2(a1^2 a2^2 a3^2);
xsurface=Energy a1;
ysurface=Energy a2;
zsurface=Energy a3;
obrazek=ParametricPlot3D[{xsurface,ysurface,zsurface},{fi,0,2\[Pi]},{teta,-\[Pi],\[Pi]},PlotStyle->{Orange,Specularity[White,10]},
ImageSize->600,PlotRange->{-1.2,1.2},Axes->None,AxesLabel->{X,Y,Z},BoxStyle->Directive[Thickness[0.01],Black]];
osdiag=Line[{{0,0,0},{1,2,0,0}}];
moment=Sphere[{1,1,1},1];

obrazekwy=Show[obrazek,Graphics3D[{Blue,Thickness[0.02],osx}],Graphics3D[{Blue,Thickness[0.02],osdiag}]]

•Cubic crystals magnetocrystalline energy surfaces* for different anisotropy coefficients:

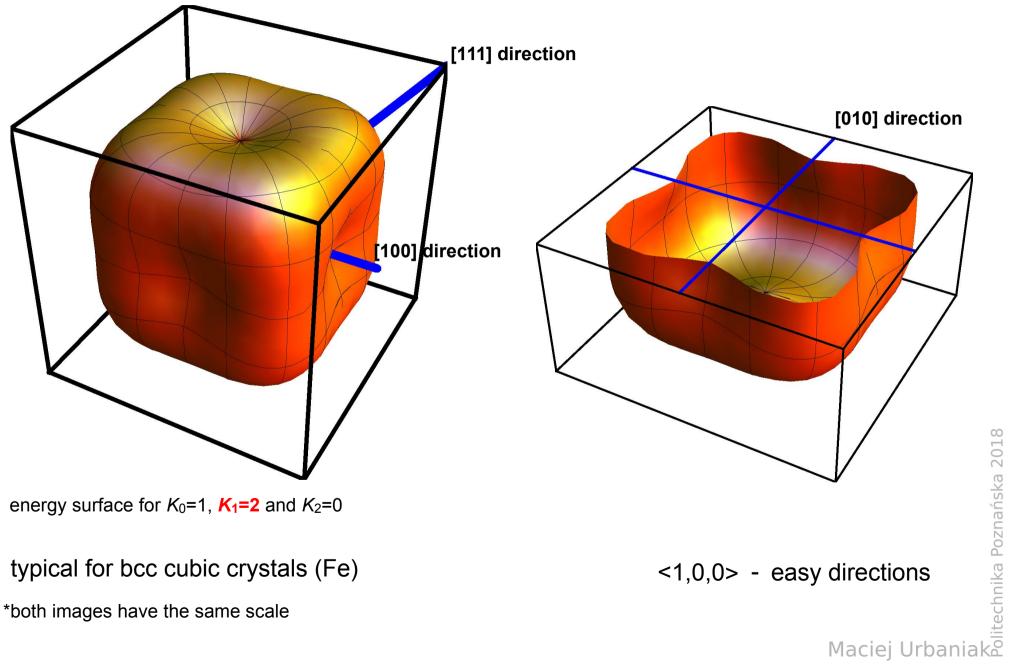


typical for bcc cubic crystals (Fe)

*both images have the same scale

typical for fcc cubic crystals (Ni)

•Cubic crystals magnetocrystalline energy surfaces* for different anisotropy coefficients:



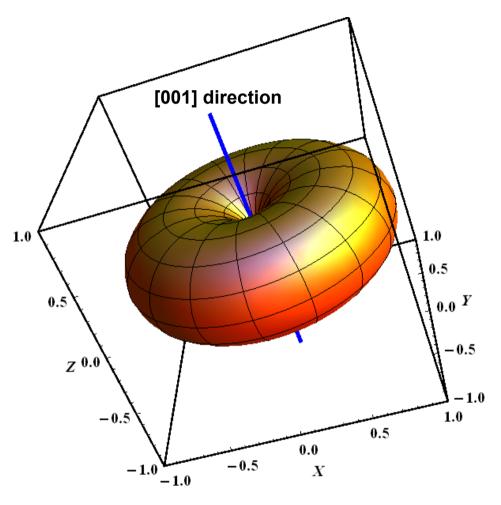
energy surface for $K_0=1$, $K_1=2$ and $K_2=0$

typical for bcc cubic crystals (Fe)

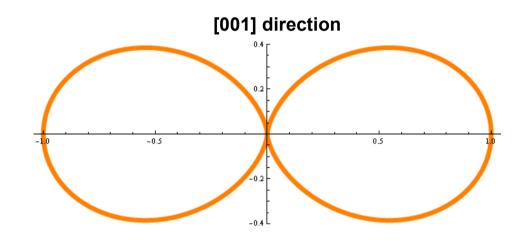
<1,0,0> - easy directions

*both images have the same scale

•Hexagonal crystals magnetocrystalline energy surfaces:



$$E_{crvstal}(\vec{M}) = K_0 + K_1 \sin^2 \theta + K_2 \sin^4 \theta$$



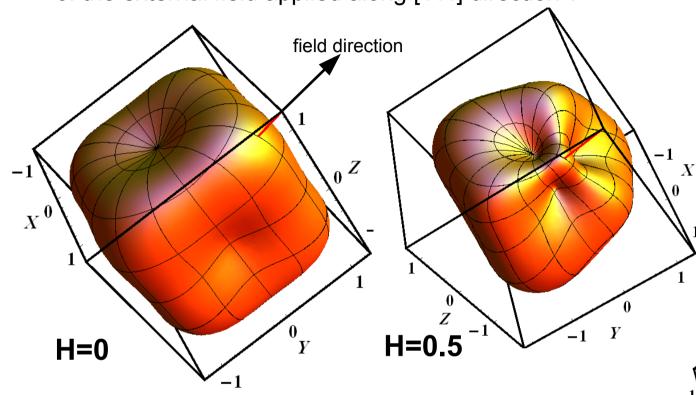
energy surface for $K_0=0$, $K_1=-1$ and $K_2=0$

typical for hcp cobalt crystals

[0,0,1] - easy direction

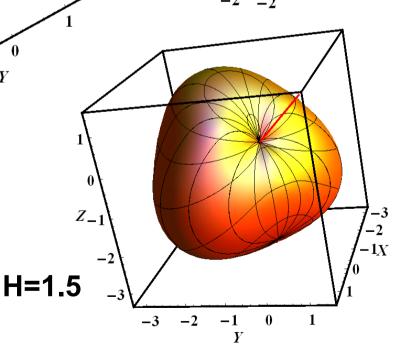
H=1

•Cubic crystals magnetocrystalline energy surfaces for different values of the external field applied along [111] direction*:



$$\begin{split} &E_{\textit{crystal}}(\vec{M}\,,\vec{H})\!\!=\!\!K_0\!+K_1(\alpha_1^{\ 2}\alpha_2^{\ 2}\!+\alpha_2^{\ 2}\alpha_3^{\ 2}\!+\alpha_3^{\ 2}\alpha_1^{\ 2})\!+\\ &K_2\alpha_1^{\ 2}\alpha_2^{\ 2}\alpha_3^{\ 2}\!+H(\alpha_1\beta_1\!+\alpha_2\beta_2\!+\alpha_3\beta_3)\\ &\beta_{1,}\ \beta_{2,}\ \beta_1\ \text{- direction cosines of H} \end{split}$$

energy surfaces for $K_0=1$, $K_1=2$ and $K_2=0$

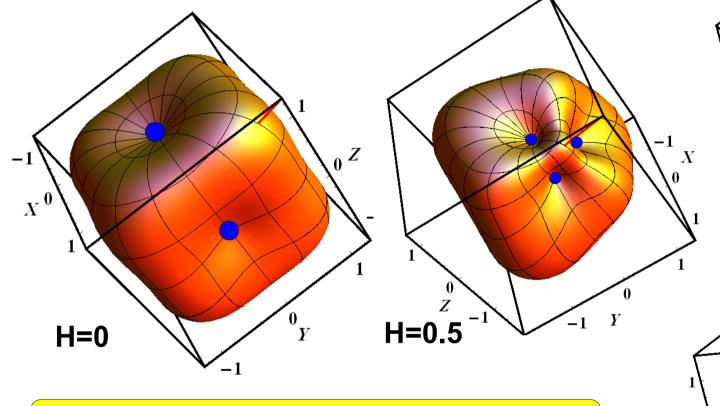


^{*}images do not have the same scale

H=1

•Cubic crystals magnetocrystalline energy surfaces for different values

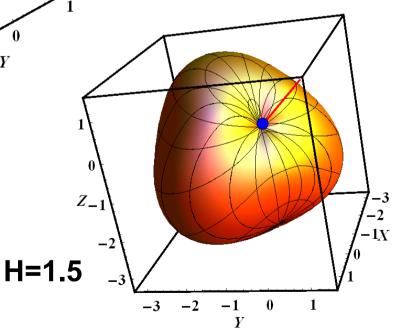
of the external field applied along [111] direction*:



•with increasing field **H** the number of local minima decreases

above saturation there is only one local minimum

energy surfaces for $K_0=1$, $K_1=2$ and $K_2=0$ *images do not have the same scale



•Bulk magnetocrystalline anisotropy constants of basic ferromagnetic elements at 4.2K [1]:

	Fe (bcc)	Co (hcp)	Ni (fcc)
K ₁ [J/m ³]	54800	760000	-126300
[meV/atom]	4.02×10 ⁻³	5.33×10 ⁻²	-8.63×10 ⁻³
K_2 [J/m ³]	1960	100500	57800
[meV/atom]	1.44×10 ⁻⁵	7.31×10 ⁻³	3.95×10 ⁻³

•Magnetocrystalline anisotropy of permalloy (Ni₈₁Fe₁₉):

K≈0 kJ/m³

•Magnetocrystalline anisotropy of rare-earth magnets [3]:

SmCo₅ K≈7.7×10⁶ J/m³

•Consider the crystal in which two uniaxial anisotropies are present together [3]. We limit our discussion to second order terms [see Eq.(2)]:

$$E_A = K_0 + K_A \sin^2 \theta$$
, $E_B = K_0 + K_B \sin^2 (90 - \theta) = K_0 + K_B \cos^2 \theta$

•The total energy of the moment is:

$$E_{total} = K'_0 + K_A \sin^2 \theta + K_B \cos^2 \theta$$

•If $K_A = K_B$ the energy is independent of θ :

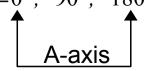
$$E_{total} = K'_0 + K_B (\sin^2 \theta + \cos^2 \theta) + (K_A - K_B) \sin^2 \theta = K'_0 + K_B$$

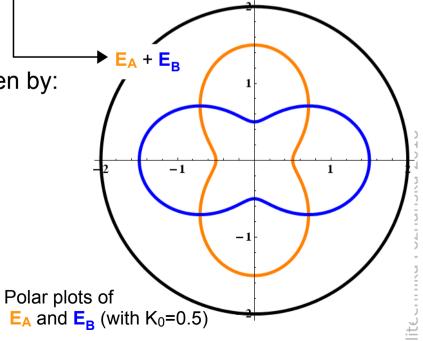
Two equal uniaxial anisotropies at right angle are not equivalent to biaxial anisotropy.

•If K_A and K_B are not equal the equilibrium angle is given by:

$$\frac{\partial E_{total}}{\partial \theta} = \frac{\partial}{\partial \theta} (K_A - K_B) \sin^2 \theta = \frac{\partial}{\partial \theta} (K_A - K_B) \left(\frac{1 - \cos(2\theta)}{2} \right) = (K_A - K_B) \sin(2\theta) = 0$$

•Solutions are $\theta = 0^{\circ}$, 90° , 180°





В

•From the second derivative (must be positive for minimum) we obtain [3]:

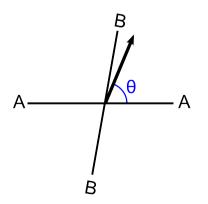
$\frac{\partial^2 E_{total}}{\partial \theta^2} = 2(K_A - K_B)\cos(2\theta) \Rightarrow$	K _A >K _B	K _A <k<sub>B</k<sub>
$\frac{\partial \theta^2}{\partial \theta^2} = 2(K_A - K_B)\cos(2\theta) \Rightarrow$	Easy axis – θ=0 Deg	Easy axis – θ=90 Deg

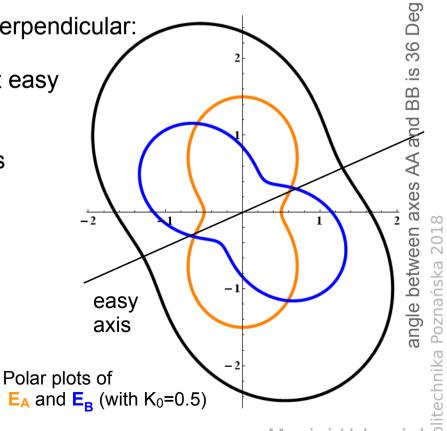
The direction of easy magnetization is not along some axis lying between AA and BB axes but is along the axis pertaining to higher anisotropy.

•Case of the two uniaxial anisotropies which are not perpendicular:

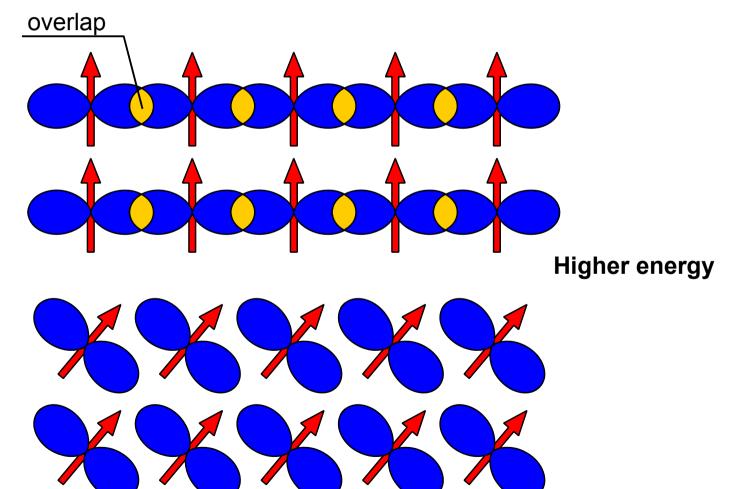
-in case of anisotropies of equal strength the resultant easy axis CC lies midway between axes AA and BB

-otherwise the CC axes makes smaller angle with axis pertaining to stronger anisotropy





- •The spin of electron interacts with the crystal structure via spin orbit coupling
- •the moment of a spin (red arrow) is strongly coupled to the electron cloud (blue orbitals) spin orbit coupling
- •when external magnetic field rotates the spin the electron "attempts" cloud follows but its energy depends on the orientation relative to neighboring atoms/orbitals
- •if the orbitals overlap there is additional energy due to coulomb repulsion (on the other hand it can lower the energy of repulsion between ions) - IT IS A QUALITATIVE DESCRIPTION!



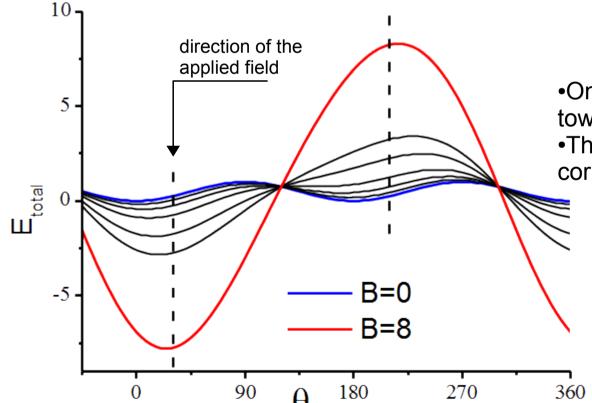
- Due to spin-orbit coupling different orientations of electron spins correspond to different orientations of atomic orbitals relative to crystal structure
- •As a consequence some orientations of the resultant magnetic moment are energetically favorable easy directions.

Easy axis

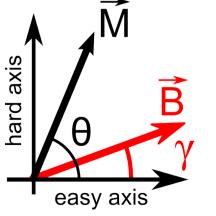
- •Describes magnetization reversal in **single domain** magnetic particles/films
- •The reversal is characterized by the orientation of single magnetic moment
- •The anisotropy may be of magnetocrystalline, shape etc. origin
- •For the uniaxial anisotropy case the energy can be described as (compare magnetocrystalline anisotropy energy expression for hexagonal system) [8]:

$$E_{total} = K_0 + K_1 \sin^2 \theta - \vec{B} \cdot \vec{M} = K_0 + K_1 \sin^2 \theta - M B \cos(\gamma - \theta) **$$
 Zeeman energy

•The energy landscape for different values of **B** ($K_0=0,K_1=1,M=1,\gamma=30^0$):



- •On increasing the field the minima shift toward its direction
- •The angle antiparallel to field corresponds to absolute maximum

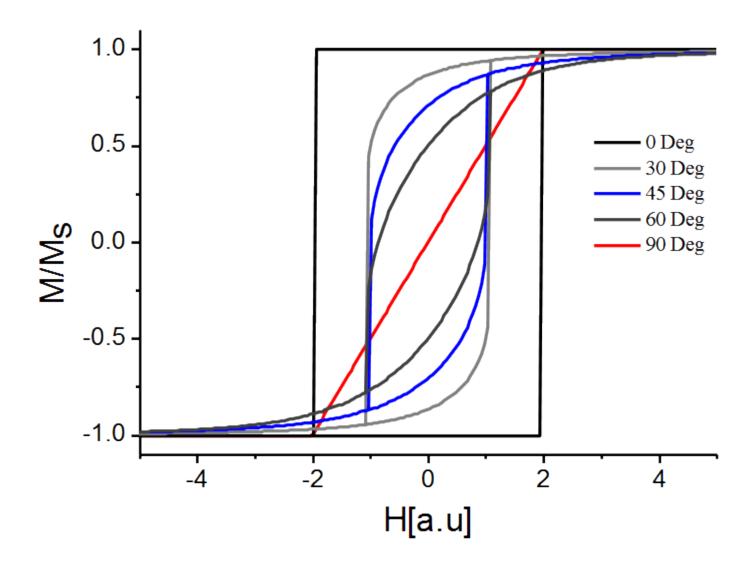


*some times called macrospin model

** this expression is for a unit volume of the material: M:=MV [Am²], K=KV [J]

Maciej Urbaniak

•The dependence angle(field) obtained from the energy landscapes of the previous slide gives hysteresis loops:



- •For field applied along easy-axis the reversal is completely irreversible
- •For field applied perpendicularly to EA direction the reversal is completely reversible
- •For field applied in arbitrary direction magnetization is "partly reversible and partly irreversible" [9]

•Hard axis reversal. We can rewrite the expression for the total energy using components of the field parallel (B_x) and perpendicular (B_y) to easy axis [9]:

$$E_{total} = K_0 + K_1 \sin^2 \theta - M B \cos(\gamma - \theta) = K_0 + K_1 \sin^2 \theta - B_x M_x - B_y M_y = K_0 + K_1 \sin^2 \theta - B_x M \cos(\theta) - B_y M \sin(\theta)$$

•Energy becomes minimum at a specific angle which can be determined setting:

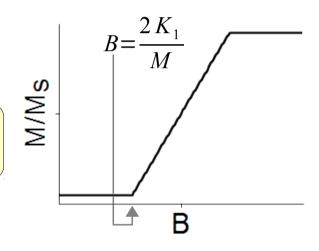
$$\frac{\partial E_{total}}{\partial \theta} = 2K_1 \sin \theta \cos \theta + B_x M \sin(\theta) - B_y M \cos(\theta) = 0$$

•With $\alpha = \frac{2K_1}{M}$ this can be written as: $\alpha \sin \theta \cos \theta + B_x \sin(\theta) - B_y \cos(\theta) = 0 \quad \text{or} \quad \frac{B_y}{\sin(\theta)} - \frac{B_x}{\cos(\theta)} = \alpha$

$$\alpha \sin \theta \cos \theta + B_x \sin(\theta) - B_y \cos(\theta) = 0$$
 or



$$\sin(\theta) = \frac{B}{\alpha}$$
 proportional to \vec{M} component parallel to \vec{B}



If field is applied perpendicularly to the easy axis the component of magnetization parallel to the field is a linear function of the external field up to saturation which happens at*:

$$B_S = \frac{2K_1}{M}$$

ika Poznańska 2018

•Depending on the value of the external field there may one or two equilibrium orientations of magnetic moment. For a given field value the two orientations collapse to one when [9]:

$$\frac{\partial^{2} E_{total}}{\partial \theta^{2}} = 0$$

$$\frac{\partial^{2} E_{total}}{\partial \theta^{2}} = \alpha \sin \theta \cos \theta + B_{x} \sin (\theta) - B_{y} \cos (\theta) = 0 = 0$$

•From the expression for derivative of energy (previous slide) we have: From previous slide:

$$\frac{\partial^{2} E_{total}}{\partial \theta^{2}} = \alpha \left(\cos^{2} \theta - \sin^{2} \theta\right) + B_{x} \cos(\theta) + B_{y} \sin(\theta) = 0$$

$$\alpha = \frac{B_{y}}{\sin(\theta)} - \frac{B_{x}}{\cos(\theta)}$$

$$\frac{\partial^2 E_{total}}{\partial \theta^2} = \cos^2 \theta \sin^2 \theta \left(\frac{B_y}{\sin^3(\theta)} + \frac{B_x}{\cos^3(\theta)} \right) = 0$$

•We are looking for the solution of the set:

$$\alpha = \frac{B_y}{\sin(\theta)} - \frac{B_x}{\cos(\theta)} \quad , \qquad \frac{B_y}{\sin^3(\theta)} + \frac{B_x}{\cos^3(\theta)} = 0 \qquad \qquad \alpha + \frac{B_x}{\cos(\theta)} = \frac{B_y}{\sin(\theta)} \cdot \frac{B_y}{\sin(\theta)} \cdot \frac{1}{\sin^2(\theta)} + \frac{B_x}{\cos^3(\theta)} = 0$$

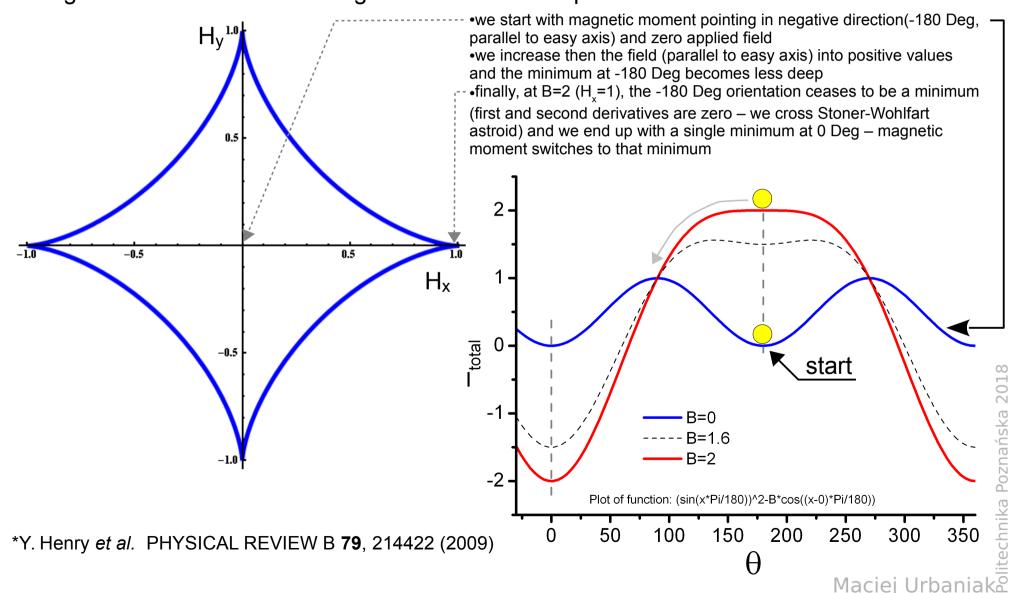
•By a direct substitution of the first equation into the second we get:

$$B_x = -\alpha \cos^3 \theta$$
, $B_y = \alpha \sin^3 \theta$

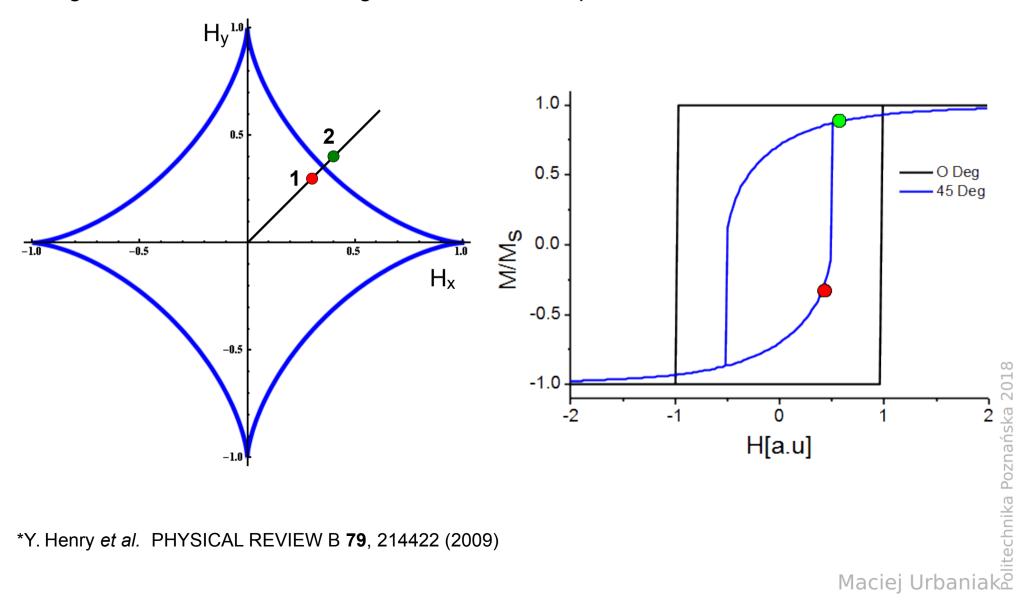
•Introducing reduced fields ($b_x = \frac{B_x}{\alpha} = -\cos^3\theta$) it may be written as:

$$b_x^{2/3} + b_y^{2/3} = 1$$
 = $\cos^2 \theta + \sin^2 \theta$ equation of astroid

- •Stoner-Wohlfarth astroid separates region, in (hx,hy) plane, with two minima of energy from that with only one minimum*
- •When the external field is changed so that the astroid is crossed the discontinuous changes of the orientation of magnetization can take place

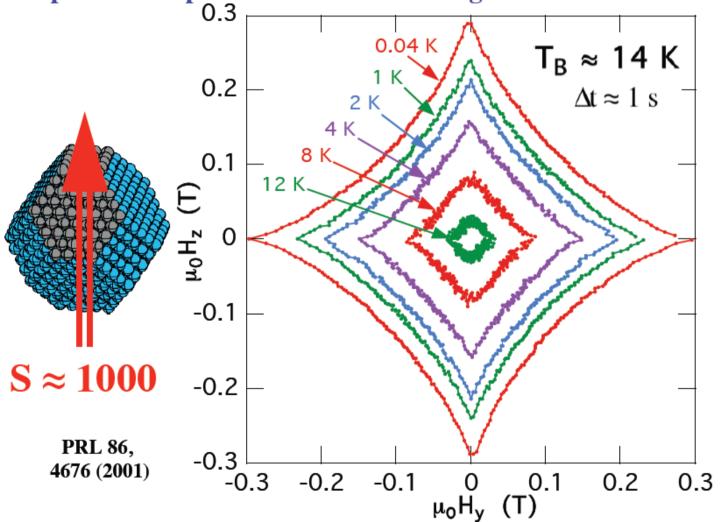


- •Stoner-Wohlfarth astroid separates region, in (hx,hy) plane, with two minima of energy from that with only one minimum*
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•Stoner-Wohlfarth astroid separates region, in (hx,hy) plane, with two minima of energy from that with only one minimum

Temperature dependence of the switching fields of a 3 nm Co cluster



M. Jamet, W. Wernsdorfer, C. Thirion, D. Mailly, V. Dupuis, P. Mélinon, and A. Pérez, Phys. Rev.Lett 86, 4676 (2001)

- •Polycrystalline samples without a preferred orientation of the grains do not show, in macroscopic experiments, any magneto crystalline anisotropy [9].
- •If the sample is not spherical the magnetostatic energy of the system depends on the orientation of magnetic moments within the sample (or macrospin in a simplified picture).
- •The effect is of purely magnetostatic origin and is closely related to demagnetizing fields (see my **lecture 2** from 2012):

If and only if the surface of uniformly magnetized body is of second order the magnetic induction inside is uniform and can be written as:

$$\vec{B} = \mu_0 (-N \cdot \vec{M} + \vec{M})$$

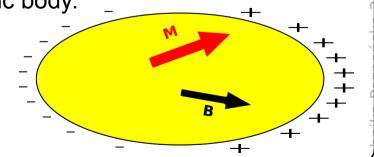
N is called the demagnetizing tensor [5]. If magnetization is parallel to one of principle axes of the ellipsoid *N* contracts to **three numbers** called demagnetizing (or demagnetization) factors sum of which is one:

$$N_x + N_v + N_z = 1$$

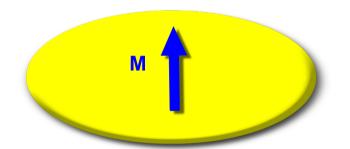
For a general ellipsoid magnetization and induction are not necessarily parallel.

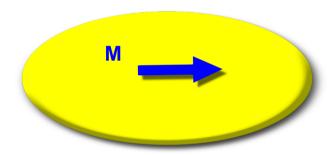
Demagnetization decreases the field inside ferromagnetic body.

 Demagnetizing field is just the name of the field produced by the body itself



- •Polycrystalline samples without a preferred orientation of the grains do not show, in macroscopic experiments, any magneto crystalline anisotropy [9].
- •If the sample is not spherical the magnetostatic energy of the system depends on the orientation of magnetic moments within the sample (or macrospin in a simplified picture).
- •The effect is of purely magnetostatic origin and is closely related to demagnetizing fields.





•The energy of the sample in its own stray field is given by the integral [9]:

$$E_{\textit{demag}}\!=\!-\frac{1}{2}\int \, \vec{B}_{\textit{demag}}\!\cdot\!\vec{M} \; dV\!=\!\!\frac{1}{2}\int \, \mu_0(N\!\cdot\!\vec{M})\!\cdot\!\vec{M} \; dV$$

$$\vec{B}_{demag} = -\mu_0 N \cdot \vec{M}$$

•If the sample is an ellipsoid the demagnetizing field is uniform throughout the sample:

$$E_{\text{demag}} = \frac{1}{2} V \mu_0(N \cdot \vec{M}) \cdot \vec{M}$$
, $V - \text{volume of the sample}$

• N is a diagonal tensor if the semiaxes of the ellipsoid coincide with the axes of the coordination system.

•For the general ellipsoid sample we have [9]:

$$E_{demag} = \frac{1}{2} V \mu_0 (N \cdot \vec{M}) \cdot \vec{M} = \frac{1}{2} \mu_0 M^2 (N_a \alpha_1^2 + N_b \alpha_2^2 + N_c \alpha_3^2) \qquad \vec{M} = M(\alpha_1, \alpha_2, \alpha_3)$$

•For a spherical sample we have:

$$N = \begin{bmatrix} 1/3 & 0 & 0 \\ 0 & 1/3 & 0 \\ 0 & 0 & 1/3 \end{bmatrix} \Rightarrow E_{demag} = \frac{1}{2} \mu_0 M^2 \frac{1}{3} (\alpha_1^2 + \alpha_2^2 + \alpha_3^2) = \frac{1}{6} \mu_0 M^2 \quad \text{no dependence on the magnetic moment orientation}$$

•For an infinitely long cylinder*
$$N_c$$
 is null: $(\alpha_1, \alpha_2, \alpha_3) = (\sin(\theta)\cos(\phi), \sin(\theta)\sin(\phi), \cos(\theta))$

$$N = \begin{bmatrix} 1/2 & 0 & 0 \\ 0 & 1/2 & 0 \\ 0 & 0 & 0 \end{bmatrix} \Rightarrow E_{demag} = \frac{1}{2} \mu_0 M^2 \frac{1}{2} (\alpha_1^2 + \alpha_2^2) = \frac{1}{2} \mu_0 M^2 \frac{1}{2} (\sin^2(\theta) \cos^2(\phi) + \sin^2(\theta) \sin^2(\phi)^2) = E_{demag} = \frac{1}{4} \mu_0 M^2 \sin^2(\theta)$$

$$U_{\text{niaxial anisotropy-characteristic for elongated particles (see Stoner-Wohlfarth model)}$$

$$E_{demag} = \frac{1}{4} \mu_0 M^2 \sin^2(\theta)$$

Wohlfarth model)

$$N_{ellipsoid} = \begin{bmatrix} N_a & 0 & 0 \\ 0 & N_b & 0 \\ 0 & 0 & N_c \end{bmatrix}$$
Maciej Urbania

^{*}polar axis is a symmetry axis

•For infinitely expanded and/or very thin ellipsoid we have [9] (θ =90° – moments in-plane):

$$N = \begin{bmatrix} 0 & 0 & 0 \\ 0 & 0 & 0 \\ 0 & 0 & 1 \end{bmatrix} \Rightarrow E_{demag} = \frac{1}{2} \mu_0 M^2 \alpha_3^2 = \frac{1}{2} \mu_0 M^2 \cos^2(\theta)$$

The in-plane orientation of magnetic moment of thin plate is energetically favorable*

•The equation can be rewritten to often used form:

$$\begin{split} E_{\text{demag}} = & \frac{1}{2} \, \mu_0 M^2 (1 - \sin^2(\theta)) = \frac{1}{2} \, \mu_0 M^2 - \frac{1}{2} \, \mu_0 M^2 \sin^2(\theta) = K_0 + K_{\text{shape}}^V \sin^2(\theta), \\ with \quad & K_{\text{shape}}^V = -\frac{1}{2} \, \mu_0 M^2 \end{split}$$

•Magnetocrystalline and thin films shape anisotropy constants for thin films of elements at 4 K**:

	Fe (bcc)	Co (hcp)	Ni (fcc)
K ₁ [J/m ³]	54 800	760 000	-126 300
K ^V [J/m ³]	1 910 000	1 290 000	171 000

Shape anisotropy in thin films usually dominates over magnetocrystalline anisotropy

^{*}in case magnetocrystalline and other anisotropies favoring perpendicular orientation are absent

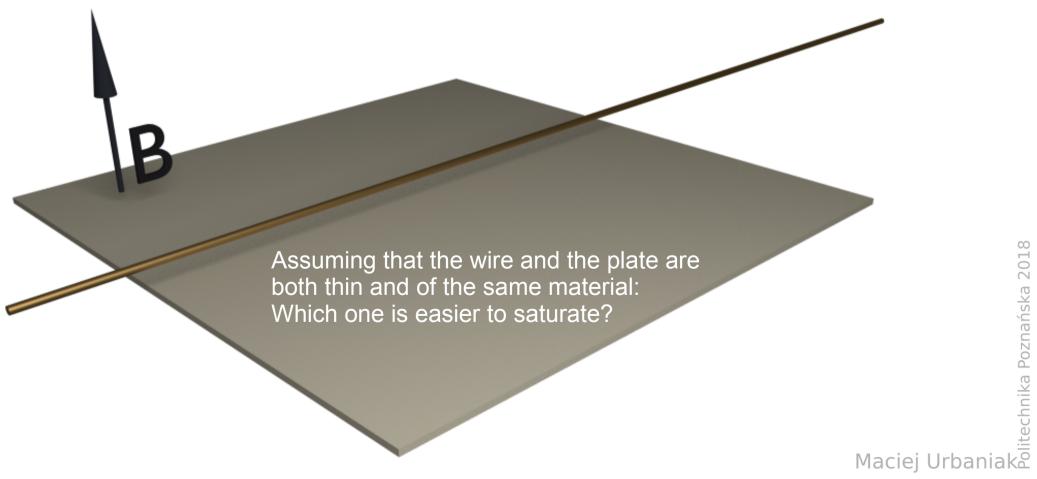
^{**}magnetization data from: Francois Cardelli Materials Handbook, Springer 2008 (p.502), http://books.google.pl

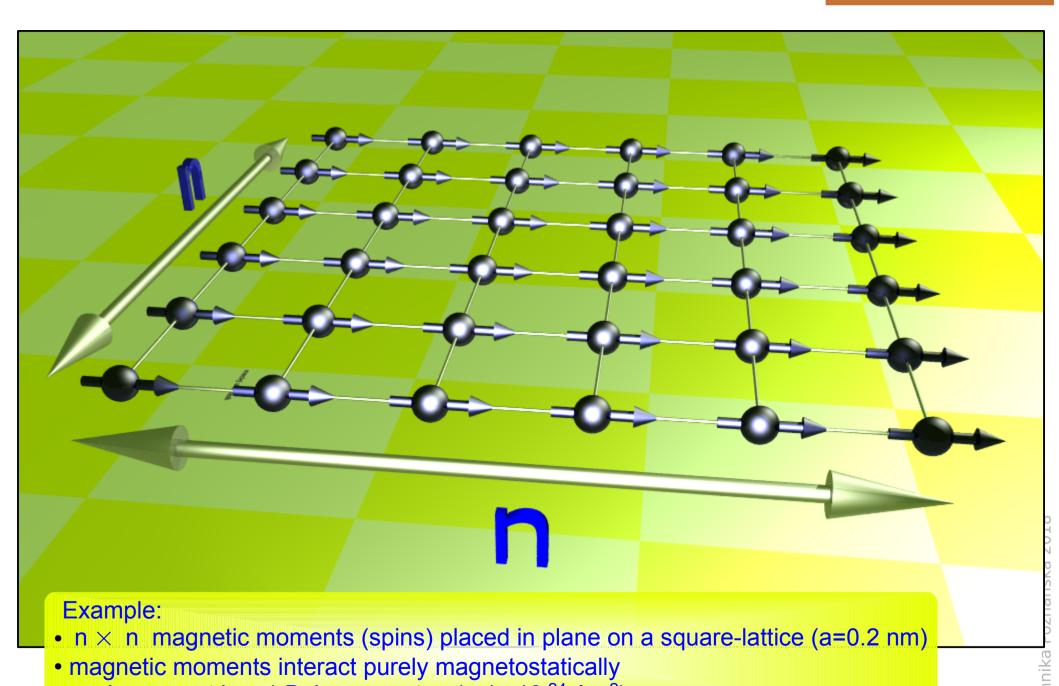
- •From Stoner-Wohlfarth model we have: $B_S = \frac{2K_1}{M}$
- •Substituting the expression for shape anisotropy of thin films $K_{shape}^{V} = -\frac{1}{2} \mu_0 M^2$ we get:

$$B_S = \mu_0 M_S$$
 or

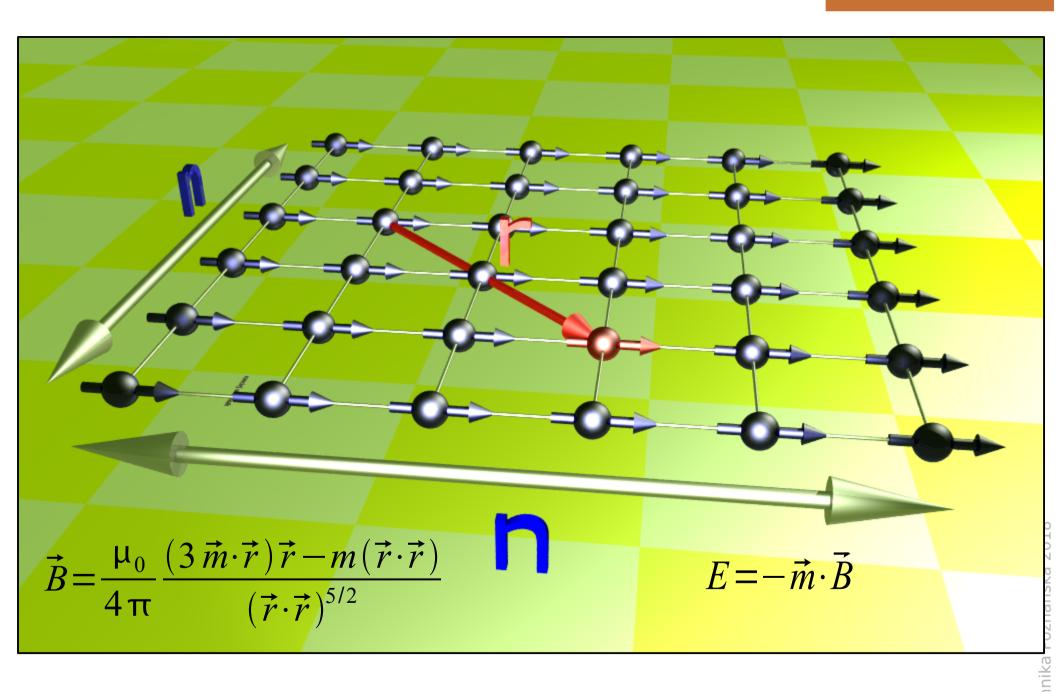
$$H_S = M_S$$

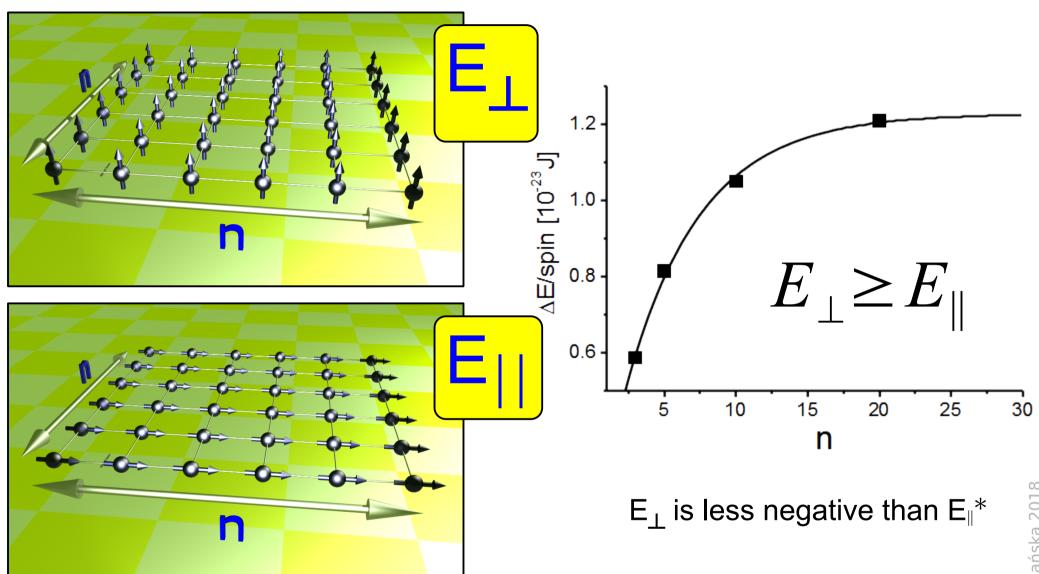
In macrospin approximation the perpendicular saturation field of thin film is equal to its magnetization.



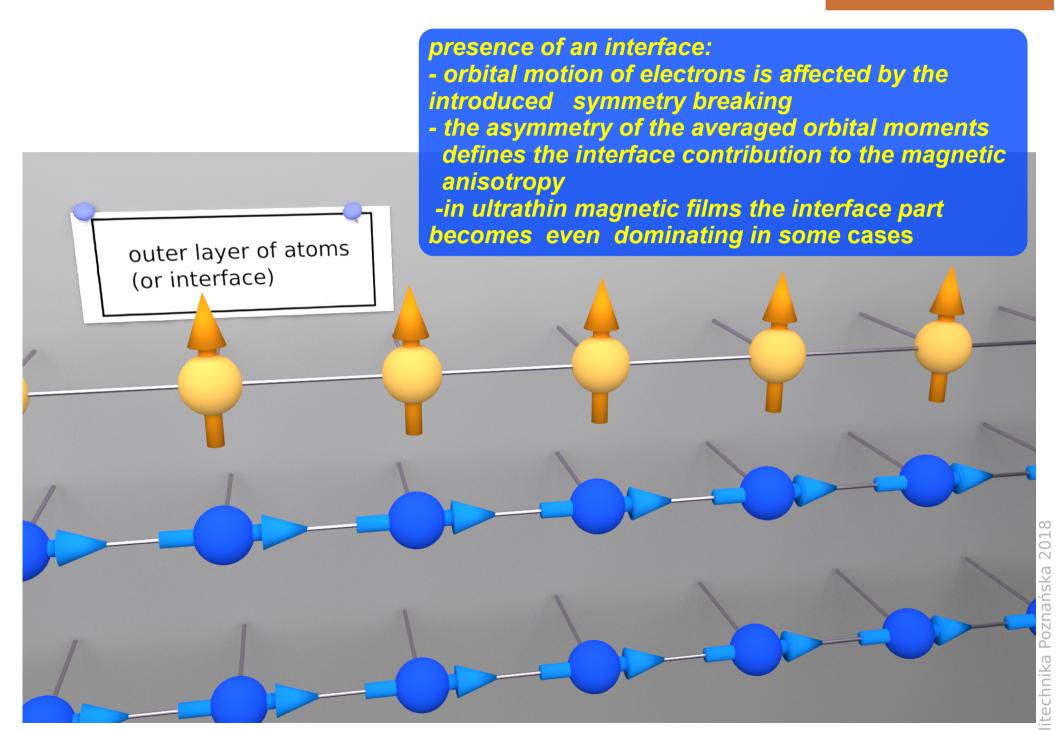


each moment is a 1 Bohr magneton (≈1×10⁻²⁴ Am²)

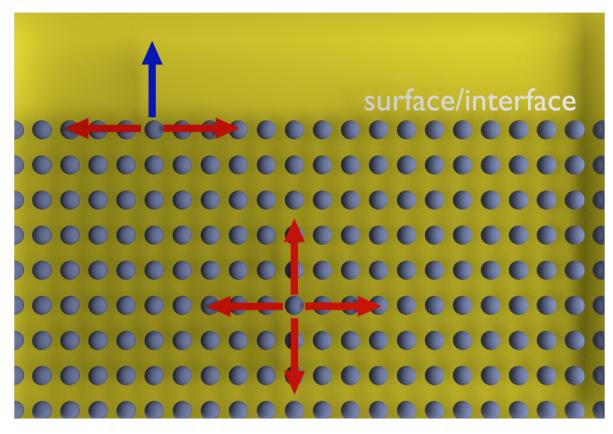




Magnetostatic interactions favor in-plane orientation of magnetic moments (spins) in thin magnetic films



•Due to broken symmetry at interfaces the anisotropy energy contains terms with lower order in direction cosines than in the infinite crystal.



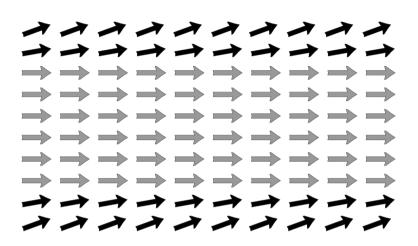
- •Energy of magnetic moments of atoms occupying lattice sites in the vicinity of the surface is different for two shown orientations
- •Each of the magnetocrystalline anisotropy constants can be phenomenologically divided into two parts, one related to volume contribution and the one to surface contribution [9]:

$$K^{eff} = K^{v} + K^{s}/t$$

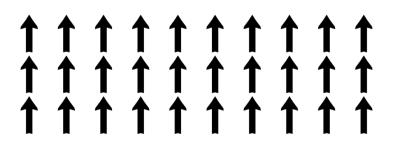
where *t* is the crystal thickness.

•Energy of magnetic moments of atoms occupying lattice sites far from the outer boundary of the crystal depends on the intrinsic symmetry of the crystal

•Let us assume that volume contribution to the anisotropy favors in-plane alignment of magnetic moments (it could be magnetocrystalline, shape, stress etc. anisotropy).



- •Due to perpendicular surface anisotropy the moments close to the surface (black arrows) are deflected out of plane
- •If the thickness of the sample/film is high the exchange coupling of the surface moments with the bulk ones keeps the overall moment of the sample nearly in plane



- •If the thickness of the film is low, and the surface anisotropy is strong enough all moments point perpendicular to plane.
- •Using macrospin approximation the total energy of the sample dependent on the orientation of magnetic moment can be written as [10] (we assume that the energy does not depend on azimuthal angle):

$$E_a = K_0 - K_2 \cos^2(\theta) - K_4 \cos^4(\theta) + \dots$$

-positive K_i favor perpendicular orientation

^{*}different notations of anisotropy constants can be encountered: R. Skomski et. al, Phys. Rev. B 58, 11138 (1998)

•Minimizing E_a with respect to θ yields the equilibrium angle:

$$\partial^2 E_a / \partial \theta^2 = 2K_2 \cos(\theta) \sin(\theta) + 4K_4 \cos^3(\theta) \sin(\theta) = 0 \quad \Rightarrow \quad \cos(\theta) \sin(\theta) (2K_2 + 4K_4 \cos^2(\theta)) = 0$$

•We have extrema for:

$$\theta = 0, \ \pi/2, \ \cos^2(\theta) = \frac{-K_2}{2K_4}$$

- •It can be shown that [10]:
- -for K₂>0 and K₄>0 the magnetization is perpendicular to the plane
- -for K₂>0 and 2K₄<-K₂ the canted magnetization is a ground state
- -the region for K₂<0 and 2K₄>-K₂ is called a coexistence region both perpendicular and in-plane orientations of magnetization correspond to local minimum; they are separated by energy barrier

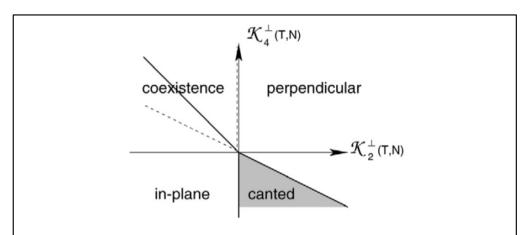


Fig. 50. Phase diagram in the $\mathcal{K}_2^{\perp}(T,N)-\mathcal{K}_4^{\perp}(T,N)$ -plane for the polar orientation of a thin film. The 'perpendicular' and the 'in-plane' phases are characterized by the polar angles $\theta=0$ and $\theta=\pi/2$, and the 'canted' phase by $0<\theta<\pi/2$. In the 'coexistence' region the perpendicular and the in-plane phase both refer to energy minima, and are separated by an energy barrier.

image source: P.J. Jensen, K.H. Bennemann, Surface Science Reports **61**, 129 (2006)

•Recalling the presence of surface anisotropy terms we get:

 $E_a = K_0 - \left(K_2^v + K_2^s/t\right)\cos^2(\%iteta) - \left(K_4^v + K_4^s/t\right)\cos^4(\%iteta) + \dots$ each anisotropy constant is divided into bulk (volume) and surface term

•Neglecting higher order terms we get the sample thickness for which the effective anisotropy is zero (neglecting constant K_0):

$$t_{RPT} = -\frac{K_2^s}{K_2^v}$$

•Usually, when considering thin films, the sample has two surfaces contributing surface anisotropy. As a consequence the multiplier 2 is added*:

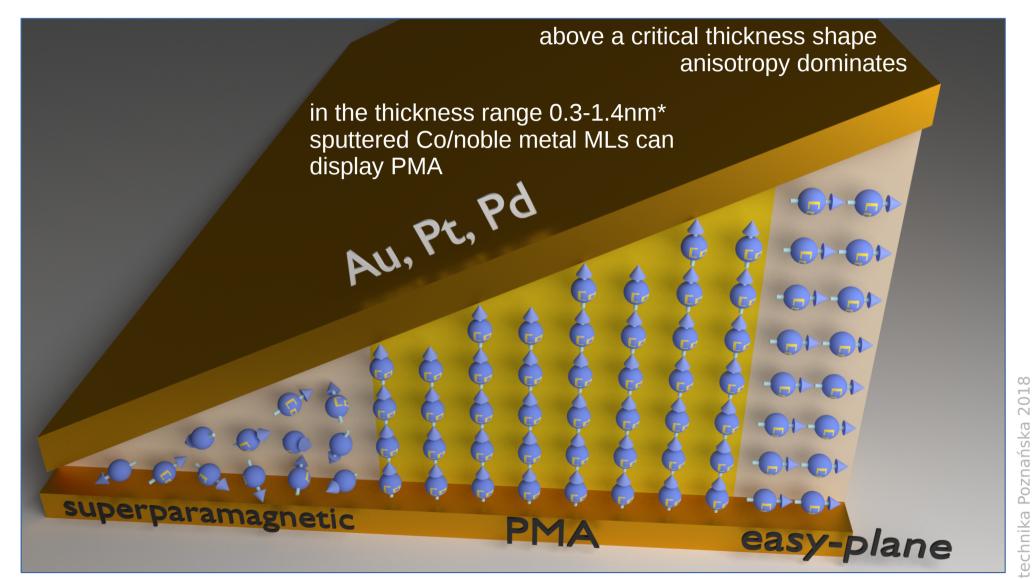
$$t_{RPT} = -\frac{2K_2^s}{K_2^v}$$

RPT – reorientation phase transition SRT -spin reorientation transition

- •For film thickness > t_{RPT} the magnetization of the film lies in-plane (if the external field is absent.
- •RPT may be caused by:
- -temperature change
- -change of the thickness of magnetic layer
- -change of the thickness of the overlayer

^{*}in general both surfaces can be characterized by different surface anisotropy constants.

Co based multilayers in which magnetic layer is sandwiched between noble metal spacer possess perpendicular magnetic anisotropy (PMA) in limited thickness range – this is due to **surface anisotropy** of the interfaces

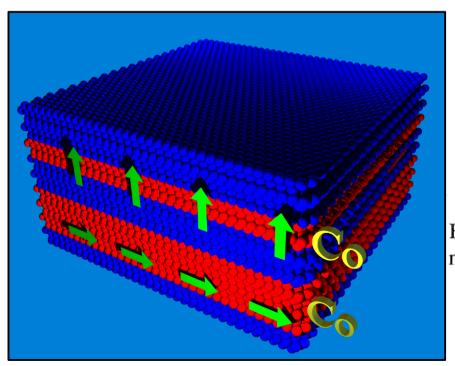


•From the expression with surface anisotropy we have:

$$K_{eff} = K_2^v + 2K_2^s/t$$

 $K_{eff} t = K_2^v t + 2K_2^s$

- •Plotting $K_{eff}t$ vs t one can determine volume and surface contributions to anisotropy with a linear fit:
- -K_V slope
- - $K_S \frac{1}{2}K_{eff} t$ for t=0



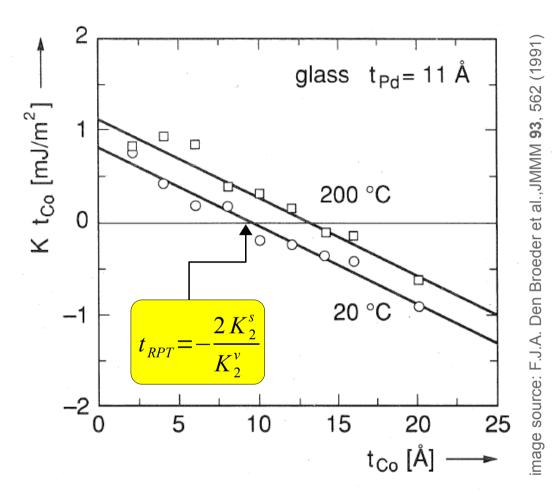


Fig. 2. Dependence of Kt_{Co} on t_{Co} for polycrystalline Co/Pd multilayers, deposited at $T_s = 20$ and 200°C.

- •RPT may be caused by:
- -temperature change
- -change of the thickness of magnetic layer
- -change of the thickness of the overlayer

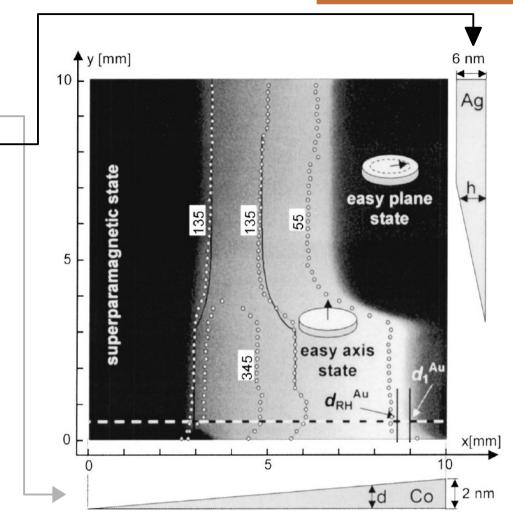


FIG. 1. Cobalt wedge remnant state image P(i,j) determined for a fully saturated sample in both $H_{\perp} > 0$ and $H_{\perp} < 0$ directions. On the basis of magnetometric analysis, localization of different magnetization states is marked. Points show the coercivity wall positions registered for different H_{\perp} field pulse ($\Delta = 900$ ms) magnitudes (measured in Oe). Solid black lines have been fitted to the coercivity wall data, registered at $H_{\perp} = 135$ Oe, using $H_{\rm C}(x,y)$ function with $h_{\rm C}^* = 0.8$ nm as the best fitting parameter. Below the horizontal dashed line in the gold region growth imperfections are clearly visible.

- •RPT may be caused by:
- -temperature change
- -change of the thickness of magnetic layer
- -change of the thickness of the overlayer

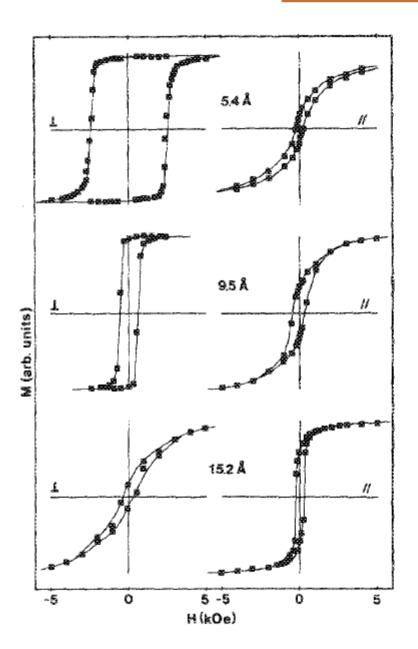


FIG. 4. Hysteresis loop with H perpendicular (1) and parallel ($\frac{1}{1}$) to the film plane, for Au/Co/Au sandwiches with t = 5.4, 9.5, and 15.4 Å, at T = 10 K.

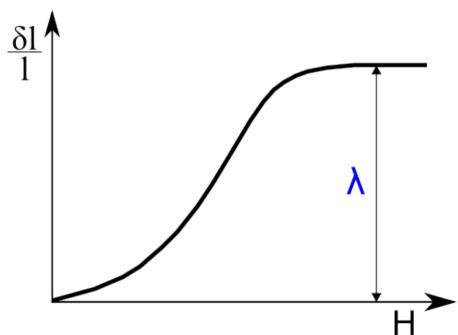
. Kakeshita et al., Appl. Phys. Lett. 77, 1502 (2000) เล zบาธ

- •Magnetostriction is a change of materials physical dimensions as a result of the change of the orientation of magnetization
- •The direction of magnetization changes under the influence of external field or temperature.

The relative deformation is usually small; of the order of 10^{-6} to 10^{-5} [6]; in Tb λ is approx. 0.002 at RT.

•The typical strain versus field dependence shows saturation which is expressed by the

value of magnetostriction constants λ :



•In giant magnetostriction materials the strain exceeds 0.5%

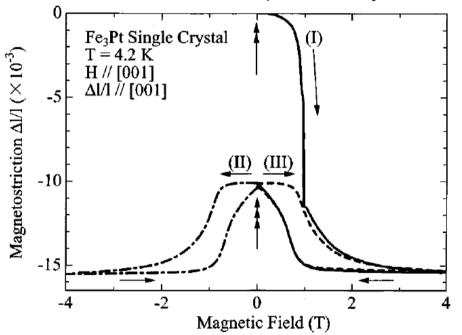


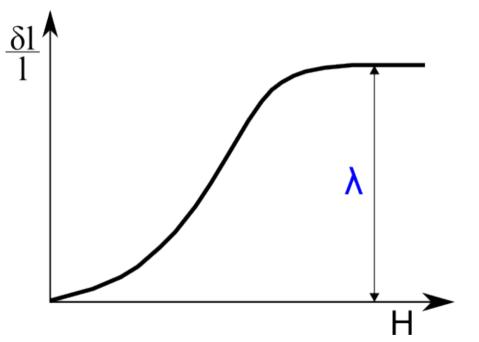
FIG. 4. Magnetostriction of an ordered Fe₃Pt. Strain of 1.5×10^{-2} is obtained by application of a magnetic field of 4 T, which is indicated by (I). The total strain comes to about 2.0×10^{-2} including the strain due to the thermally induced martensitic transformation shown in Fig. 3. The reversible strain is 5×10^{-3} by applying and removing the magnetic field, which is indicated by (II) and (III).

- •Magnetostriction is a change of materials physical dimensions as a result of the change of the orientation of magnetization
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The relative deformation is usually small; of the order of 10^{-6} to 10^{-5} [6]; in Tb λ is approx. 0.002 at RT.

•The typical strain versus field dependence shows saturation which is expressed by the

value of magnetostriction constants λ :



•The dependence dl/l(H) is different for different orientations of applied field relative to crystal axes

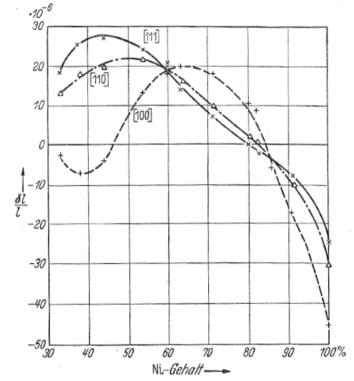


Abb. 186. Sättigungsmagnetostriktion von Einkristallen der Nickel-Eisen-Legierungen zwischen 30% und 100% Nickel für die drei kristallographischen Hauptrichtungen. [Nach F. Lichtenberger: Ann. Phys., Lpz. V, Bd. 10 (1932) S. 45.]

•In most practical applications the saturation distortion can be described by expression with small number of constants [11]:

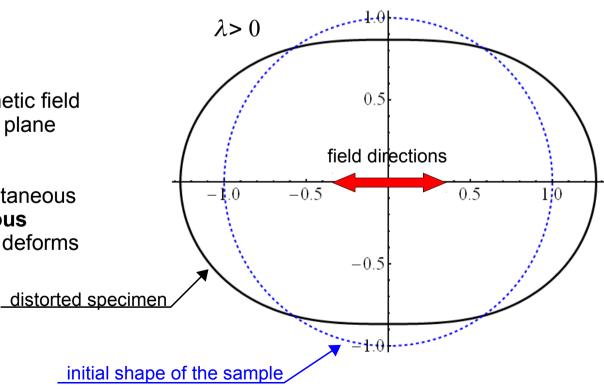
$$\lambda = \frac{3}{2} \lambda_{100} (\alpha_1^2 \beta_1^2 + \alpha_2^2 \beta_2^2 + \alpha_3^2 \beta_3^2 - \frac{1}{3}) + 3 \lambda_{111} (\alpha_1 \alpha_2 \beta_1 \beta_2 + \alpha_2 \alpha_3 \beta_2 \beta_3 + \alpha_3 \alpha_1 \beta_3 \beta_1),$$

where α_1 , α_2 , α_3 – direction cosines of magnetic moment direction; β_1 , β_2 , β_3 - direction cosines of the direction along which the deformation is measured.

•In amorphous and polycrystalline materials (without the texture) the above expression simplifies to:

$$\lambda = \frac{3}{2} \lambda_S (\cos^2 \theta - \frac{1}{3})$$

- Distortion along the external magnetic field direction is twice that observed for plane perpendicular to the field (see the drawing→)
- •Below Curie temperature the spontaneous magnetization leads to spontaneous distortion of lattice [9]: cubic cell deforms to tetragonal system



initial shape of the sample

- •Stress applied to a ferromagnetic body will affect the orientation of magnetization through magnetostriction [6].
- •The applied stress changes the magnetization reversal characteristics:

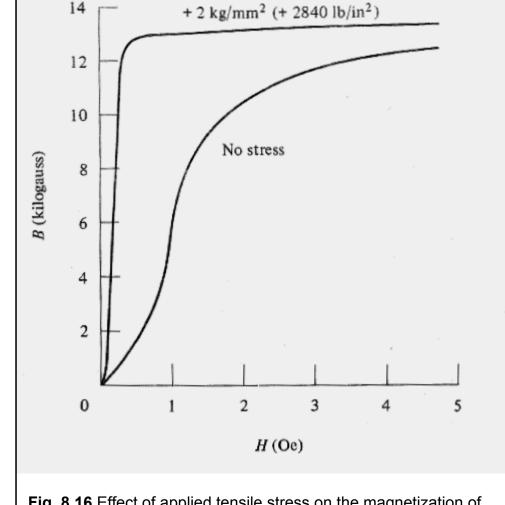


Fig. 8.16 Effect of applied tensile stress on the magnetization of 68 Permalloy. After Bozorth [G.4].

image from: B. D. Cullity, Introduction to magnetic materials, Addison-Wesley, Reading, Massachusetts 1972 chnika Poznańska 2018

^{*}called inverse magnetostrictive effect, too

•The part of the energy of a cubic crystal depending on magnetic moment orientation and the stress applied to crystal can be shown to be [3]:

$$E = K_1(\alpha_1^2\alpha_2^2 + \alpha_2^2\alpha_3^2 + \alpha_3^2\alpha_1^2) + \dots - \frac{3}{2}\lambda_{100}\sigma(\alpha_1^2\gamma_1^2 + \alpha_2^2\gamma_2^2 + \alpha_3^2\gamma_3^2)$$

$$-3\lambda_{111}\sigma(\alpha_1\alpha_2\gamma_1\gamma_2 + \alpha_2\alpha_3\gamma_2^2 + \alpha_3\alpha_1\gamma_3\gamma_1), \qquad \qquad \gamma_{1,} \gamma_{2,} \gamma_{3} \quad \text{- direction cosines of the external stress } \sigma$$

•When the magnetostriction is isotropic ($\lambda_{100} = \lambda_{111} = \lambda_{si}$) the last two terms reduce to*:

$$E_{stress} = -\frac{3}{2} \lambda_{si} \sigma \cos^2 \theta$$

where θ is the angle between macrospin (magnetization) and the the stress directions

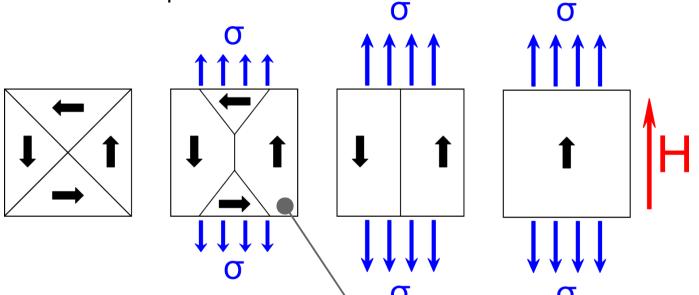
•The effect of stress on isotropic sample depends on the sign of the $\lambda_{si}\sigma$ product

•The effect of stress is to introduce additional anisotropy to the ferromagnetic system

- •The effect of the stress on magnetization reversal for positive $\lambda_{si}\sigma$ product [3]:
- 1) the magnetic moments within the specimen point in one of four easy directions
- 2) the application of tensile stress causes domains with magnetic moment perpendicular to the stress to dwindle
- 3) still higher stress leaves only magnetic moments parallel to the stress

4) Application of the weak magnetic field is sufficient to move 180 Deg domain wall and

saturate the specimen



$$E_{stress} = -\frac{3}{2} \lambda_{si} \sigma \cos^2 \theta$$

lower energy under stress

- •If compressive stress was applied instead "vertical domains" would disappear and the field would initially (for small H) be perpendicular to magnetic moments.
- •In Ni samples the stress of 6.4×10^6 Pa [3] causes stress anisotropy to be roughly equal to magnetocrystalline anisotropy.

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